

# ***Project 33B-L: In-Situ Studies of Strain Rate Effects on Phase Transformation and Microstructural Evolution in Multi-Principal Element Alloys***

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Colorado School of Mines, Golden, CO  
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*Other Participants: Francisco Coury (UFSCAR), Jonah Klemm-Toole (Mines)*



# Project 33B-L: In-Situ Studies of Strain Rate Effects on Phase Transformation and Microstructural Evolution in Multi-Principal Element Alloys



- Student: John Copley (Mines)
- Advisor(s): Amy Clarke (Mines)

**Project Duration**  
MS: September 2018 to Summer 2020

- **Problem:** The effects of strain rate and state and temperature on the TRIP/TWIP behavior exhibited by MPEAs are not well understood.
- **Objective:** Determine the relationship between alloying, strain rate and strain state effects on the evolution of deformation twins and deformation induced phase changes.
- **Benefit:** Improved understanding of TRIP/TWIP behavior seen in other materials, alloy design for specific applications, especially blast resistance.

- Recent Progress**
- Material production and characterization of effects of processing (rolling, heat treatment) on grain size and phase.
  - Determination of methods for better sample preparation for in-situ testing at the Advanced Photon Source

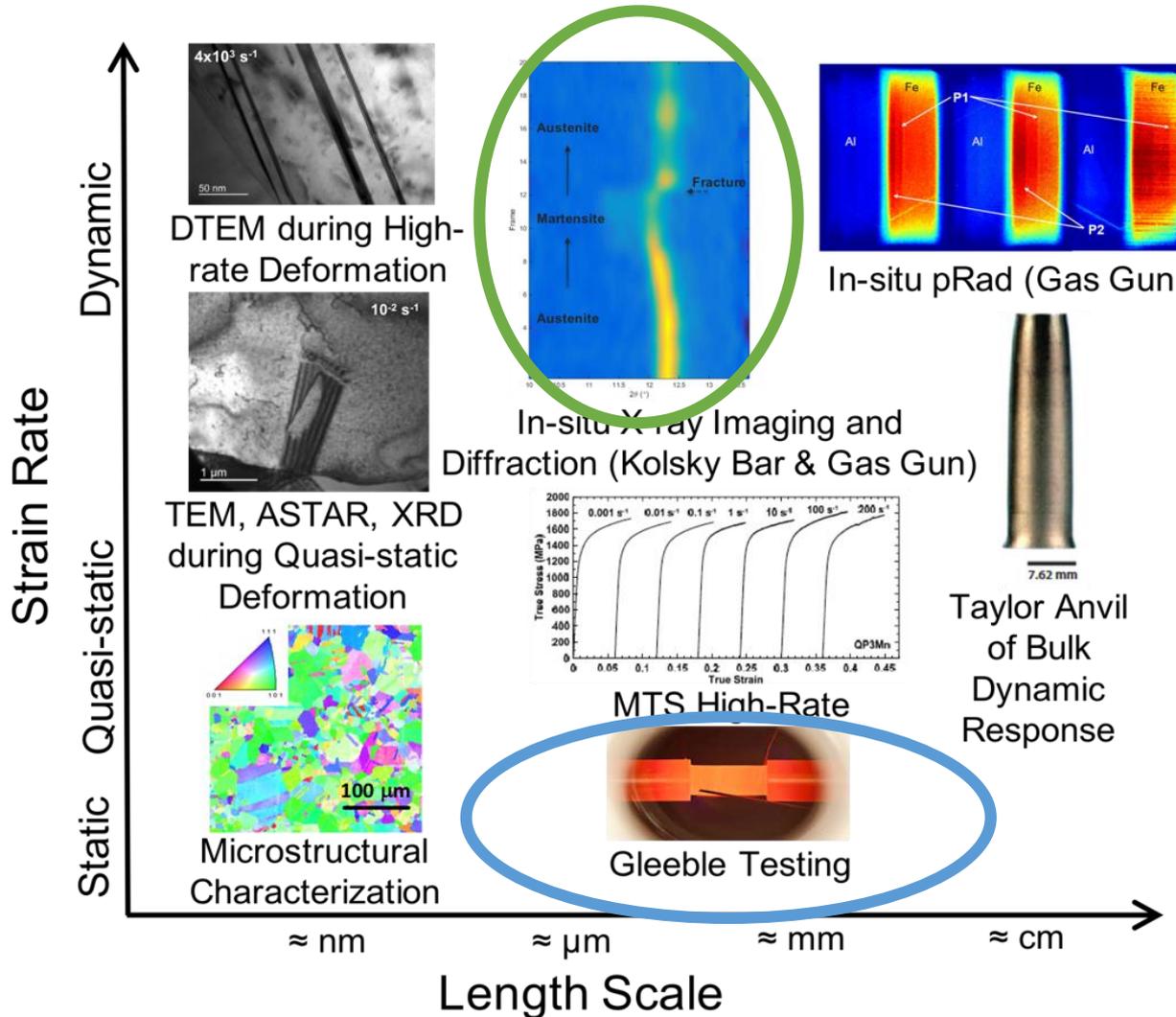
Metrics		
Description	% Complete	Status
1. Literature review	75%	●
2. Quasi-Static Testing	30%	●
3. Dynamic Testing	25%	●
4. Multi-scale in-situ imaging and diffraction	35%	●

# Industrial Relevance

- Understanding of TRIP/TWIP of MPEAs during high rate deformation
  - New strategies to design deformation mechanisms
  - Drive development of alloys for blast-resistance and performance in extreme environments
- Fundamental understanding of TRIP/TWIP
  - Applications to more commonly used Advanced High Strength Steels and some Ti alloys



# Project Vision

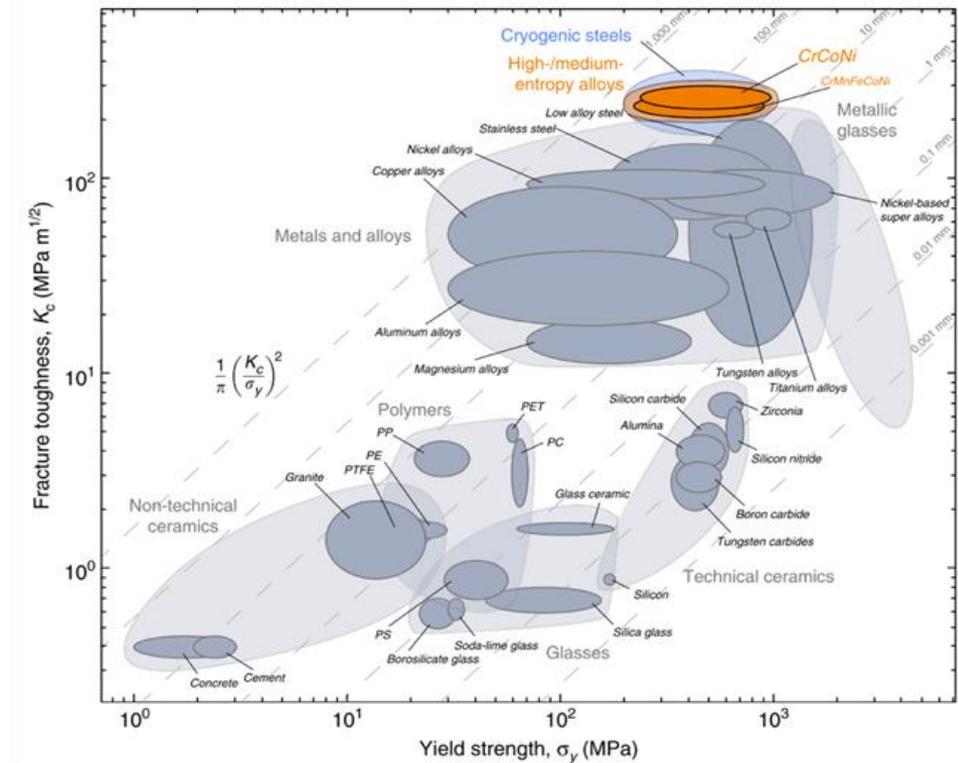


State-of-the-art, multi-scale microstructural characterization with electrons, x-rays, and protons of TRIP/TWIP in MPEAs for blast resistance

Figure courtesy of Dr. Amy Clarke

# MPEAs

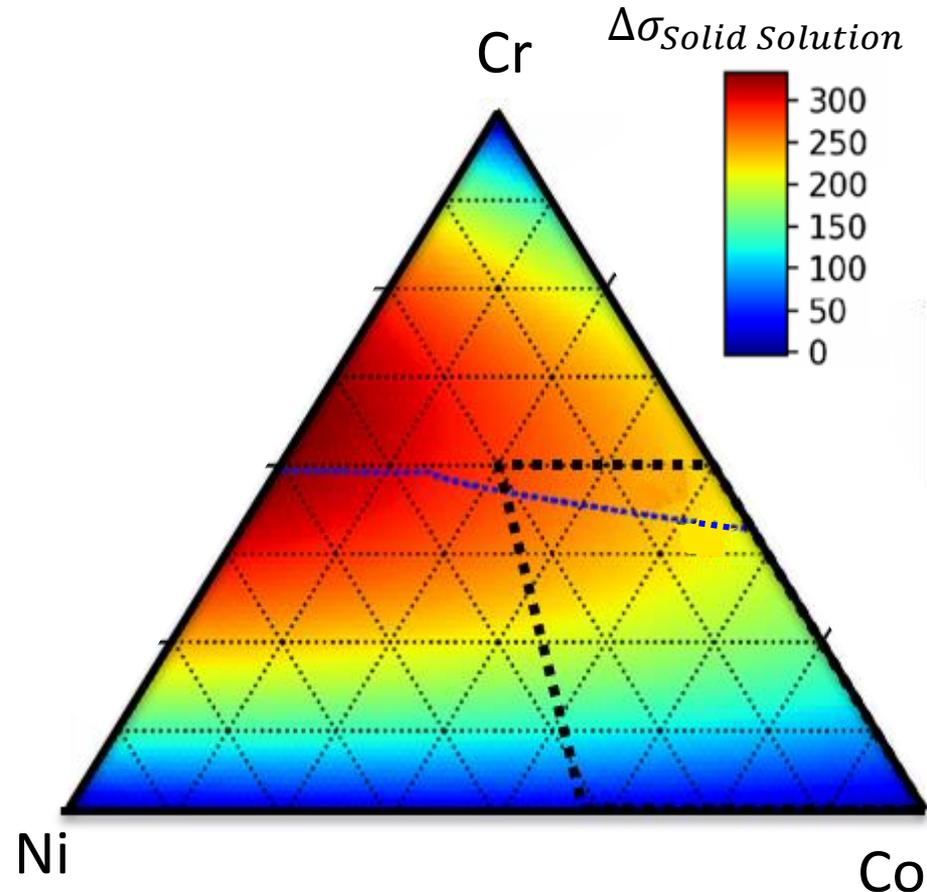
- No definable main element
  - Equiatomic, or
  - Several (>2) components present in very high concentrations
- Almost infinite combinations
- MPEA vs HEA
  - Broader definition than HEAs
    - Strength and toughness do not scale with entropy
  - CoCrNi Family
    - Fails HEA criteria
    - Toughest known CCAs



B. Gludovatz, et al., Nature Communications, 2016, 7:10602

# MPEAs and Solid Solution Strengthening

- Initial HEA/MPEA studies:
  - Equiatomic compositions
  - Solid solid solution strengthening
- Effective Atomic Radii for Strengthening (EARS)
  - Atoms in solution do not have the radii conventionally used to predict strength
  - EARS radii act as better strength predictors
  - Shows that optimal properties are not correlated with maximum entropy



# Twinning and Transformation Induced Plasticity

- Deformation accommodated by change in local atomic stacking
- Increased work hardening rates
  - Burgers vectors are not conserved at twin or phase interfaces
  - The “Dynamic” Hall-Petch Effect
  - High work hardening rates delay instability
- Delayed Instability
  - Increased UTS, elongation
  - Improved toughness

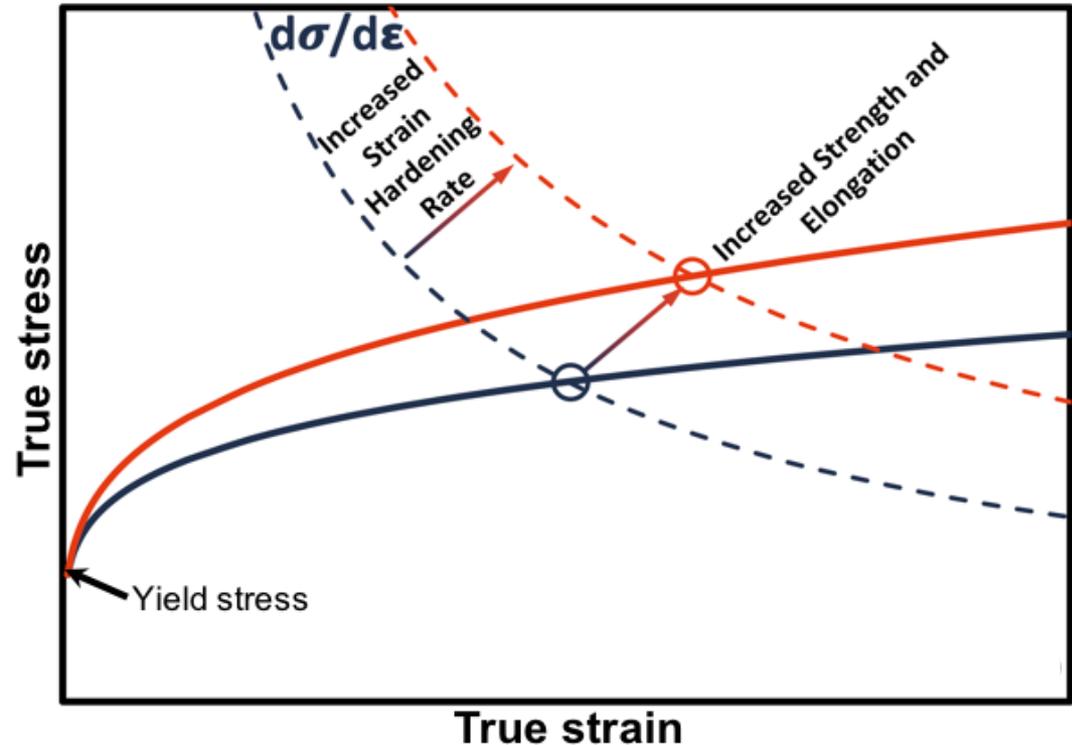
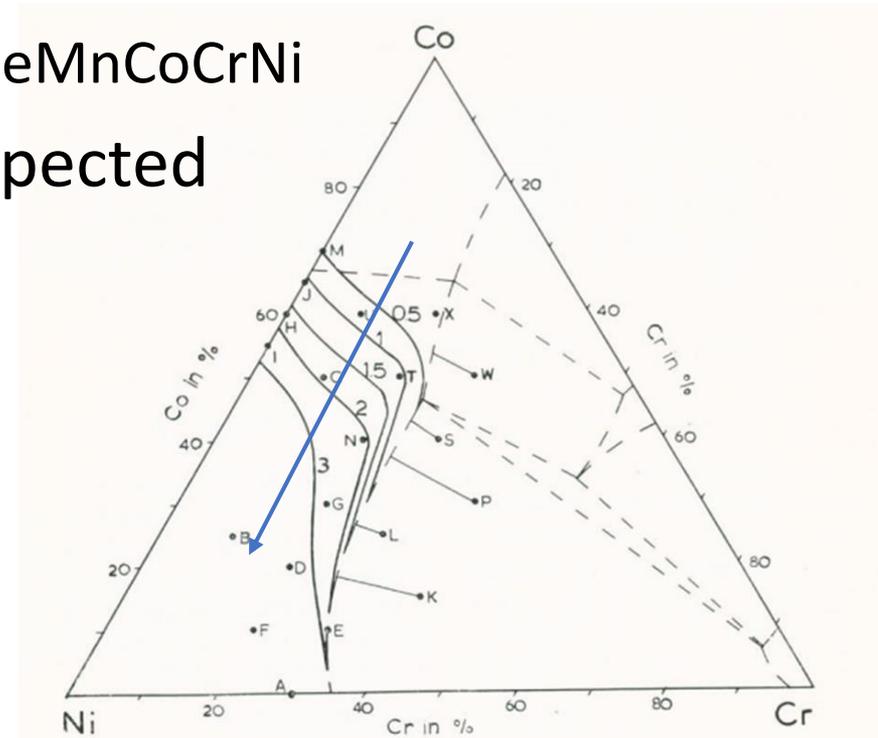
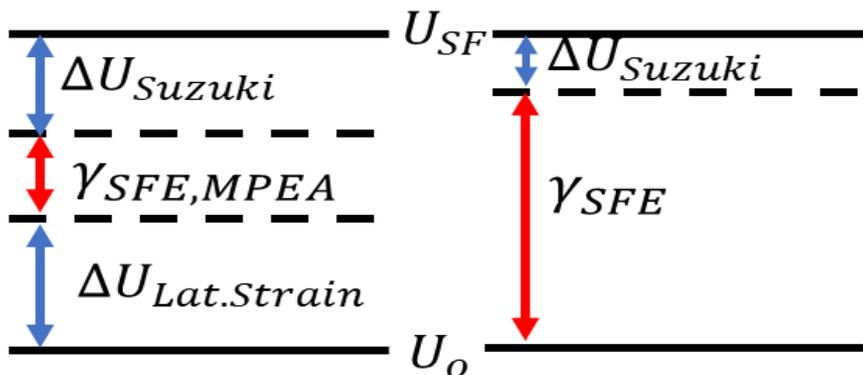


Figure courtesy of Dr. Kester Clarke

# Why are MPEAs Good Candidates for TRIP/TWIP

- Shown to occur in some MPEAs
  - CoCrNi, FeCoCrNi, FeMnCoCr, FeMnCoCrNi
- High occurrence of twins is expected (low SFE)
  - Suzuki Interaction
  - Lattice Distortion

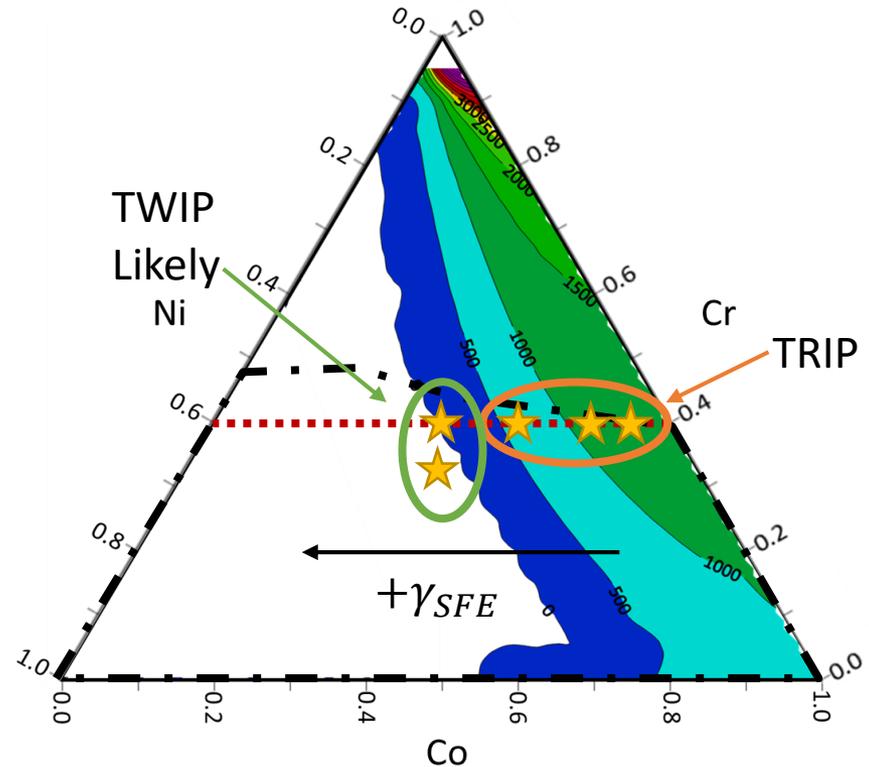


Isopleths in  $\gamma_{SFE}$  for CoCrNi Ternary Ranging from  $9 \times 10^{-7} - 5.6 \times 10^{-6} \text{ J/cm}^2$  (0.5-3Gb)

From: E.H. Koster et al. Stacking Fault Energies of Ni-Co-Cr Alloys, The Philosophical Magazine, Series 8, Vol 10 (1964)

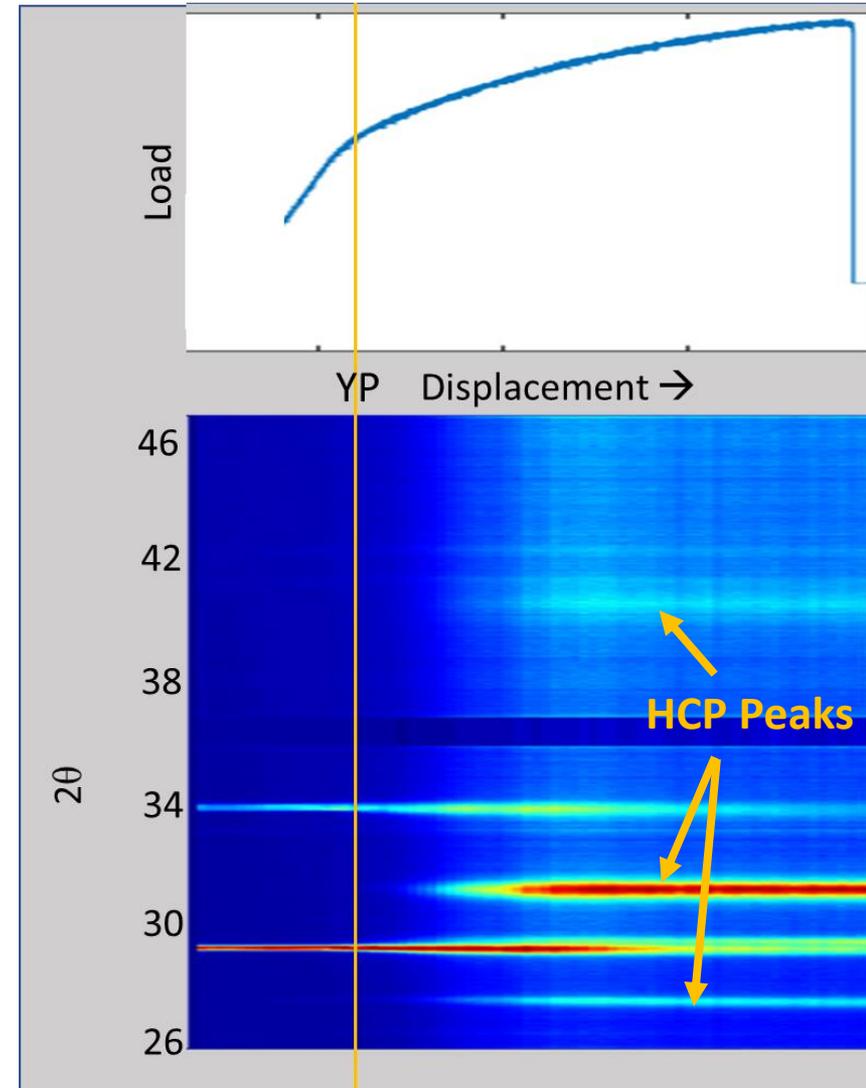
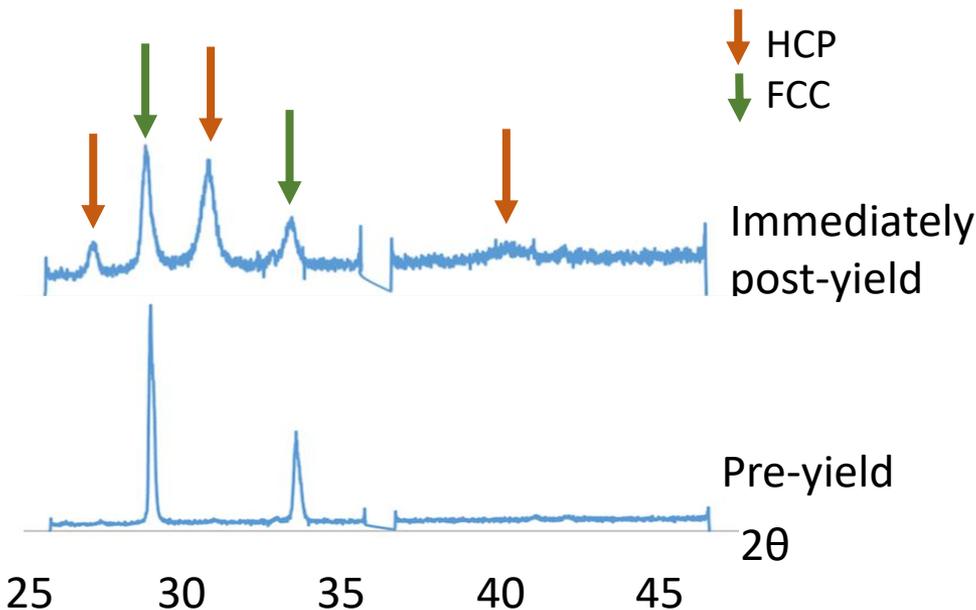
# Alloy Design for TRIP and TWIP

- $T_0$ : The diffusionless transformation temperature
  - Free energy and composition of HCP and FCC phases are the same
  - High  $T_0$  indicates the possibility of TRIP behavior at elevated temperatures
- Alloy Selection
  - High chromium alloys for higher  $T_0$  values → possibility of TRIP at high temperatures
    - Constrained by FCC single phase region (at  $1100^\circ\text{C}$ , black dashed line)
      - Surrounded by  $\sigma$ -phase
  - Alloys for study have been selected along  $\text{Co}_x\text{Cr}_{40}\text{Ni}_{60-x}$  (dotted red line)
    - Values of  $X=0.55, 0.50, 0.40, 0.30$



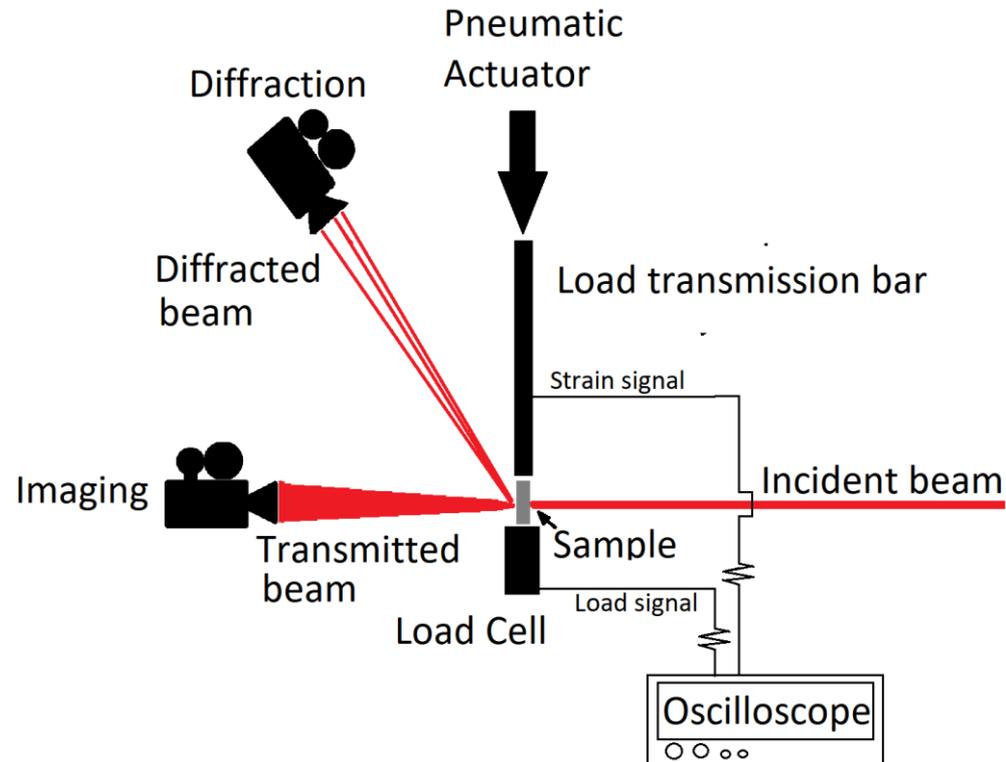
# Project Review: Evidence of TRIP in $\text{Co}_{55}\text{Cr}_{40}\text{Ni}_5$ at $60^\circ\text{C}$

- Low Temperature Deformation ( $\text{Co}_{55}\text{Cr}_{40}\text{Ni}_5$ ):
  - Transformation from FCC  $\rightarrow$  HCP as strain increases
  - Onset of transformation at or near yielding
  - Similar behavior from  $-100^\circ\text{C}$  to  $450^\circ\text{C}$



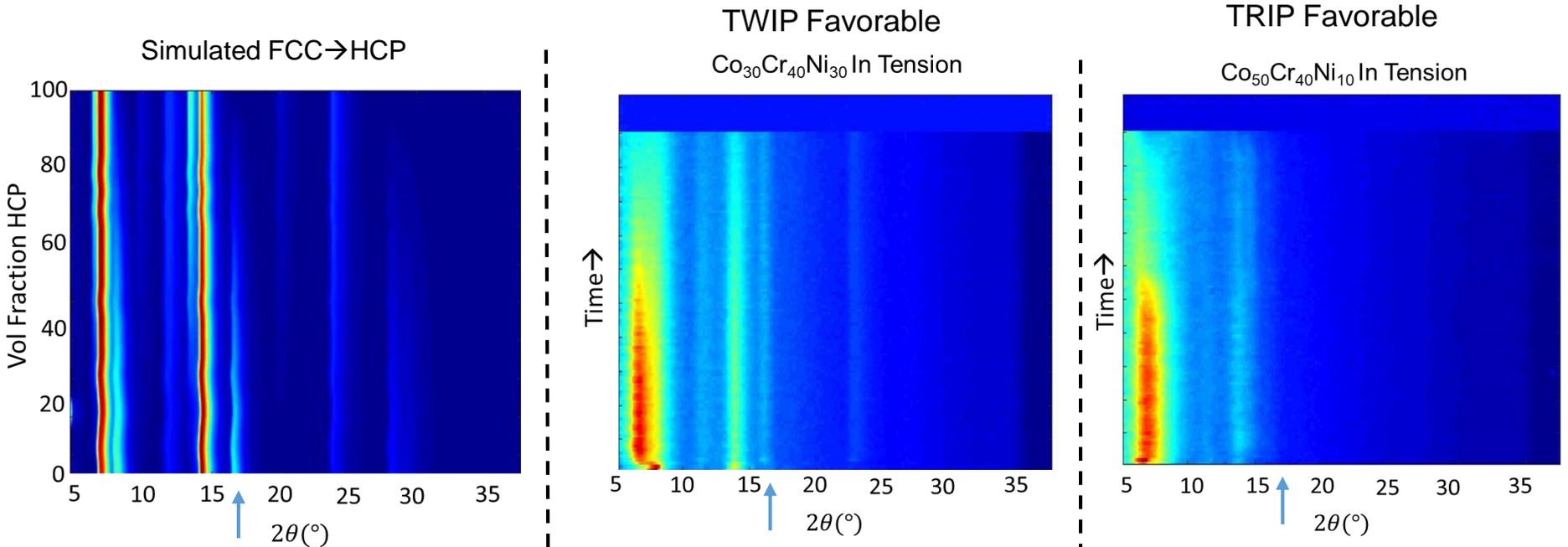
# In-situ Dynamic Testing at APS

- In-situ Kolsky Bar, X-ray imaging and diffraction experiments
  - $\dot{\epsilon} \sim 10^3 \text{ s}^{-1}$
- Alloys tested:
  - $\text{Co}_{50}\text{Cr}_{40}\text{Ni}_{10}$
  - $\text{Co}_{40}\text{Cr}_{40}\text{Ni}_{20}$
  - $\text{Co}_{30}\text{Cr}_{40}\text{Ni}_{30}$
  - $\text{Co}_{33.3}\text{Cr}_{33.3}\text{Ni}_{33.3}$
- Loading in both tension and compression



Thanks to Ben Ellyson for the figure

# Project Review: APS Simulation vs Experimental Results



- No apparent change in FCC (200) 1<sup>st</sup> Harmonic peak intensity
  - Expected for Co30
  - Peak is difficult to discern in Co50, but transformation may have occurred
    - High intensity from 2<sup>nd</sup> harmonic washes out 1<sup>st</sup> harmonic peaks

# Project Review: Sample Geometry Effects

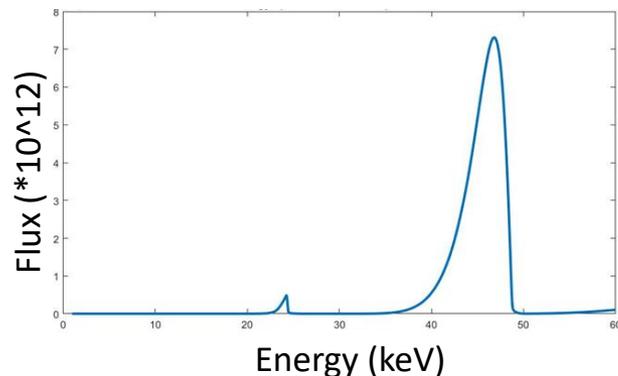


Thickness

Modified Energy Spectrum

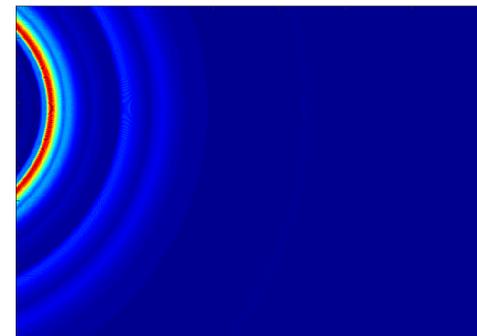
Ratio of  $I_{1st}:I_{2nd}$

500 $\mu m$

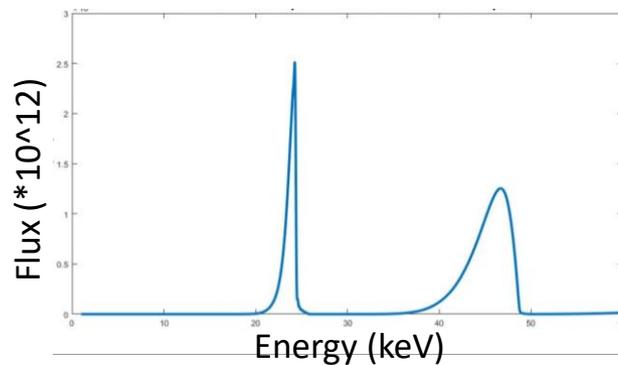


1:10

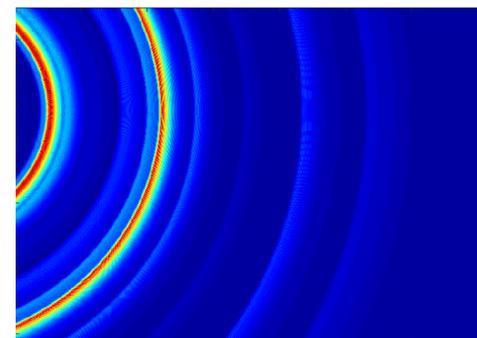
Tilting of the sample improved ratio by having some thinner sections



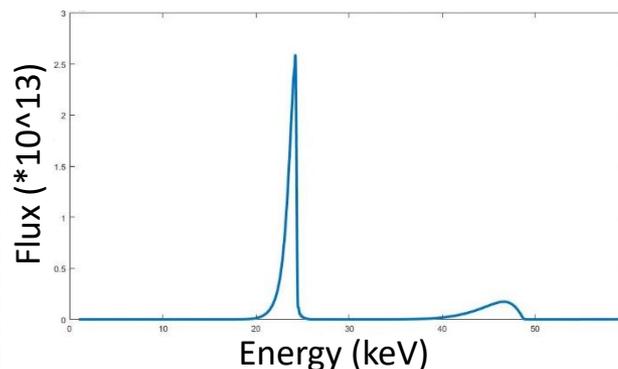
250 $\mu m$



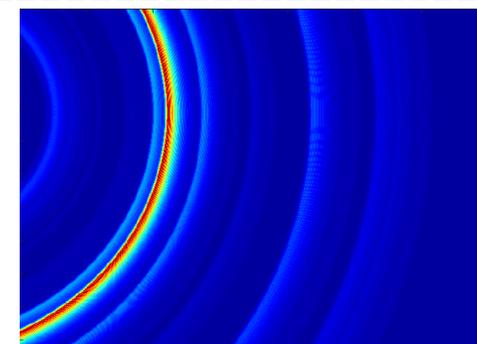
2:1



100 $\mu m$

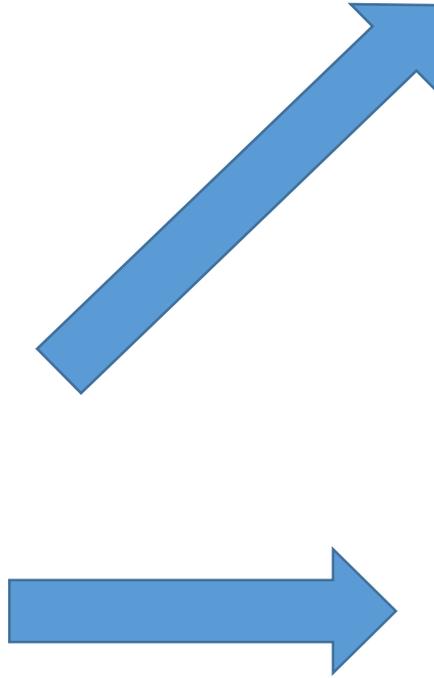
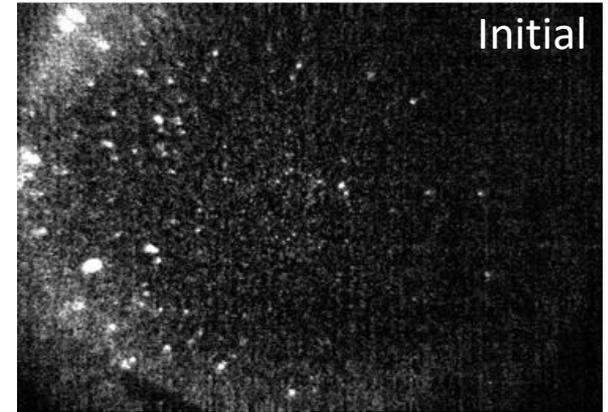


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# Project Review: Twinning Seen by Domain Refinement

- Evolution of full rings
  - Crystallite refinement  $\rightarrow$  twinning  $\rightarrow$  TWIP
  - Seen in all alloys and strain rates tested at APS

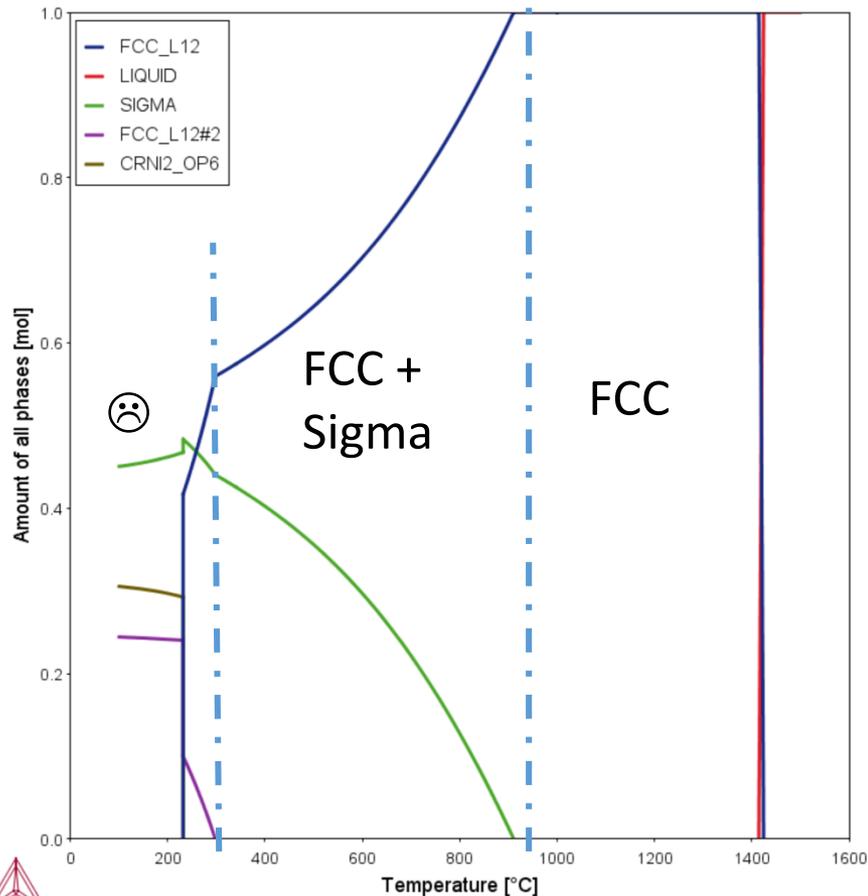


# Recent Work

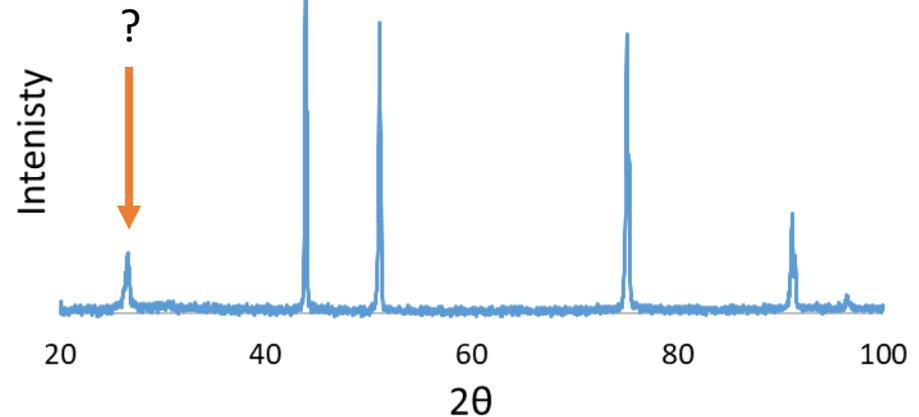
- Making material → making material into samples
  - Cast material (from Dr. Coury Co55, Co40)
  - Arc melted material (all other alloys)
    - Needs to be rolled
    - 50g limit
- Recent Findings
  - $\text{Cr}_2\text{O}_3$  particles in cast material
  - Multi-phase microstructures in low Co content alloys



# Multi-Phase Buttons ☹️

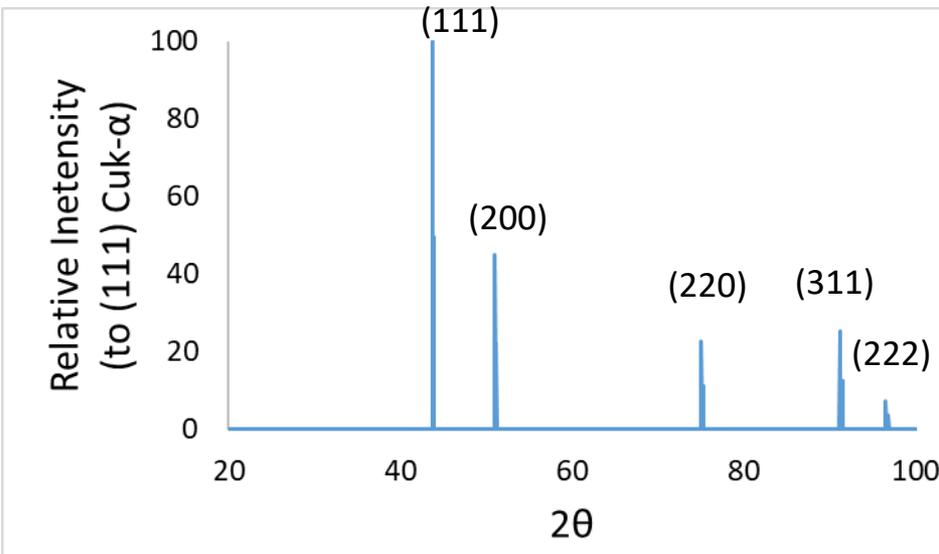
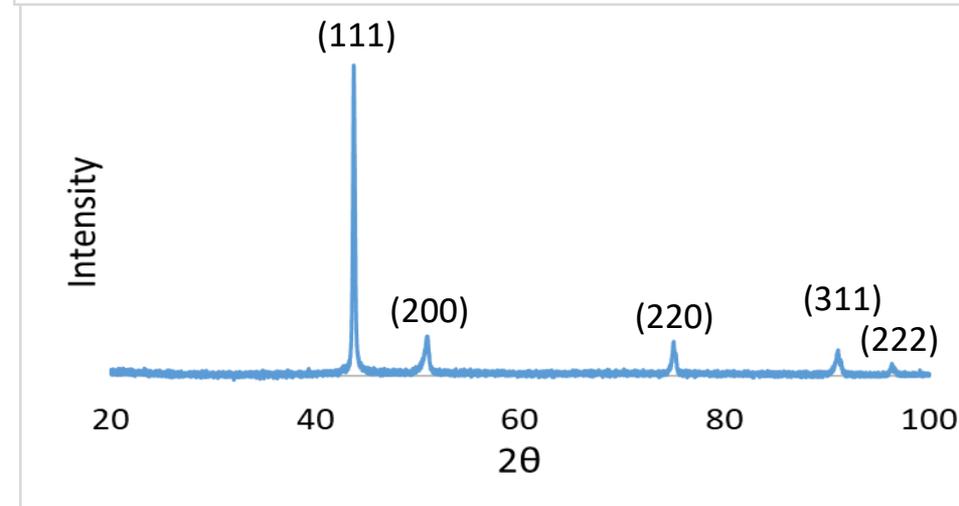
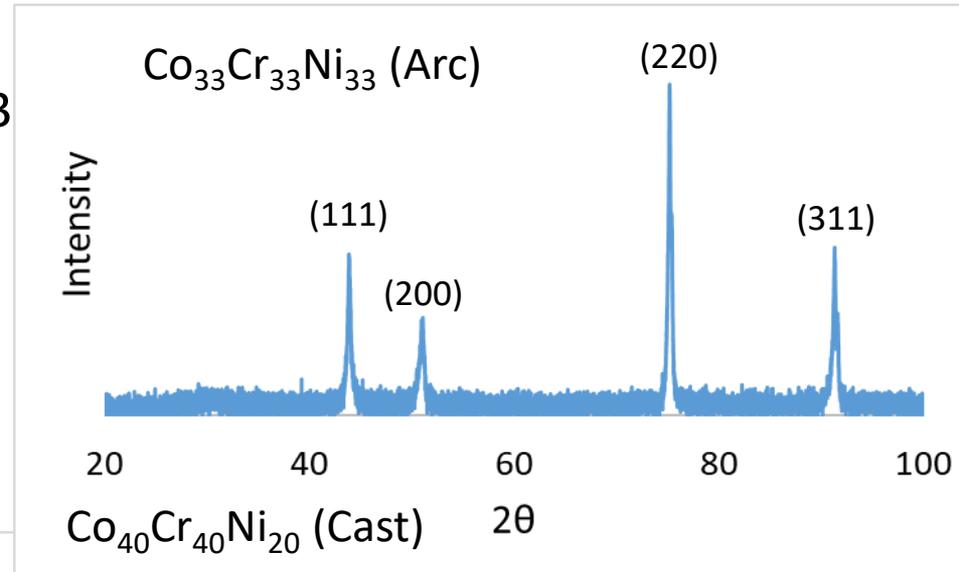


- Preparation:
  - Cold rolling to 75%
  - Furnace 1150°C for 2 hours
    - Needed to prevent  $\sigma$ -phase
  - Polish → XRD
- Observed for Co30, CoCrNi
- Buttons did not fully transform back to FCC



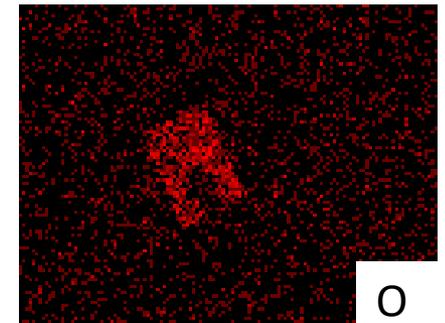
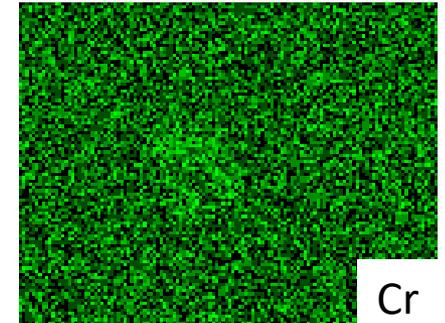
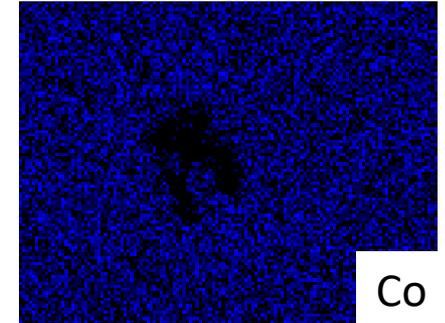
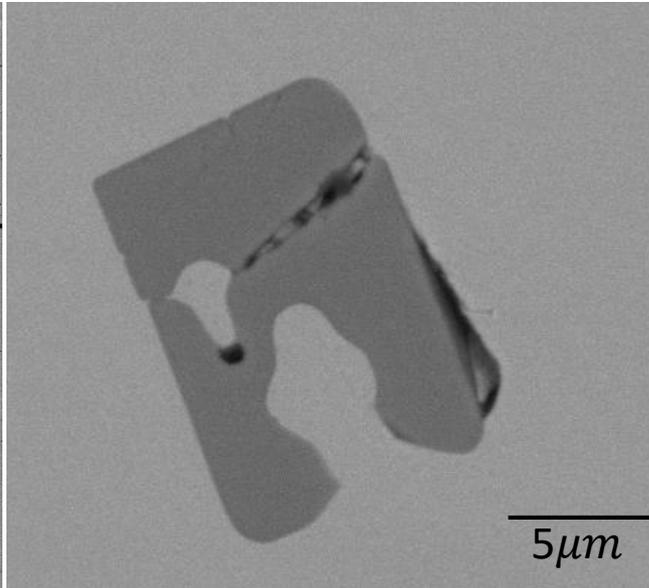
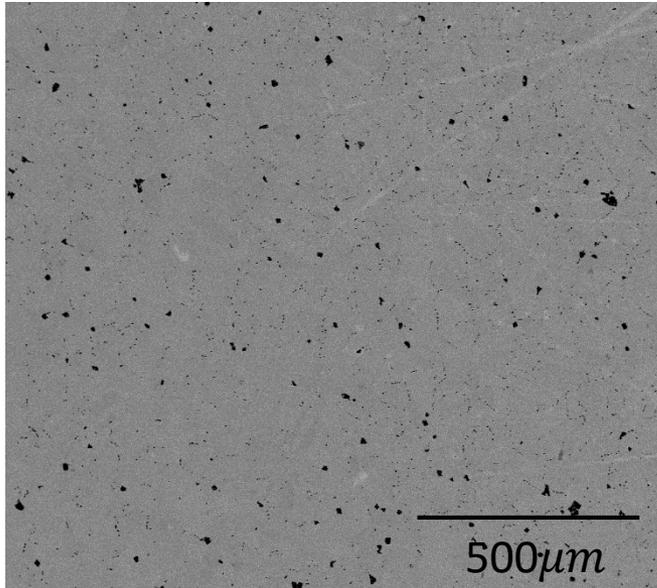
# Single Phase Achieved

- 2 hours @ 1200°C
  - Additional heat treatment for Co33
  - Initial heat treat for Co40
- Both appear textured
  - Co33 → Rolling ≈ 70%
  - Co40,55 → Rolled only 20%, but as cast



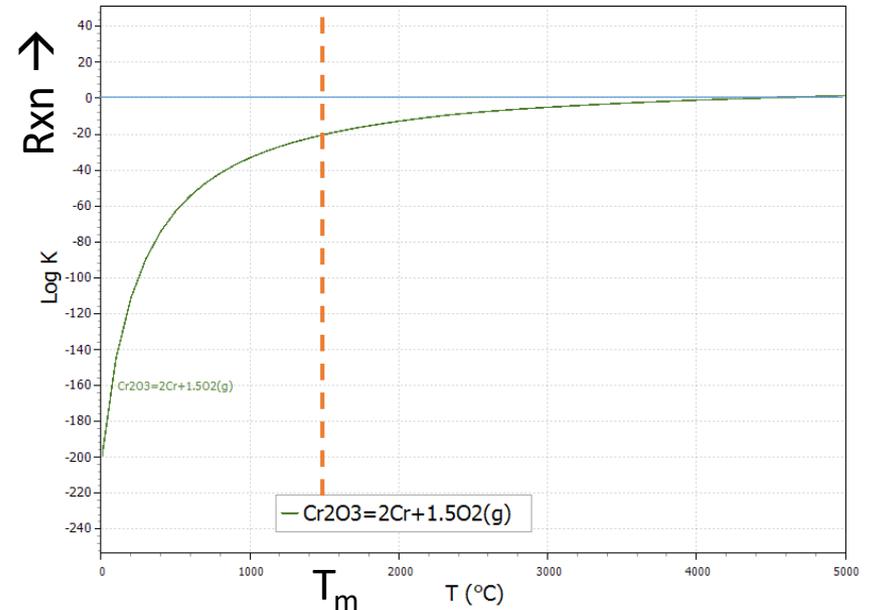
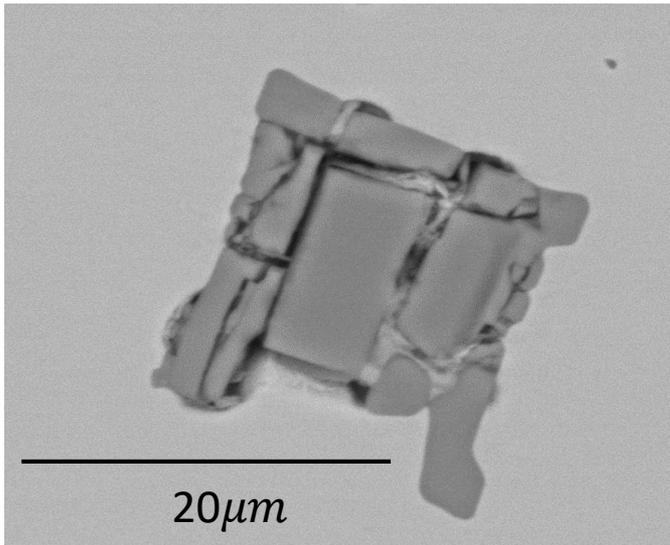
# Chromium Oxides in Cast Material

- Co40 and Co55 ingots have Cr rich oxides
  - Potentially detrimental to mechanical properties
    - Reduced rolling tolerance from 75% to 25%



# Removing Oxides?

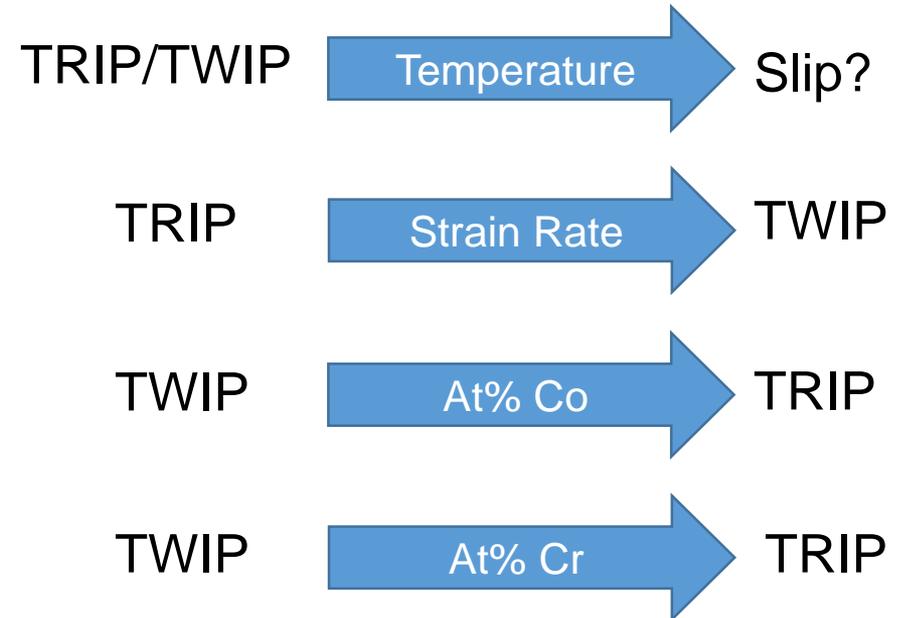
- $2\text{Cr} + 3/2\text{O}_2 \rightarrow \text{Cr}_2\text{O}_3$ 
  - Favorable beyond melting point of alloy—in air
  - Cannot be dissolved
- May be able to re-melt in vacuum to drive reaction leftwards



Plot of reaction constant vs temperature for the dissociation of  $\text{Cr}_2\text{O}_3$  (green). The blue line shows the point at which the reaction becomes favorable, the orange dashed line shows the approximate melting point of the alloy.

# TRIP and TWIP Observed in CoCrNi

- TRIP behavior seen:
  - At low strain rates ( $10^{-2} s^{-1}$ )
  - Temperatures from  $-100^{\circ}C$  to  $450^{\circ}C$
  - In  $Co_{55}Cr_{40}Ni_5$ , predicted to TRIP by Thermo-Calc modelling
- TWIP behavior seen:
  - At high strain rates ( $10^3 s^{-1}$ )
  - Across a range of CoCrNi alloys



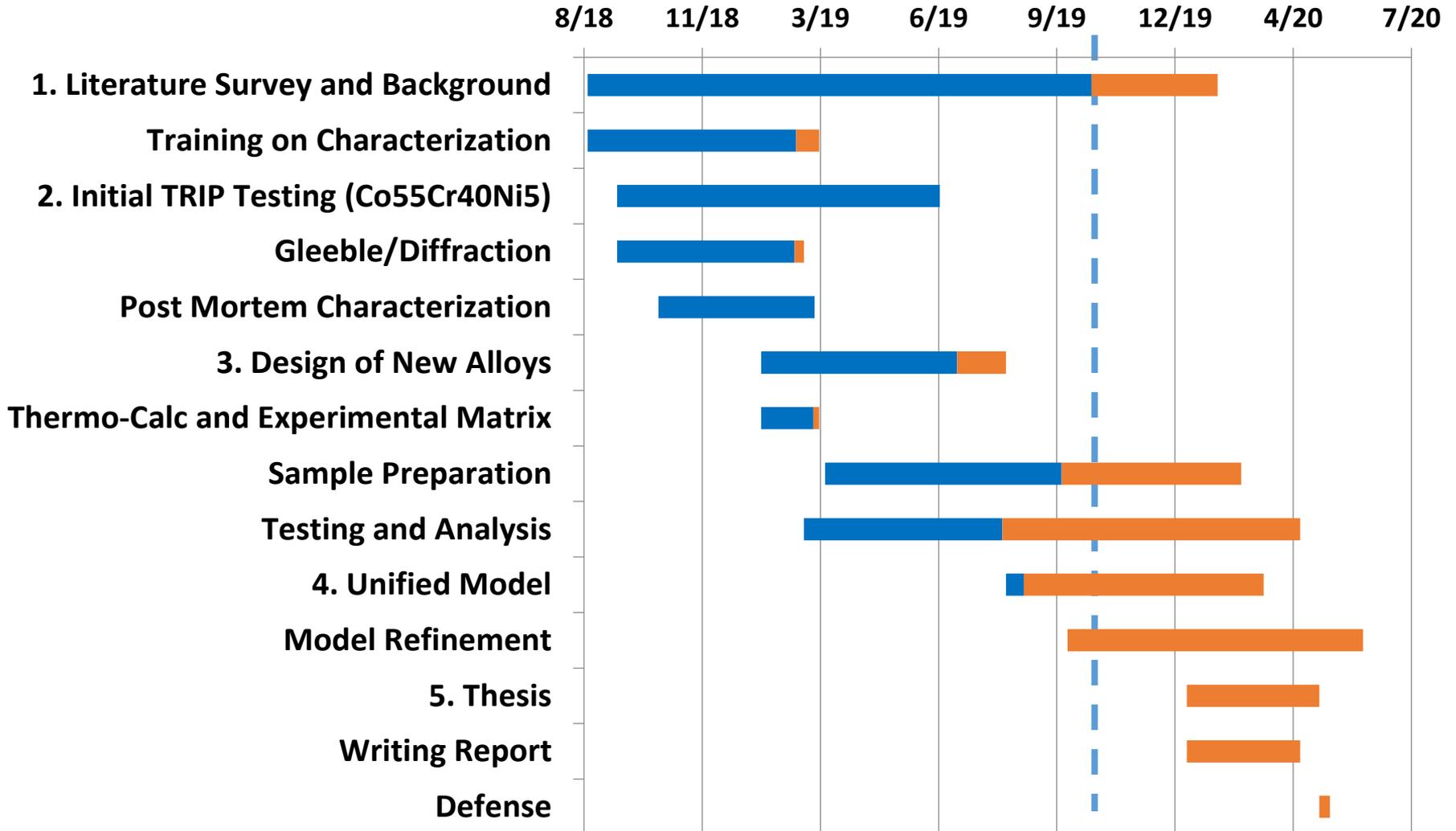
Alloy composition in CoCrNi can be controlled to activate specific deformation mechanisms given knowledge of use conditions.

# Continuing Work



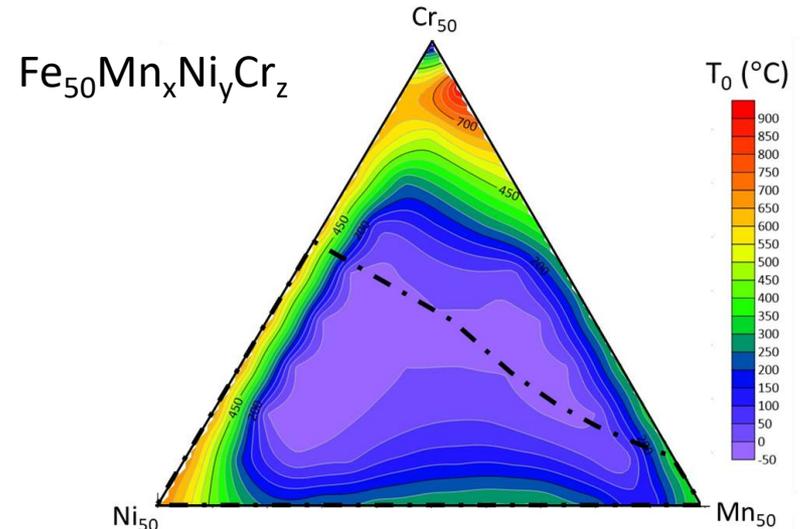
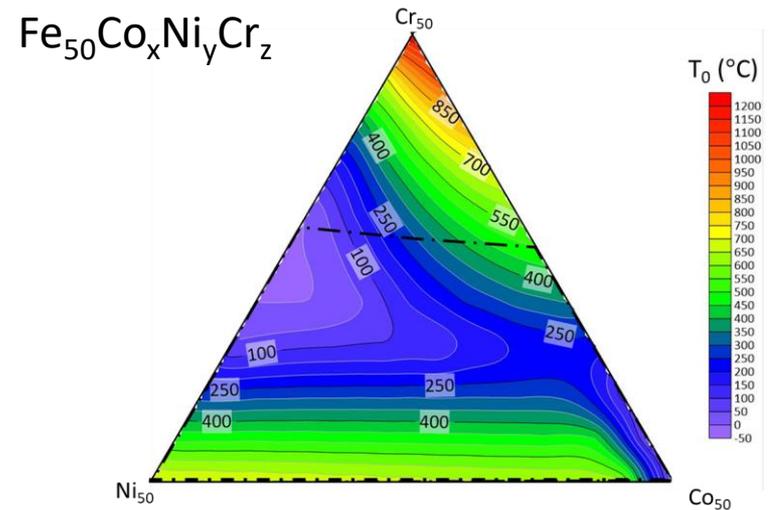
- Developing methods of material processing
  - Aiming for single phase, fine-grained material for mechanical testing
- Upcoming testing at CHESS and APS (Early 2020)
  - For APS new sample geometries and improved heat treatment methods have been designed
- Mechanical testing at Mines
  - Strain rates  $10^{-2} - 10^2 s^{-1}$
  - Interrupted testing
  - Post-mortem XRD and EBSD to observe microstructural evolution

# Progress



# Challenges and Opportunities

- Challenges:
  - Making material is difficult
    - Arc melted buttons require rolling → texture evolution
    - Spray forming → Porous material
    - Casting → large Cr-oxide inclusions
- Opportunities
  - Understanding of designing TRIP and TWIP into CoCrNi can be extended to other alloy systems



*Thank you!*

*John Copley*  
*[jacopley@mines.edu](mailto:jacopley@mines.edu)*

### Project 33B-L: In-Situ Studies of Strain Rate Effects on Phase Transformation and Microstructural Evolution in Multi-Principal Element Alloys

Student: *John Copley*

Faculty: *Amy Clarke*

#### Goals

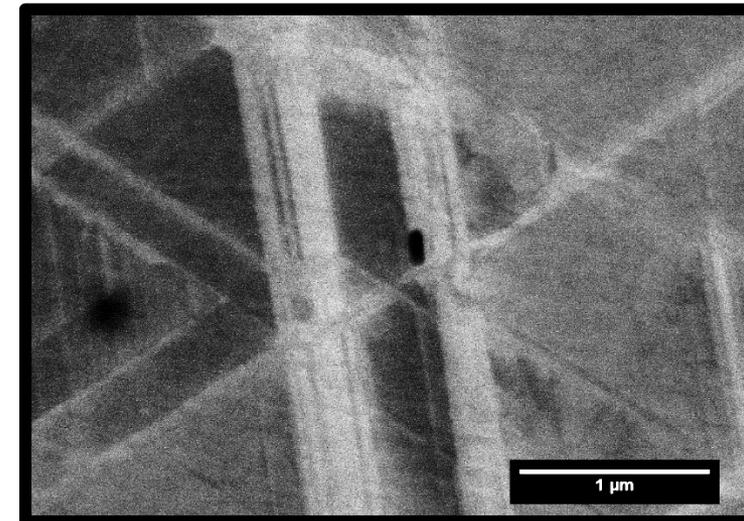
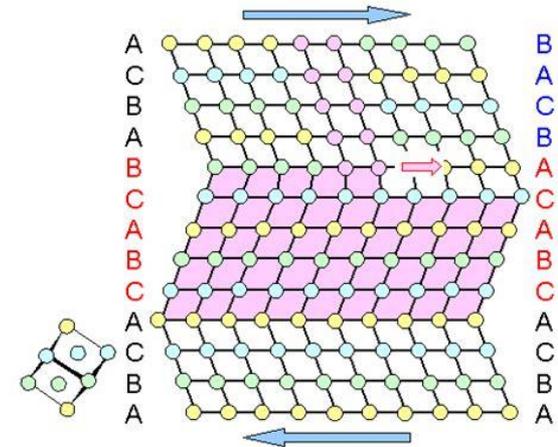
- Modeling of deformation behavior as it pertains to microstructural changes in MPEAs

#### Significance and Impact

- The TRIP/TWIP behavior seen in some MPEAs results in high work hardening behavior, resulting in an increased ductility, toughness and blast resistance.

#### Research Details

- In-situ diffraction tests will show microstructural evolution (twins or HCP phase in a FCC matrix) which can be compared to the strain rate, strain state and alloy composition to allow alloy design for specific applications



# Project 33B-L: In-Situ Studies of Strain Rate Effects on Phase Transformation and Microstructural Evolution in Multi-Principal Element Alloys

Student: *John Copley*

Faculty: *Amy Clarke*



## Goals

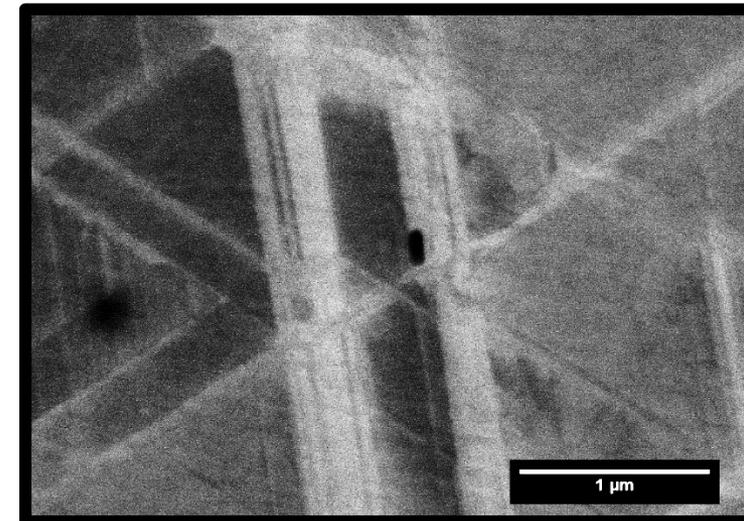
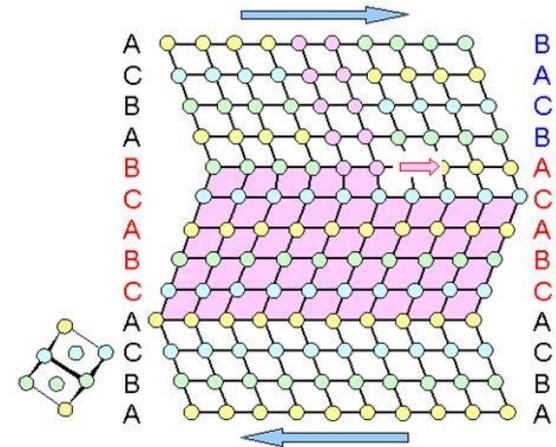
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Faculty: *Amy Clarke*

Project Duration: *Sept. 2018 – May 2020*

## Program Goal

- Modeling of deformation behavior as it pertains to microstructural changes in MPEAs

## Approach

- In-situ diffraction tests will show microstructural evolution (twins or HCP phase in a FCC matrix) which can be compared to the strain rate, strain state and alloy composition to allow alloy design for specific applications

## Benefits

- The TRIP/TWIP behavior seen in some MPEAs results in high work hardening behavior, resulting in an increased ductility, toughness and blast resistance.

