

#### Center for Advanced Non-Ferrous Structural Alloys An Industry/University Cooperative Research Center

#### **Project 52-L: Data Driven Qualification (DDQ) Framework for Metals Additive Manufacturing (AM)**

## Semi-annual Fall Meeting October 2021

Student: Charles Smith (Mines)

Faculty: Jonah Klemm-Toole (Mines)

COLORADOSCHOOLOFMINES.

Amy Clarke (Mines)

Participant: Craig Brice (Mines)

Sponsor: National Center for Defense Manufacturing and Machining (NCDMM)



**IOWA STATE UNIVERSITY** 

#### **Project 52-L: Data Driven Qualification** (DDQ) Framework for Metals Additive Manufacturing (AM)



<ul> <li>Student: Charles Smith (Mines)</li> <li>Advisor: Jonah Klemm-Toole (Mines)</li></ul>	Project Duration
Amy Clarke (Mines)	M.S. January 2021 to December 2022
<ul> <li><u>Problem</u>: The range of equipment suppliers that use their own proprietary feedstock and process parameters makes each AM system and qualification protocol unique.</li> <li><u>Objective</u>: Use a data driven qualification approach to form relationships across platforms and alloy systems using intelligent machine learning algorithms and physics-based modeling.</li> <li><u>Benefit</u>: Accelerated qualification and adoption of AM parts into military vehicles.</li> </ul>	<ul> <li><u>Recent Progress:</u></li> <li>Basic thermodynamic (heat transfer) modeling to model melt pool geometry and fusion characteristics has continued.</li> <li>Simulations to model thermal gradients and heating characteristics has been furthered developed.</li> <li>Microstructure characterization has begun.</li> </ul>

Metrics			
Description	% Complete	Status	
1. Literature review	40%	•	
2. Development of metallographic preparation techniques	90%	•	
3. Preliminary thermodynamic modeling/simulations	50%	•	
4. Microstructure Characterization	10%	•	
5. Development of G (temperature gradient) and V (solidification velocity) diagram	0%	•	

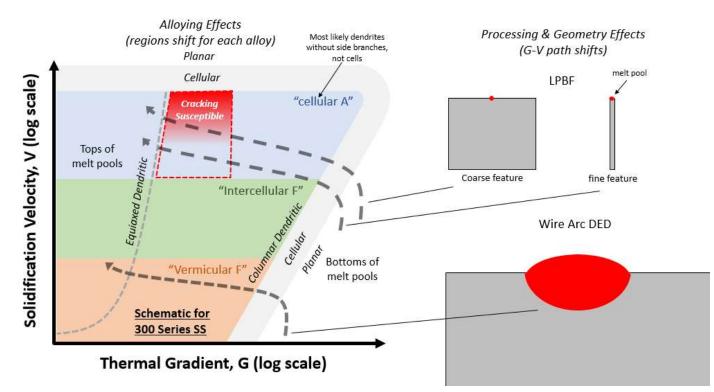
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#### **Overview**

- Proposed Project Methodology
- Modeling and Simulation
  - Effects of laser powder bed fusion processing parameters on microstructure and defects in the as-built conditions
  - Heat transfer/thermodynamic process maps to predict process windows
  - Moving point heat source to model laser powder bed fusion processes
- Metallography and Characterization
- Challenges & Opportunities

### Proposed Processing – Microstructure Mapping Methodology



The map can predict the resulting microstructure (solidification + solid state transformation) if the solidification velocity and thermal gradient are known for a process.

J. Klemm-Toole, unpublished, 2020 CANFSA FALL MEETING – OCTOBER 2021

Center Proprietary – Terms of CANFSA Membership Agreement Apply

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#### Process Parameters Effect on Build Properties

- Optimal energy density (E<sub>d0</sub>) of about 100-105 J/mm<sup>3</sup> for 316L
  - Results in the least porosity, good surface finish, and hardness
- $E_d < 0.5 E_{d0}$  reduction in density
- $0.5E_{d0} < E_d < 0.7E_{d0}$  inconsistent density
- $0.7E_{d0} < E_d < 1.3E_{d0}$  optimum processing

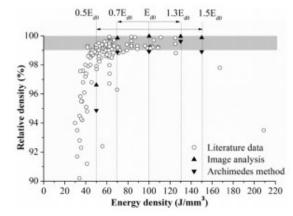
100,0	
99,8	
	SS 1400,HD 0.12
$\smile$	SS 1400,HD 0.09
A 99,4 99,2 99,0	A SS 1100,HD 0.09
6 99,21	SS 800, HD 0.12
□ 99,0+	SS 1400,HD 0.06
00.0	/ SS 800, HD 0.09
98,8	SS 1100,HD 0.06
98,6	SS 800, HD 0.06
+	Energy Density (J/mm <sup>3</sup> )



$$E_d = \frac{P}{\nu * h * l}$$

Where:

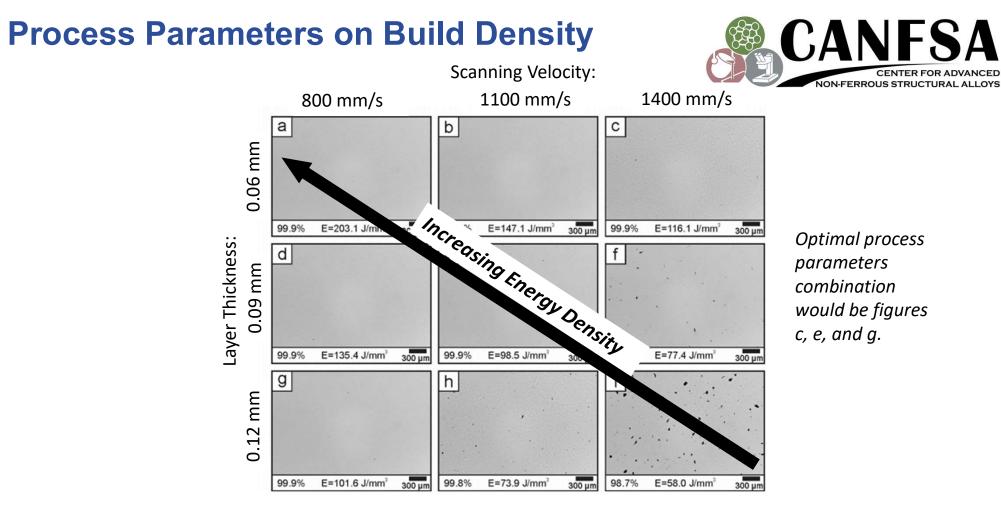
- $E_d$  = total volumetric energy (J/mm<sup>3</sup>)
- P = laser power (W)
- v = scan speed (mm/s)
- *h* = hatch distance (mm)
- *l* = layer thickness



#### Increasing energy density leads to a decrease in porosity

A. Leicht et al, Materials Characterization, 2019 M. Zhang et al, Materials Science and Engineering A, 2017

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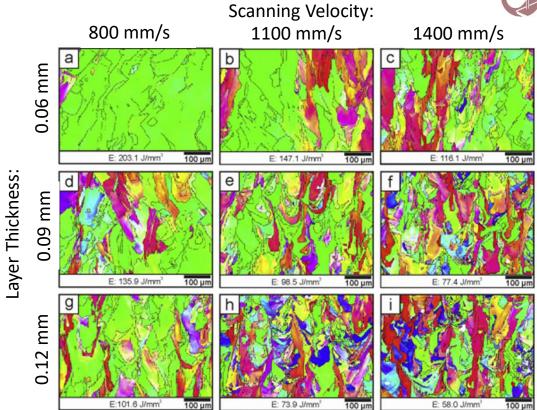


#### Negligible changes in porosity are observed above an energy density of ~ 98 J/mm<sup>3</sup>

A. Leicht et al, Materials Characterization, 2019

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#### **Process Parameters on Grain Orientation**



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Higher energy densities are correlated with coarser microstructures likely leading to lower strengths in the as-built condition

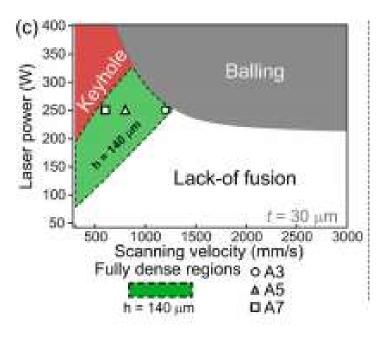
A. Leicht et al, Materials Characterization, 2019

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### **Defect Maps**

- Each region represents parameter sets that could experience solidification defects
- These maps were developed using the Rosenthal Model along criteria proposed through literature





By using process parameters along with predicted melt pool geometry, it is possible to predict and control defects and solidification behavior

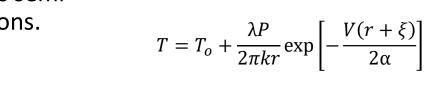
J. Zhu et al, Additive Manfacting, 2021

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## **Rosenthal Model**

#### Assumptions

- Thermophysical properties are temperature independent.
- Scanning speed and power input are constant.
- Point heat source
- The heat transfer is governed purely by conduction.
- The Rosenthal model predicts semicircular melt pool cross-sections.



#### The Rosenthal model predicts semi-circular melt pool cross-sections whose depths and widths are deeper and narrower respectively compared to many additive manufacturing techniques.

2.02

2.01

2.00

1.99

1.98

1.97

(mm) 1.96 1.95 1.94 1.93 2 1.92

1.91

1.90

1.89

1.88

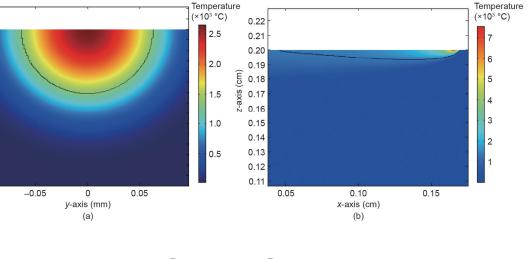
1.87

1.86

P. Promoppatum et al, Engineering, 2017

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## **Criteria for Defect Maps**

• Balling:

$$\frac{\pi W}{L} < \sqrt{\frac{2}{3}}$$

• Lack of fusion:

$$\left(\frac{h}{W}\right)^2 + \frac{t}{t+D} \ge 1$$

• Keyhole-induced pore formation:

$$\frac{\Delta H}{h_s} = \frac{AP}{\pi h_s \sqrt{\alpha v a^3}} > \frac{\pi T_b}{T_m}$$

Where:

- *W* = melt pool width
- *D* = melt pool depth
- *L* = melt pool length
- *h* = hatch distance
- *t* = layer thickness
- *P* = laser power
- *a* = beam radius
- v = scanning velocity

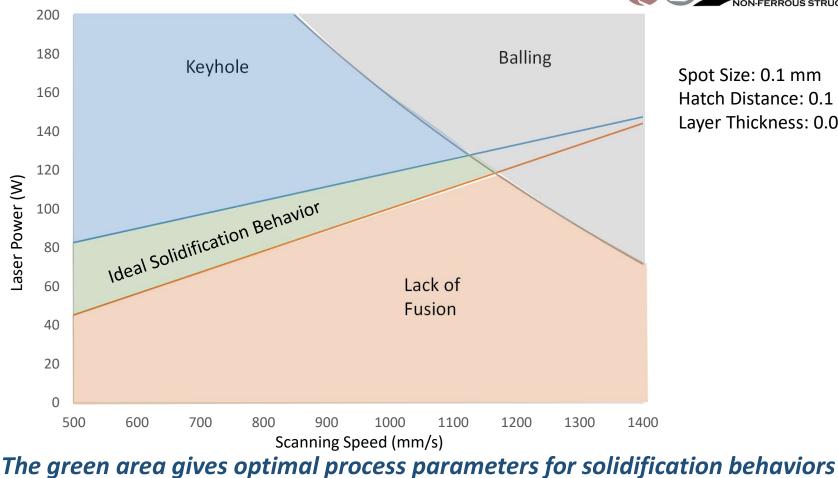
#### The solidification behavior and defect formation can be predicted using process parameters and melt pool geometry

J. Zhu et al, Additive Manfacting, 2021

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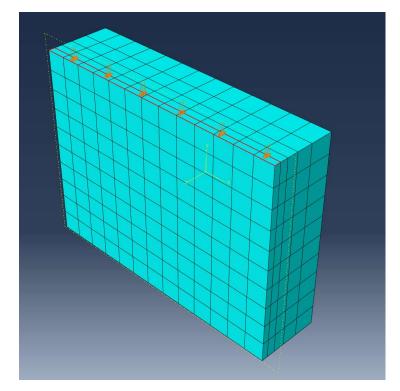
> Spot Size: 0.1 mm Hatch Distance: 0.1 mm Layer Thickness: 0.04 mm

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## **ABAQUS Simulations**

- Simulates a single-track moving heat source on a 316L steel plate
- Uses a Goldak heat distribution to model a moving heat source
- Accounts for convection and radiation loss to the environment, but only conduction in the melt pool
- Possible to do more complex processes including material depositing, multi-passes, and different scanning strategies.



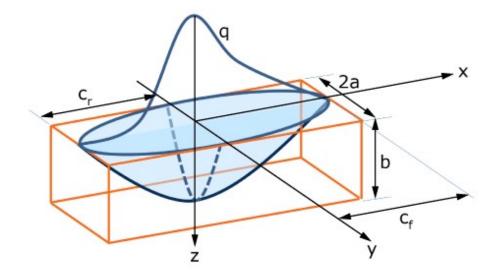


ABAQUS simulations are used to model thermal gradients and temperature histories during the build process

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# Goldak Distribution and ABAQUS Simulation

- Developed in 1984 by John Goldak for welds
- Aimed to expand on the Rosenthal model to overcome some of the limiting assumptions
- Model is based on a double ellipsoid configuration



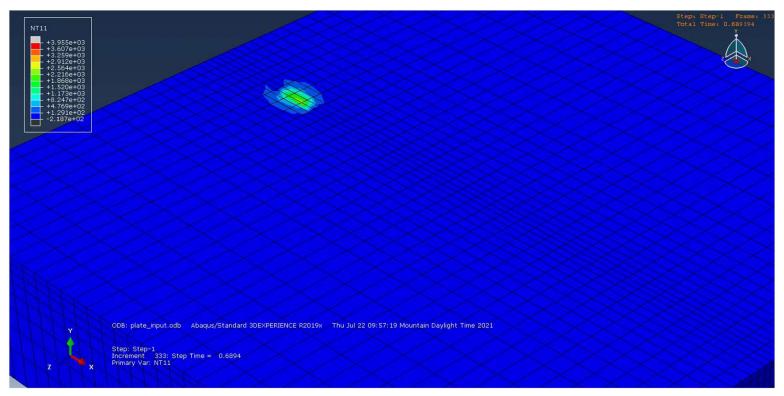
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## The Goldak distribution was developed for larger melt pool geometries and the applicability to LPBF is being evaluated

J. Goldak et al, Metallurgical Transaction, 1984 Abaqus software documentation CANFSA FALL MEETING – OCTOBER 2021

#### **Example Simulations – Preliminary**

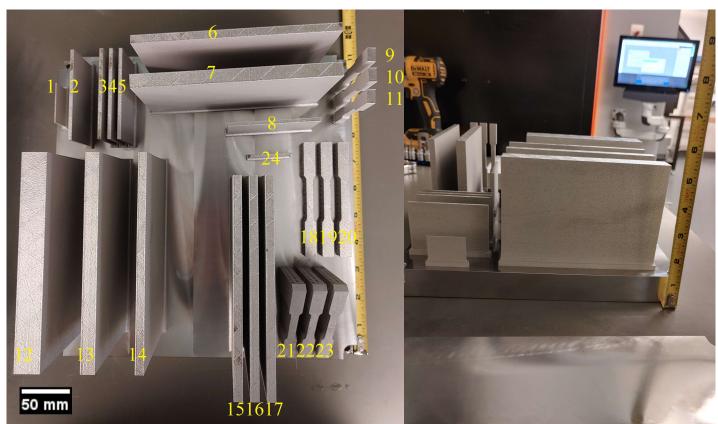




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## First 316L Build

- First build was designed to evaluate effects of part sizes and proximity of nearby builds on microstructure
- This build was done using baseline parameters from 3D Systems using the Mines DMP Flex 350



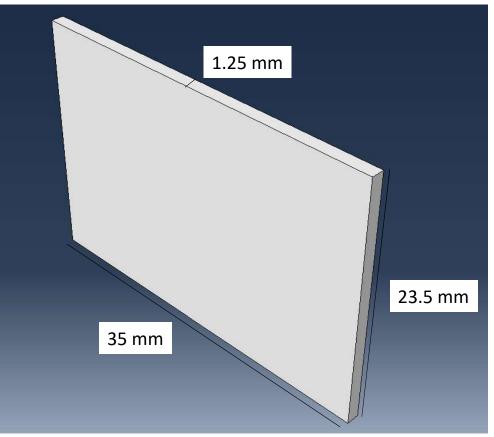


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## Plate 1

- Plate 1 35 mm x 1.25 mm x 23.5 mm
- Plate 1 was isolated in the build and designed to assess the thinnest feature that would be made
- The top section or top layer would experience the smallest amount of remelting and reheating which would give the most meaningful results to compare to solidification models
- ABAQUS simulations were done to predict the thermal gradients and solidification velocity





## Plate 1 Melt Pool Analysis

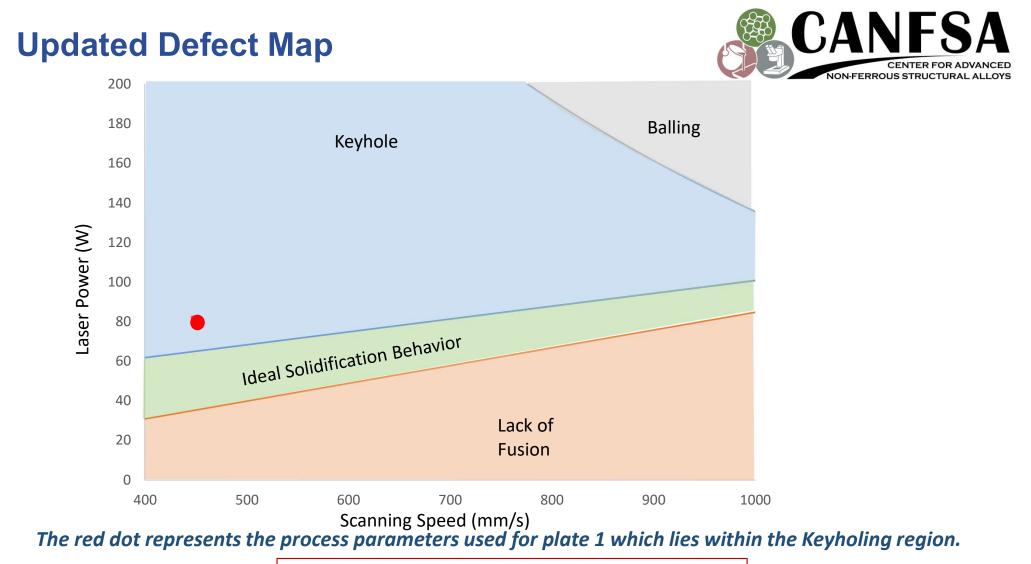
- The melt pool dimensions were predicted using the Rosenthal model
- The melt pool is predicted, using the Rosenthal model, to have a width of 108 μm and a depth of 34.4 μm
- Measurements show an average width of 156 μm and an average depth 48 μm respectively





Discrepancies between Rosenthal model predictions and experiments are likely due to a combination of sectioning effects and the limiting assumptions in the model.

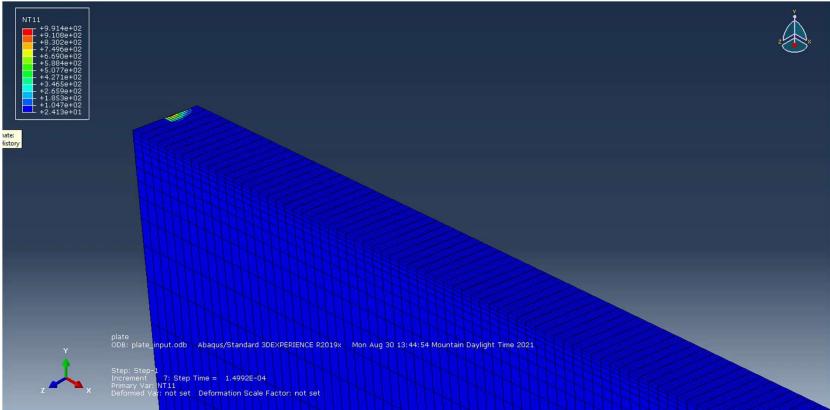
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#### **Simulation of Plate 1**



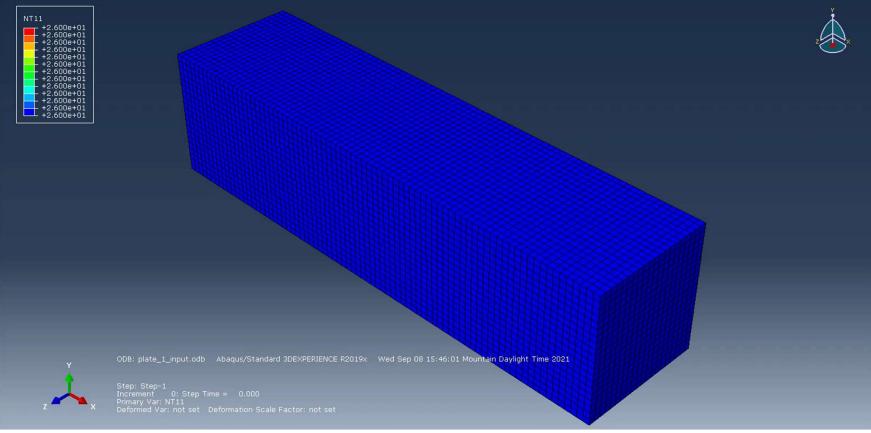


#### Scale model of plate 1 from build 1 showing a moving heat source across the top layer

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#### **Scaled Down Model of Plate 1**



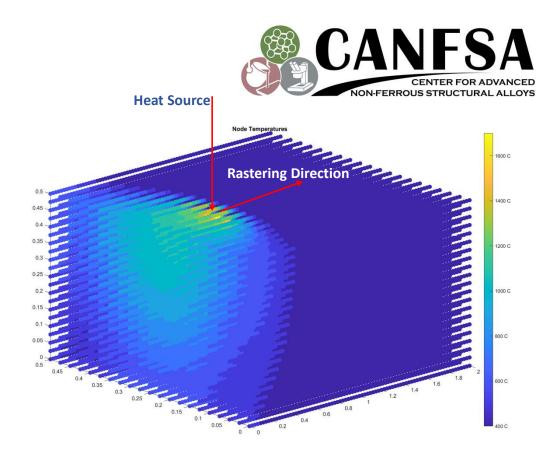


#### A scaled down model allows quicker simulations and a finer mesh

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## **Predictions of G and V**

- The thermal gradient (G) and solidification velocity (V) can be calculated by observing the melt pool boundary
- The nodal temperatures of the model were plotted at a set time
- The surface of the melt pool was also plotted

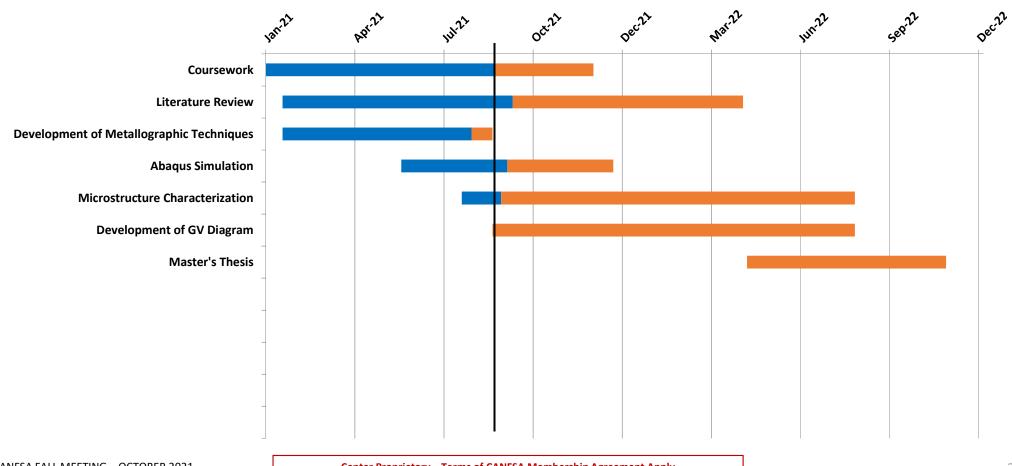


The next step is to overlay the melt pool surface with the nodal temperature plots and observe the movement of the solidification boundary to determine V and observe the thermal gradients across the solidification boundary to determine G.

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## **Challenges & Opportunities**



- The defect maps should provide insight on ideal process parameters, but they do not provide insight on effects of component geometry and solidification conditions.
- Progress is being made with ABAQUS simulations, and more work needs to be done to simulate more complex situations.
- Other simulation programs such as SYSWELD and/or Flow3D will be evaluated to compare the thermal gradients and history.
- Metallography and melt pool analysis has begun
  - Are there any particular microstructural features of interest to the sponsors (other than dendrite arm spacing, grain size, and grain morphology)?
- There are many other aspects of AM that these microstructure maps will not take into consideration.
  - Are there any recommendations from the sponsors about critical variables or microstructure features that should be included?

Thank you! Charles Smith ctsmith@mines.edu

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#### References



- A. Leicht, M. Rashidi, U. Klement, and E. Hryha, "Effect of process parameters on the microstructure, tensile strength and productivity of 316L parts produced by laser powder bed fusion," *Mater. Charact.*, vol. 159, no. August 2019, p. 110016, 2020, doi: 10.1016/j.matchar.2019.110016.
- M. Zhang *et al.*, "Fatigue and fracture behaviour of laser powder bed fusion stainless steel 316L: Influence of processing parameters," *Mater. Sci. Eng. A*, vol. 703, no. April, pp. 251–261, 2017, doi: 10.1016/j.msea.2017.07.071.
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- Goldak, J., Chakravarti, A. & Bibby, M. A new finite element model for welding heat sources. Metall Mater Trans B 15, 299–305 (1984). https://doi.org/10.1007/BF02667333