

# ***Project 36E-L: In-Situ Characterization of Microstructural Evolution During Simulated Additive Manufacturing in Model Alloys***

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# Project 36E-L: In-Situ Characterization of Microstructural Evolution During Simulated Additive Manufacturing in Model Alloys



- Student: Brian Rodgers (Mines)
- Advisor(s): Amy Clarke (Mines)

**Project Duration**  
PhD: September 2019 to March 2023

- **Problem:** Aerospace components are difficult to produce conventionally, but the effects of additive manufacturing (AM) on microstructural evolution are not understood enough to replace conventional manufacturing.
- **Objective:** Develop an understanding of solidification behavior in model alloys under AM conditions by *in-situ* characterization.
- **Benefit:** Microstructural control for additive manufacturing of aerospace components.

- Recent Progress**
- Analysis of *in-situ* radiography from APS on Ni and Al base alloys
  - Top-down imaging of as-solidified melt pools

Metrics		
Description	% Complete	Status
1. Literature review	15%	●
2. Analysis of APS beam line data	30%	●
3. Analysis of Dynamic Transmission Electron Microscopy (DTEM) of rapid solidification	0%	●
4. Simulation of experimental conditions	15%	●
5. Complementary <i>ex-situ</i> microstructural characterization	5%	●

# Industrial Relevance

Enhanced microstructure/properties:

- Control during AM

Applications:

- Turbine components
- Aerospace components

Benefits:

- Faster production time
- Greater geometric flexibility
- Reduced processing cost
- Enhanced performance
- Low volume / lean manufacturing and customization



Left photo courtesy of  
wikimedia, right photo  
courtesy of NASA

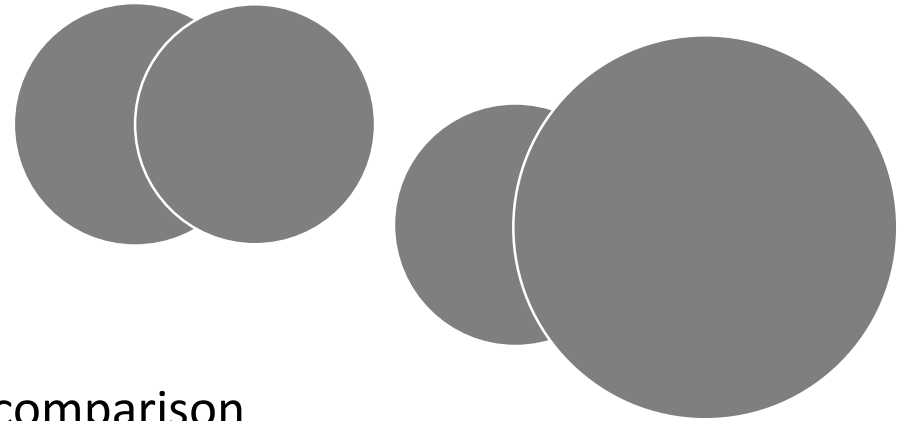
# Model Alloys of Choice



- Two alloy systems:
  - Ni-Al-Mo
  - Al-Ag
- Ni-base system consists of two alloys with same equilibrium  $\gamma'$  volume fraction supplied as single crystals
  - R2: Ni-6.6Al-1.9Mo (wt%)
  - R4: Ni-2.8Al-22.2Mo (wt%)
- Al-Ag binary system consists of two different fractions of silver
  - Al-10Ag (wt%)
  - Al-18Ag (wt%)

# APS Experiments

- R2 and R4 alloys:
  - Rasters at constant heat input
  - Overlapping spot melts
- Al-Ag system:
  - Rasters and re-rasters for DTEM comparison



Alloy	Beam power [W]	Raster speed [m/s]	Notes
R2 [110], R2 [111], R4[100], R4 [110]	253.9	1.6	
	139.4	0.5	
	47.8	0.1	
	82.1/82.1	Spot melt	Edge of second pool intersects middle of first
	82.1/253.9	Spot melt	
R2 [110]	517.1	1.6	
	253.9	1	
R4 [100]	517.1	1.6	
	368.3	1.6	
	253.9	Spot melt	
Al-10Ag & Al-18Ag	282.5/282.5	0.1/0.1	Re-rasters with 100% overlap
	368.3/368.3	2/2	
	282.5	0.1	
	368.3	2	

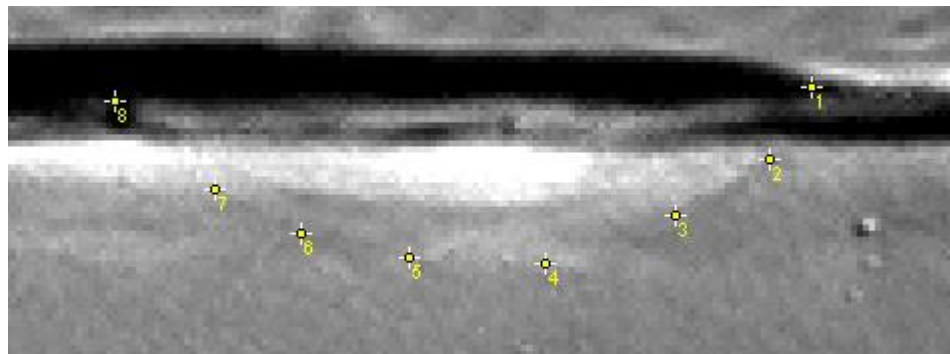
# Recap of Previous Report



- Wrote a MATLAB program to perform Rosenthal-type simulations of conditions from APS
- Molten pool shape results differed between simulations and real melt pools
  - R2 and R4 matched for intermediate raster velocity, mismatched for low and high velocity
  - Al-Ag mismatched for most conditions
  - Non-steady-state behavior, especially keyholing, caused most extreme mismatches

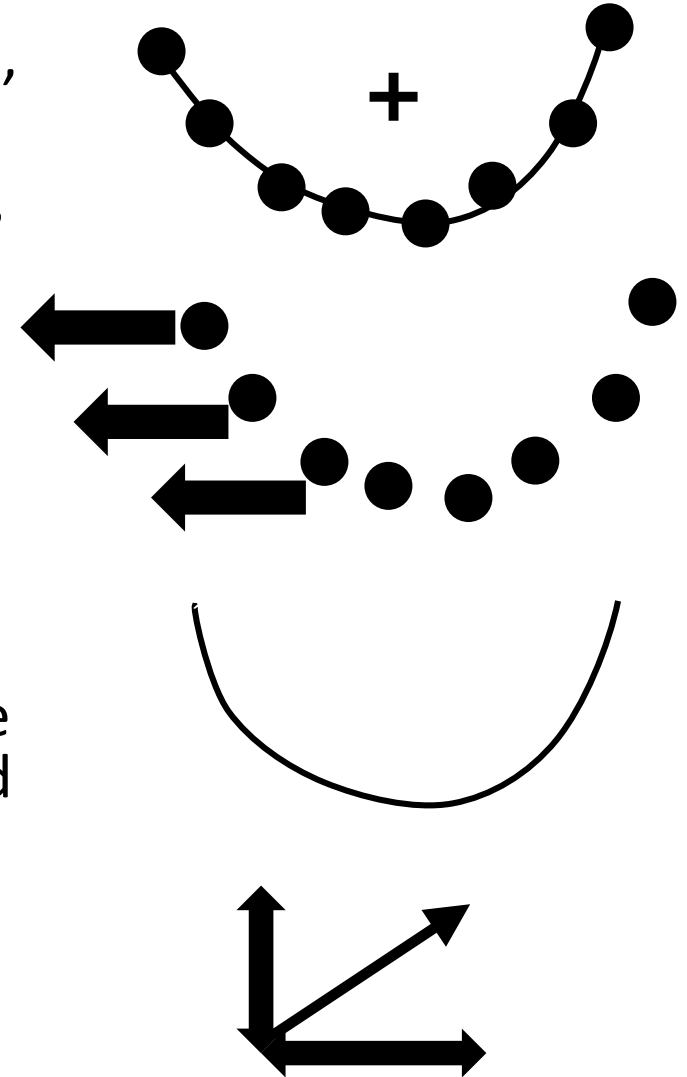
# Point Clicking for Interface Tracking

- Solid-liquid interface is tracked manually by clicking points along interface at different frames
  - Program developed by Gus Becker (thank you Gus!)
- Program outputs .csv file with x-y coordinates and frame numbers
- Conversion to physical distances and actual time stamps only requires a short unit conversion
- Finding the solidification velocity itself requires more data analysis



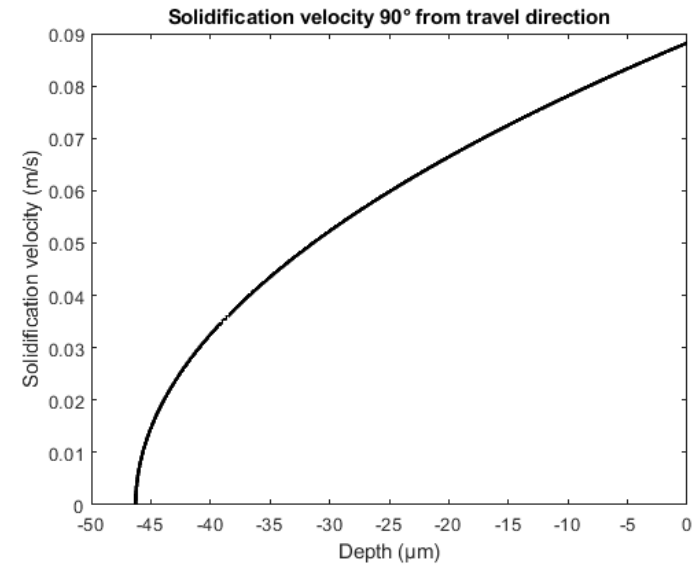
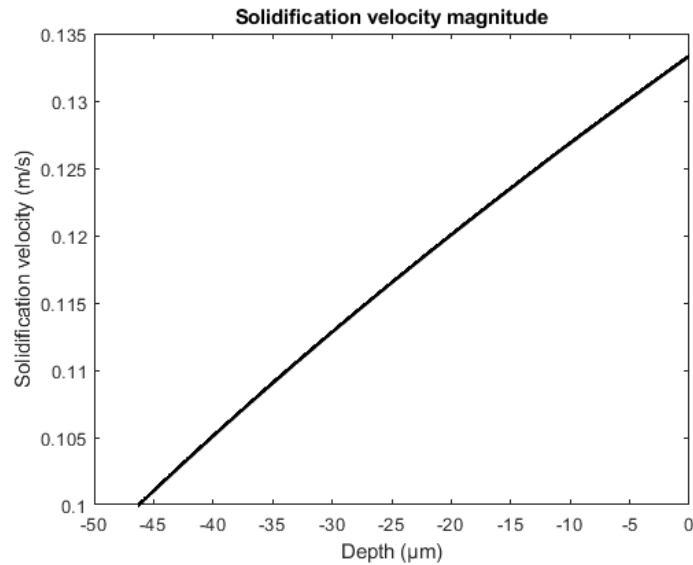
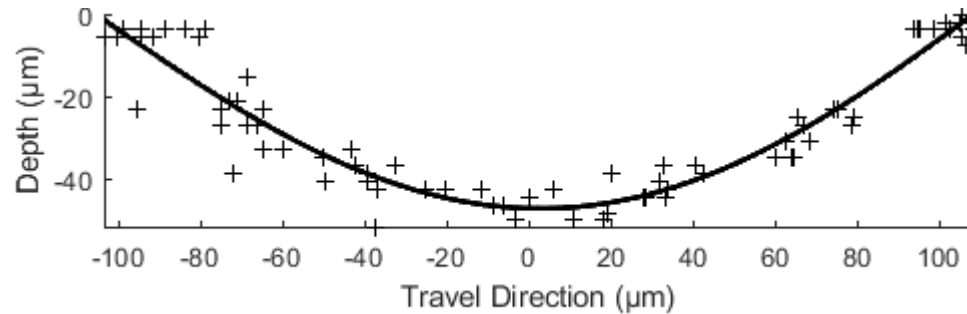
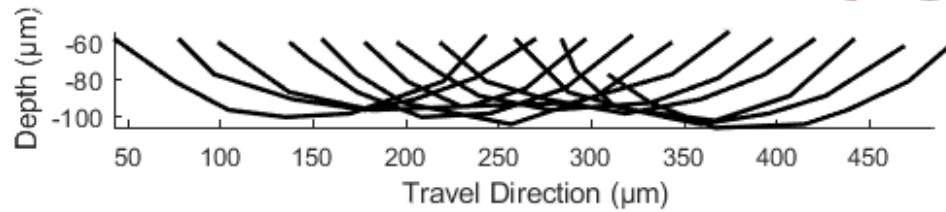
# Analysis of Point Clicking for Rasters

- Find a polynomial fit to each set of points, and use it to compute the centroid
- Translate all points so that their centroids are about  $x=0$
- Find a new fit including every data point
- Extract vectors for x and y points of polynomial fit to the left of the absolute minimum along depth
- Find the change in x and y between each set of points and adjust vector magnitude so change in x is equal to the raster speed
- If needed for single crystals, compute projection of result onto unit vector of given angle



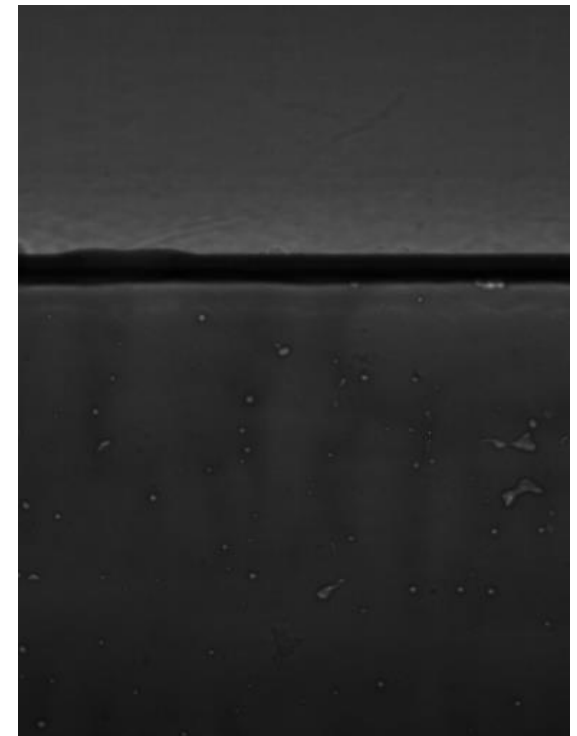
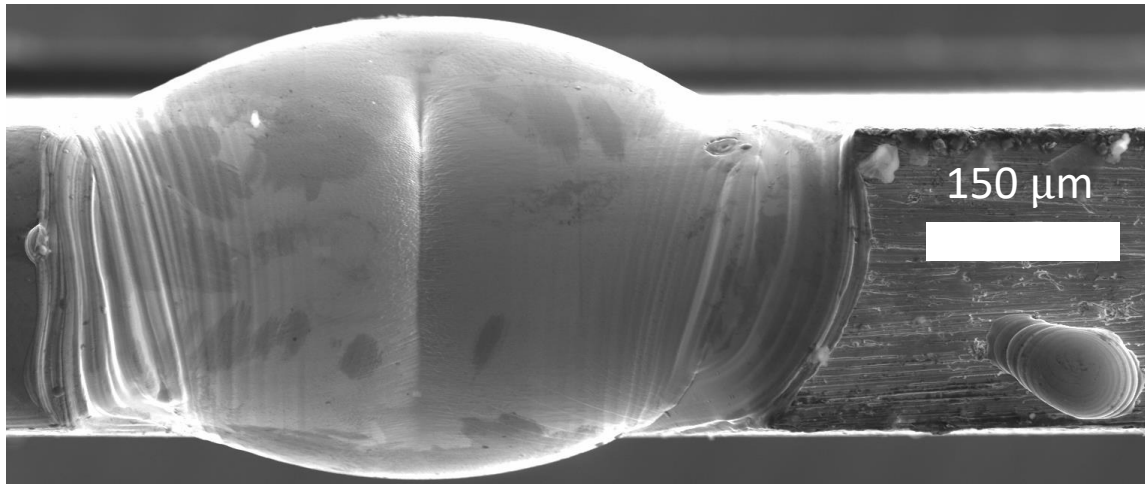


# Example Results



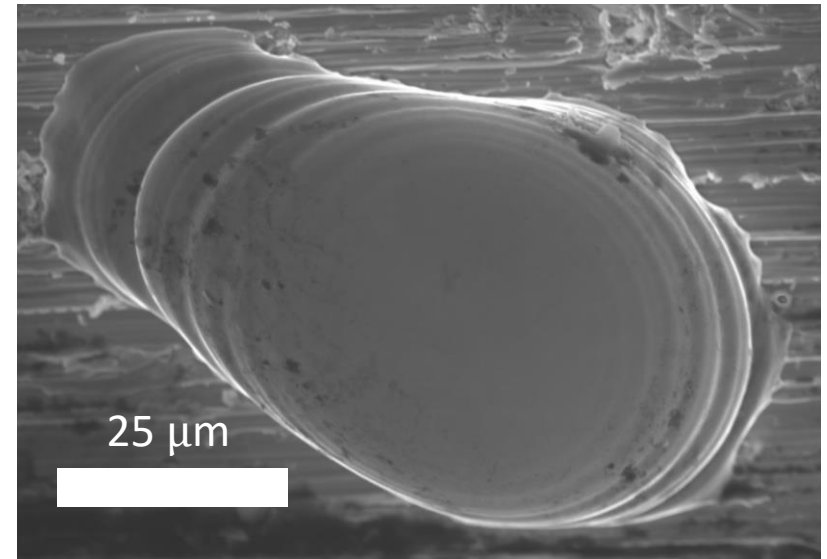
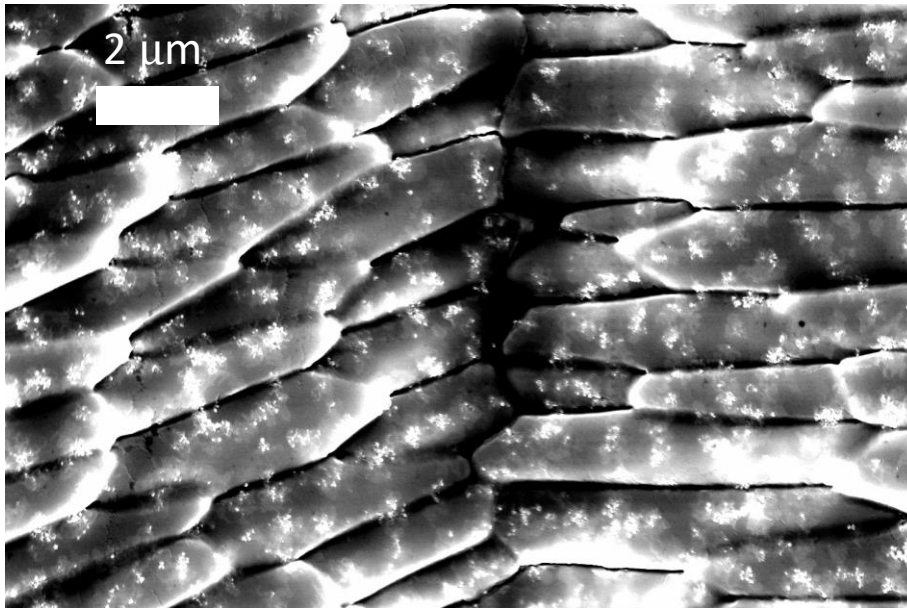
# Top Down Imaging

- Beginning of correlation between microstructural features and *in-situ* radiography
- For example, finding spatter that landed on the sample surface during keyholing



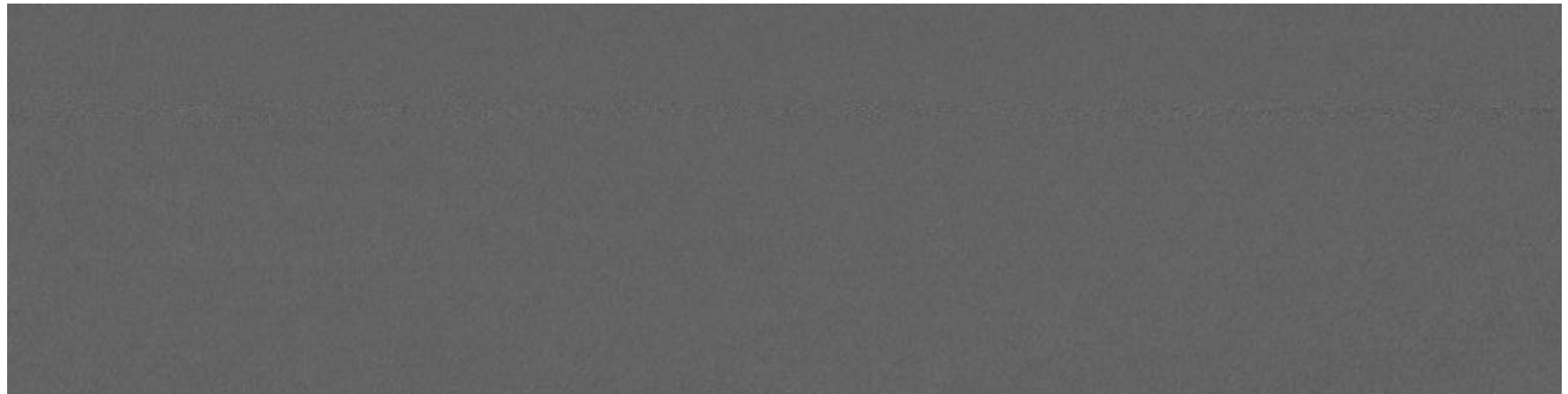
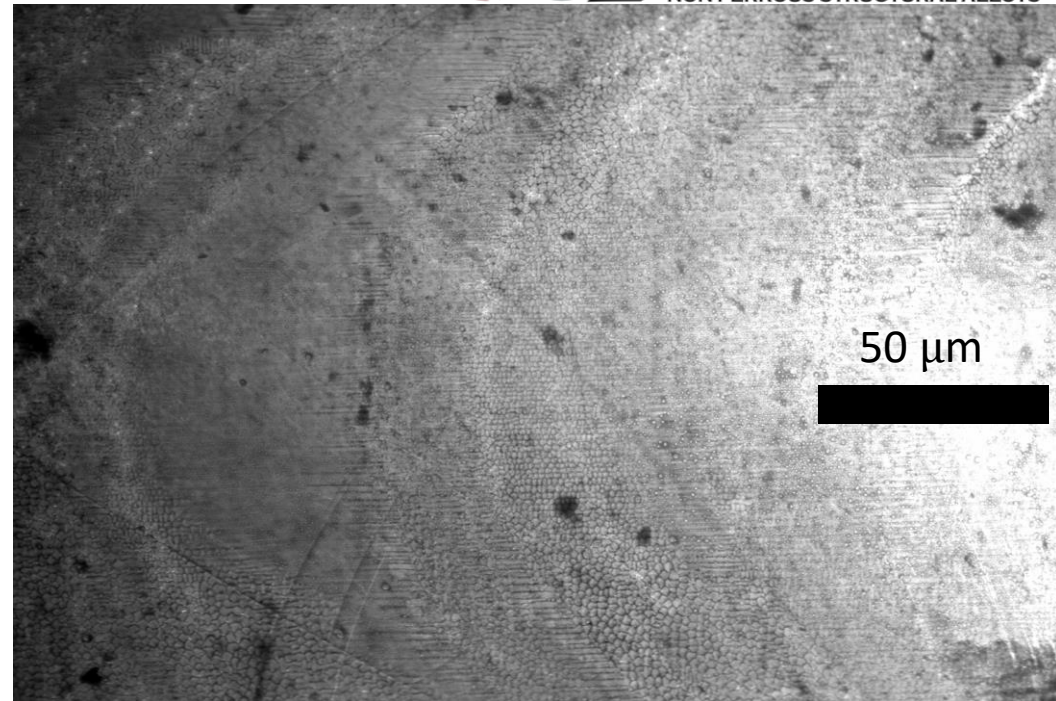
# Spatter Solidifies under Uncontrolled Conditions

- Dendrites indicate single crystal / highly textured material near centerline
- No such indication within spatter



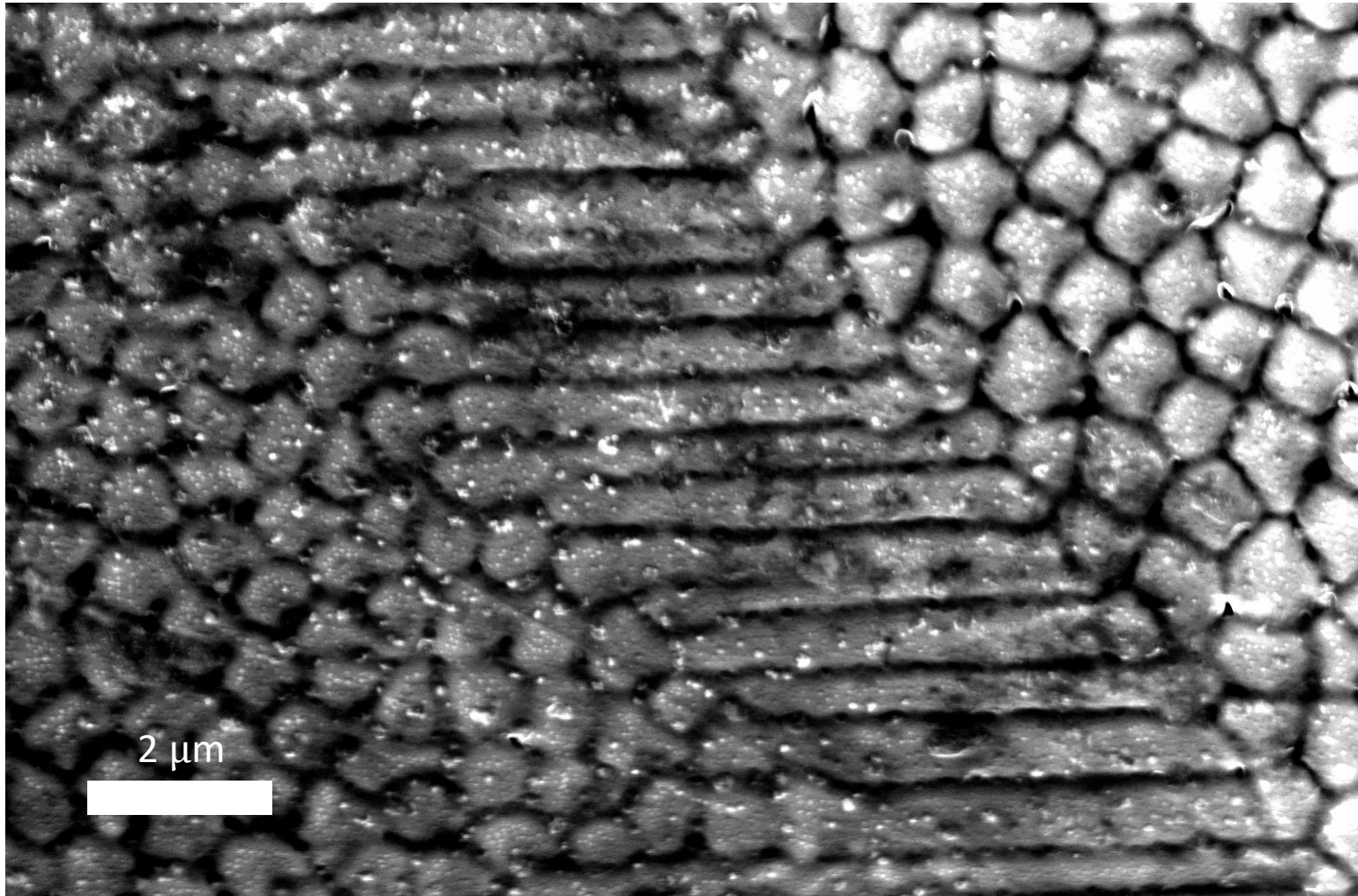
# Dendrites Switching Growth Direction during Rasters

- Sample has a  $\{100\}$  orientation with respect to the travel direction
- Remaining a 'single crystal' requires dendrite growth along  $\langle 100 \rangle$ , but multiple growth directions in  $\langle 100 \rangle$  can be active



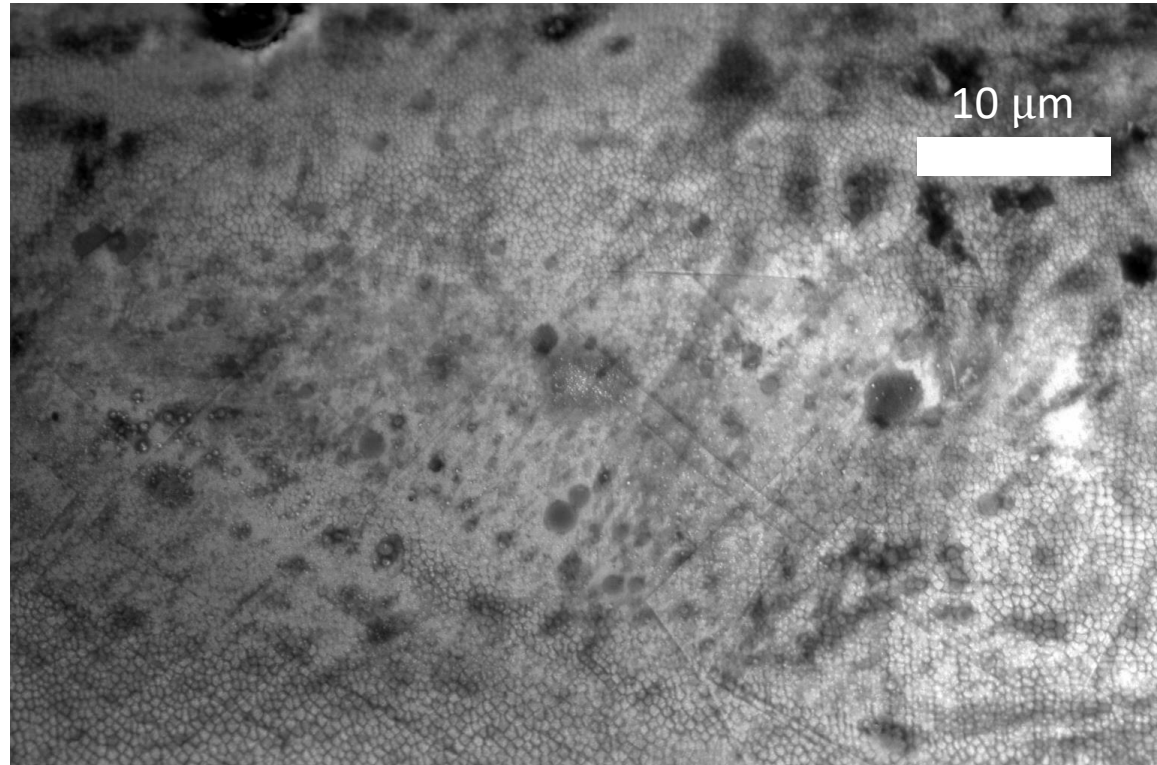


# Same Dendrites at Higher Magnification



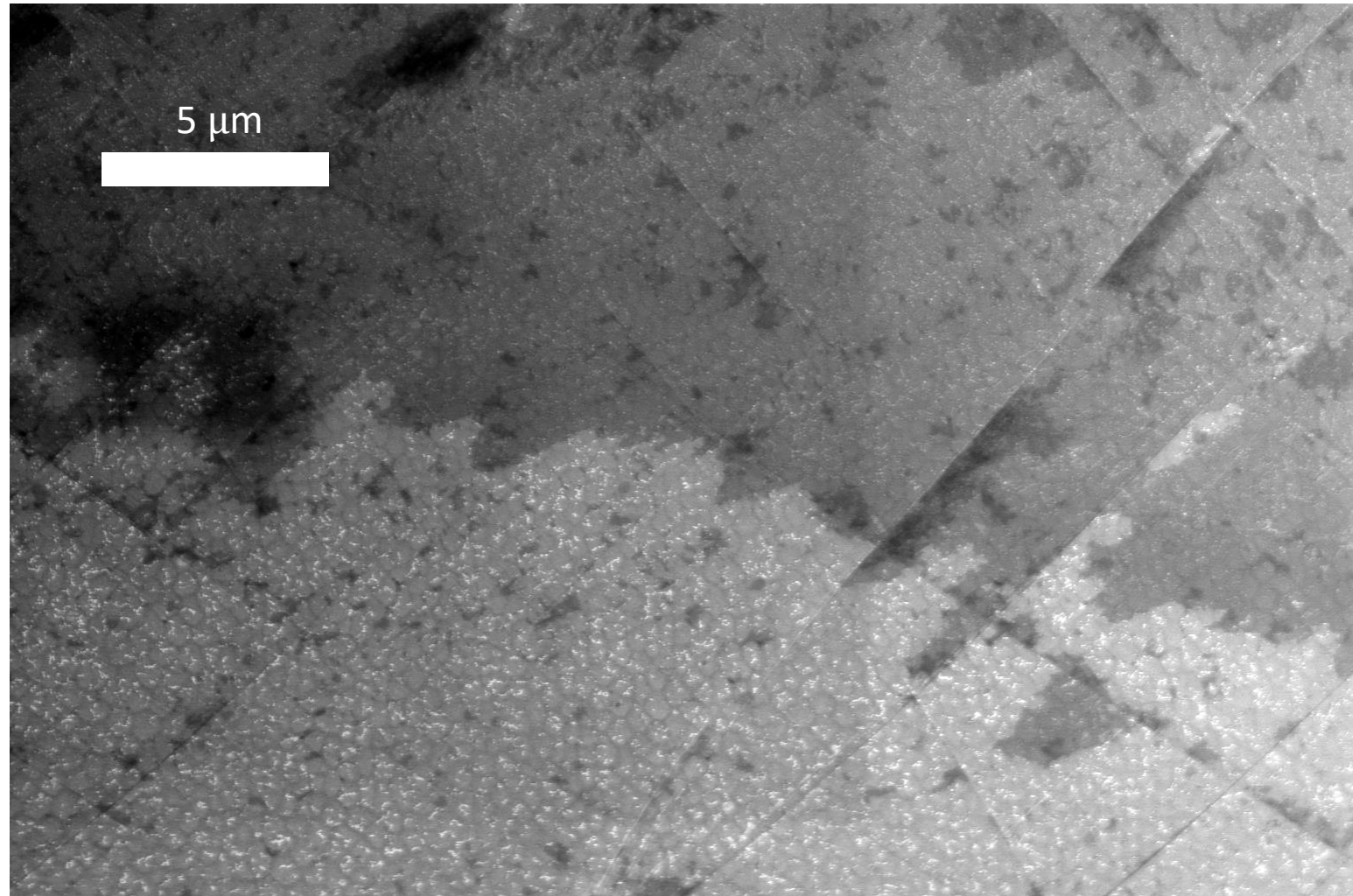
# High Speed Raster

- Still {100} orientation but dendrite orientation remains consistent



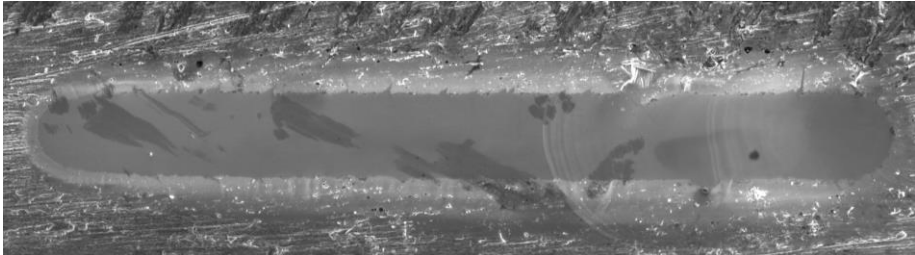


# Dendrite Growth Direction Consistent for Stray Grains at High Raster Speed

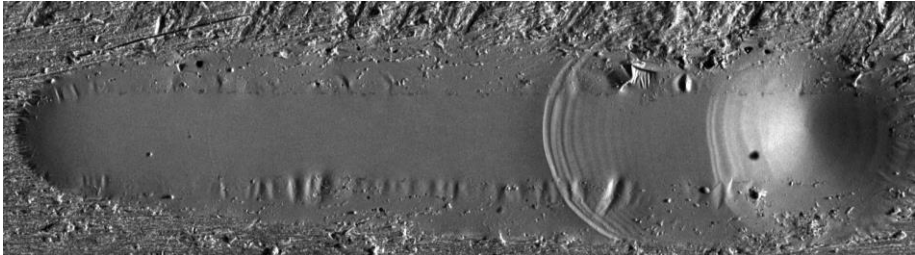


# Slow Raster in Al – Ag

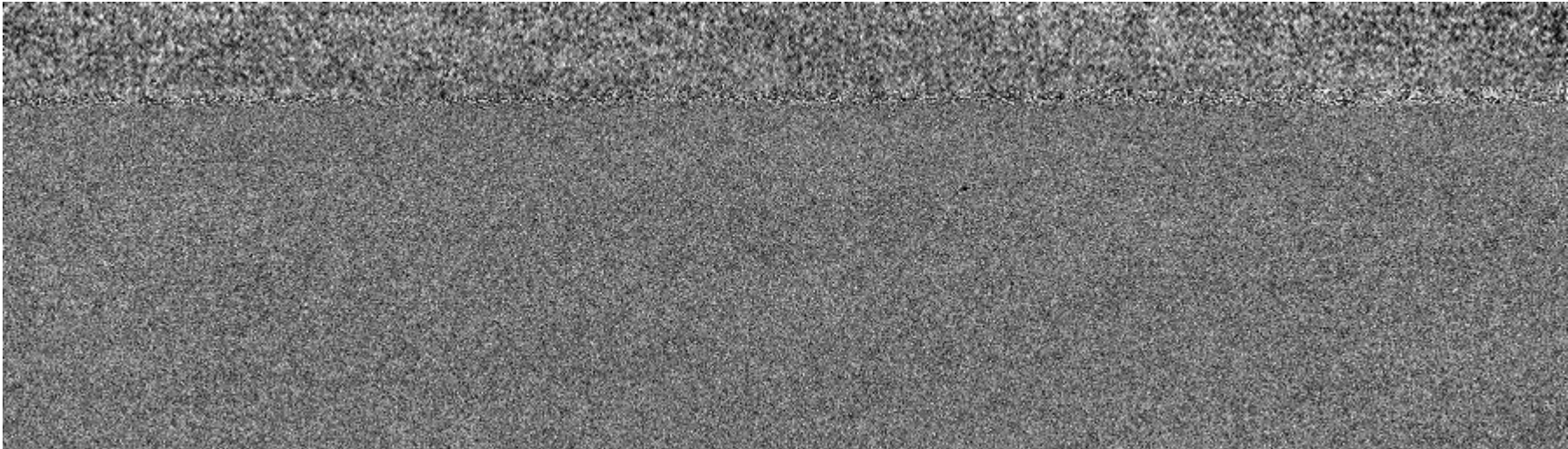
Secondary electron



Backscatter



250  $\mu\text{m}$





# Microstructure Differs between Steady State and End of Raster

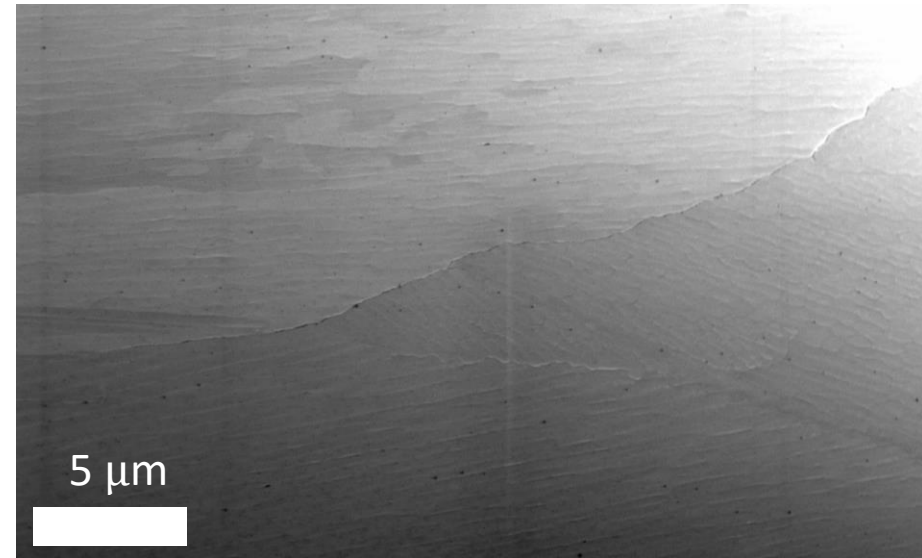


- Appears more seaweed like in steady state regime and columnar towards the end
- Microstructure at beginning difficult to see due to oxide layer

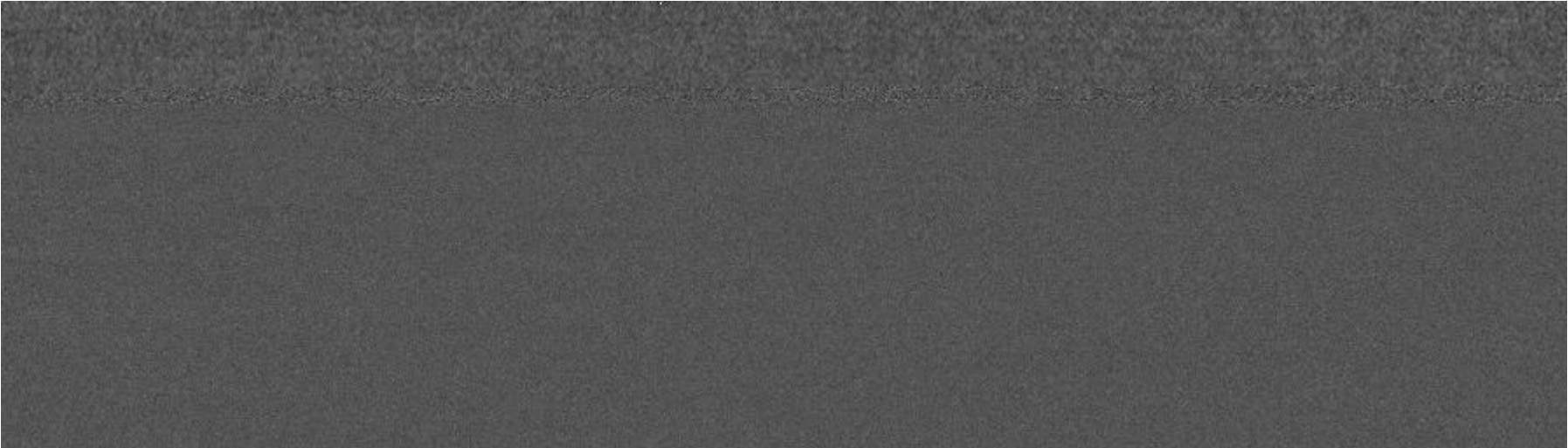
Steady state



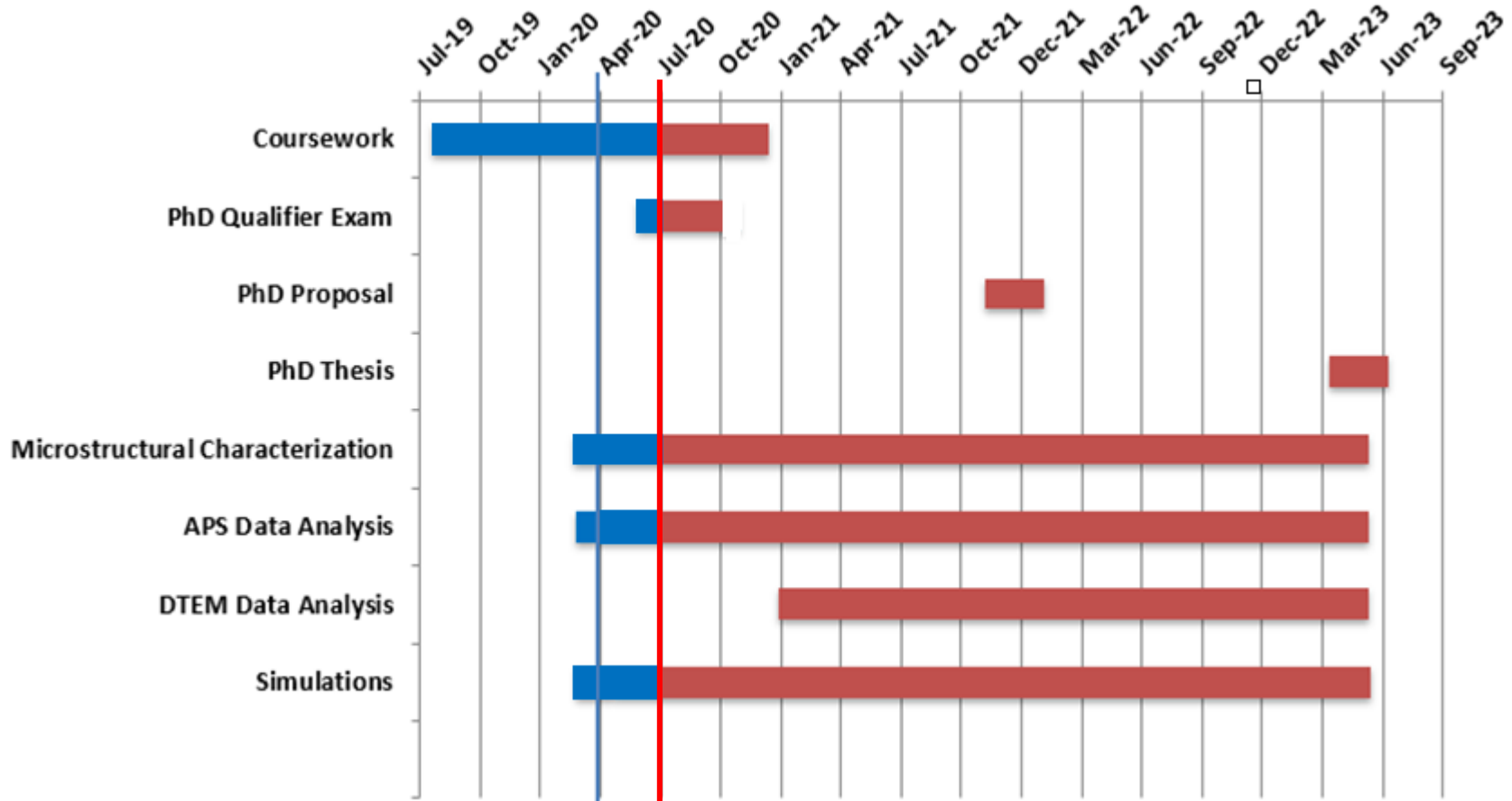
End



# Fast Raster in Al-Ag



# Progress



# Challenges & Opportunities



- Analysis of spot melt velocities
  - Extension of normal lines
  - Intercepts along a line at a given angle
- Imaging Al – Ag with oxide layer present
  - Cannot see microstructure underneath oxide layer in some areas
  - Might need a method to remove oxide without affecting metal
- FLOW-3D simulations
  - Learning software
- Qualifying exam

Thank you!  
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