

Center for Advanced **Non-Ferrous Structural Alloys** An Industry/University Cooperative Research Center

Project 37b-L: Rationalization of Liquid/Solid and Solid/Solid Interface Instabilities During Thermal-Mechanical Transients of Metal Additive Manufacturing (ISU)

Fall Meeting

October 13th – 15th 2020

- ISU team: Katie O'Donnell, Matt Kenney, Maria Quintana ٠
- Faculty: Dr. Peter Collins (ISU)
- MURI members: S. Babu (UTK), A. Clark (CSM), J. R. Jinschek (OSU), T. Pollock (UCSB), Z. Kong (VT), S. Ringer and X. Liao (Univ. of Sydney), S. Primig (Univ. of South Wales) and TEAM
- Industrial Mentors: Billy Short and Jennifer Wolk (ONR)



Project 37b-L: Rationalization of Liquid/Solid and Solid/Solid Interface Instabilities During Thermal-Mechanical Transients of Metal Additive Manufacturing (ISU)



 ISU team: Matt Kenney, Katie O'Donnell,	Project Duration
Maria Quintana Advisor(s): Prof. Peter Collins (ISU)	August 2018 to August 2021
 <u>Problem</u>: Understand the thermal gradients in an AM build as a function of different scan strategies by studying the microstructure. <u>Objective</u>: To understand the science behind the relation between thermal gradients and the microstructure and texture evolution. <u>Benefit</u>: Optimize the final cost and mechanical properties of the AM component. 	 Recent Progress Investigations into spherical porosity CT analysis to locate spherical pores SEM imaging to determine distribution between and within samples EDS analysis around selected defects Finite element modeling of the effect of thermal stresses on spherical pores

Metrics					
Description	% Complete	Status			
1. Sample preparation for optical, SEM-BSE, EBSD and TEM	60%	•			
2. Literature review	70%	•			
3. Texture scans – EBSD, SRAS, and ASTAR PED		•			
4. 3D analysis		•			
5. Relate thermal gradients to microstructure and the final mechanical properties	30%	•			

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Industrial Relevance



- Understanding underlying behavior of different AM strategies on resulting microstructure and mechanical properties of metallic printed parts
- Build a scientific basis into Integrated Computational Materials Engineering (ICME) predictions of AM knowledge gap areas (nano and micro scale regimes of length and time)
- Reduce trial and error phase of AM design and manufacture curve

Progress





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Research Interests



 Three different AM scan strategies are selected to understand fundamental research questions. The different scan strategies will change the thermal gradient: Raster, Dehoff, and Random



Ordering of Raster Fill

Ordering of Dehoff Fill

Ordering of Random Fill

Outline of the Project



 Ti64 AM and Inconel 738 builds with different scan strategies are provided by ONRL – Raster(L), Random (R) and Dehoff (D)

• TASKS:

- Imaging Macro, Optical and SEM-BSE
 - 1. Texture (across length scales)
 - A. SRAS Spatially Resolved Acoustic Spectroscopy (macro-texture)
 - B. EBSD Electron Back Scattered Diffraction (SEM) (micro-texture)
 - C. PED Precession Electron Diffraction (TEM) (nano-texture)
 - 2. Analysis of the 2D and 3D data
 - 3. Develop the understanding to relate thermal gradient to the microstructural evolution

Material



- 3 cuboid Ti64 AM builds Raster (L5), Random (R5), and Dehoff (D5)
- 2 cuboid Inconel 738 builds Random and Raster
- Z is the build direction for all the samples
- 3 new Inconel samples cube, pyramid, spiral









strategies had varying amounts of porosity

Porosity in Ti-6AI-4V Samples

All three scanning





Spherical



Lack of fusion (LOF)

Porosity in Ti-6AI-4V Samples



 Porosity can serve as "tattle tales", revealing information about defect formation mechanisms, processing-microstructure relationships in AM builds, as well as much about the dynamics of the process





Spherical Porosity in Our Samples



- Spherical pores in EBM can arise from two general sources:
 - Retained gas porosity, present in the powder used in the process
 - Pores are comprised of argon
 - Keyholing/vaporization of elements during the process
 - Vaporized elements solidify on the surface of formed pores, leading to pores at a partial vacuum
- The different scanning strategies showed a distinct difference in volume fraction and size of spherical porosity present
- Larger spherical pores, present only in the raster scan strategy, had ripples on their free surfaces

Spherical Porosity: Determining the Cause

- Retained gas porosity
 - Entrapped gases (likely argon from powder processing) would still be present in the build
 - Potential analysis technique: SEM/EDS of a spherical pore just below the surface
 - Complications: electron beam penetration and locating spherical pores
- Keyholing/vaporization porosity
 - Vaporized elements would leave a thin layer on the surface of resulting pores
 - Potential analysis technique: Auger electron spectroscopy (AES) of exposed spherical pores



CANES

X-ray CT data of unused powder

Spherical Porosity: Determining CANFSA the Cause

- Electron beam penetration:
 - Rough approximation:

$$x \ (\mu m) = \ \frac{0.1 \ E_0^{1.5}}{\rho}$$

- Using E = 30keV and a density of $4.4g/cm^3$, this gives an electron penetration depth of ~3.7µm
- Matches approximation from Casino software



http://academic.uprm.edu/pcaceres/Courses/CHAMINA/HO5.pdf

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Spherical Porosity: Determining the Cause

- Coupling data from CT experiments with SEM/EDS analysis:
 - CT analysis on a needle sample is being utilized to determine the location of spherical pores
 - Targeted polishing will bring candidate pores close to the surface
 - MultiPrep[™] polishing system allows for precise polishing (1µm resolution)
 - Given exact positions, EDS analysis can then be conducted





Allied MultiPrep[™] Polishing System

Spherical Porosity: Distribution

- L5 sample had three times as many pores as the spot melting samples
- Larger pores (>25µm) were only seen in the L5 sample





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Spherical Porosity: Observations in Larger Pores









Compression and buckling

No evidence of deformation in microstructure

- CANFSA CENTER FOR ADVANCED NON-FERROUS STRUCTURAL ALLOYS
- Goal: Using finite element analysis (COMSOL) to investigate the effect of thermal stresses on argon gas pores in a Ti64 matrix
- General initial conditions:
 - Geometry:
 - Cube of varying sizes
 - 15µm radius sphere, centered
 - Materials:
 - Cube: Ti64
 - Sphere: Argon
 - Fixed surface: Bottom of cube





- COMSOL Physics
 - Thermal expansion:
 - Couples Solid Mechanics with Heat Transfer in Solids
 - Domain: cube
 - Nonlinear Structural Mechanics Module allows for the application of plasticity to the cube as well
 - Laminar Flow
 - Domain: sphere
- COMSOL Study: Stationary



- Initial thermal gradients were generated from a top temperature of 1600°C and a bottom temperature of 985°C
 - 985°C was the lower temperature limit (to stay above the beta transus, all models were kept above this temperature)
 - 1600°C was the upper limit (approximately the melting temperature)

		1.2°C/μm	0.62°C/μm	0.12°C/μm
	500μm x 500μm x 500μm	Top Temp: 1600°C Bottom Temp: 985°C	Top Temp: 1295°C Bottom Temp: 985°C	Top Temp: 1047°C Bottom Temp: 985°C
	1000μm x 1000μm x 1000μm	Top Temp: 2223°C* Bottom Temp: 985°C	Top Temp: 1600°C Bottom Temp: 985°C	Top Temp: 1109°C Bottom Temp: 985°C
	5000μm x 5000μm x 5000μm	Top Temp: 7175°C* Bottom Temp: 985°C	Top Temp: 4080°C* Bottom Temp: 985°C	Top Temp: 1600°C Bottom Temp: 985°C







- Allows for local variation of properties to either side of pore
 - Instabilities can lead to buckling

500μm x 500μm x 500μm cube top temperature: 1600°C bottom temperature: 985°C (thermal gradient: 1.2°C/μm)



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• Ti-6Al-4V Young's modulus: 113.8e9 Pa







- Papers:
 - One paper on LOF (raster sample)
 - One paper on spherical porosity causes and observations
 - One paper on distribution of porosity
- Continued defect analysis
- Analysis of new MURI samples

Progress





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Thank you!

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