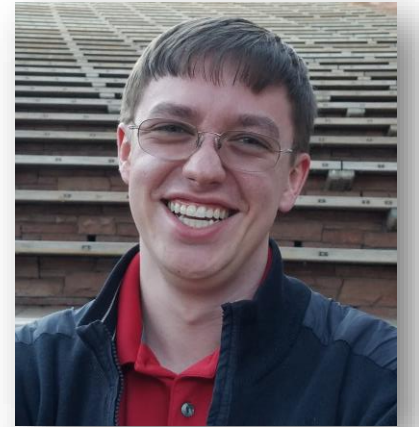


***Project #36A-L: Microstructural Evolution in
Titanium Alloys Under Additive Manufacturing
Conditions***

***Semi-annual Spring Meeting
April 12-14, 2022***



- Student: Alec Saville (Mines)
- Faculty: Amy Clarke, Kester Clarke (Mines)
- Industrial Mentors: Adam Pilchak (MRL), S. Lee Semiatin (AFRL), Jessica Buckner & Andrew Kustas (SNL)
- Other Participants: Sven Vogel (LANL), Adam Kreuziger & Jake Benzing (NIST)

Project 36A-L: Microstructural Evolution in Titanium Alloys Under Additive Manufacturing Conditions



- Student: Alec Saville (Mines)
- Advisor(s): Amy Clarke (Mines)

Project Duration
PhD: 2018 - 2022

- **Problem:** Control of material properties in metallic additive manufacturing (AM) is difficult due to a lack of background knowledge on material evolution within AM production methods.
- **Objective:** Understand microstructural evolution of $\alpha + \beta$ and binary alloys under AM conditions.
- **Benefit:** Greater understanding of microstructural evolution in AM will inform predictive capabilities and improve performance of AM parts.

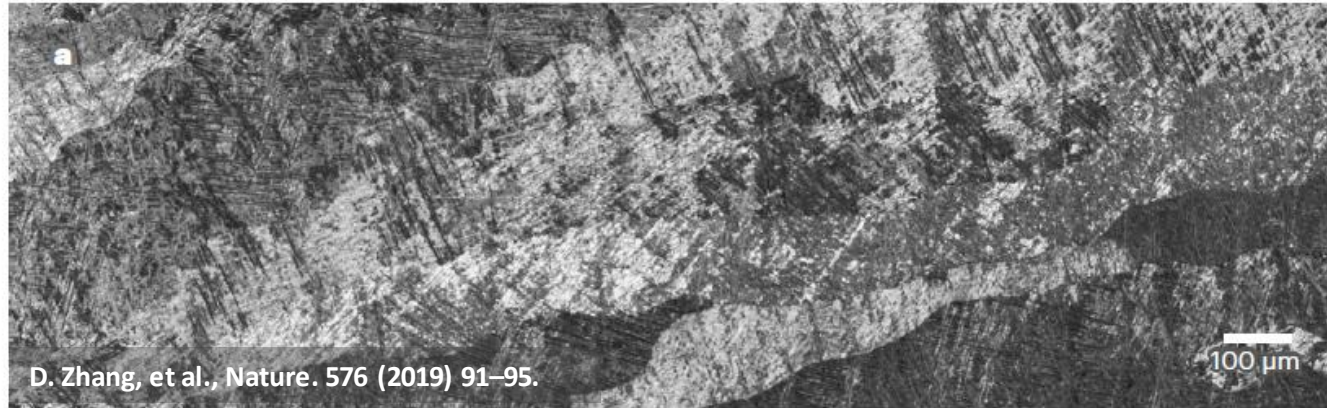
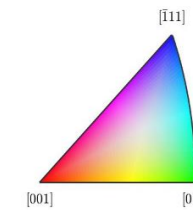
- Recent Progress**
- Publishing WAAM Ti-6Al-4V texture and microstructure work
 - Correlating solid state and parent grain size texture-microstructure relationships to EBM-PBF and WAAM Ti-6Al-4V
 - Mechanical testing of EBM-PBF Ti-6Al-4V specimens
 - Collating research for PhD defense

Metrics		
Description	% Complete	Status
1. EBM-PBF Ti-6Al-4V Microstructure, Texture, and Solidification	100%	●
2. MAUD Rietveld Refinement Tutorial	100%	●
3. EBM-PBF Elastic Modulus and Mechanical Testing	90%	●
4. WAAM Ti-6Al-4V Microstructural and Texture Evolution	85%	●
5. Ti-10-2-3 Parent Phase Reconstruction	75%	●

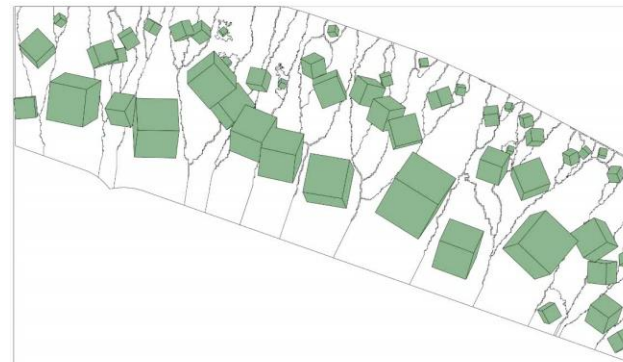
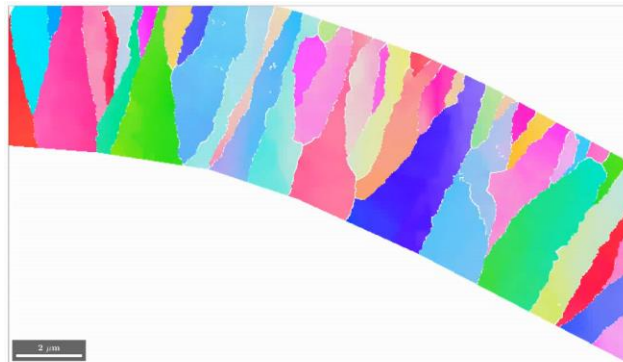
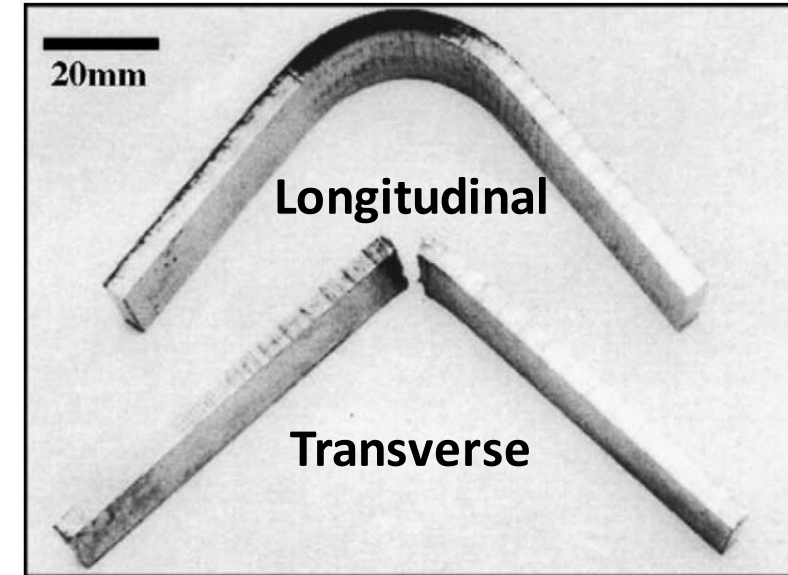
Background and Previous Work

Challenges in AM – There Are Many

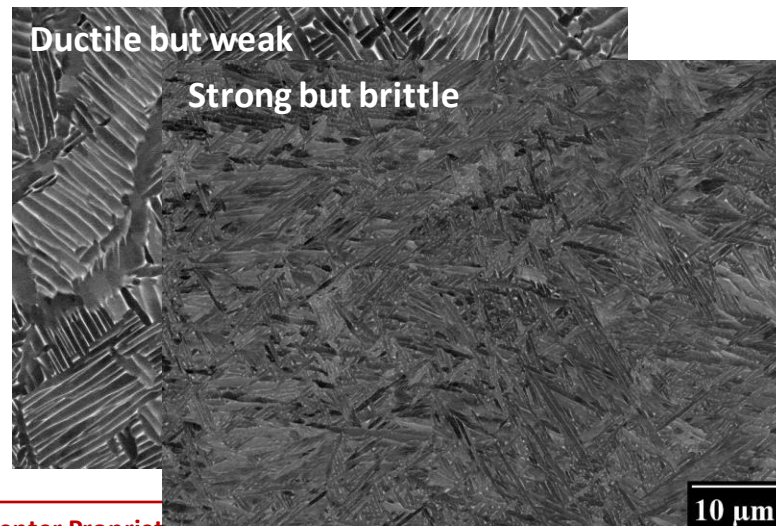
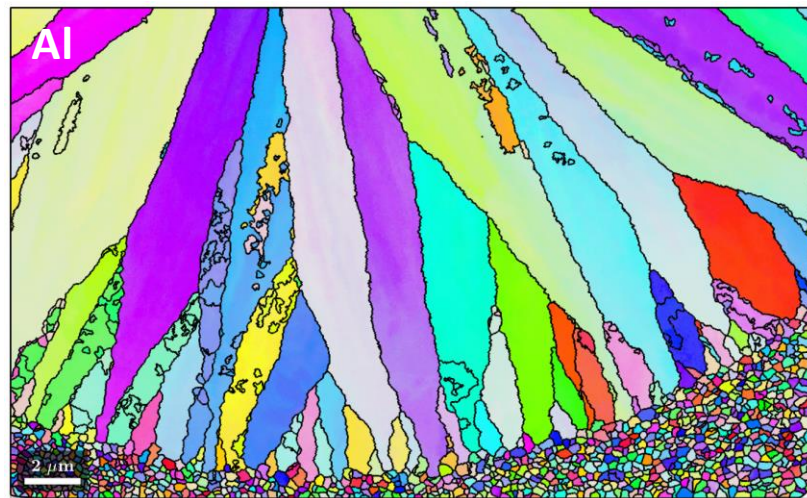
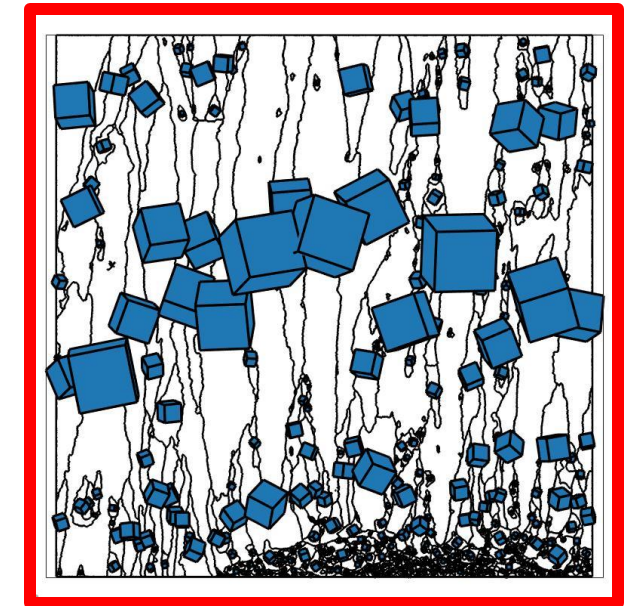
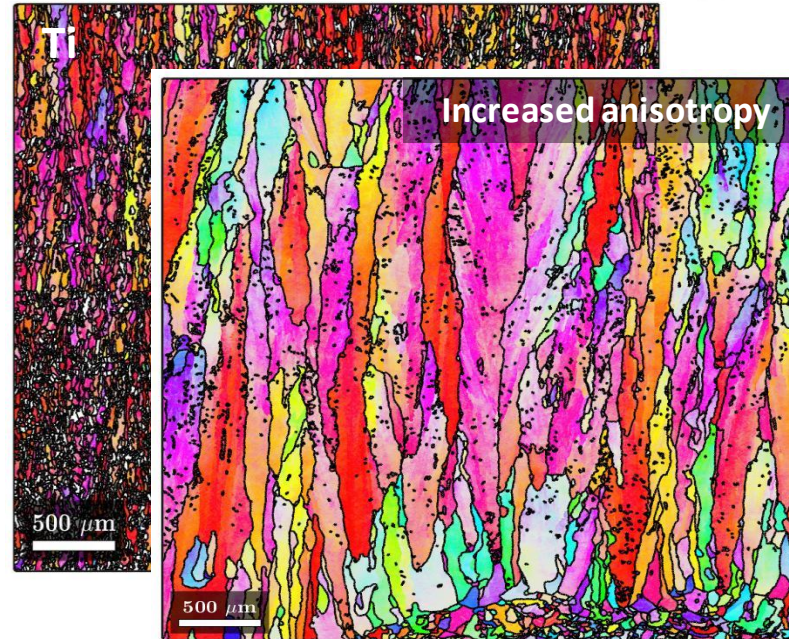
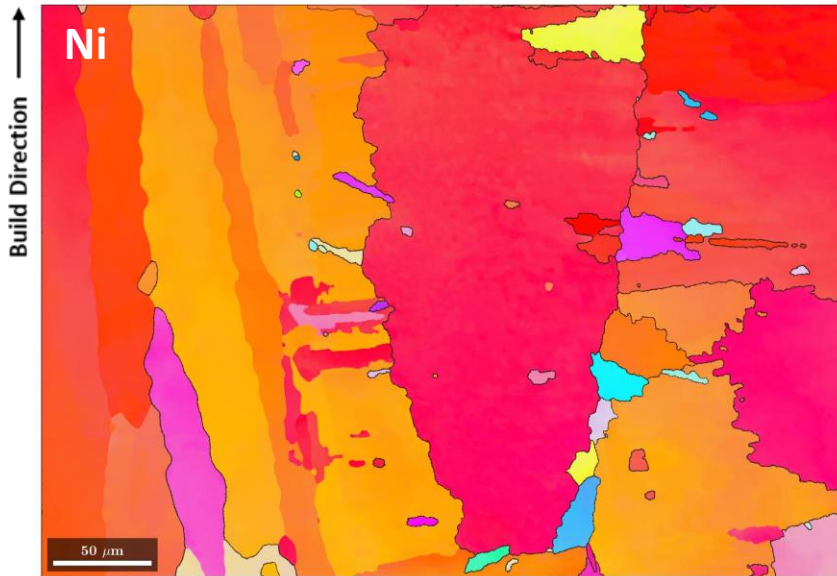
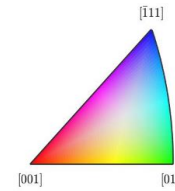
Microstructural Control



D. Zhang, et al., Nature. 576 (2019) 91–95.

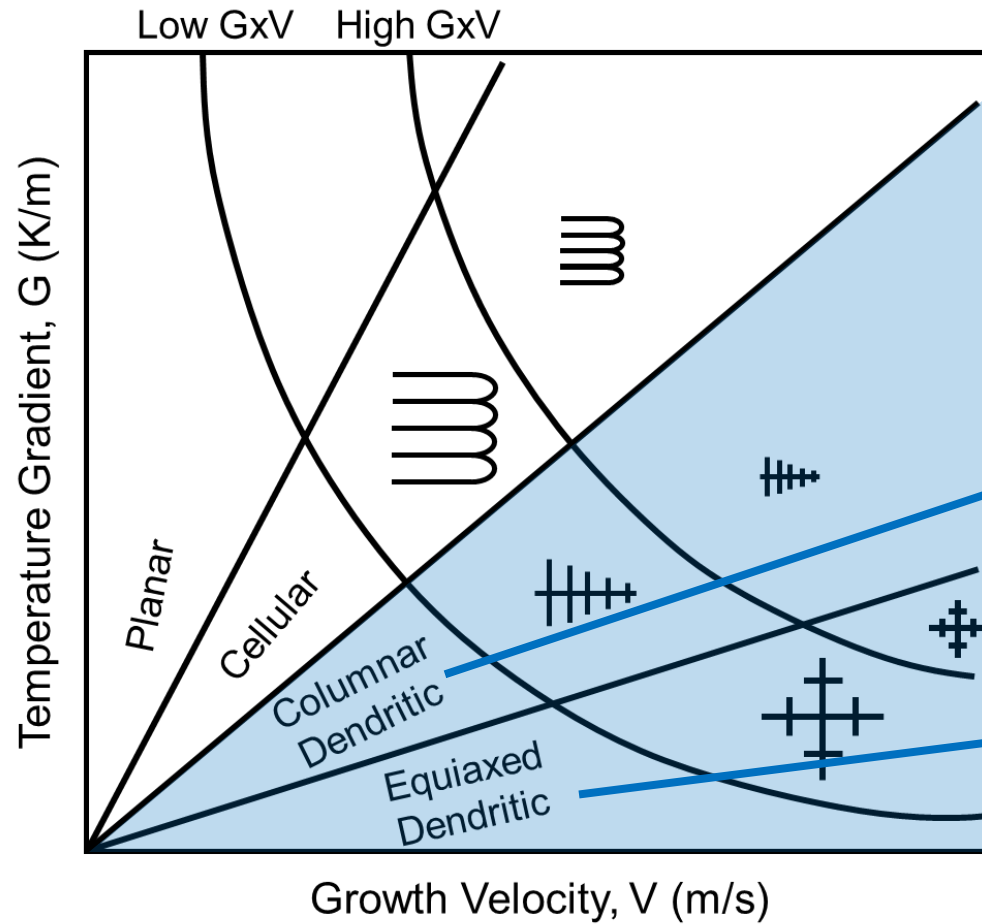


Microstructural Evolution Is Complex

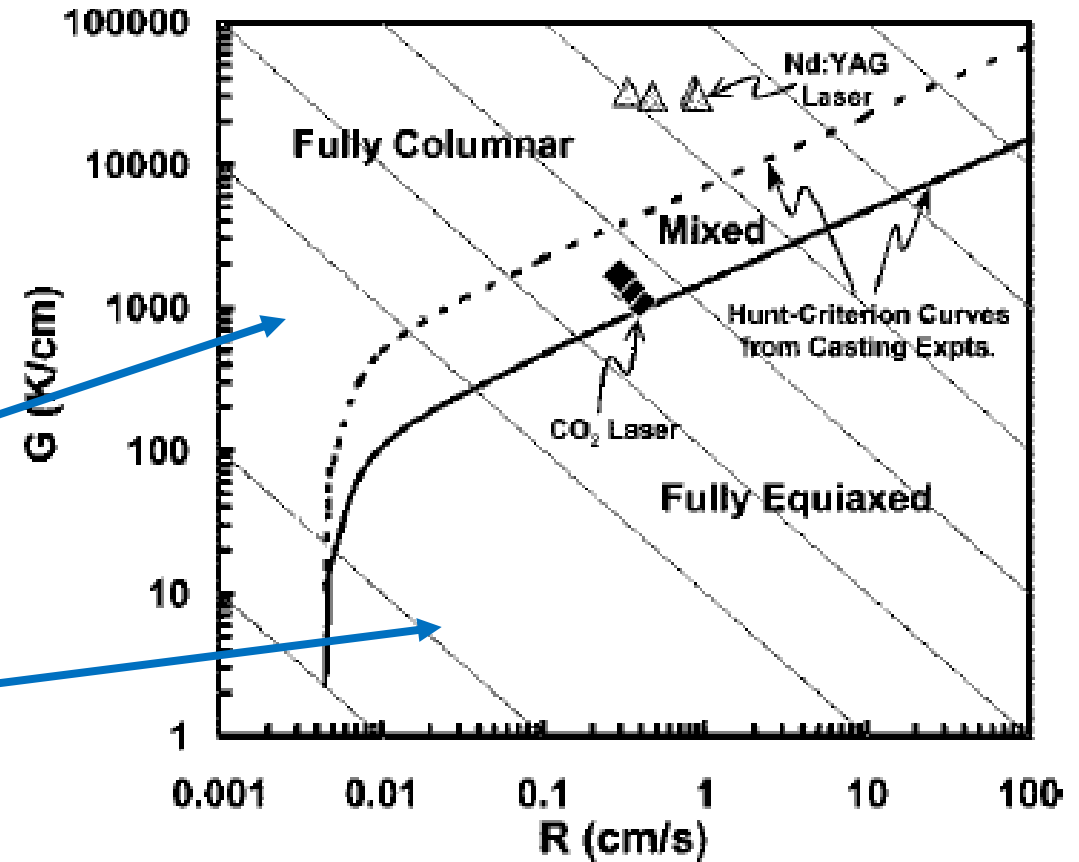


Microstructural control is required during solidification AND in the solid state

Solidification Microstructure in AM



Adapted from, S. Kou, Welding Metallurgy, John Wiley & Sons, Inc., 2002.

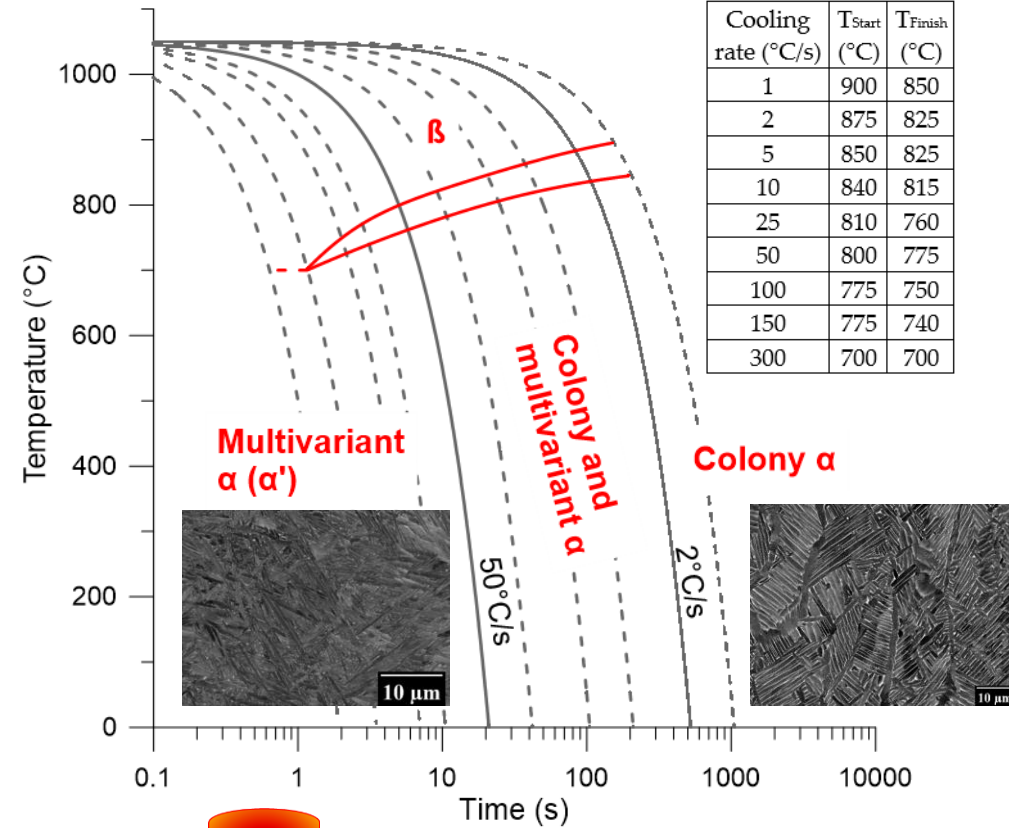
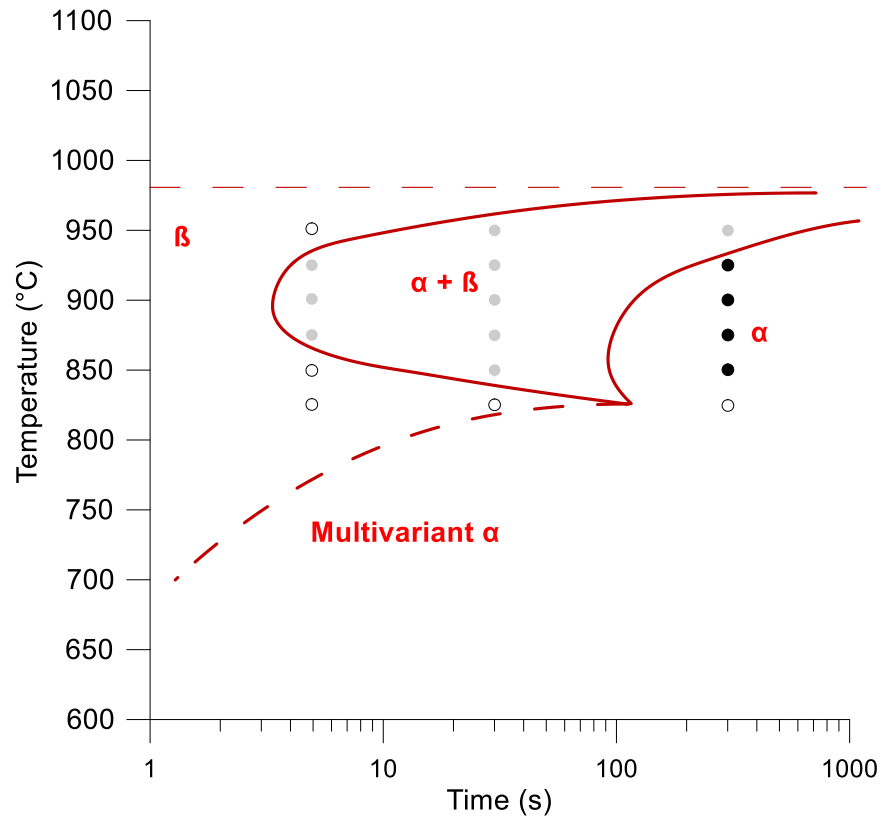


P.A. Kobryn, S.L. Semiatin, Microstructure and texture evolution during solidification processing of Ti-6Al-4V, Journal of Materials Processing Technology. 135 (2003) 330-339.

Solid State Microstructure in AM: Ti-6Al-4V

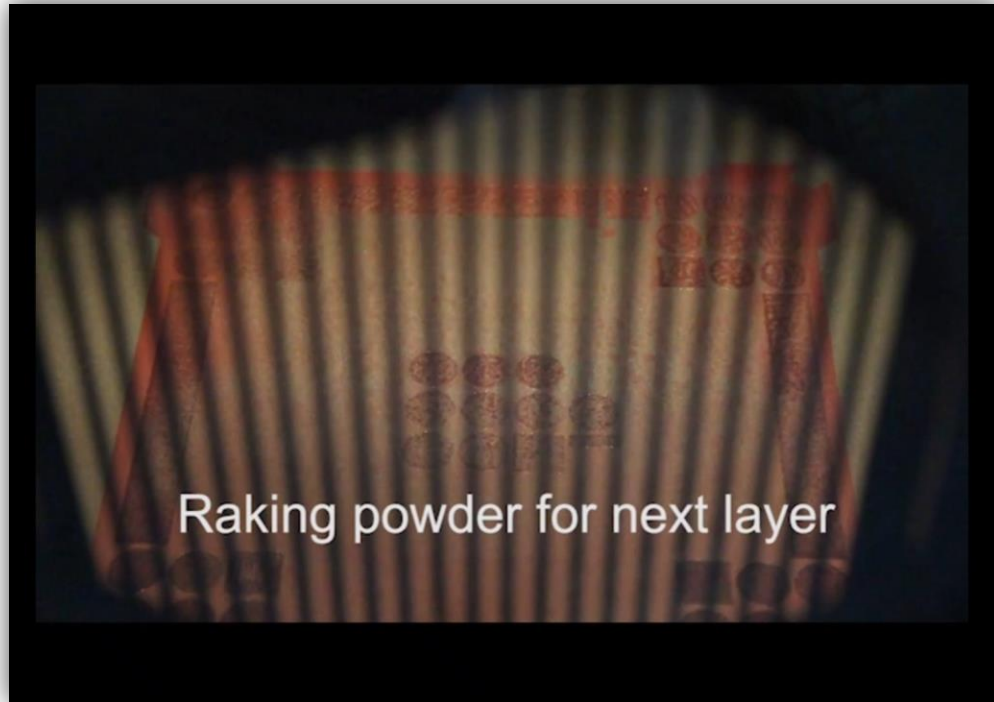


B. McArthur, Effects of Thermal Processing Variations on Microstructure and High Cycle Fatigue of Beta-STOATi-6Al-4V, (2017), 82.



Exploring Two Different Build Processes

Electron Beam Powder Bed Fusion



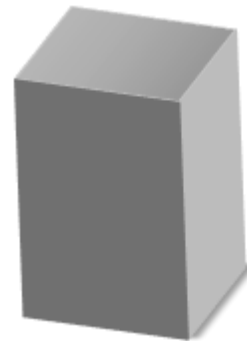
Directed Energy Deposition



Electron Beam Powder Bed Fusion



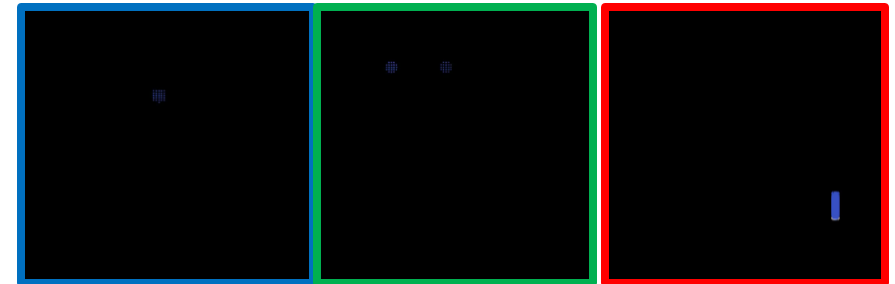
Credit: Arcam



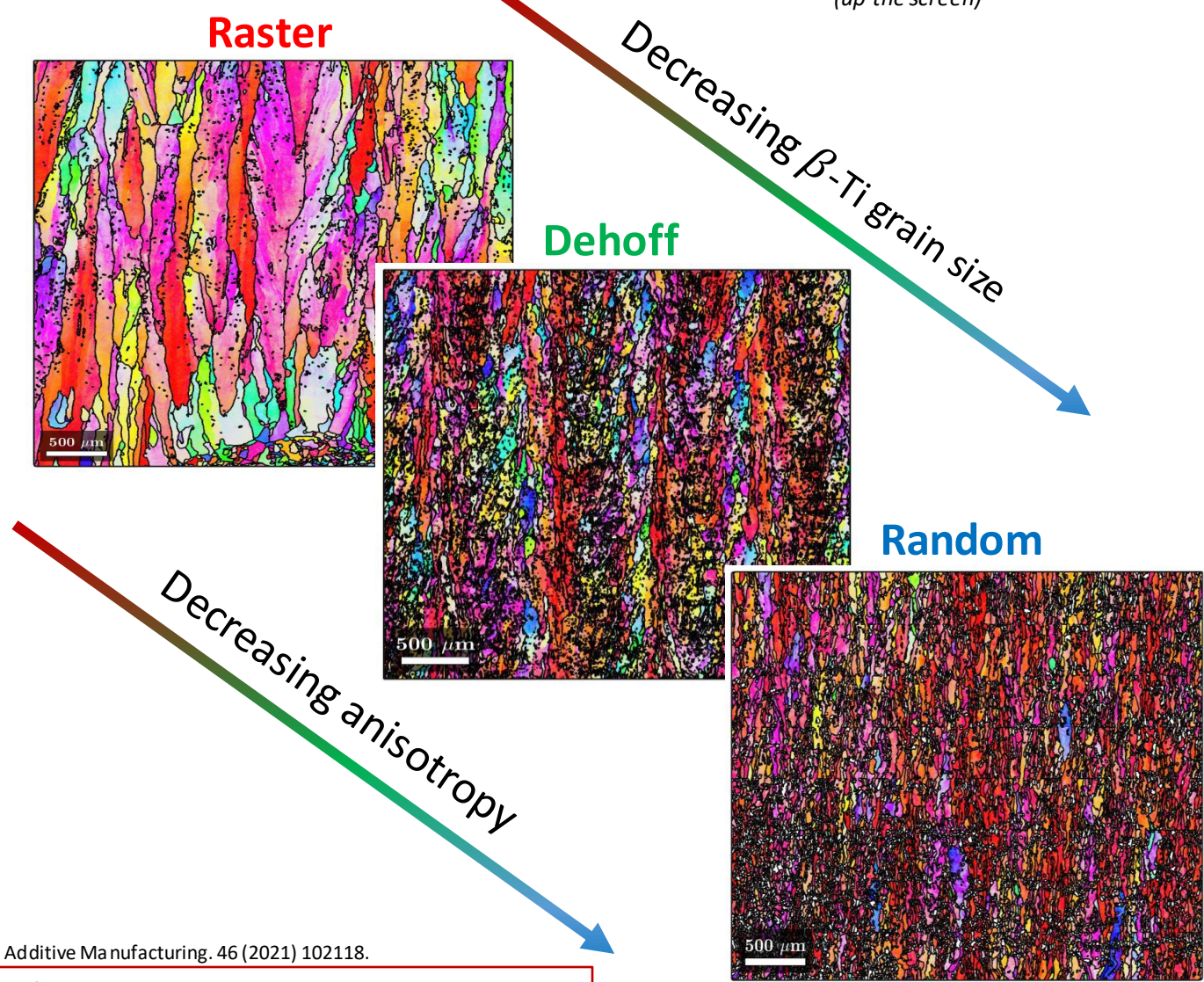
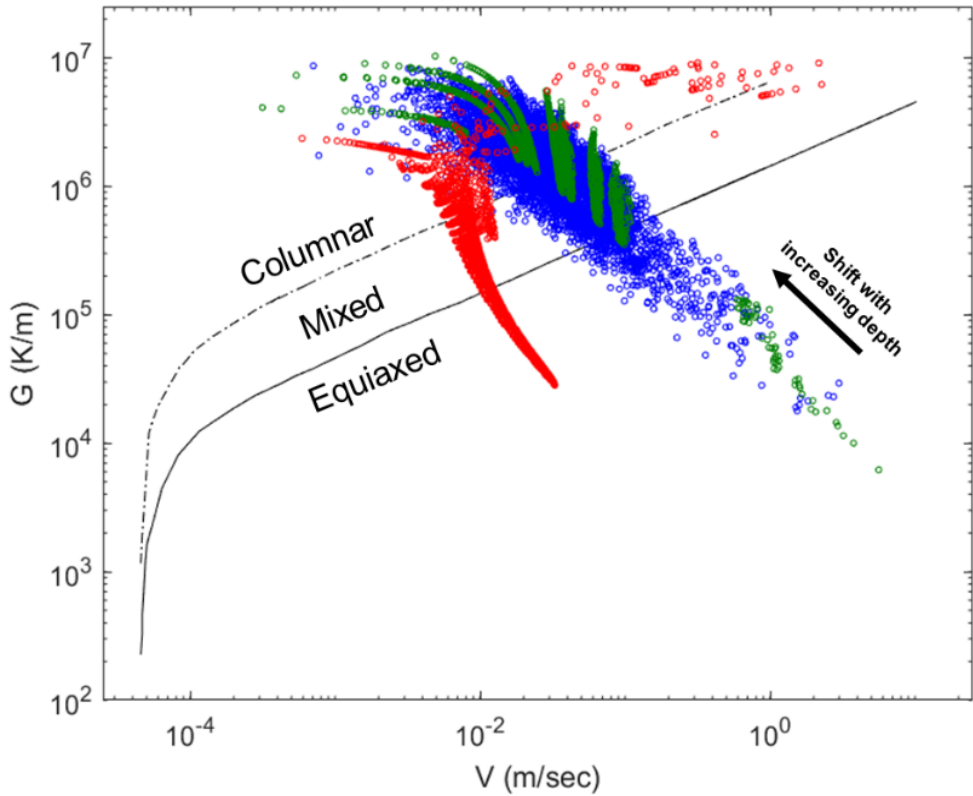
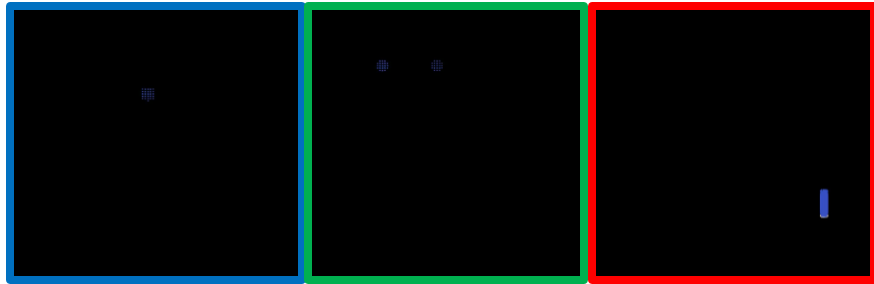
Random

Dehoff

Raster

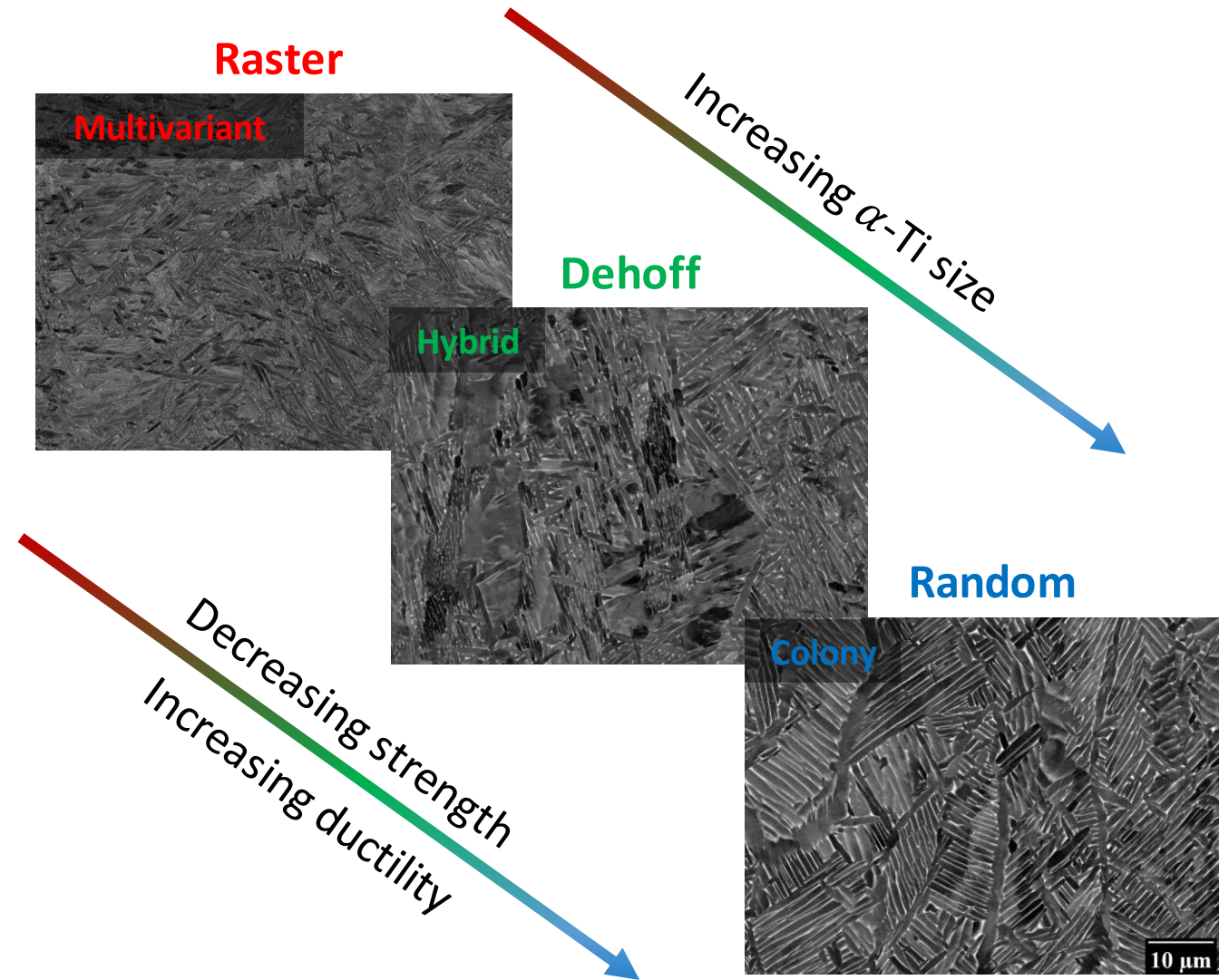
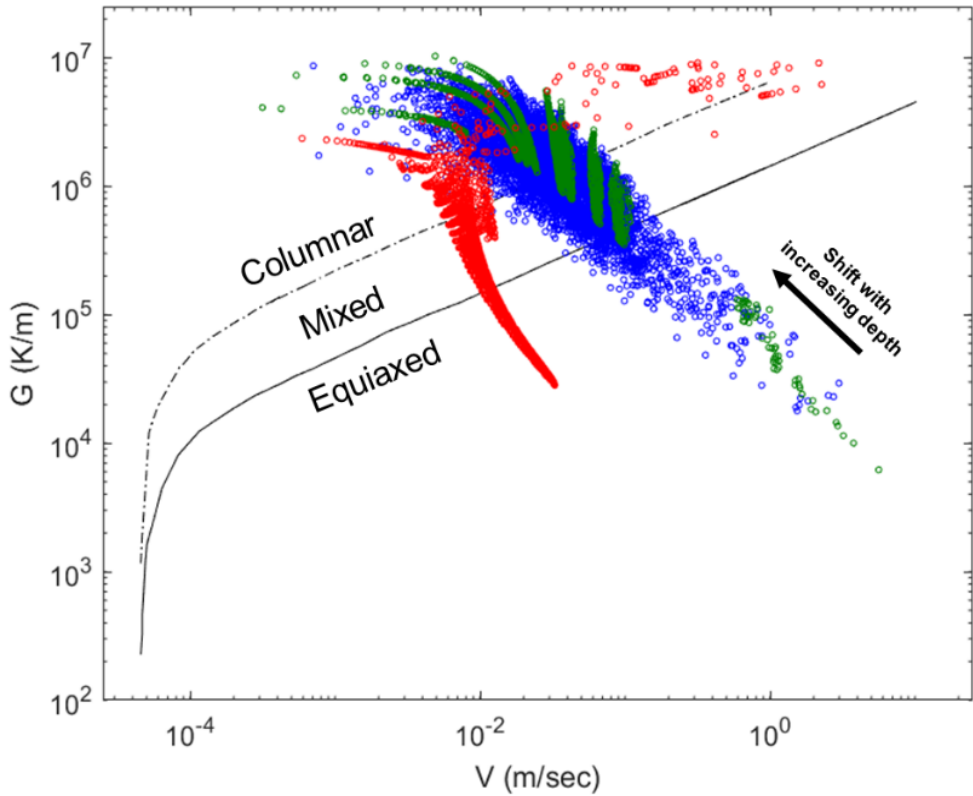
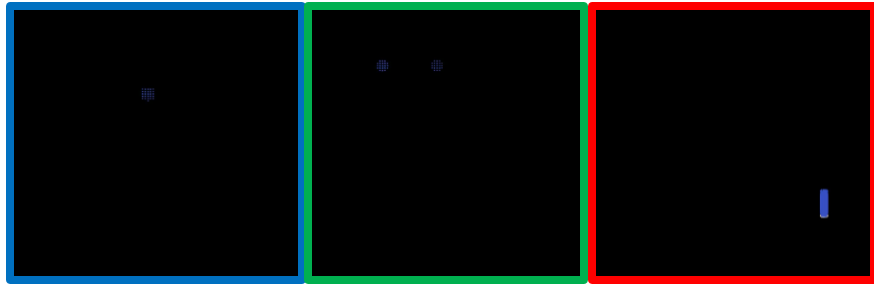


Changing Scan Strategy: Solidification

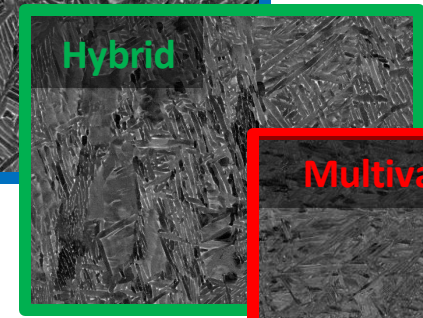
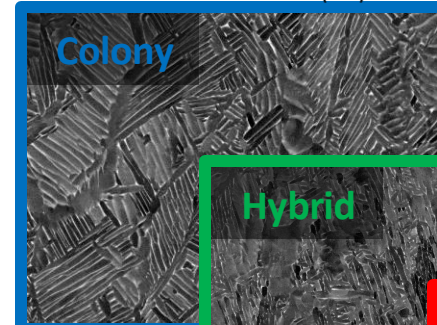
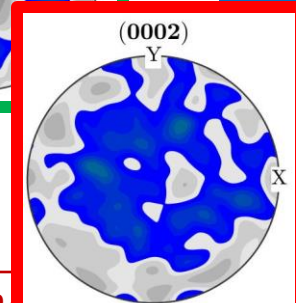
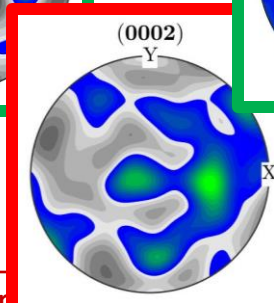
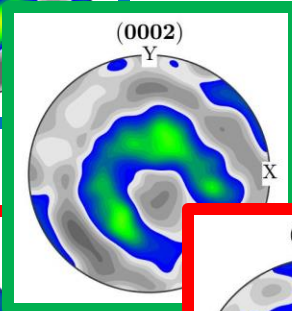
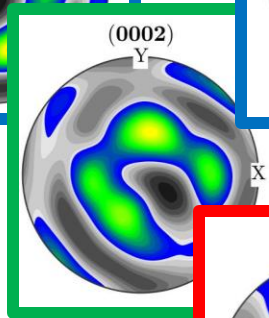
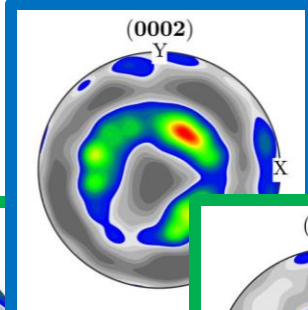
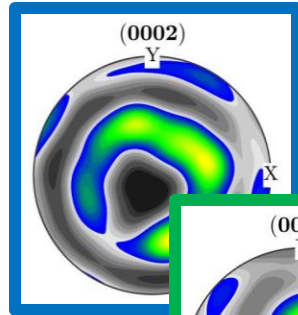
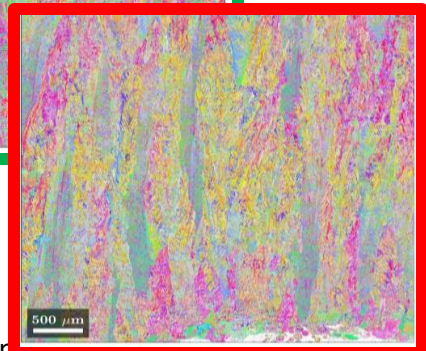
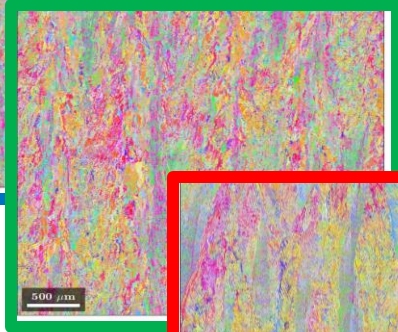
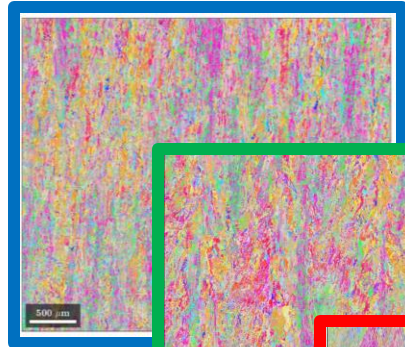
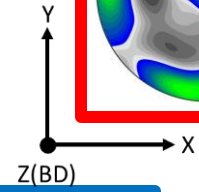
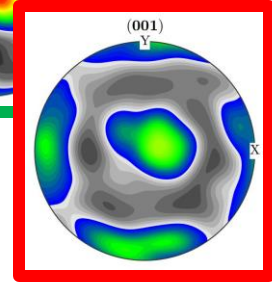
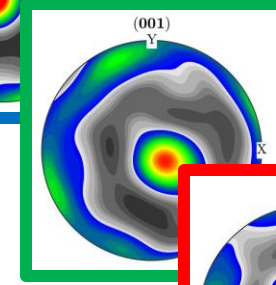
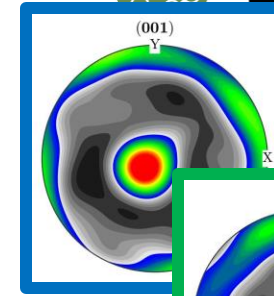
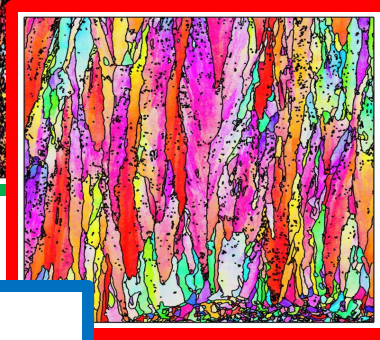
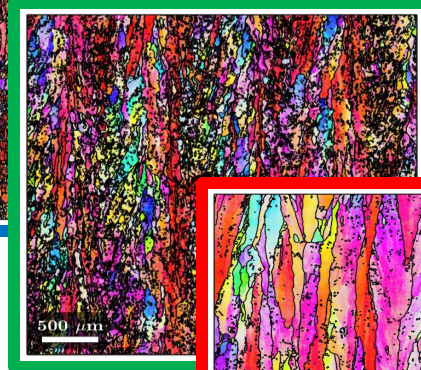
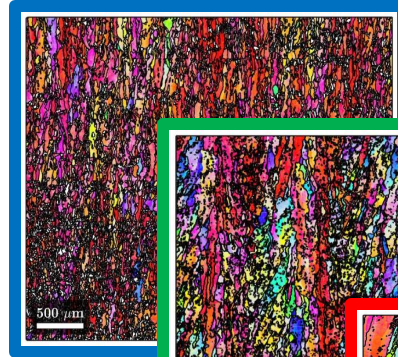
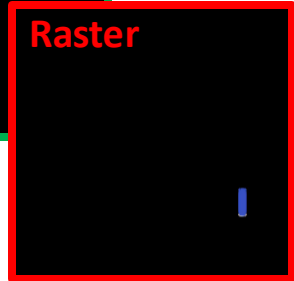
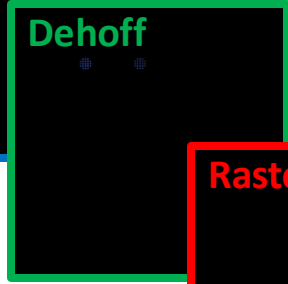
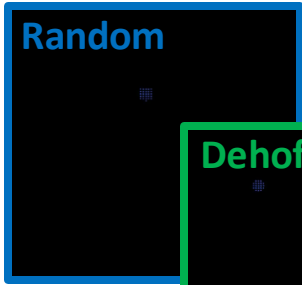


A.I. Saville, et al., Texture evolution as a function of scan strategy and build height in electron beam melted Ti-6Al-4V, Additive Manufacturing. 46 (2021) 102118.

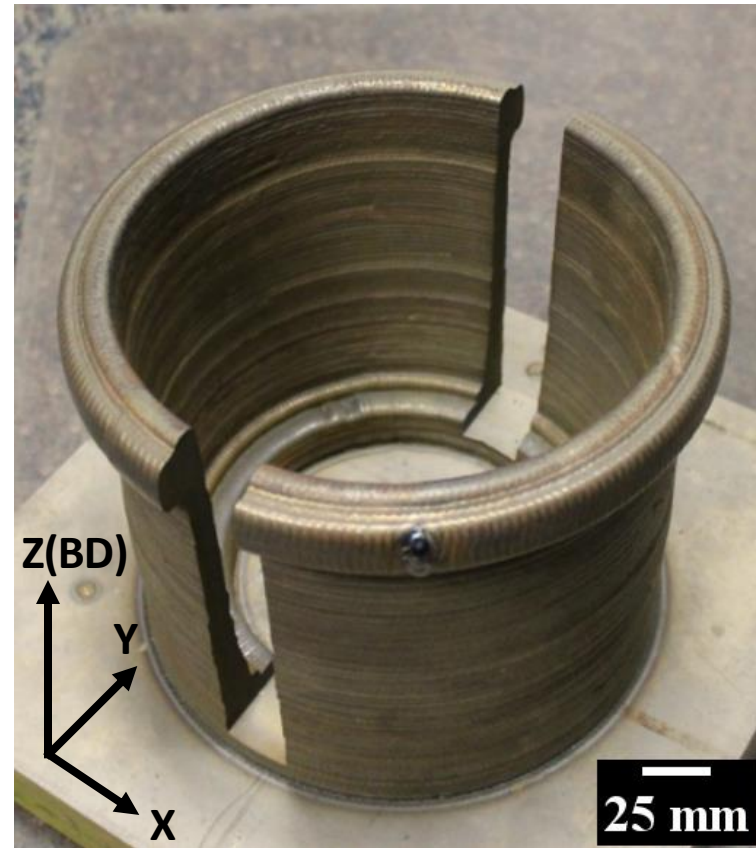
Changing Scan Strategy: Solid State



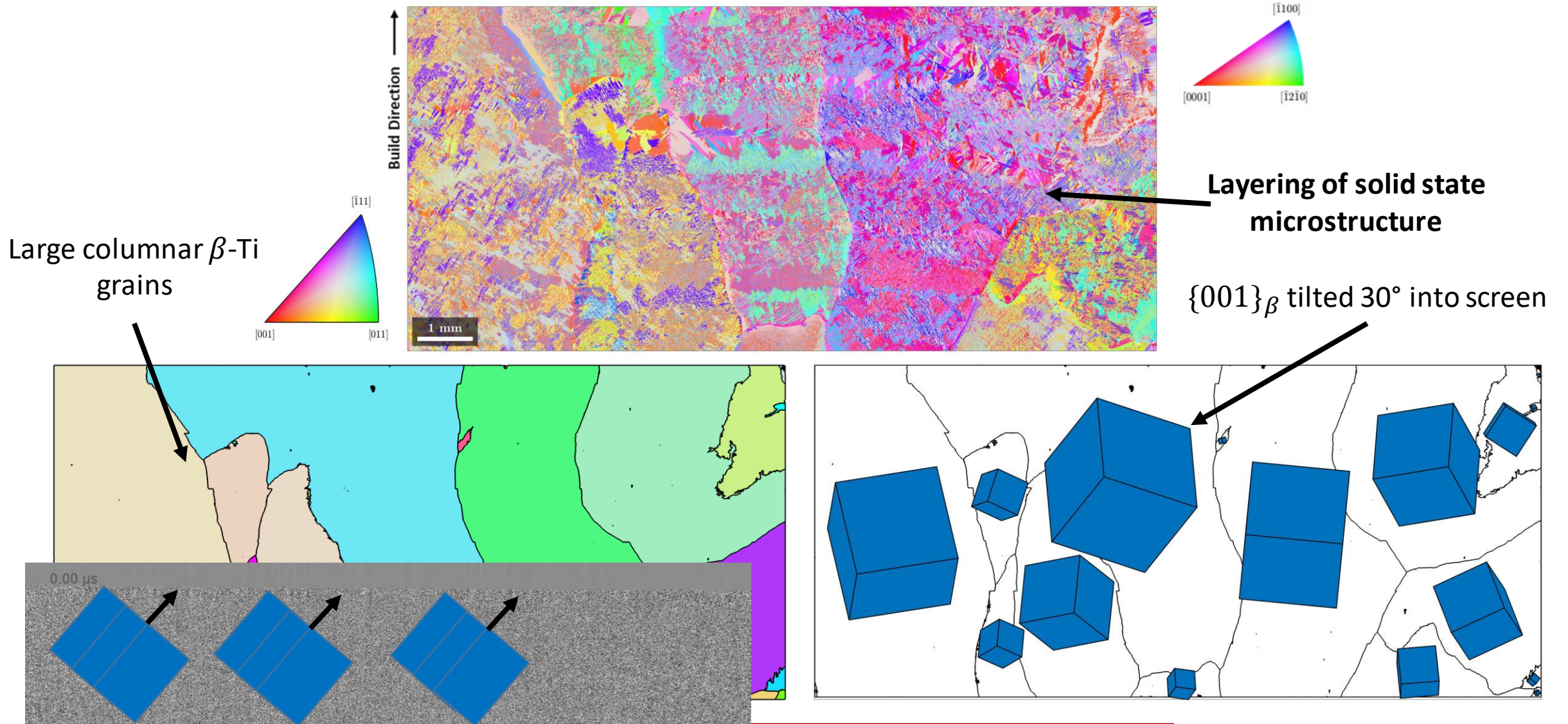
Texture and Microstructural Scale



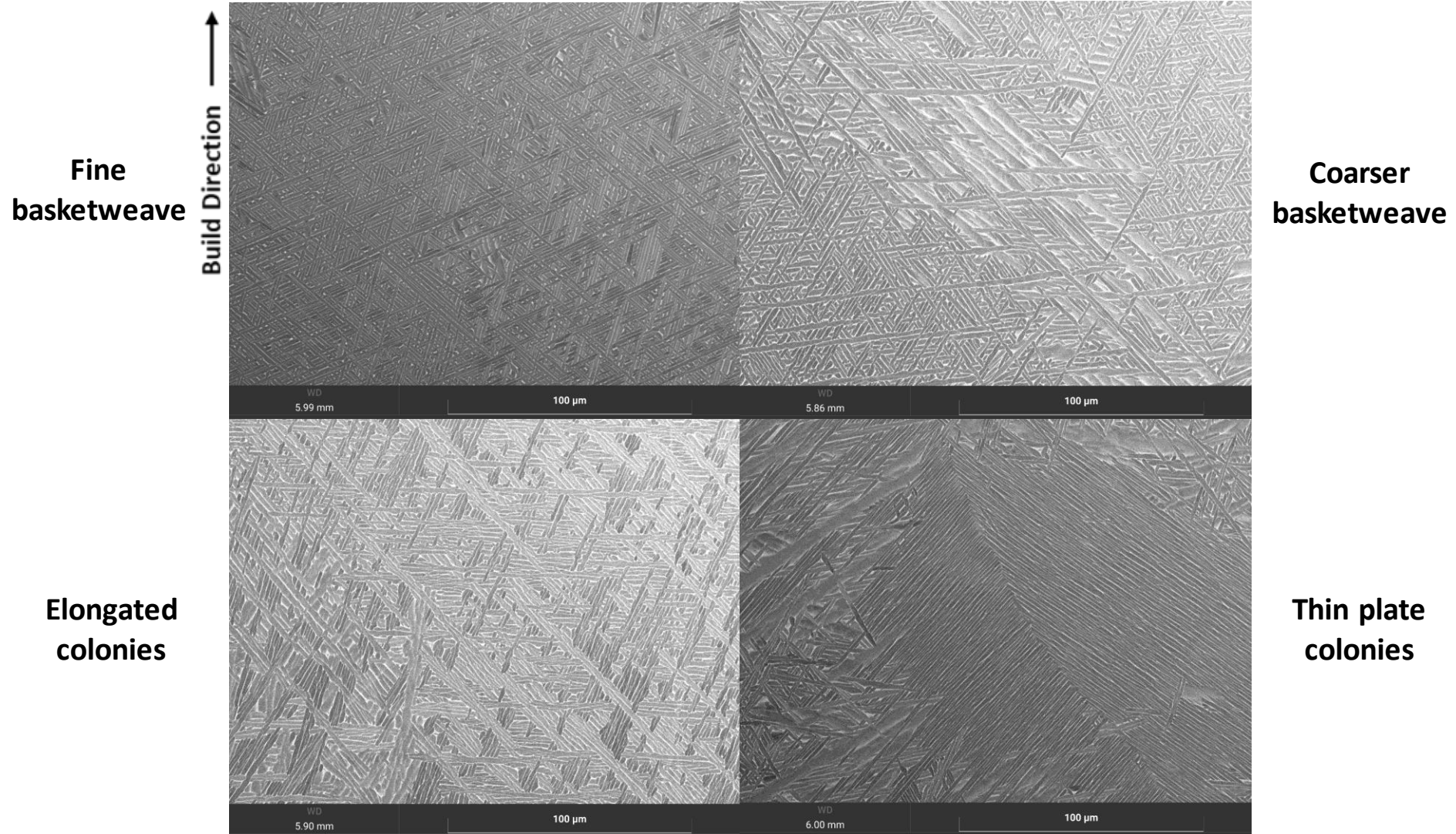
Directed Energy Deposition



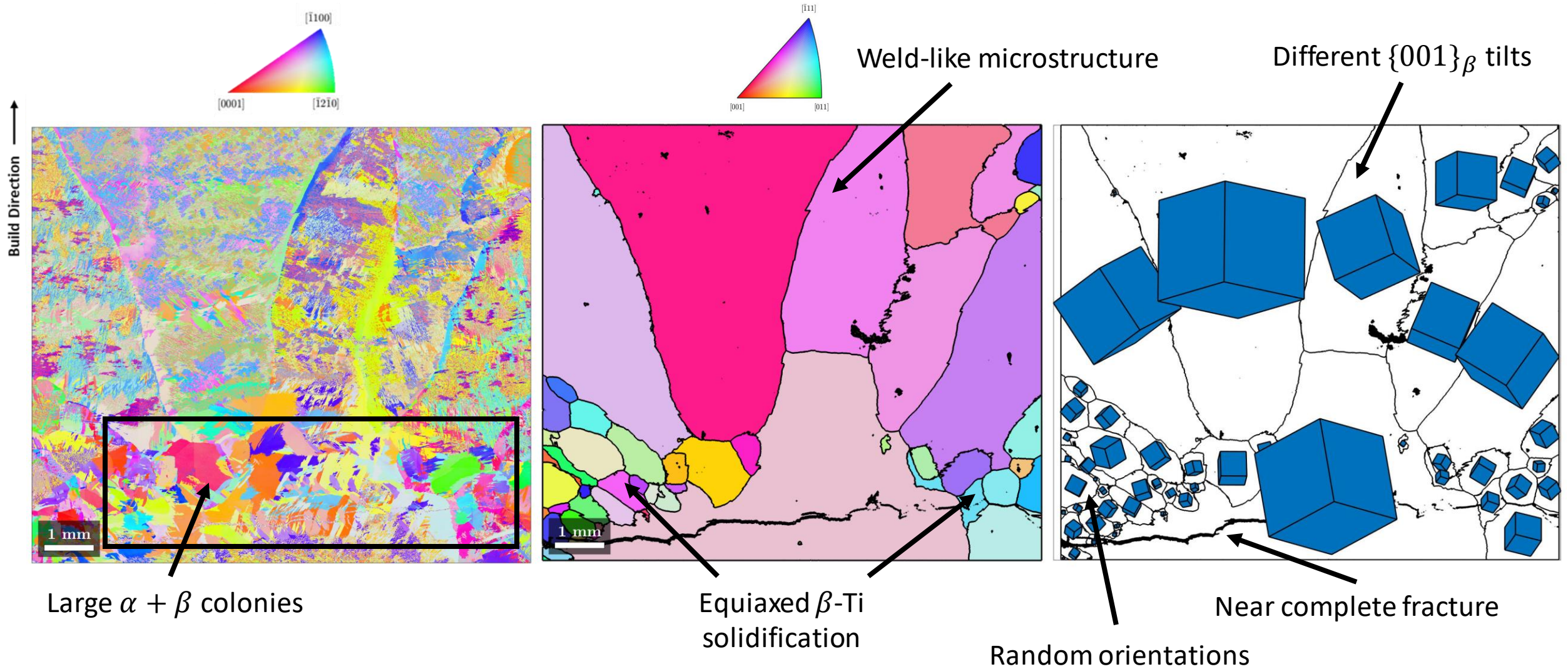
Typical DED Ti-6Al-4V Microstructure



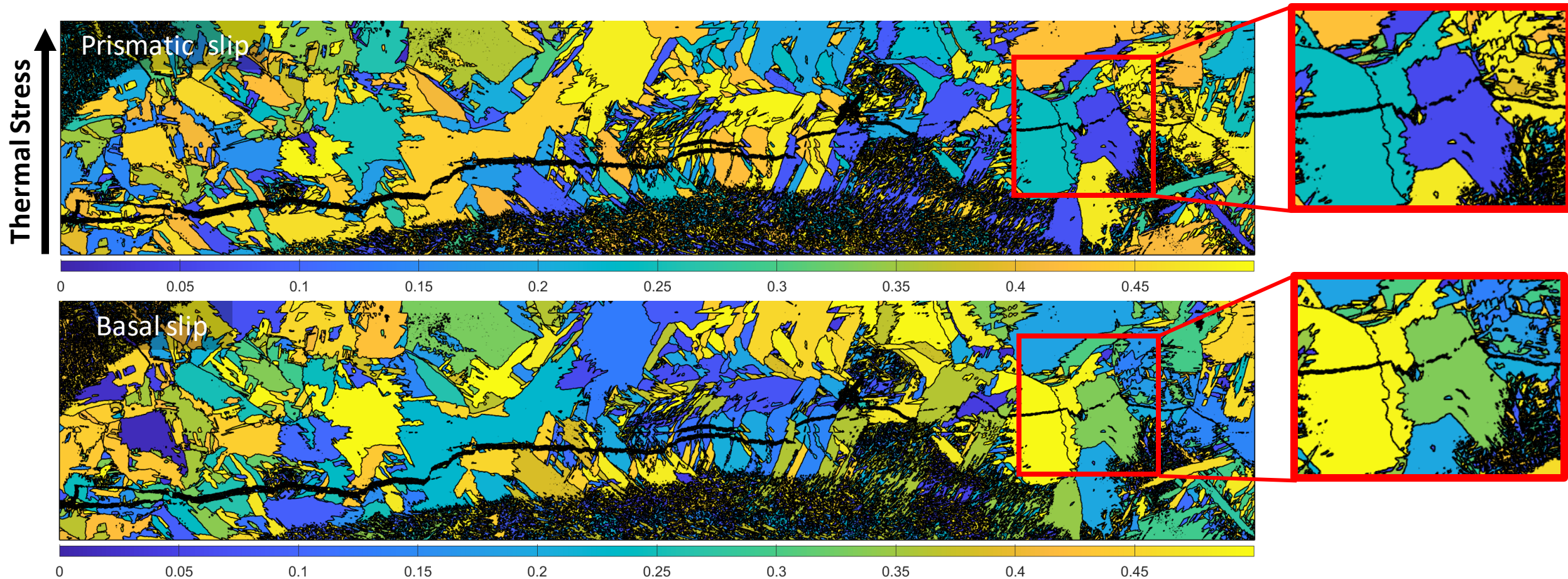
Typical DED Solid State Microstructures



Non-Typical DED Microstructure



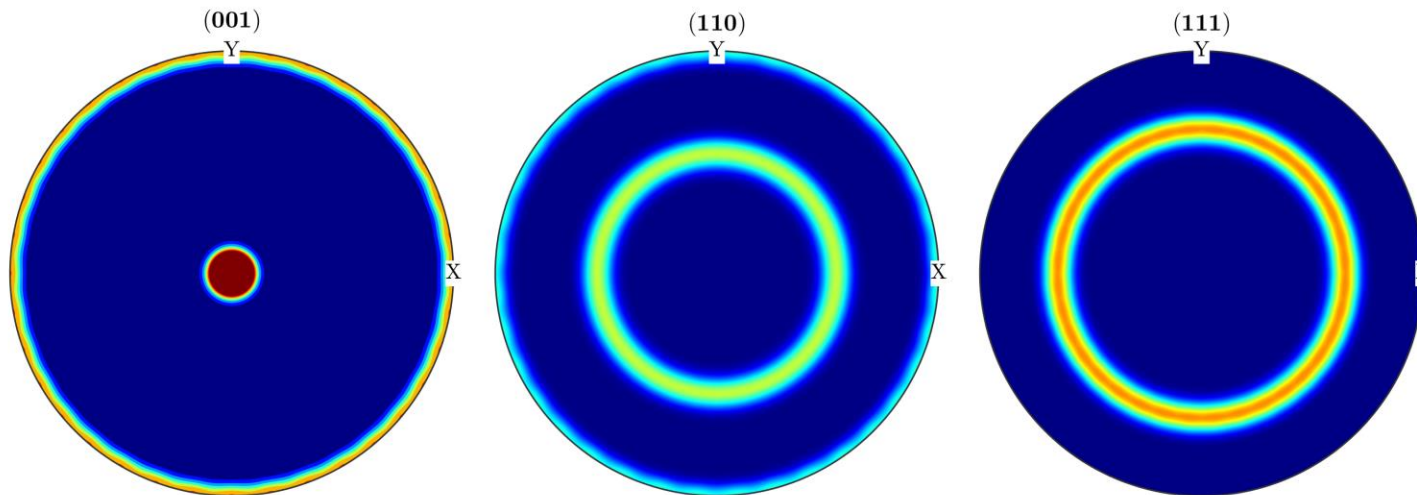
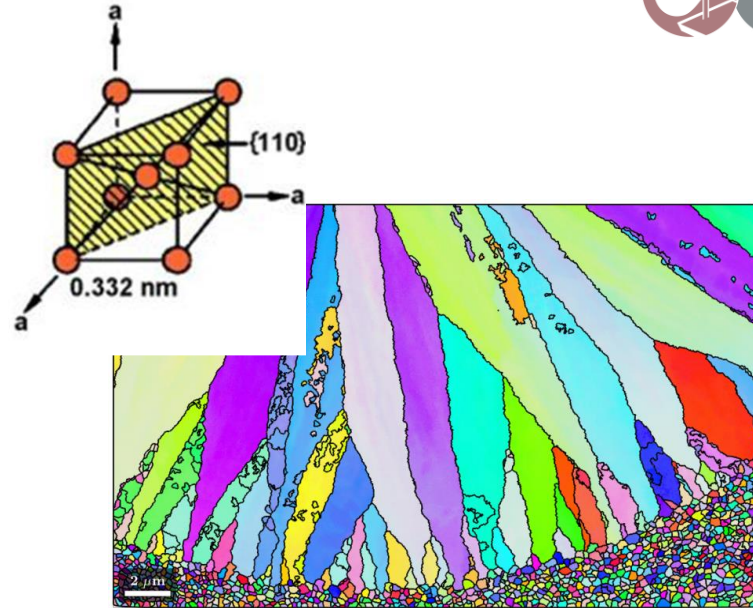
Schmid Factors For A Thermal Stress



Recent Work

Quick Texture Crash Course

- Crystallographic texture
 - Preferential orientation of crystal planes
 - Modify material behavior
- Fiber texture
 - A ring of orientations around a real space vector



Texture ↔ Microstructure Relationships?



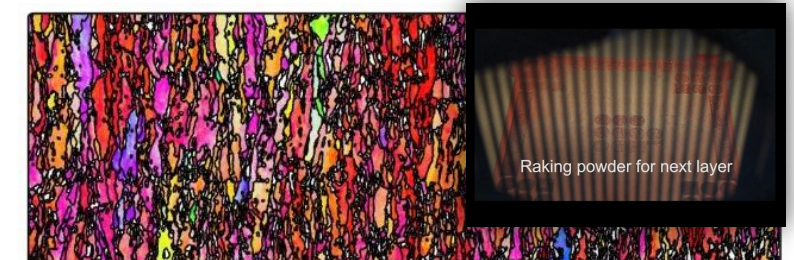
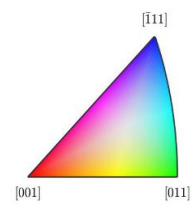
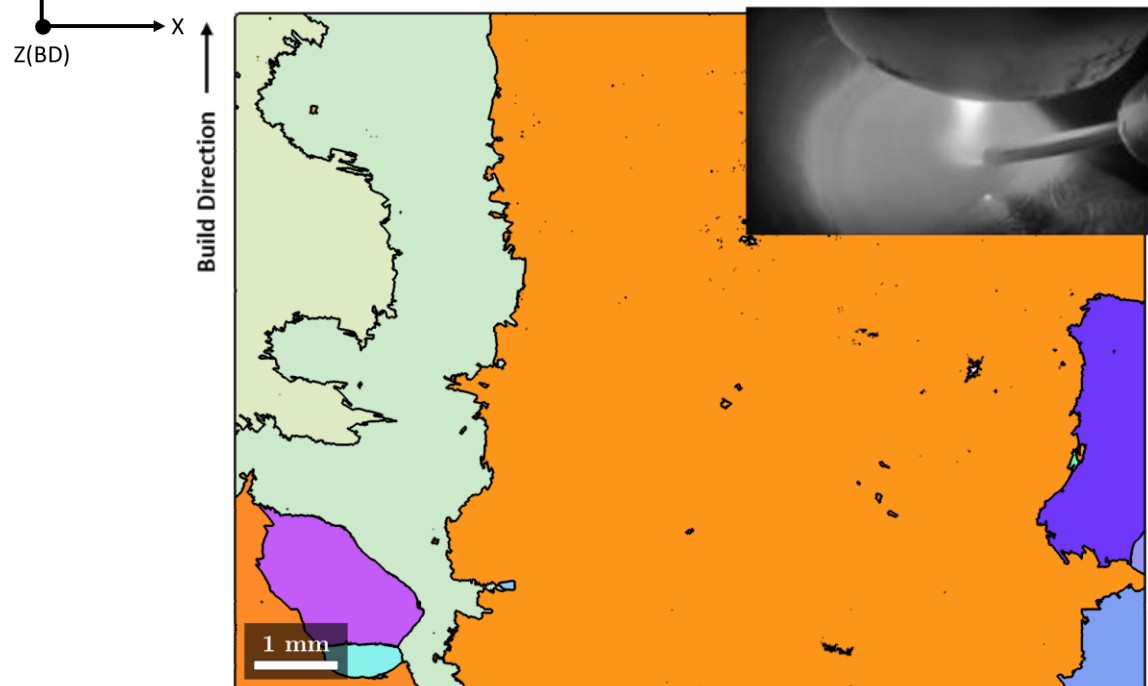
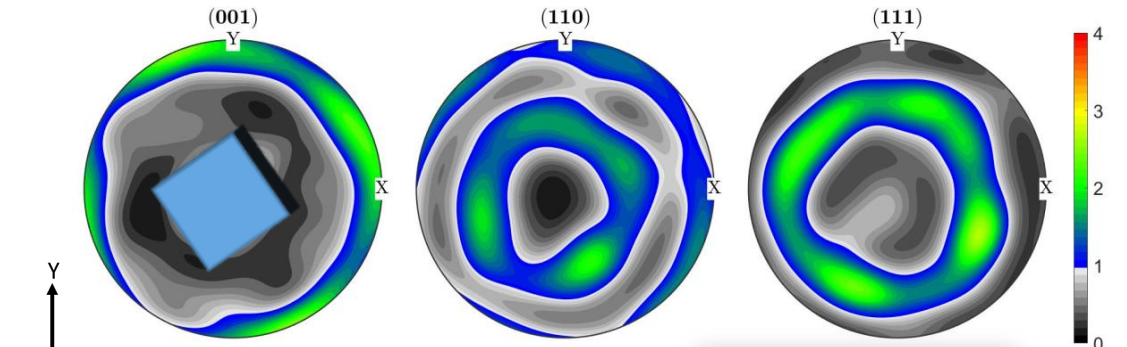
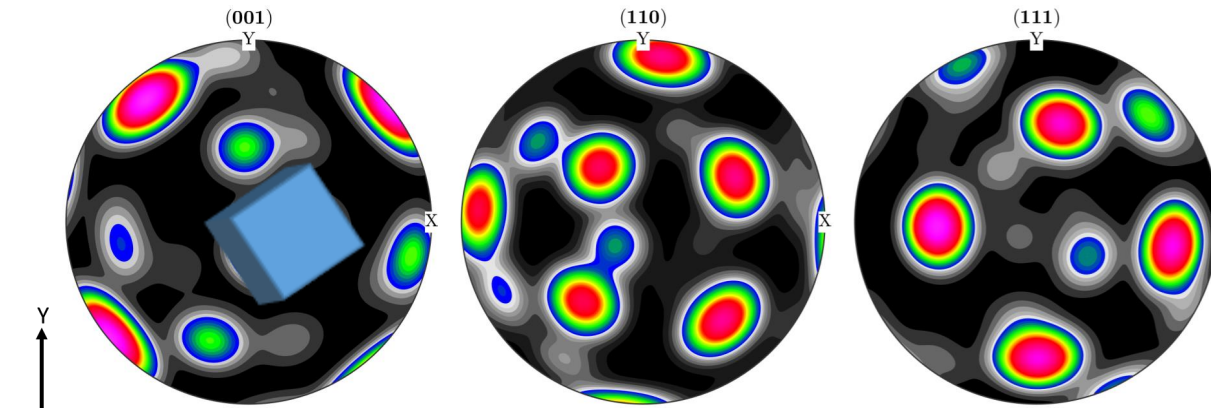
CANFSA

CENTER FOR ADVANCED
U.S. STRUCTURAL ALLOYS

β -Ti

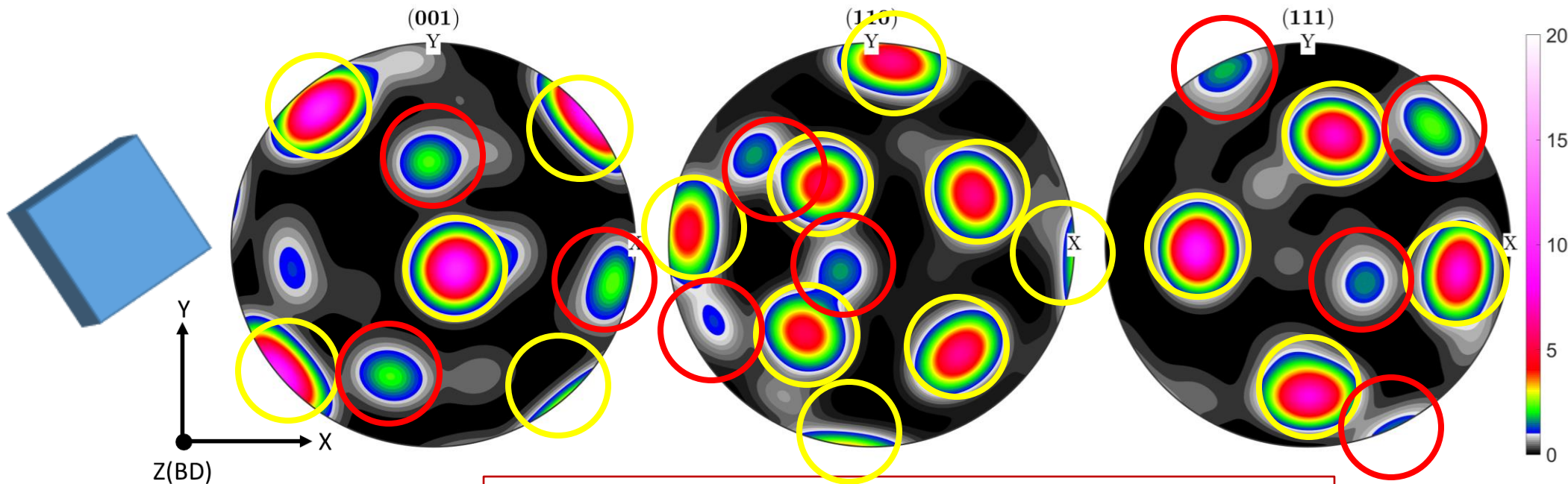
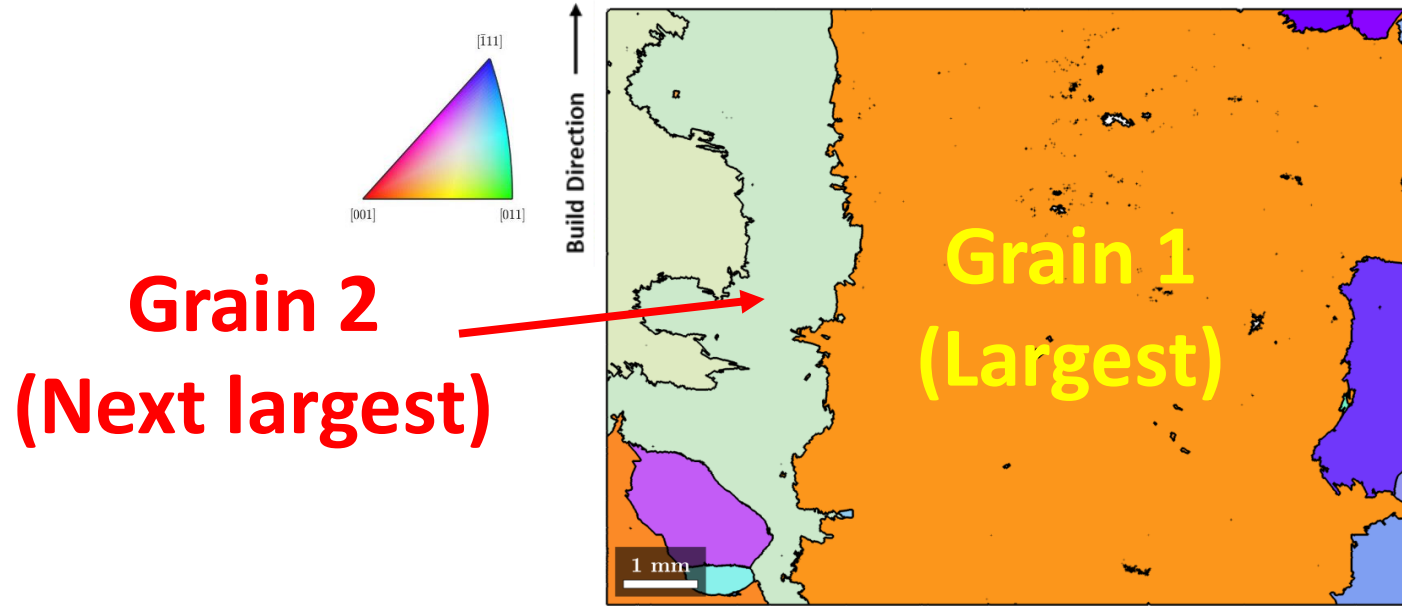
DED

EBM-PBF

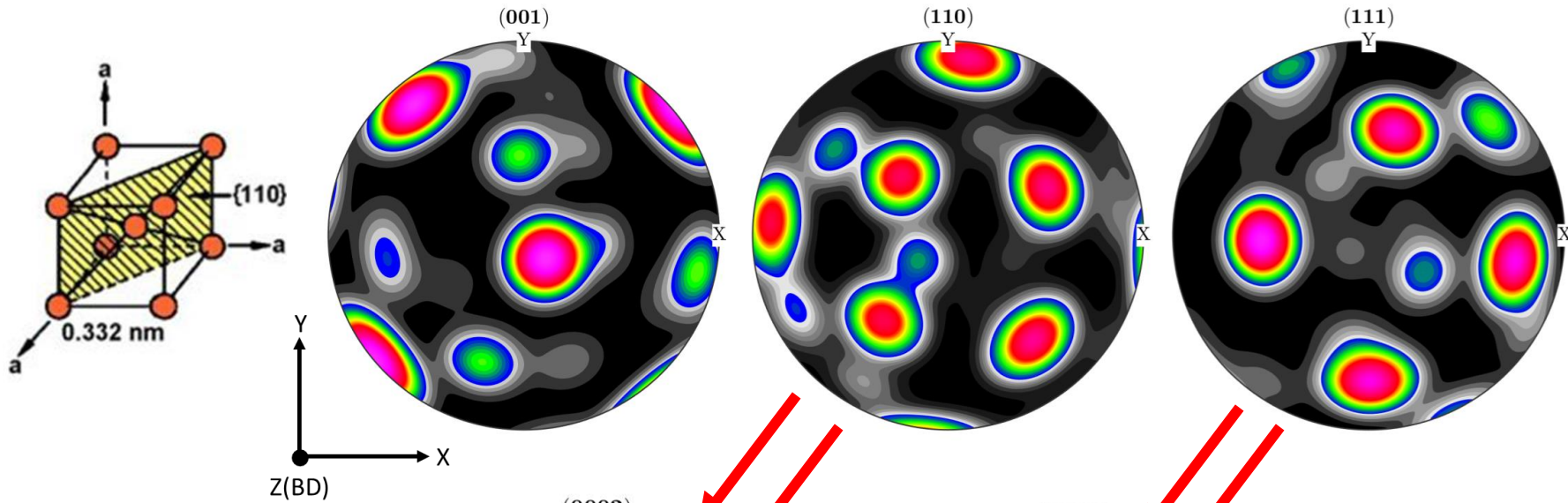


What gives rise to the fiber texture? Does it mean anything?

DED = Large β -Ti Grains



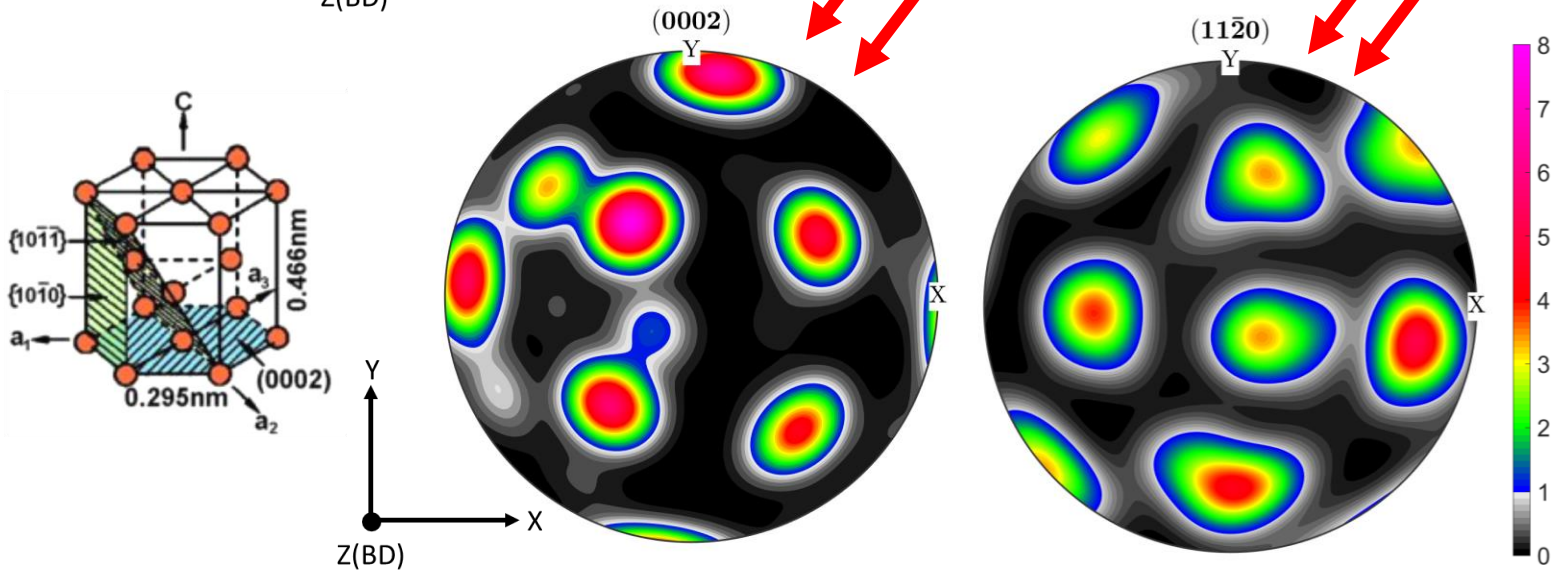
Specific α -Ti Texture Results



Burgers OR

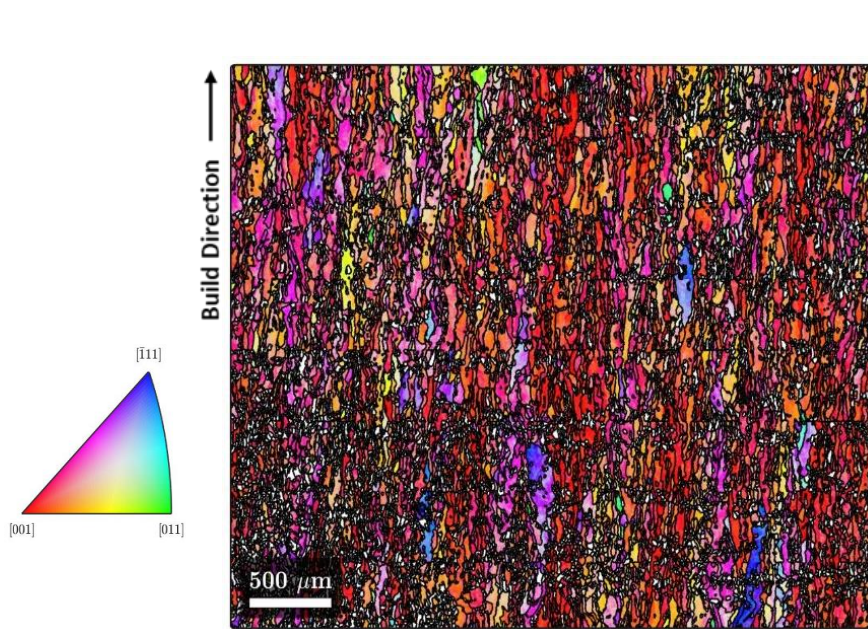
$\{0001\}_\alpha \parallel \{011\}_\beta$

$\langle 11\bar{2}0 \rangle_\alpha \parallel \langle 111 \rangle_\beta$

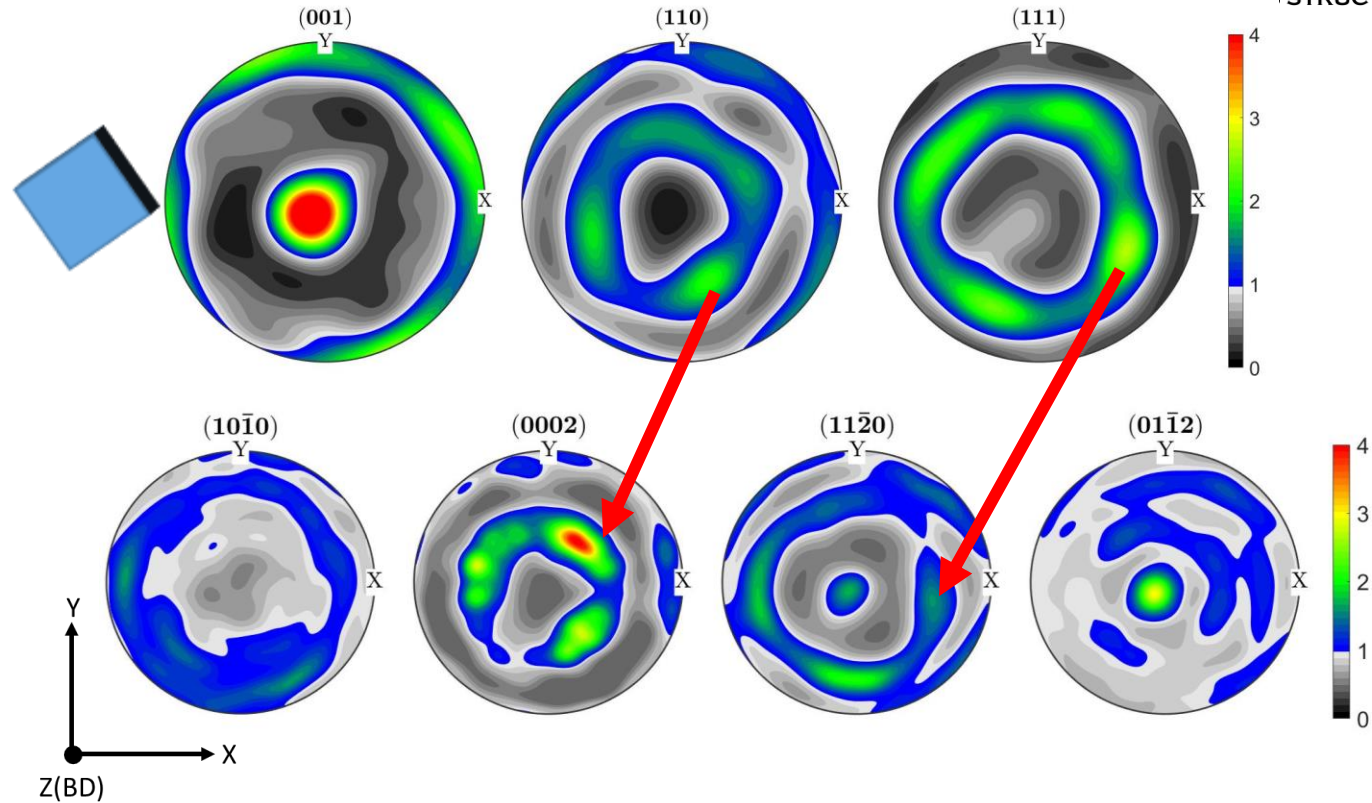


No fiber texture present in either phase, signifying larger parent β -Ti grains

EBM-PBF = Smaller β -Ti Grains



No large grains observed



Do fiber textures indicate finer as-solidified grains?

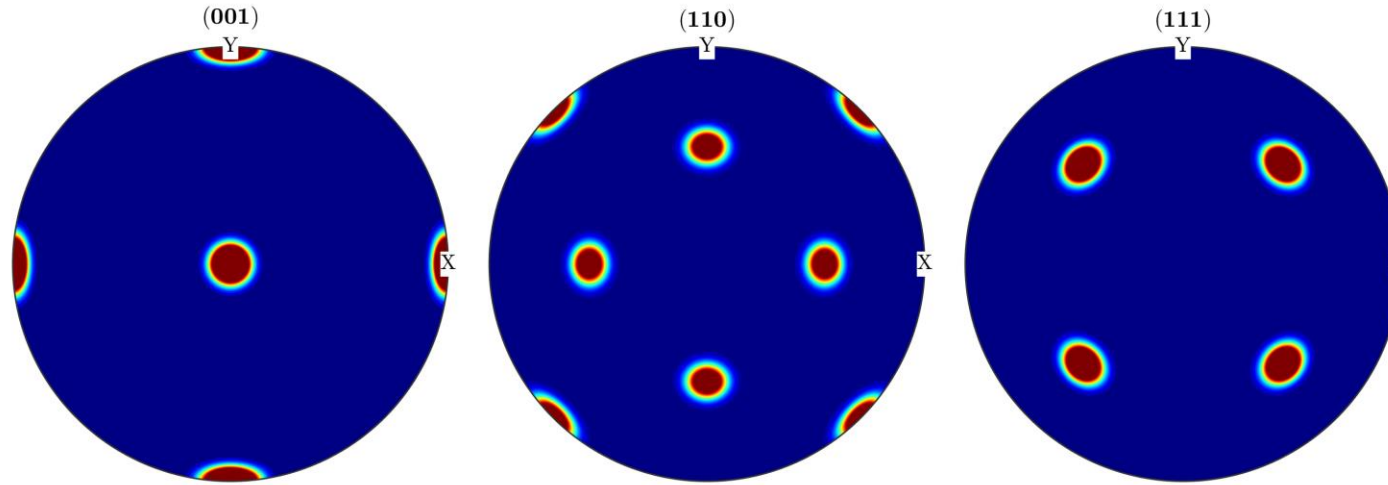
Simulations



Simulating Single β -Ti Grain Solidification



Single β -Ti Grain



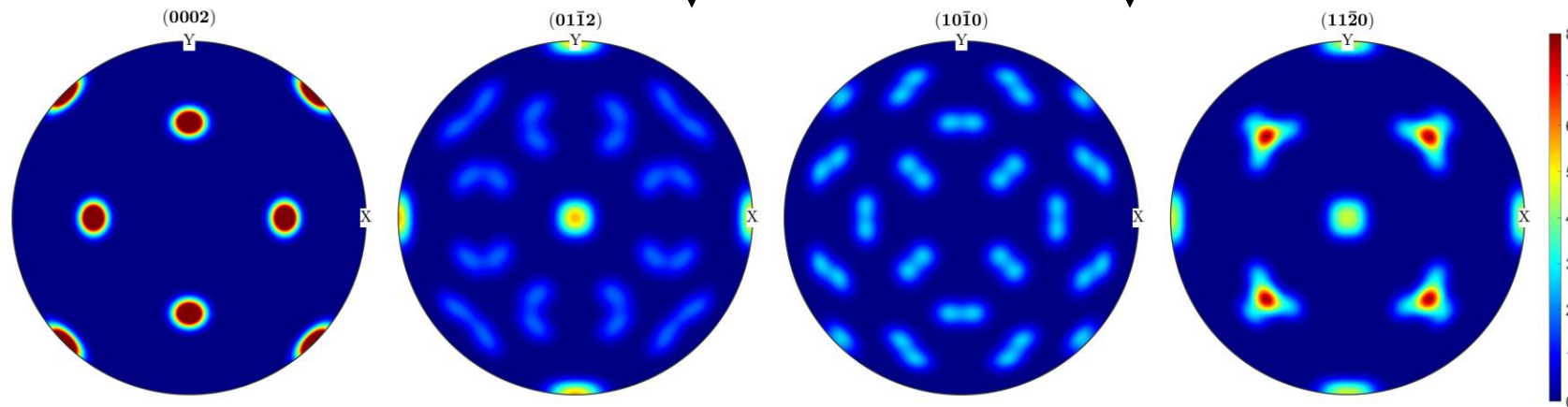
Applying
Burgers OR

$$\{0001\}_\alpha \parallel \{011\}_\beta$$

$$\langle 11\bar{2}0 \rangle_\alpha \parallel \langle 111 \rangle_\beta$$



12 α -Ti Variants



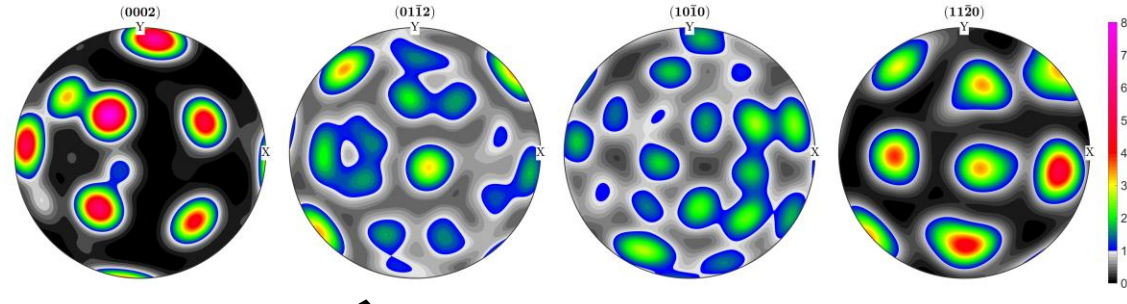
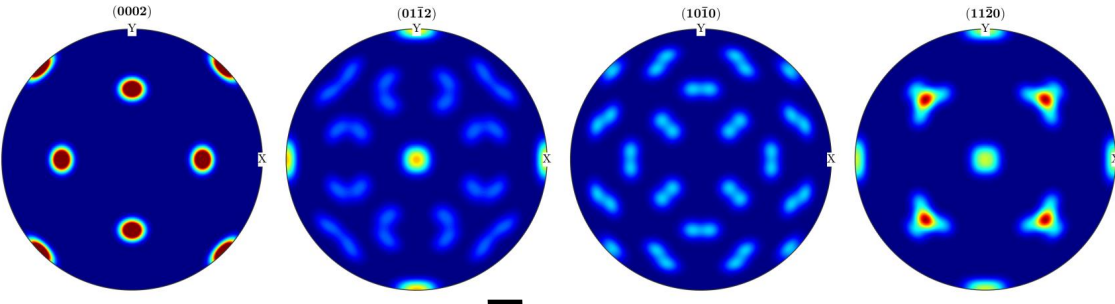
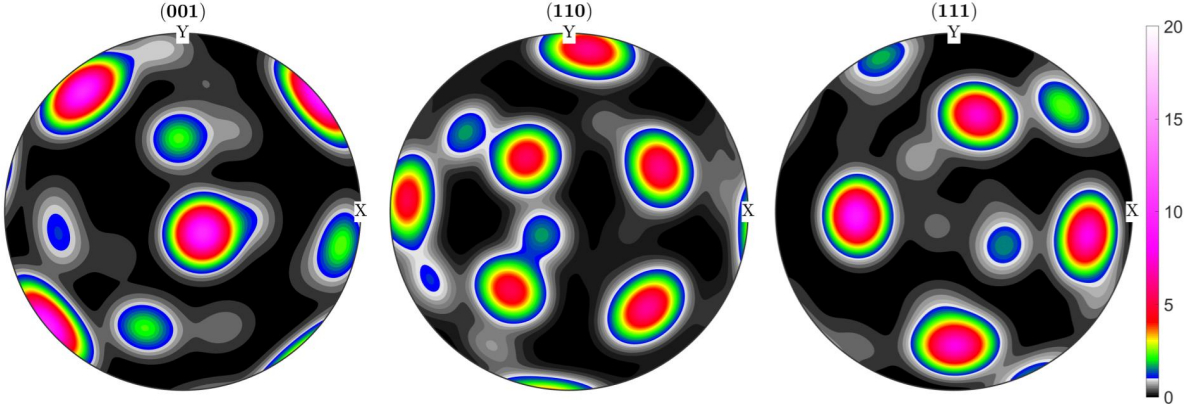
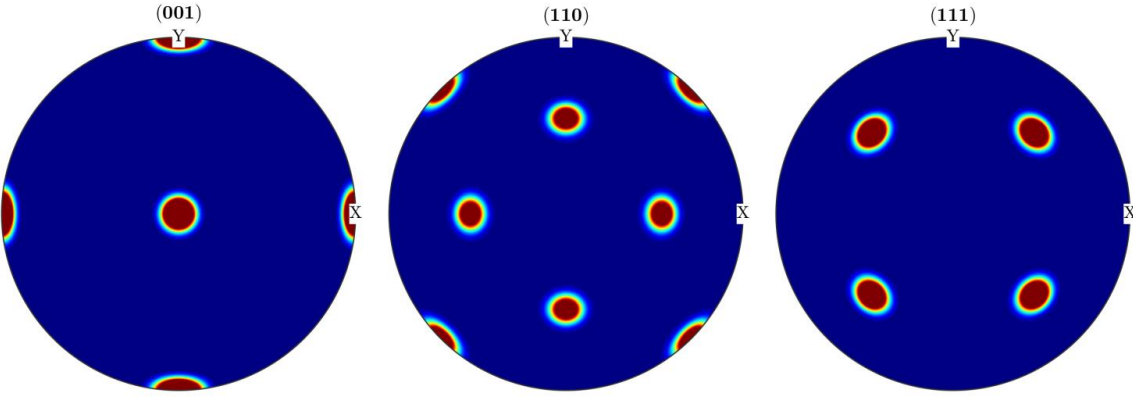
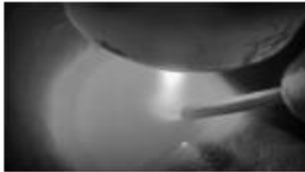
Comparing Simulations to DED Textures



Simulation



DED

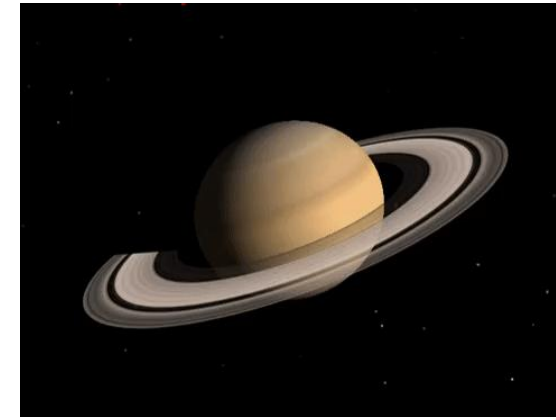
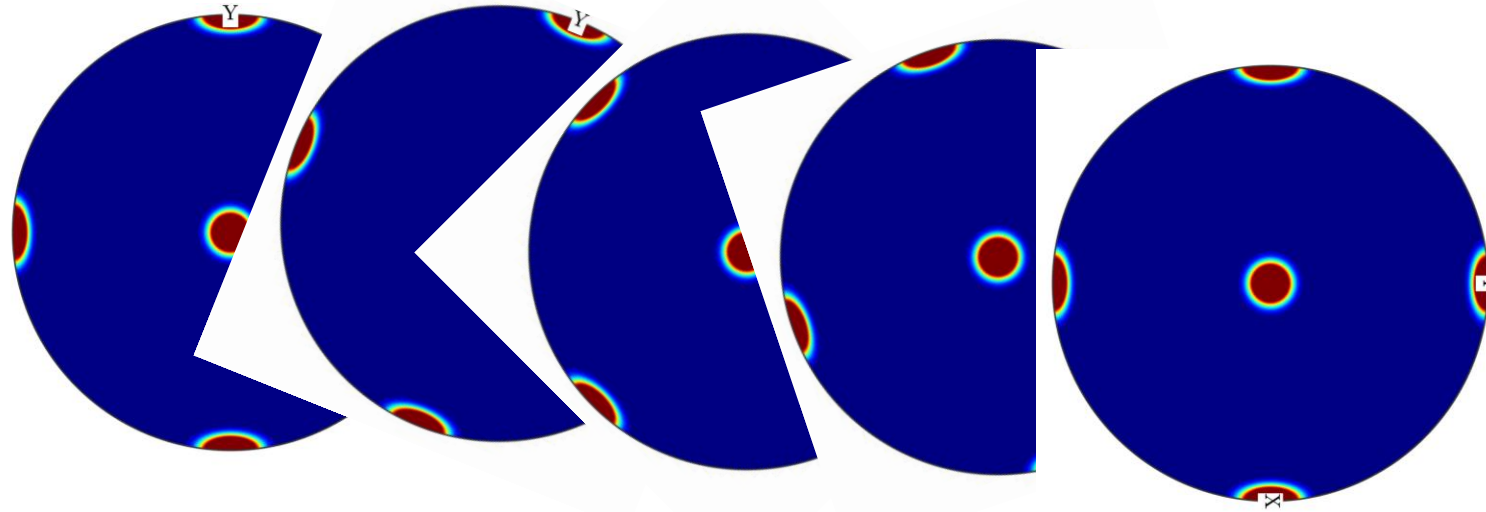
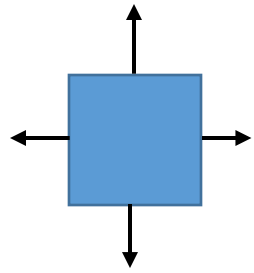


Single Grains Can't Form Fibers

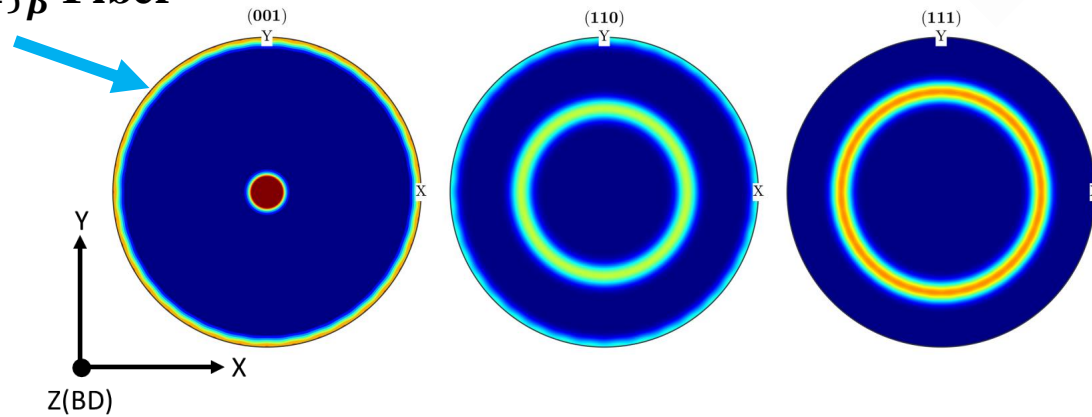
Simulating Multi-Grain β -Ti Solidification



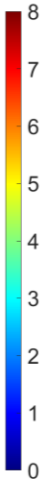
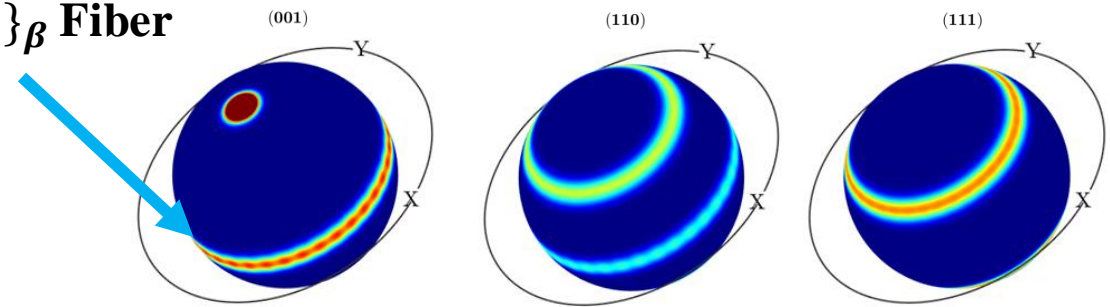
“The healthier option”



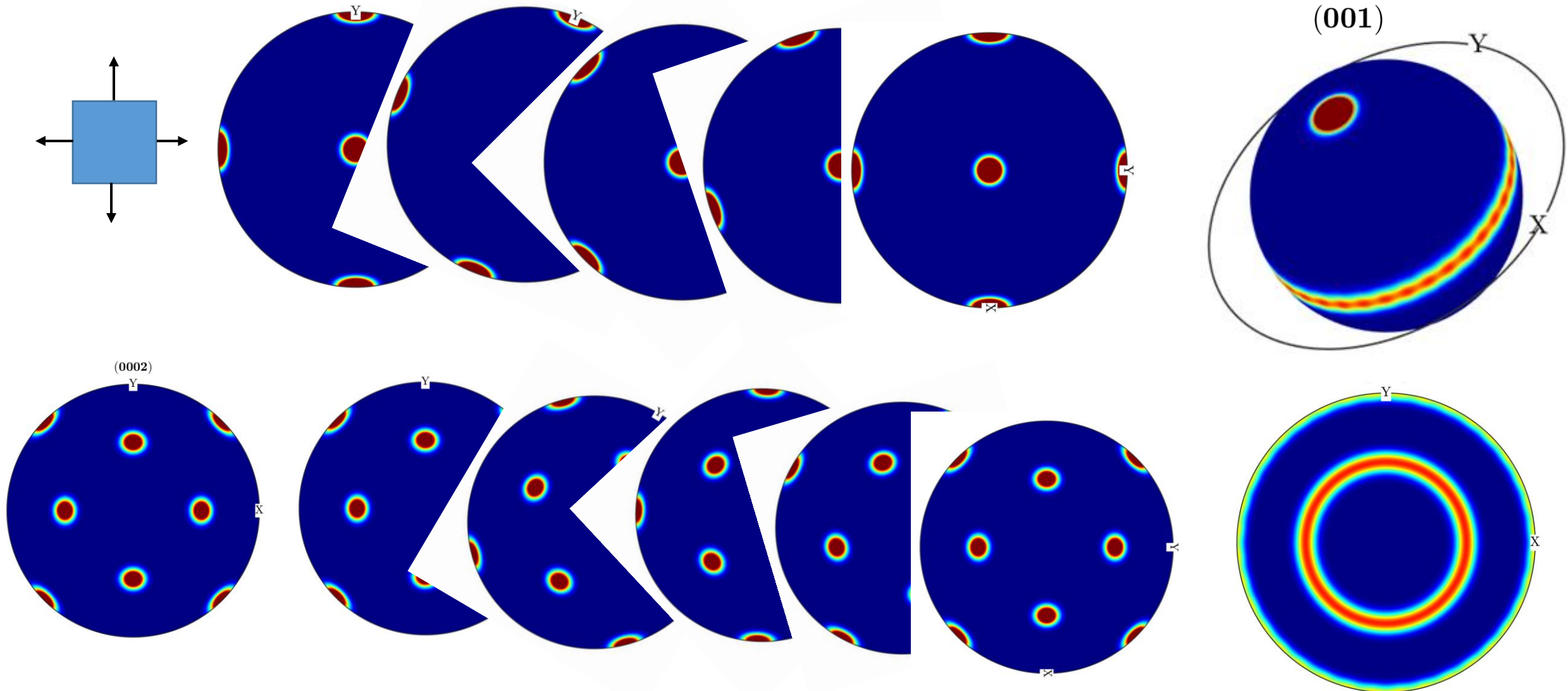
$\{001\}_\beta$ Fiber



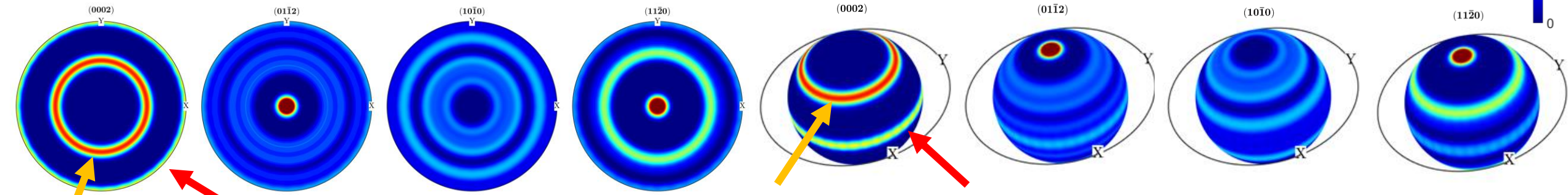
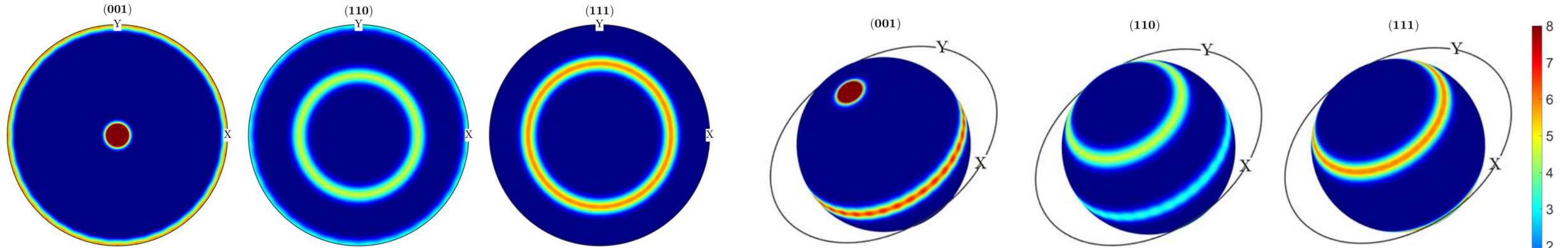
$\{001\}_\beta$ Fiber



How Does This Affect α -Ti Orientations?



Continuing Our Multi-Grain Diet

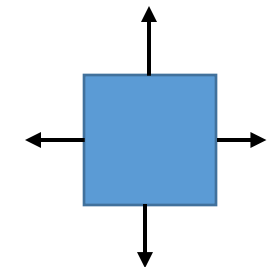


$\{01\bar{1}2\}_\alpha$ Fiber $\{11\bar{2}0\}_\alpha$ Fiber

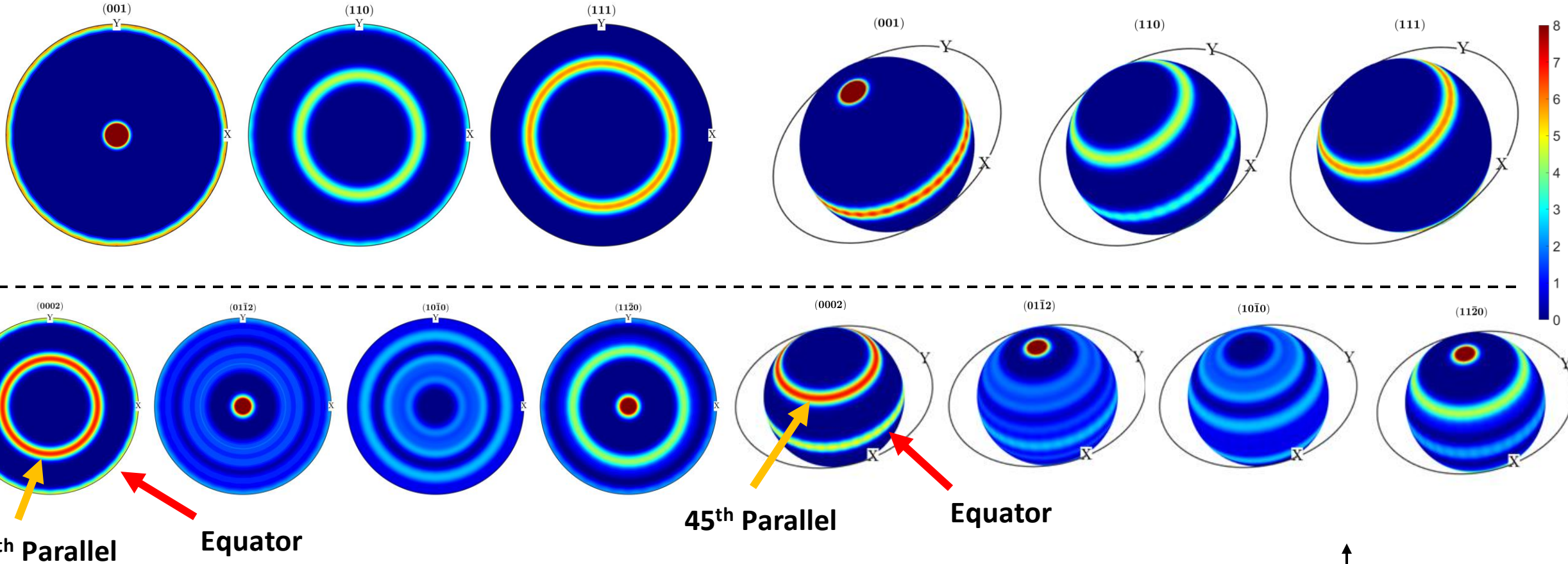
$\{01\bar{1}2\}_\alpha$ Fiber

$\{11\bar{2}0\}_\alpha$ Fiber

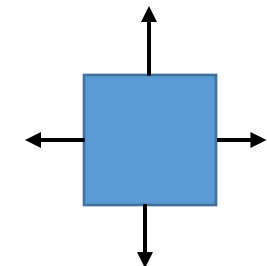
Finer β -Ti grains = Fiber textures in AM solidification
More random orientation of planes

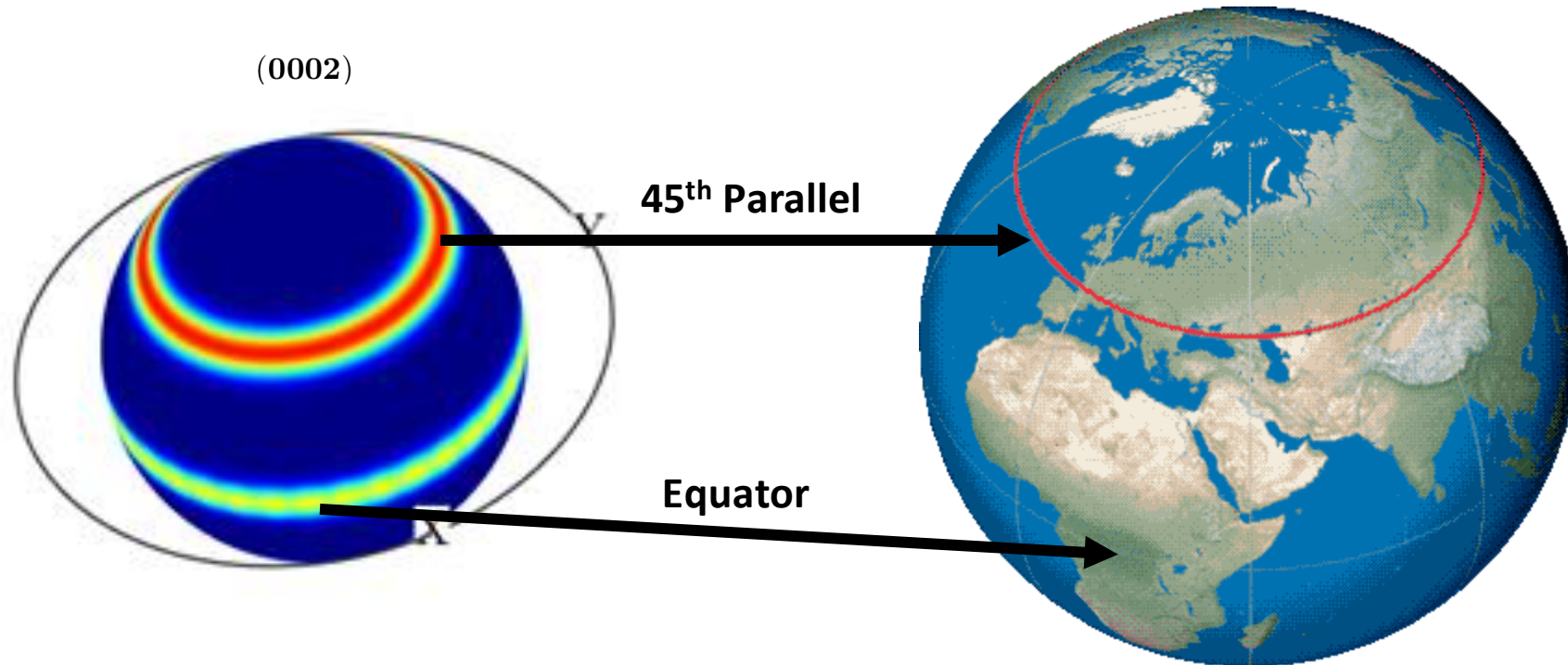


Continuing Our Multi-Grain Diet



Finer β -Ti grains = Fiber textures in AM solidification
More random orientation of planes

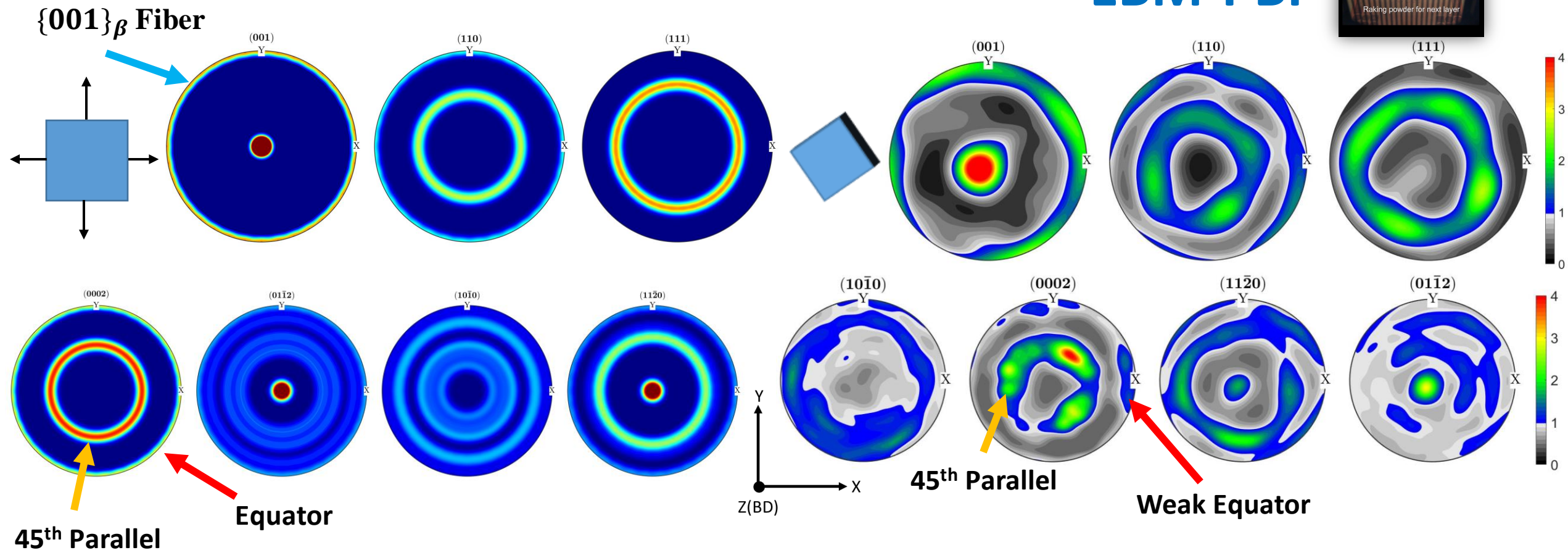
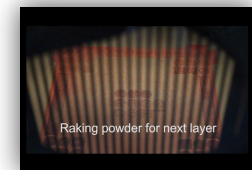




Comparison to EBM-PBF Textures

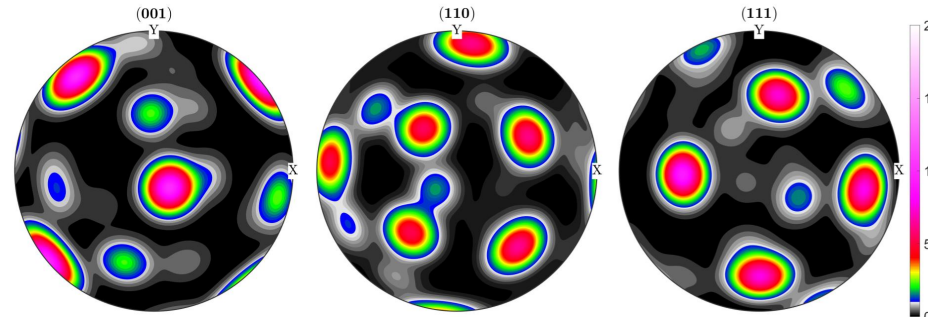
Simulation

EBM-PBF



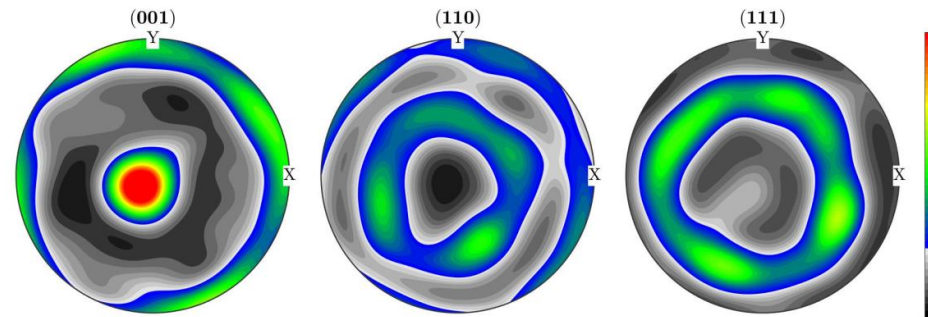
Texture-Microstructure Takeaways

Large As-Solidified β -Ti Grains

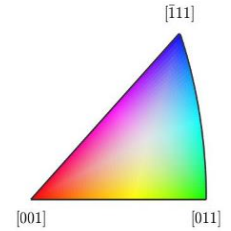
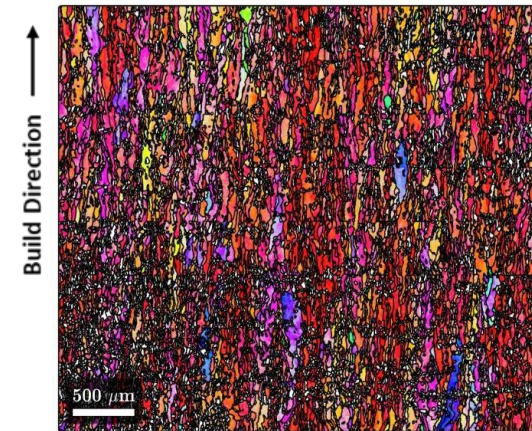
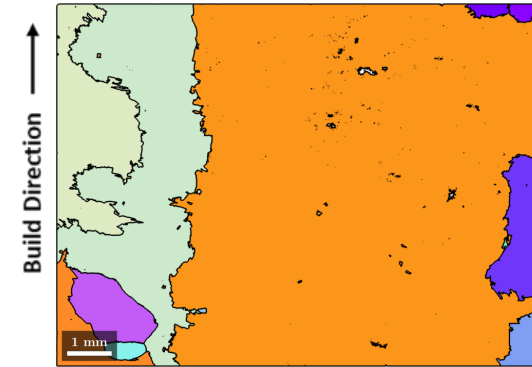


No fiber textures

Finer As-Solidified β -Ti Grains

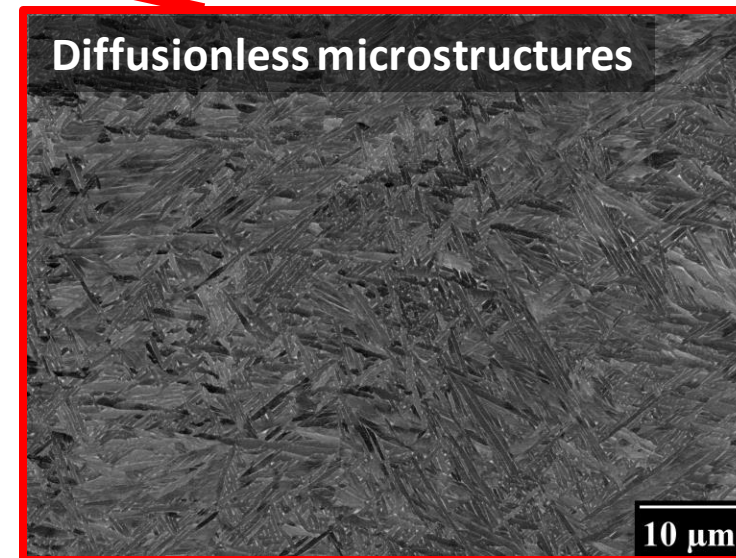
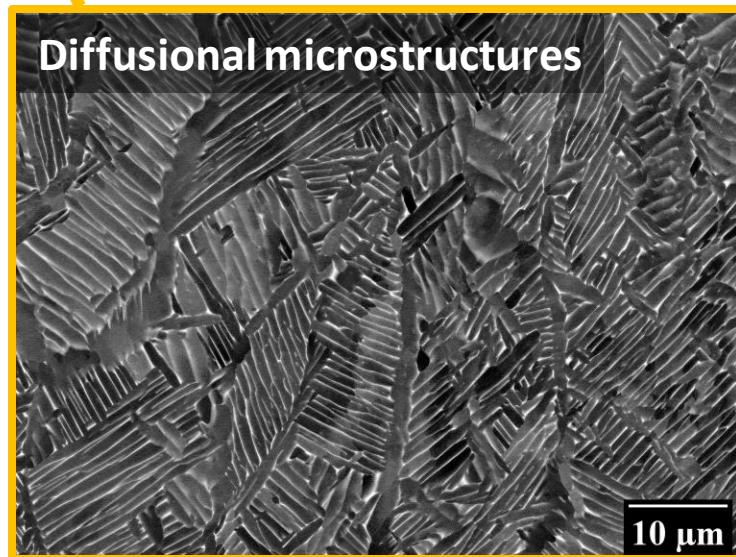
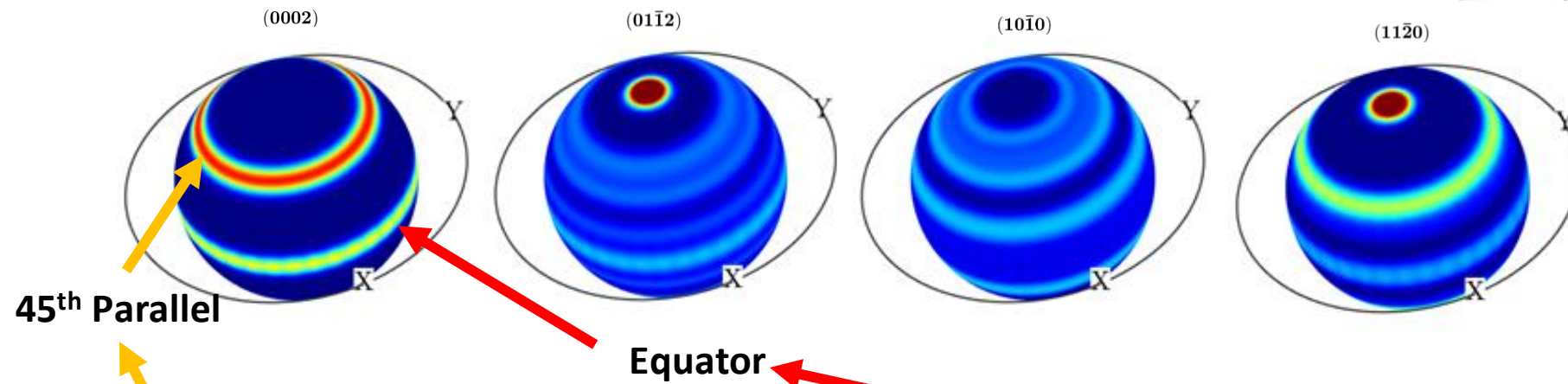


Fiber textures

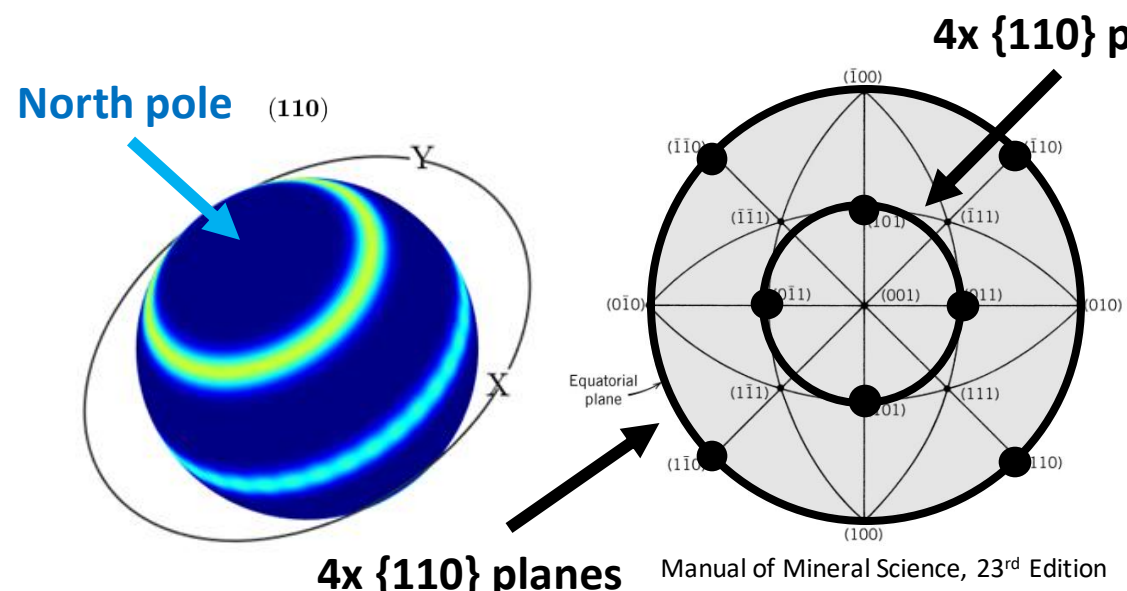
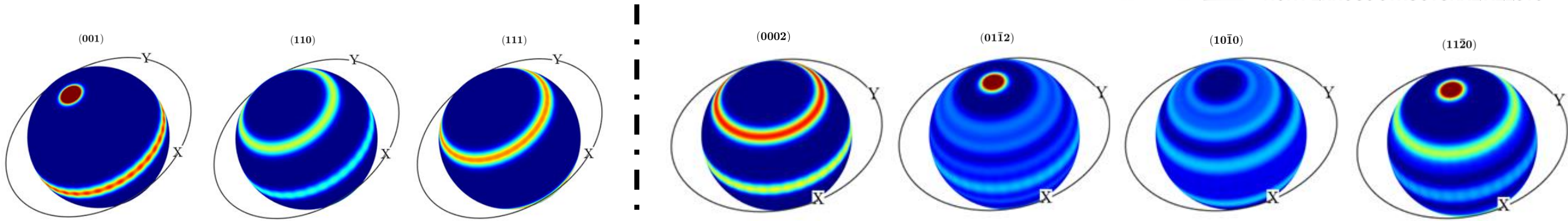


What about the solid state microstructure?

Cooling Rate and Favored Orientations



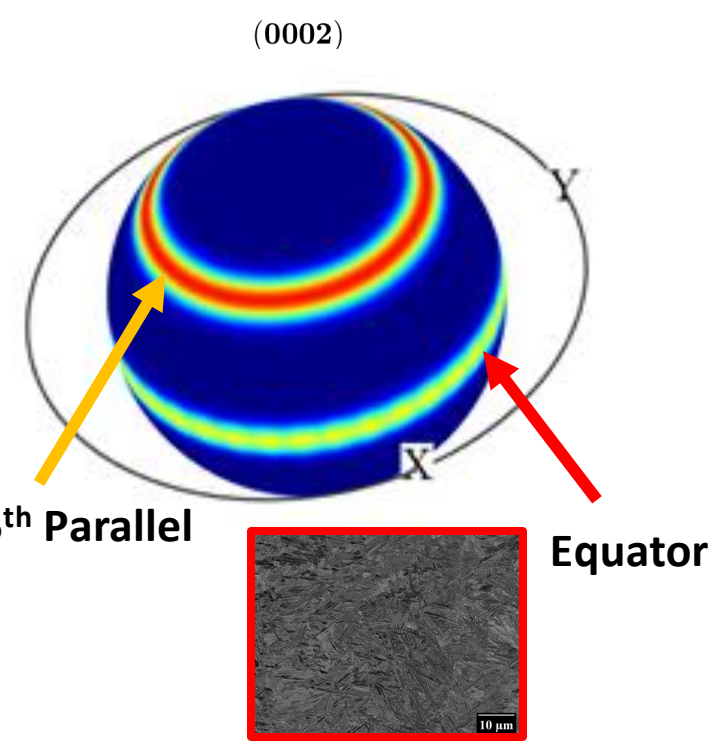
Why Different Orientations?



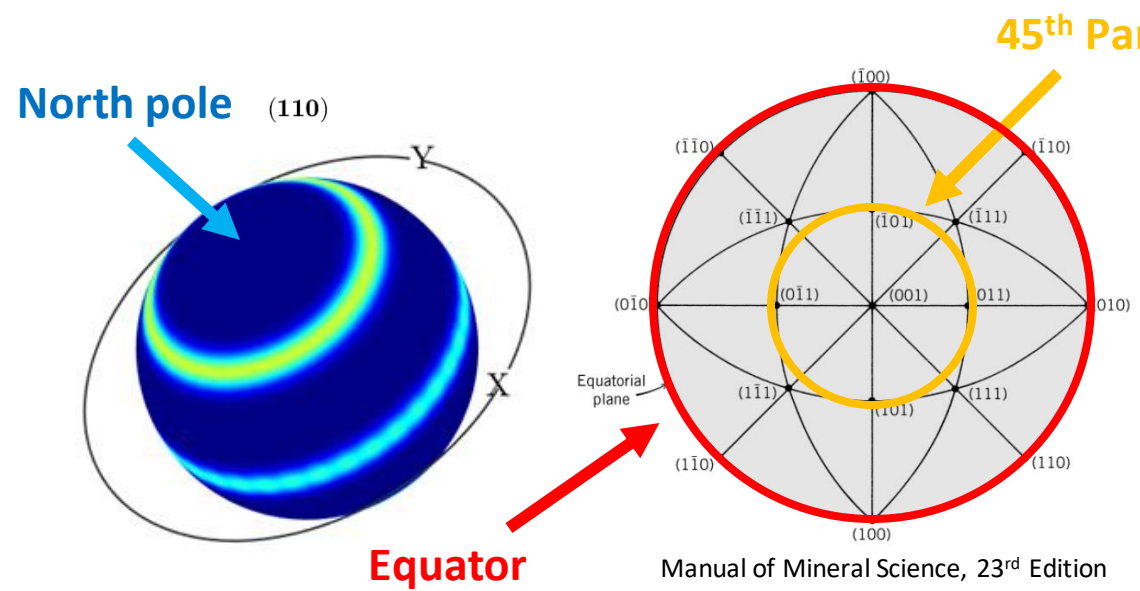
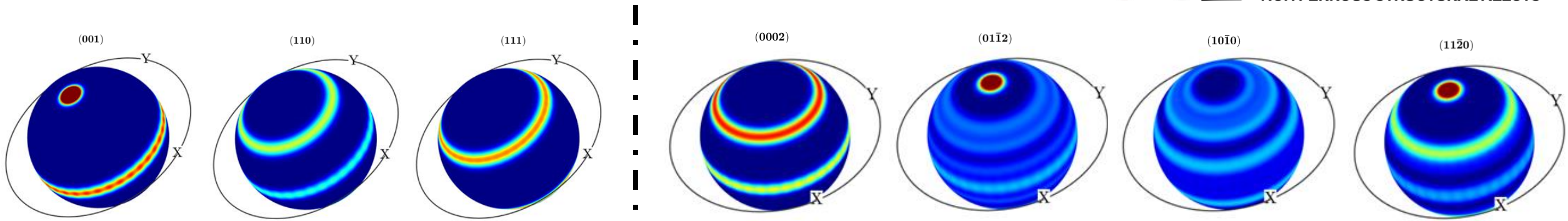
Burgers OR

$$\{0001\}_\alpha \parallel \{011\}_\beta$$

$$\langle 11\bar{2}0 \rangle_\alpha \parallel \langle 111 \rangle_\beta$$

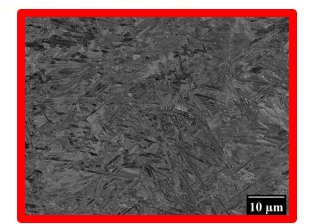
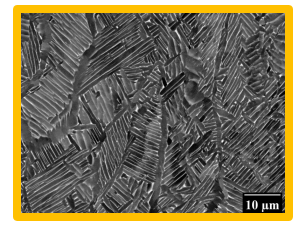
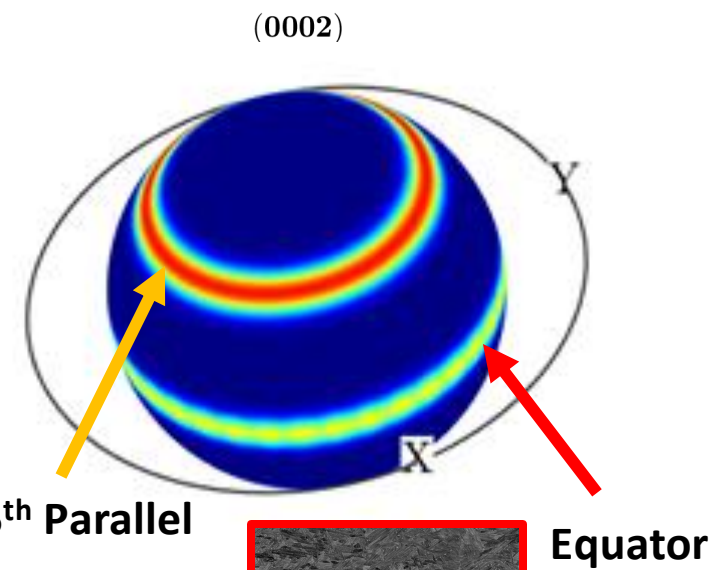


Why Different Orientations?



Burgers OR

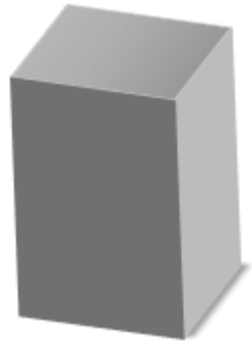
$$\{0001\}_\alpha \parallel \{011\}_\beta$$

$$\langle 11\bar{2}0 \rangle_\alpha \parallel \langle 111 \rangle_\beta$$


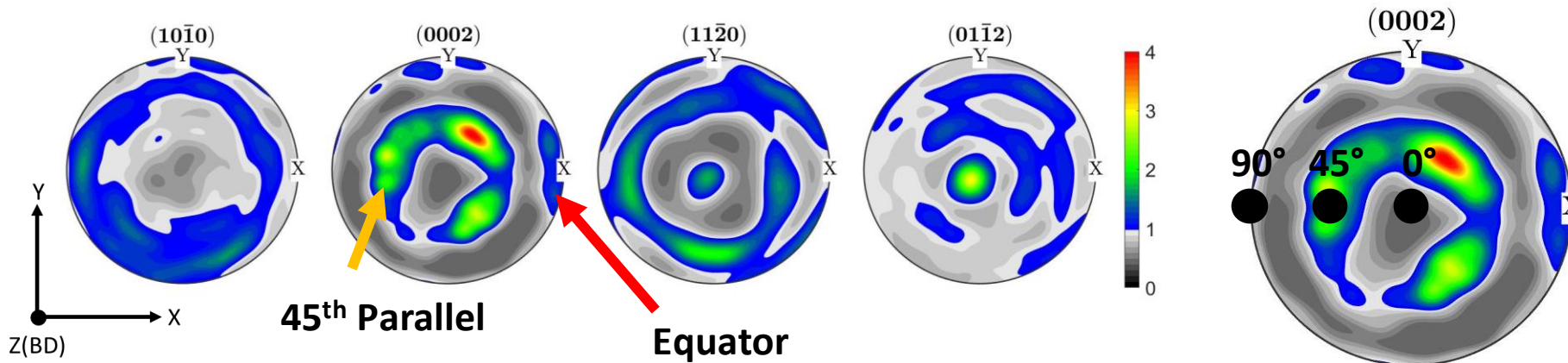
Different variants preferred depending on solid state cooling rate in AM Ti-6Al-4V

Does Texture Impact Mechanical Properties?

Texture → Mechanical Properties



EBM-PBF

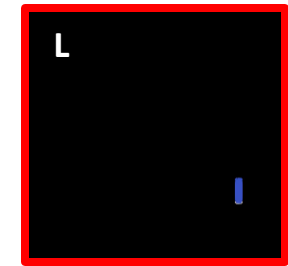
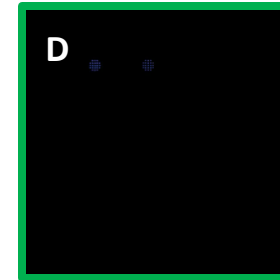
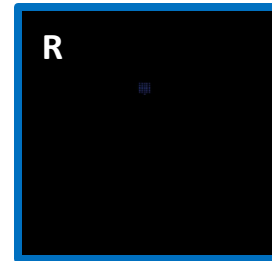


Evaluate influence of texture on mechanical properties

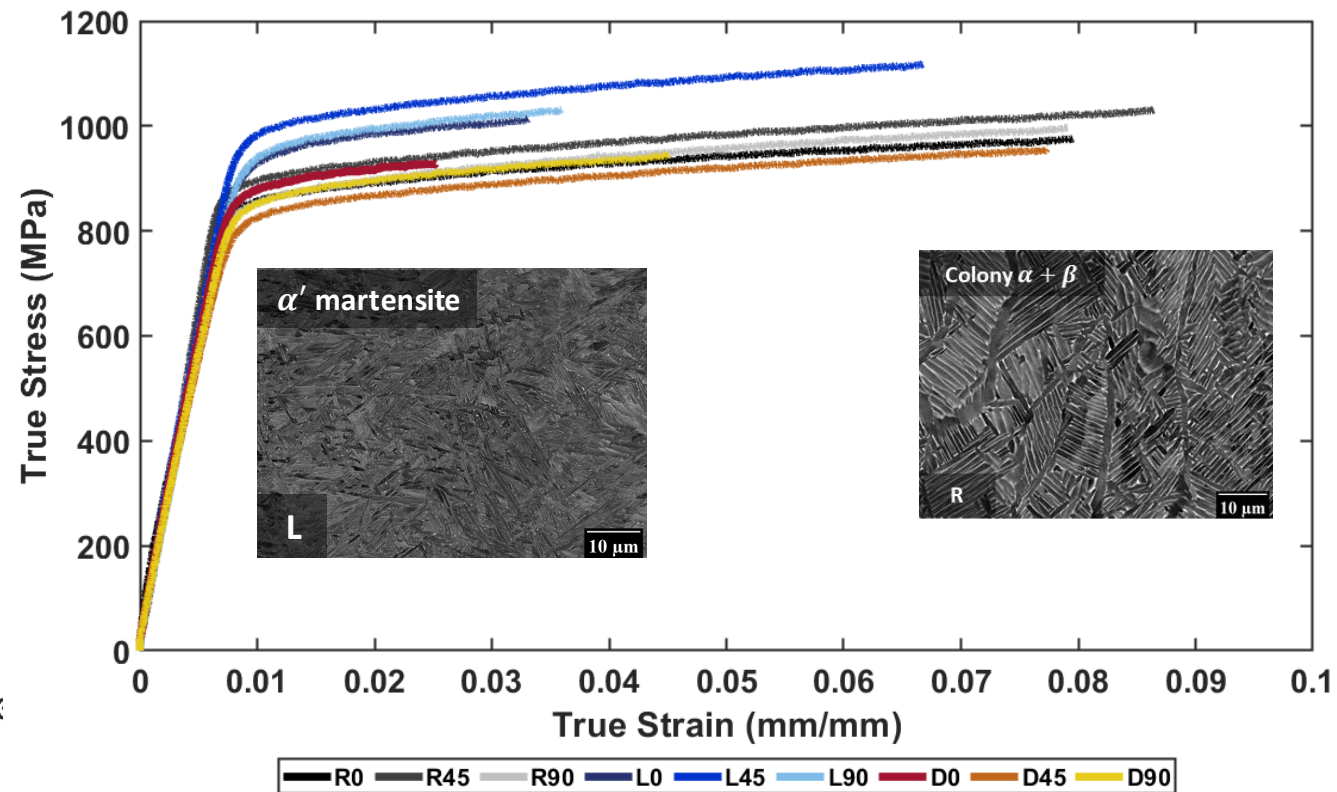
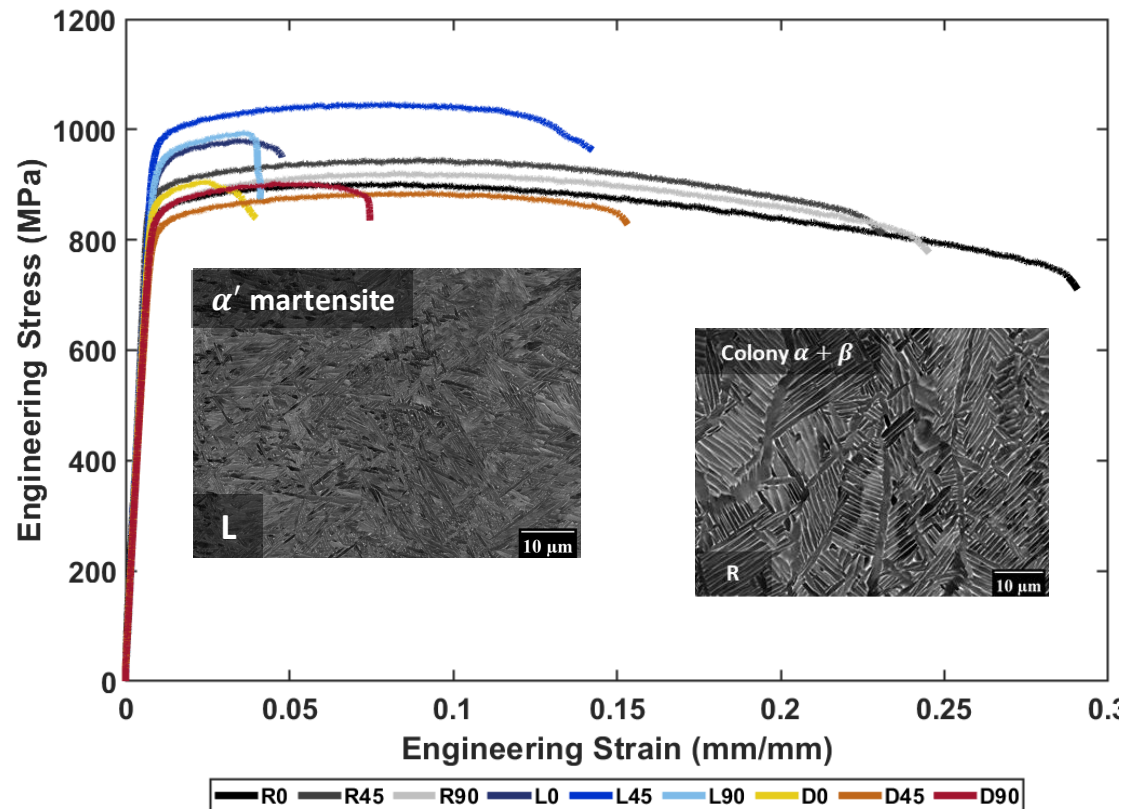
Influence of Texture and Microstructure on Mechanical Properties



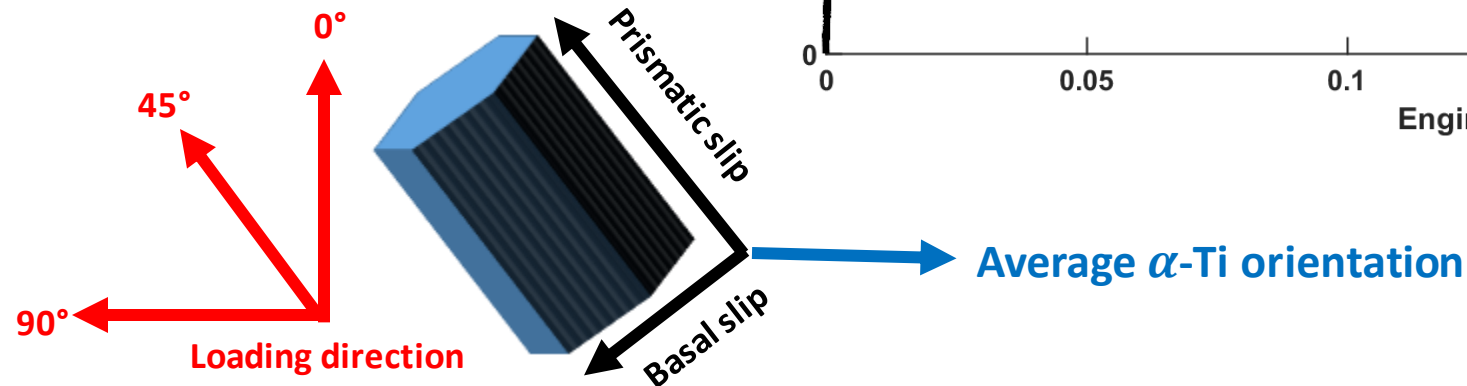
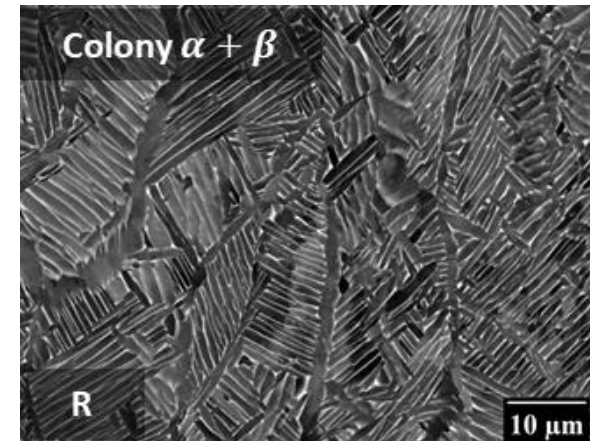
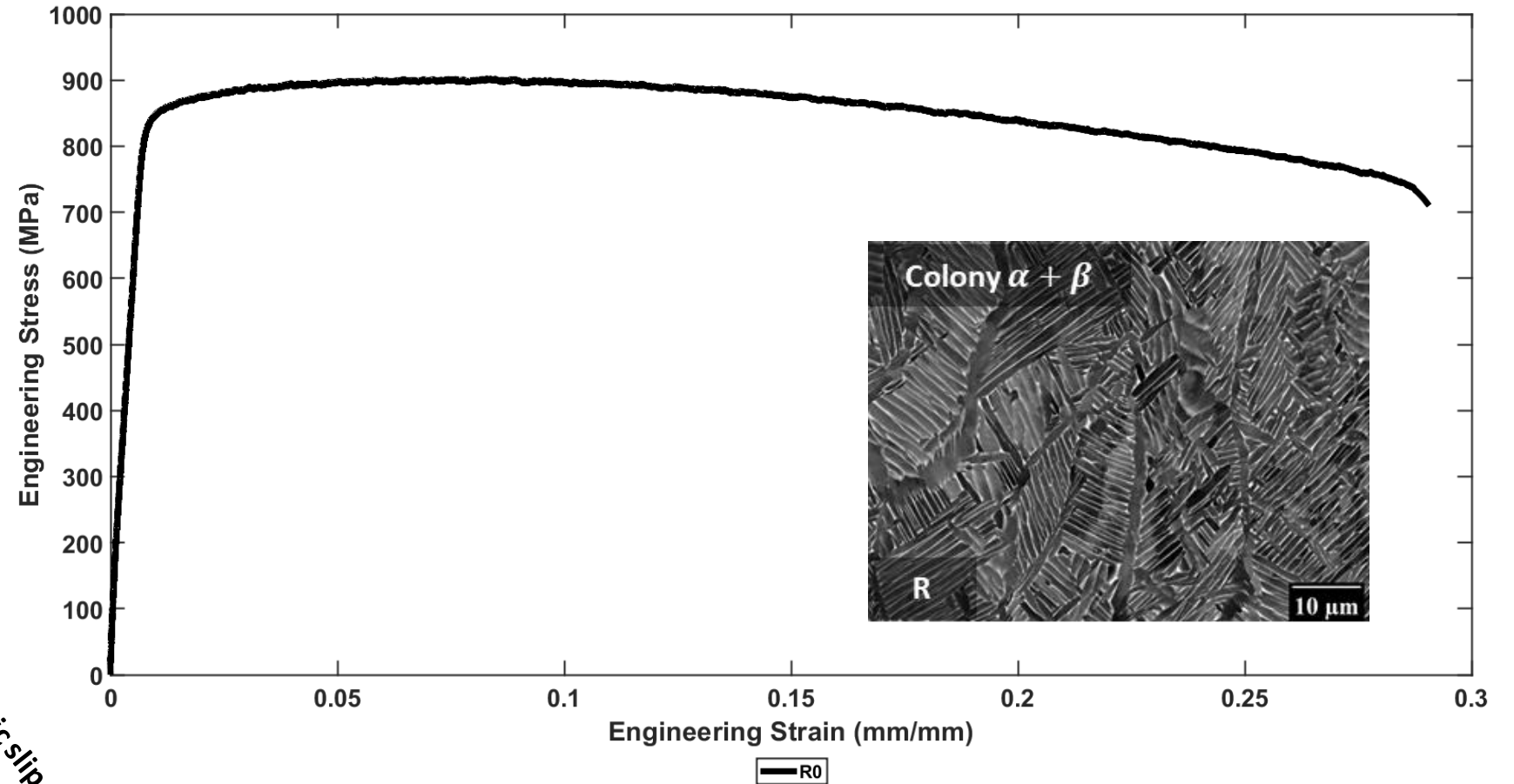
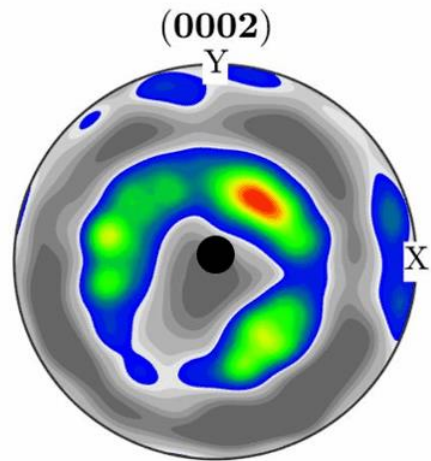
Engineering Stress-Strain



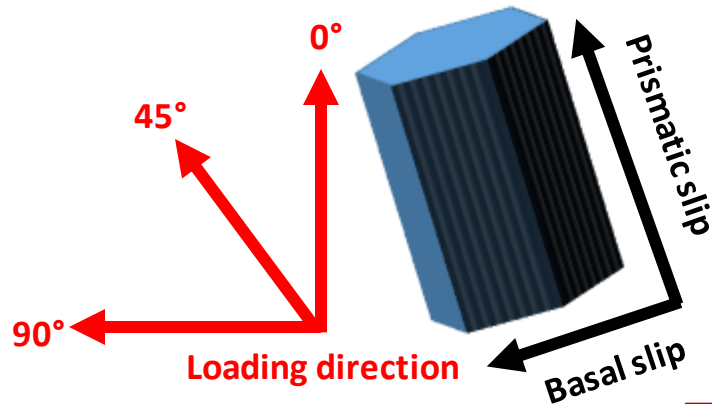
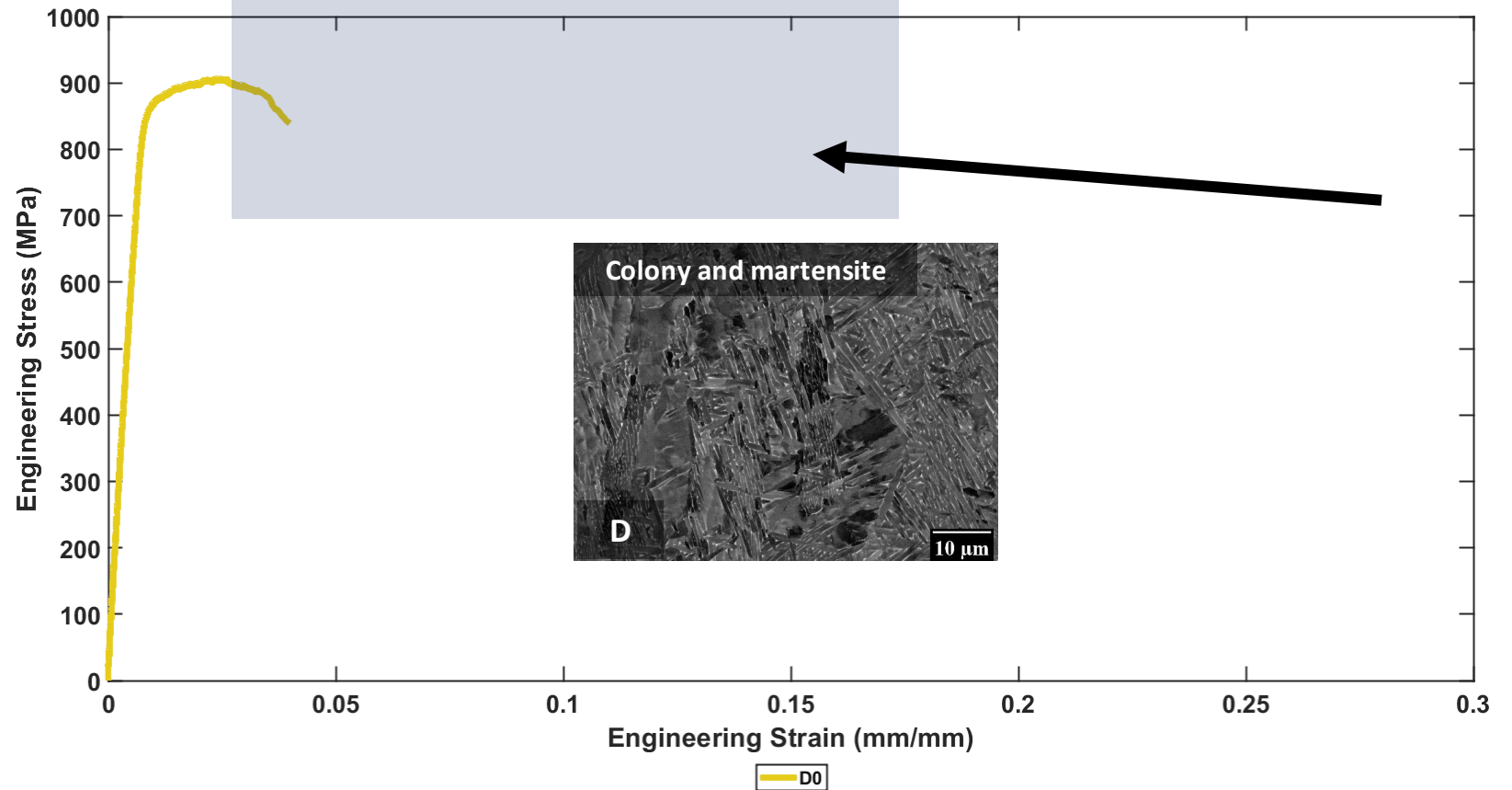
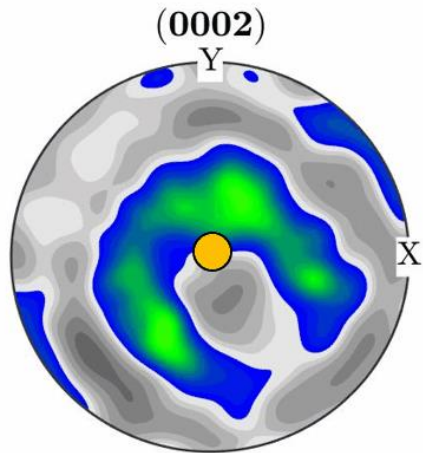
True Stress-Strain



Texture and Mechanical Response: R

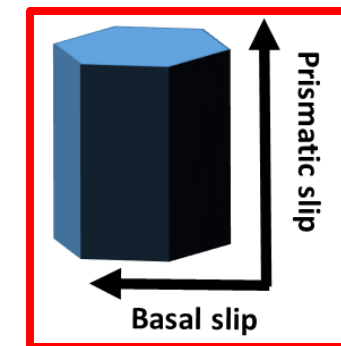
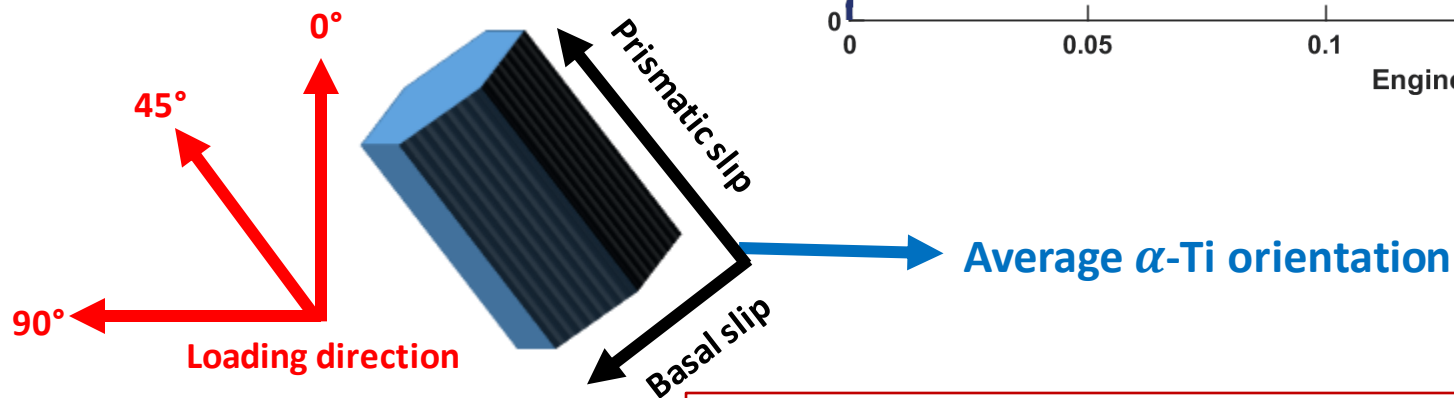
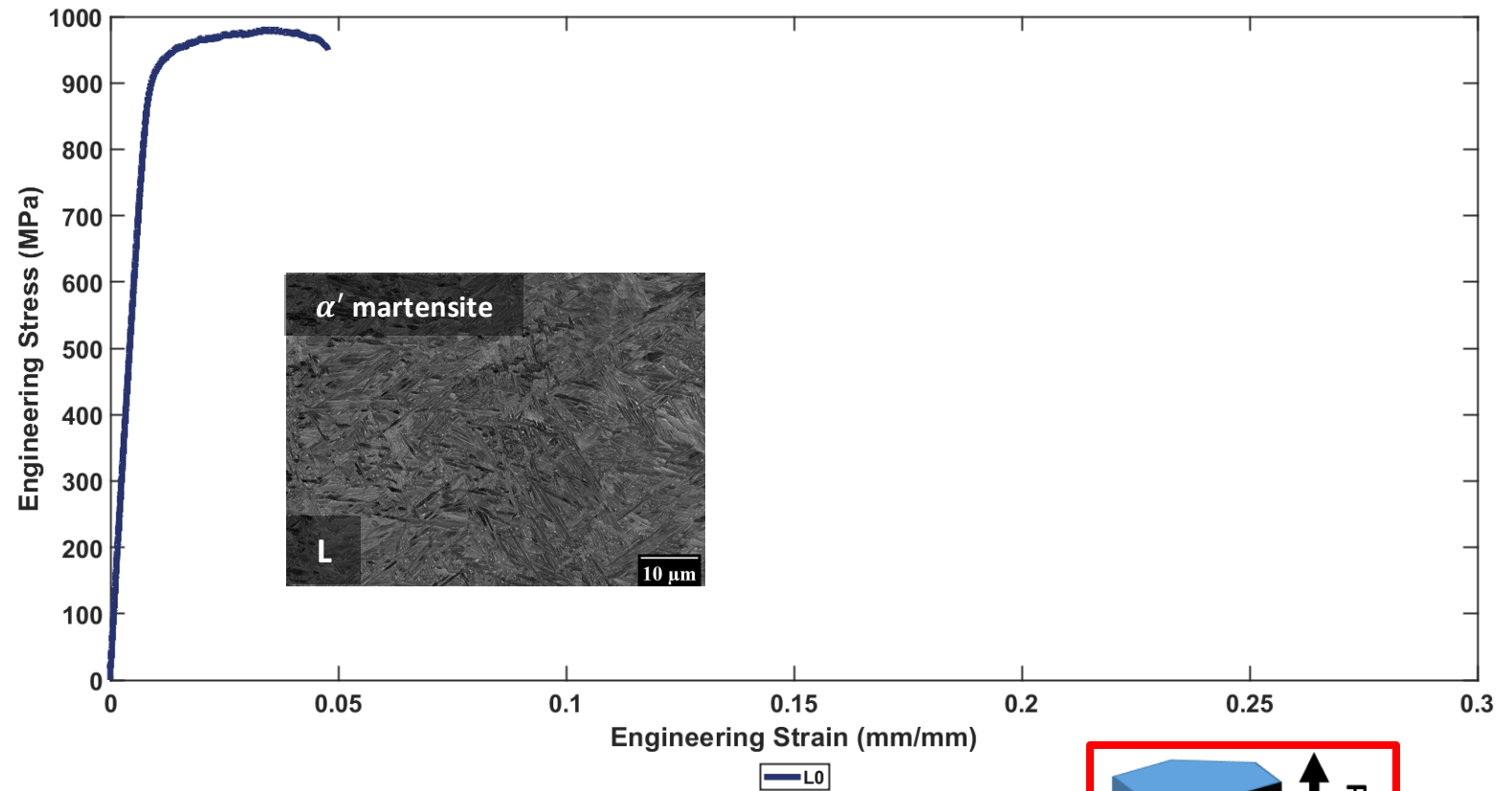
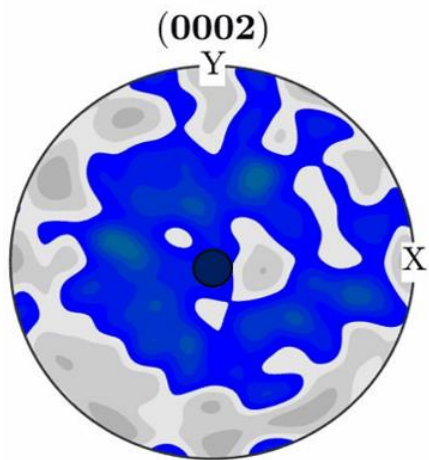


Texture and Mechanical Response: D



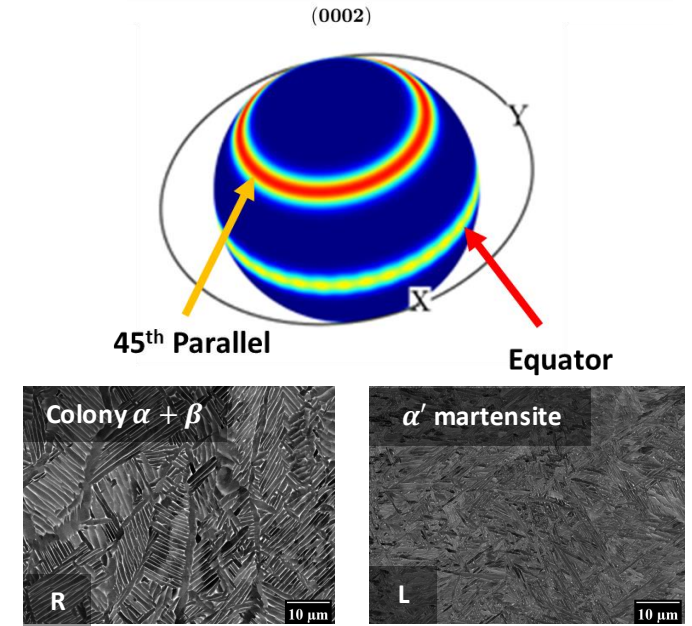
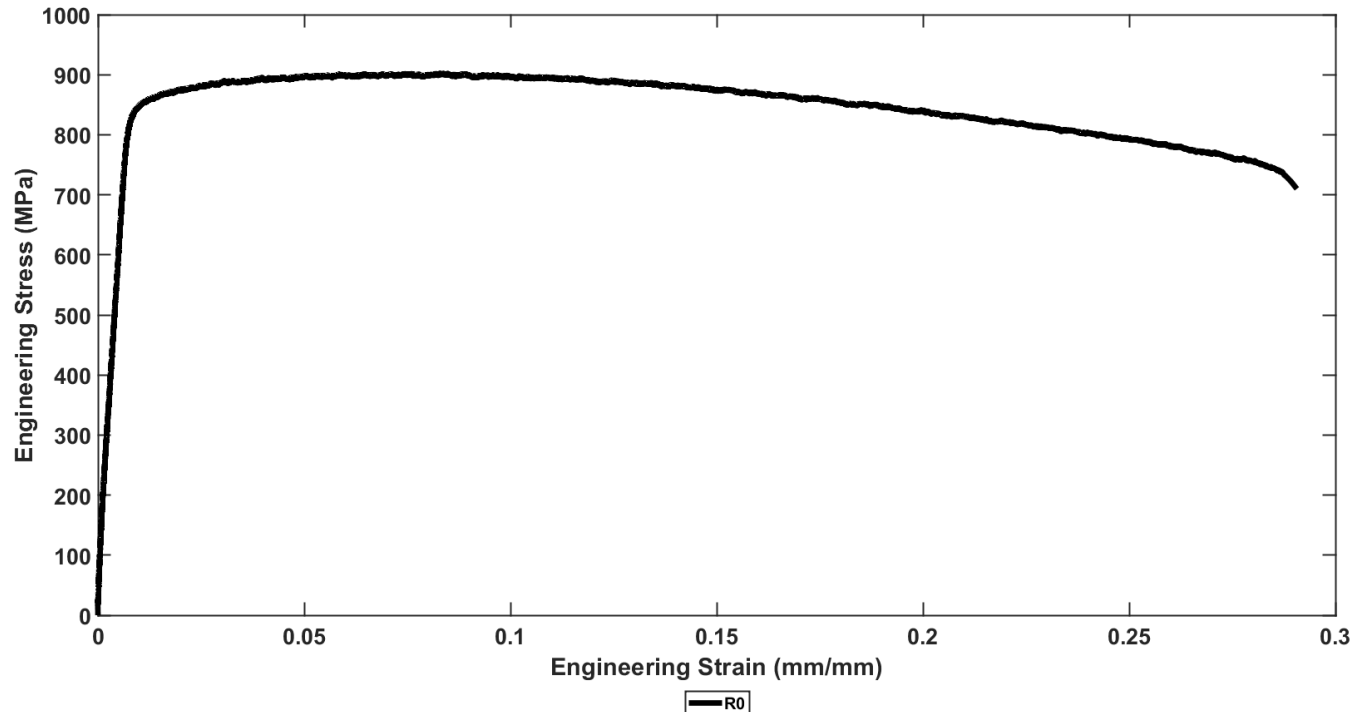
Average α -Ti orientation

Texture and Mechanical Response: L



Summary

- AM Ti-6Al-4V
 - Texture can be used as a marker of microstructural condition
 - Fiber texture = Finer as solidified β -Ti grains
 - Increased 45^{th} Parallel orientations = *Diffusional microstructures*
 - Increased *Equator* orientations = *Diffusionless microstructures*
 - Orientations demonstrate noticeable influence on properties
 - Explanation still being developed

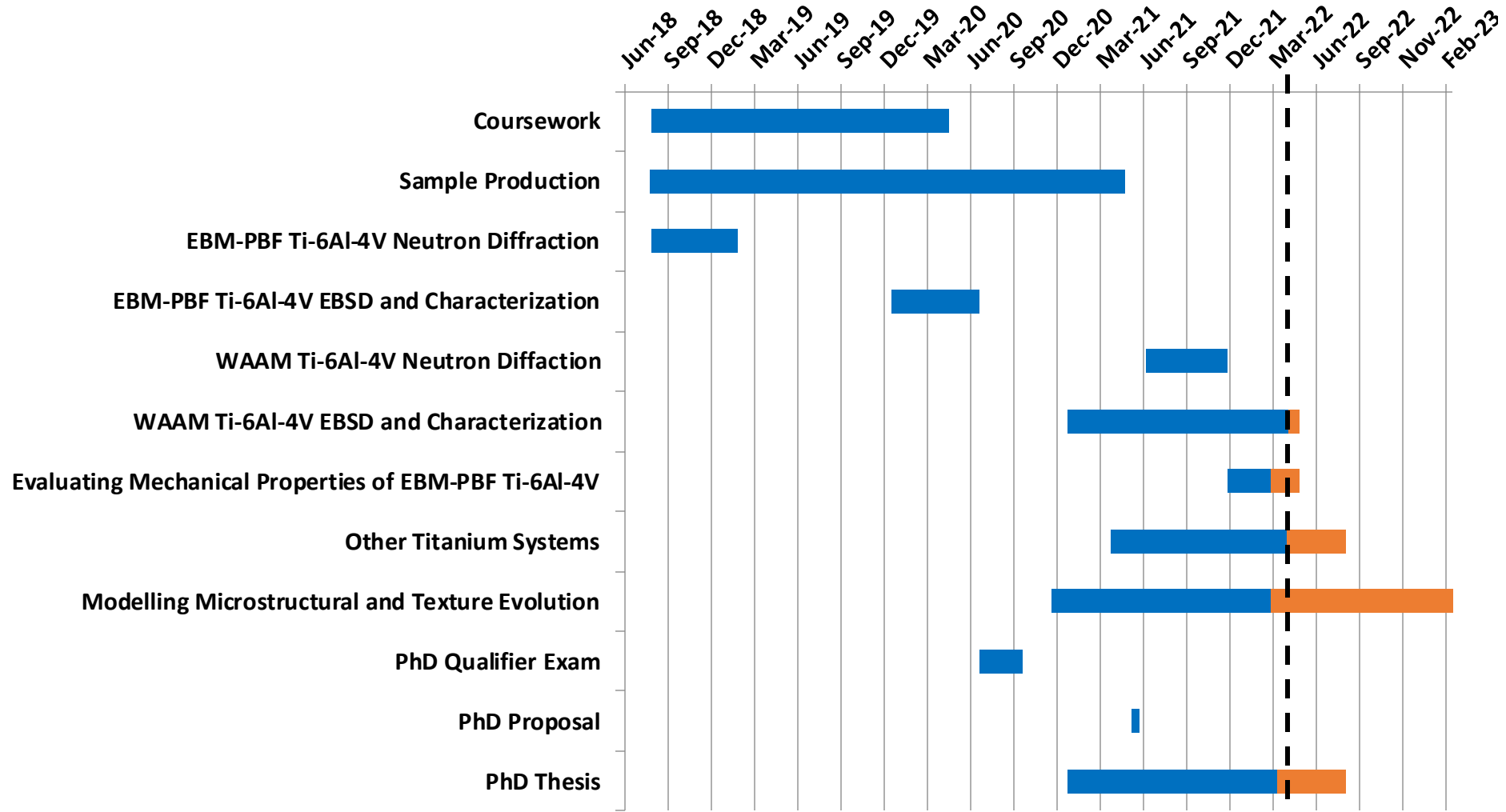


Challenges & Opportunities

- Limited material for follow-up tensile tests
 - No duplicates
- Non-ASTM specimen size
 - Restricted volume available for testing



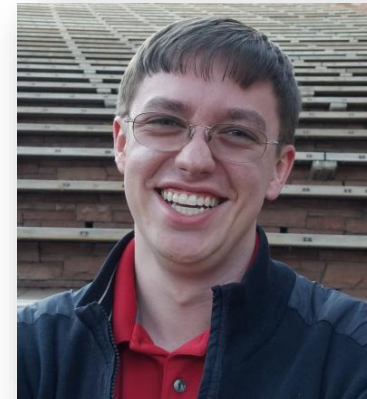
Project Timeline



**Thank you for listening! Any questions,
comments, or concerns?**

Thank you Jake Benzing for all the EBSD help!

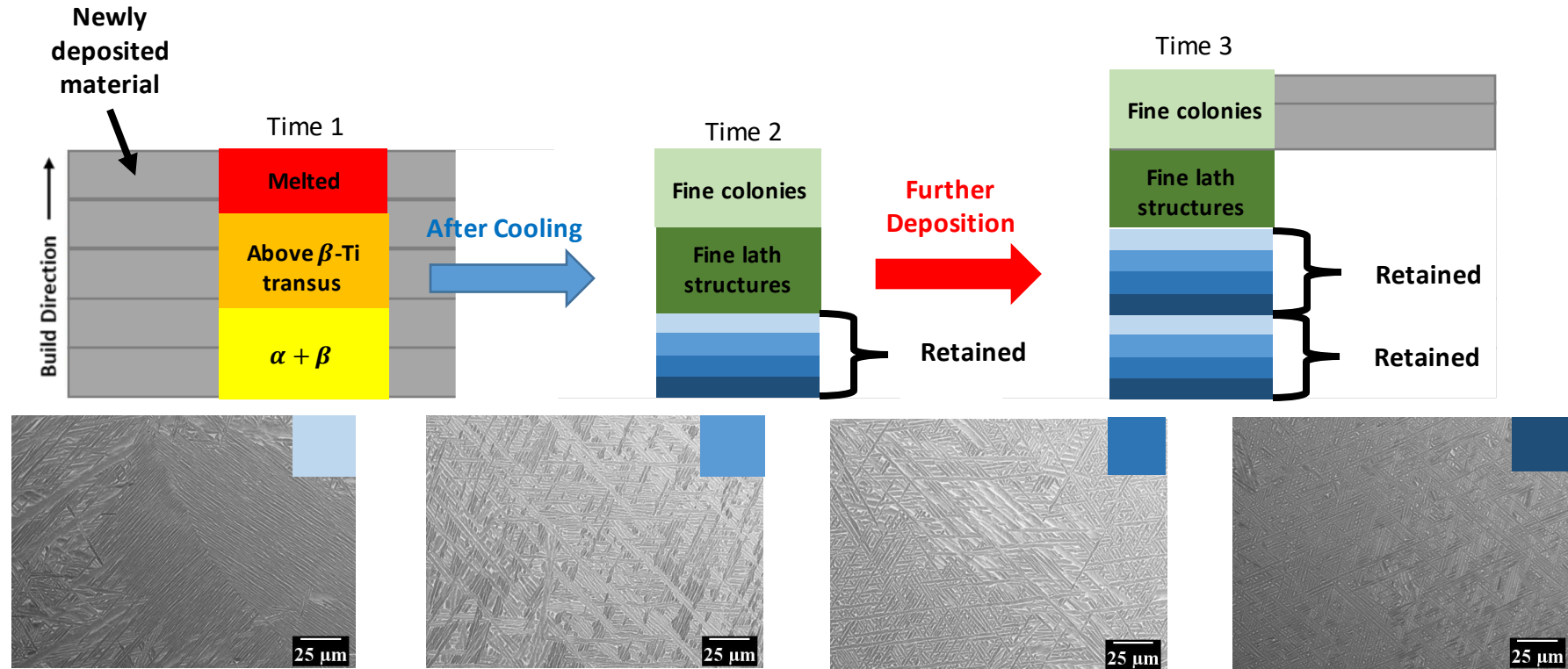
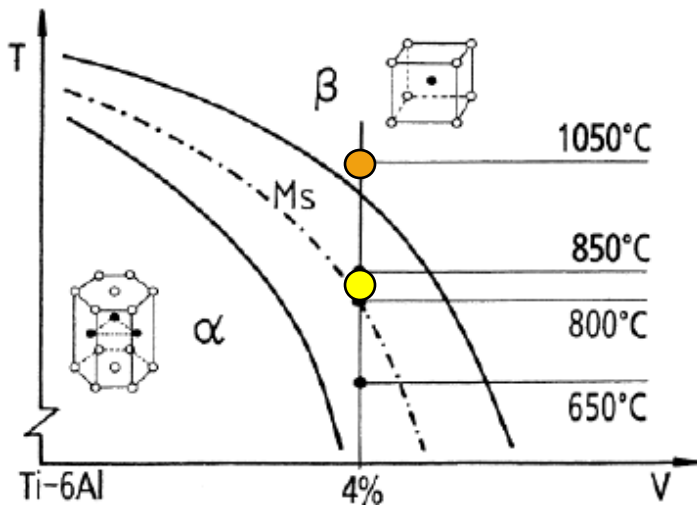
Alec Saville
asaville@mymail.mines.edu



Extra Slides



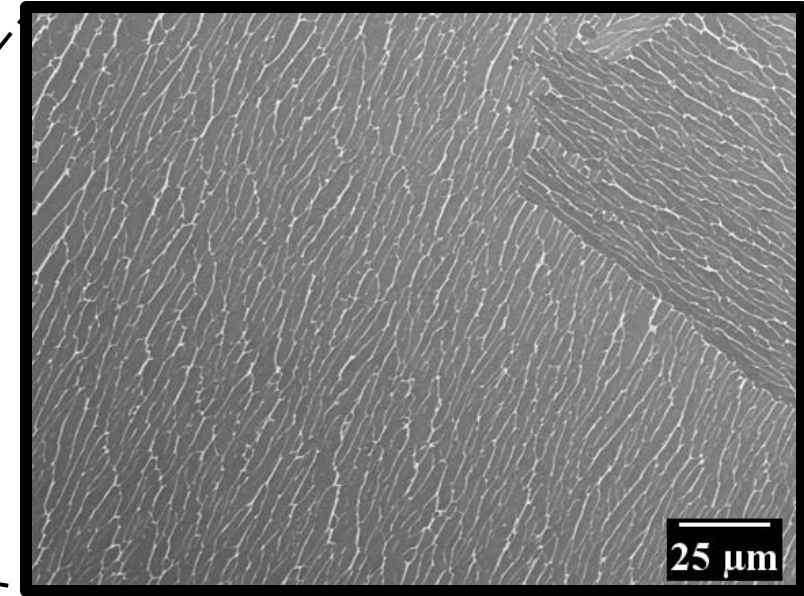
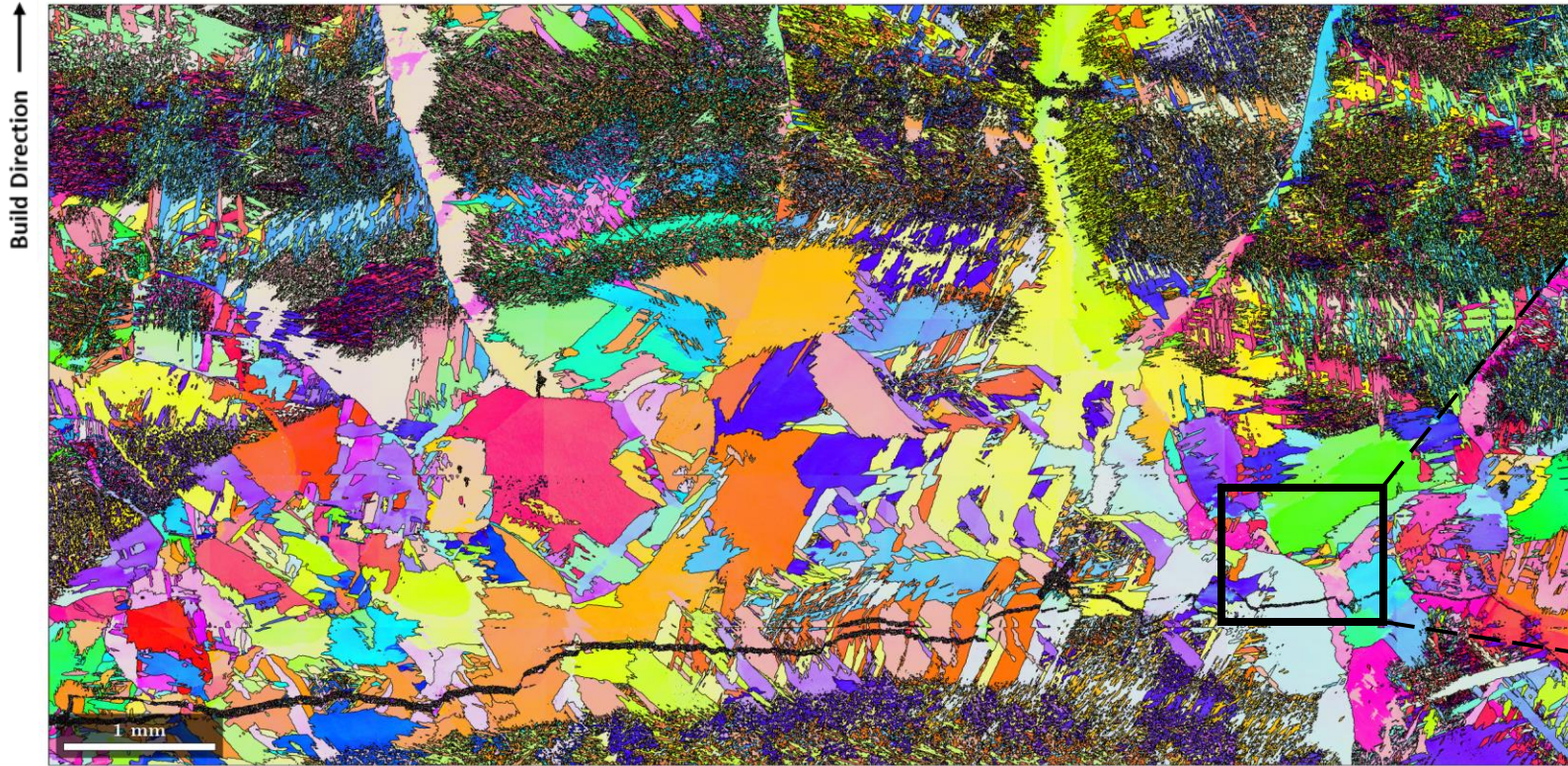
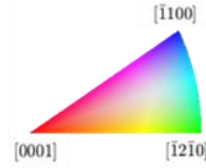
Explaining DED Microstructures



A. Ducato, et al., An Automated Visual Inspection System for the Classification of the Phases of Ti-6Al-4V Titanium Alloy, Springer Berlin Heidelberg, Berlin, Heidelberg, 2013: pp. 362–369..

Kelly, S.M., Kampe, S.L., 2004. Microstructural evolution in laser-deposited multilayer Ti-6Al-4V builds: Part I. Microstructural characterization. Metall and Mat Trans A 35, 1861–1867. <https://doi.org/10.1007/s11661-004-0094-8>

Crack Propagation



Such crack propagation was *not* observed in four-layer microstructure

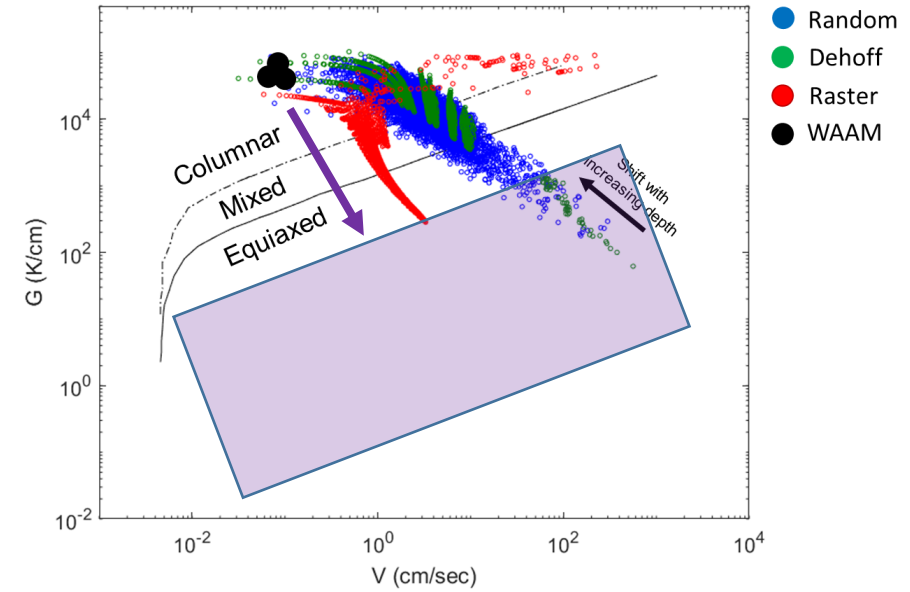
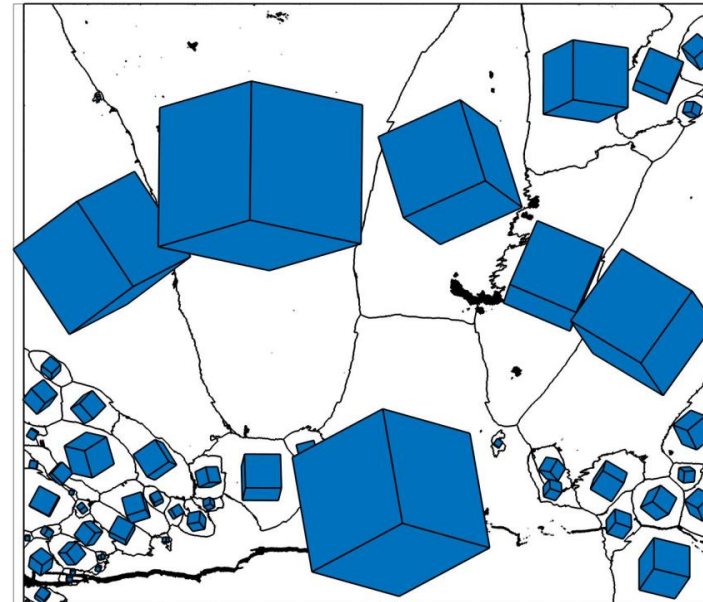
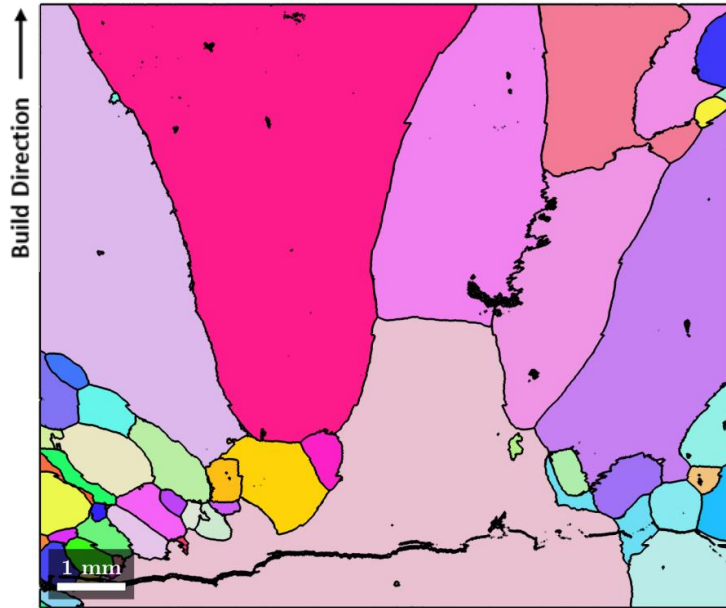
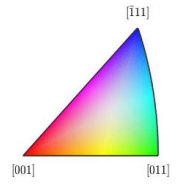


Why did cracking propagate so far through $\alpha + \beta$ colonies?



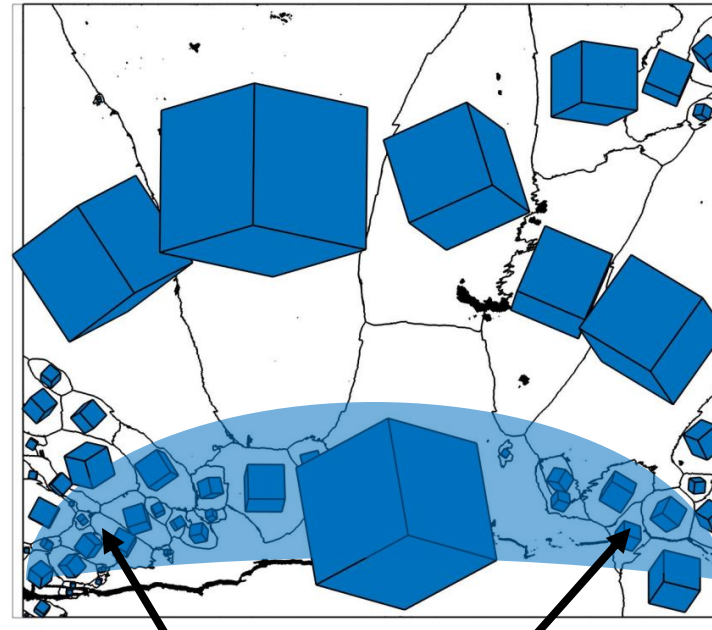
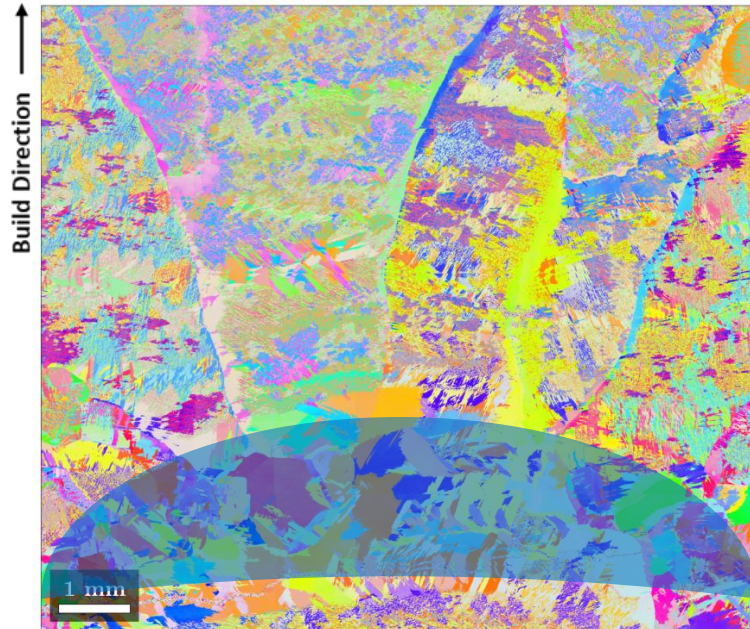
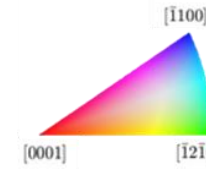
Colonies acted as super-highways for crack propagation

The Smoking Gun - Solidification

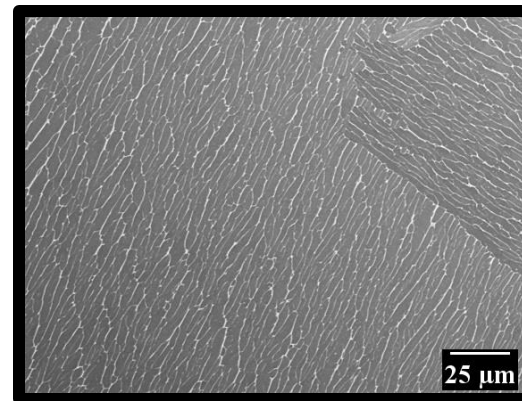
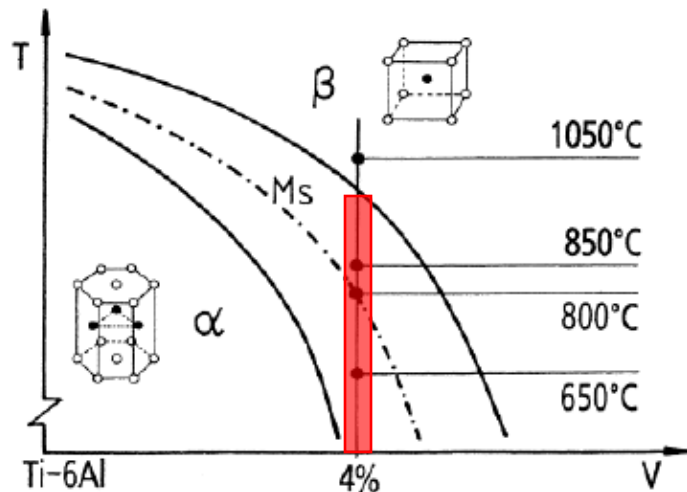
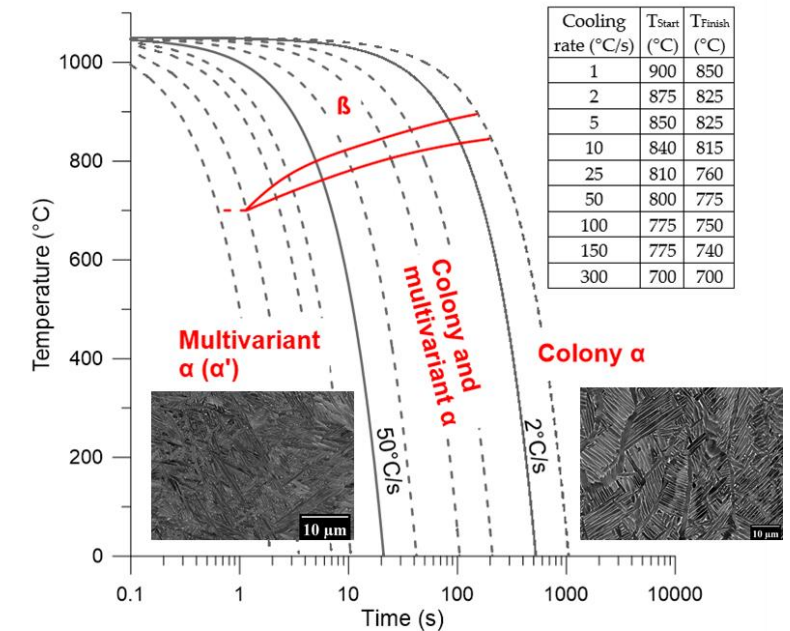


Why did the solidification conditions suddenly change?

The Second Smoking Gun – Colonies



Equiaxed β -Ti grains form in solidification!



Theory 1: Colonies formed from heat buildup reducing solid state cooling rate

Push colonies above the β -Ti transus

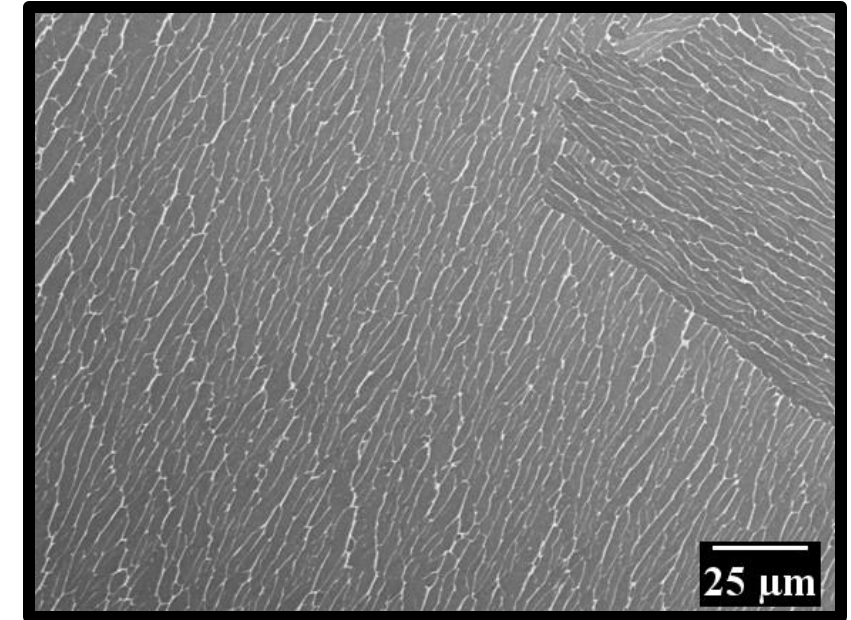
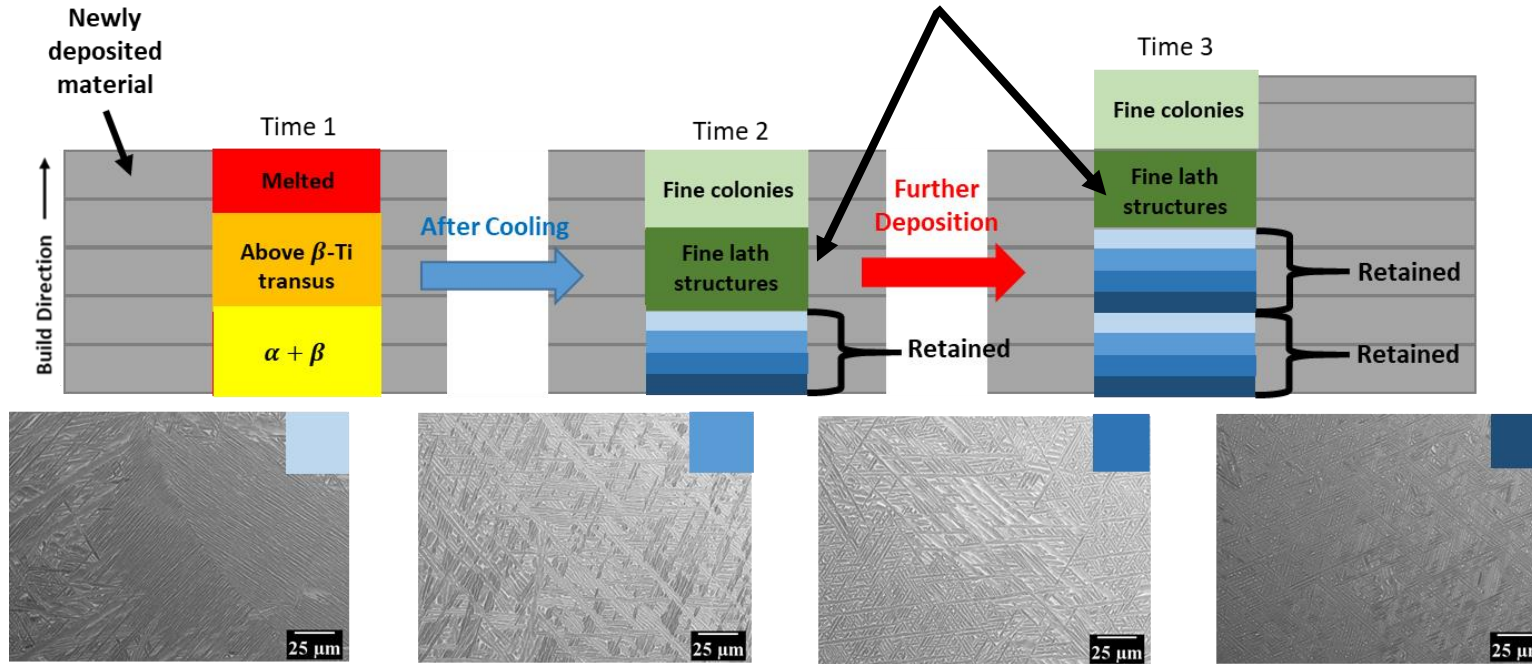
Theory 2: Colonies formed due to a build pause during production

Build volume allowed to cool, subsequent passes unable to push initial colonies over β -Ti transus

Heating enables colony coarsening

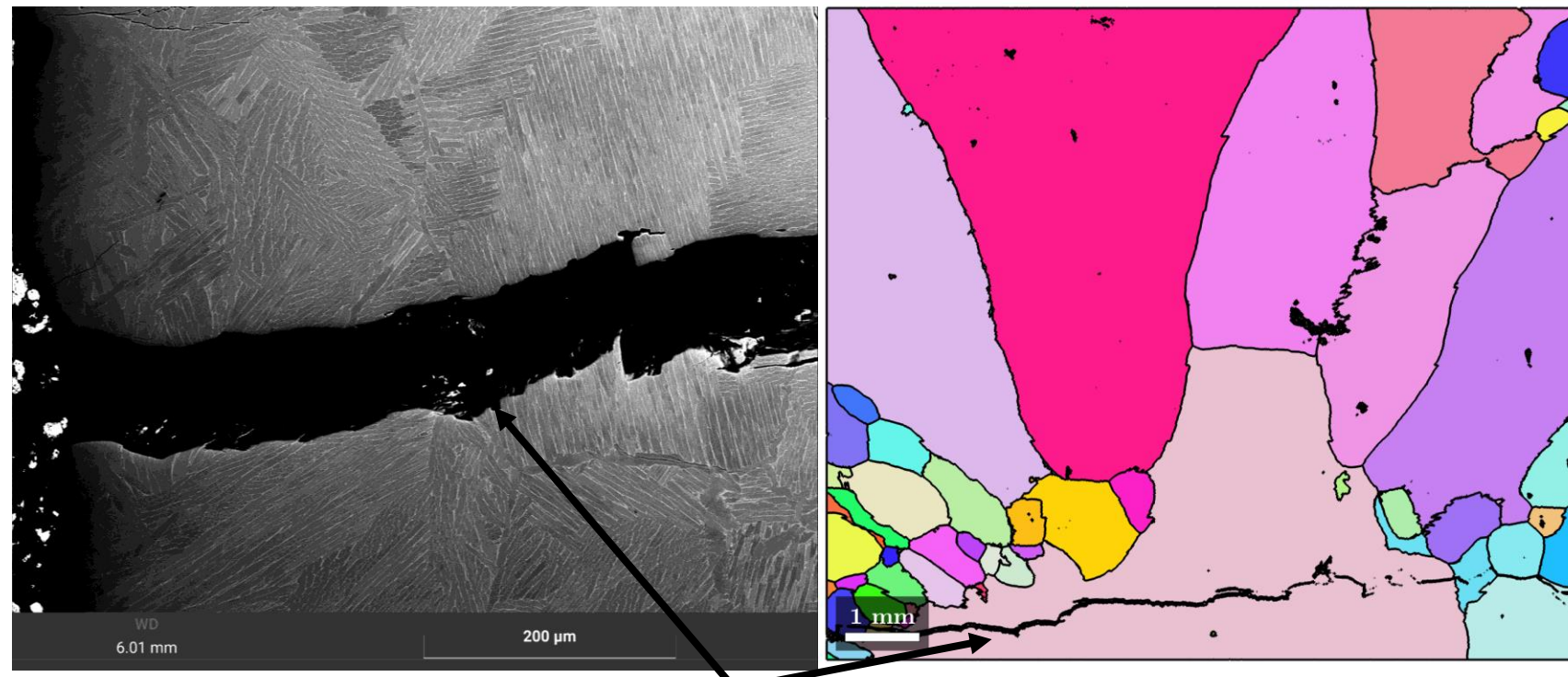
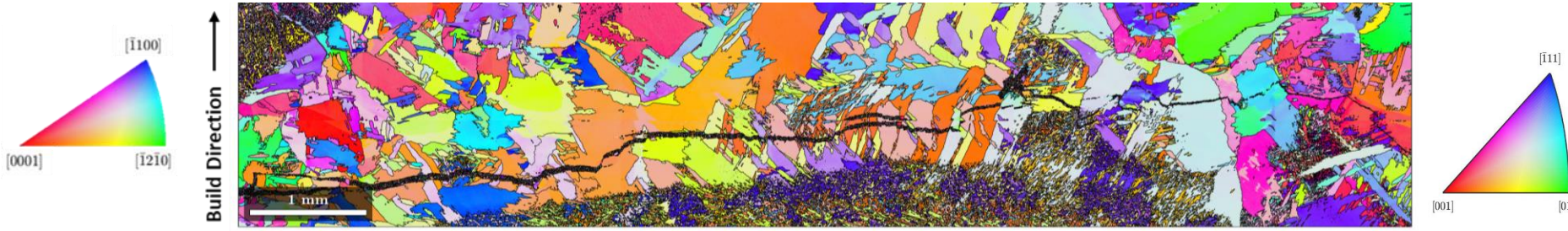
Colonies Are Expected In Fresh β -Ti

Colonies do not form lath structures

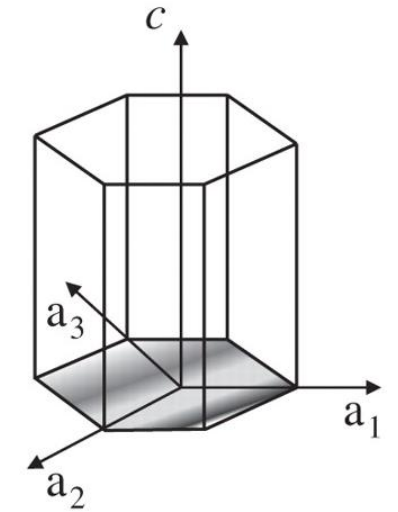


Why do we see near complete fracture *ONLY* in this region?

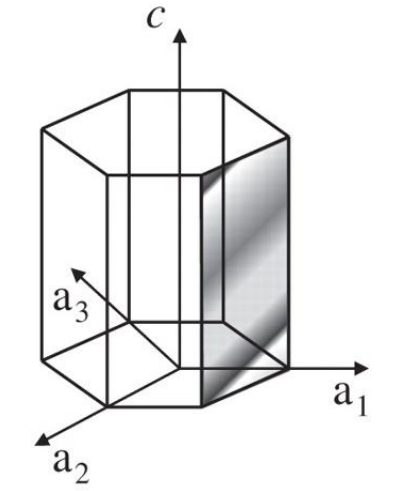
The Third Smoking Gun – Crack Path



Crack propagates through colonies, not β -Ti grains



basal $\langle a \rangle$, 3



prismatic $\langle a \rangle$, 3