

### ***Project 33a-L: In-situ Studies of Strain Rate Effects on Phase Transformations and Microstructural Evolution in $\beta$ -Titanium***

***Fall 2018 Semi-Annual Meeting  
Colorado School of Mines, Golden, CO  
October 2-4, 2018***

*Student: Benjamin Ellyson (Mines)*

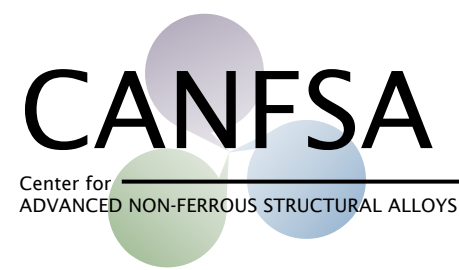
*Faculty: Amy Clarke (Mines)*

*Industrial Mentors: Austin Mann (Boeing)*

*Other Participants: Yaofeng Guo (Mines)*



# Project 33a-L: In-Situ Studies of Strain Rate Effects on Phase Transformations and Microstructural Evolution in $\beta$ -Titanium



- Student: Benjamin Ellyson (Mines)
- Advisor(s): Amy Clarke (Mines)

**Project Duration**  
PhD: September 2017 to May 2021

- **Problem:** Uniform elongation and work hardening of titanium alloys restricts applications.
- **Objective:** Fundamentally understand microstructural evolution in metastable  $\beta$  titanium alloys to develop an alloy design methodology and tailor microstructures and properties.
- **Benefit:** Novel titanium alloys for blast and crash resistant applications

## Recent Progress

- Ti-1023 (wt.%) quasi-static compression study completed
- Initial Ti-1023 quasi-static tensile testing underway
- Deformation mechanism of Ti-1023 identified by TEM
- Initial microstructural characterization of Ti-15Mo (wt.%)

## Metrics

Description	% Complete	Status
1. Literature review	70%	●
2. Quasi-static mechanical characterization of Ti-1023 and Ti-15Mo	40%	●
3. Dynamic testing of Ti-1023 and Ti-15Mo	0%	●
4. Microstructural characterization of pre- and post- deformed samples	30%	●
5. <i>In situ</i> characterization of microstructural evolution during deformation	0%	●

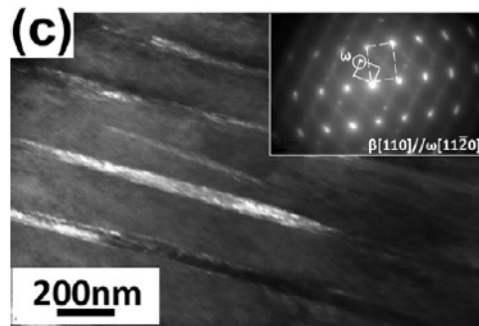
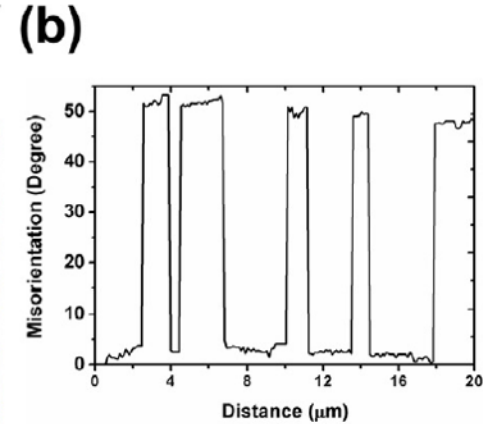
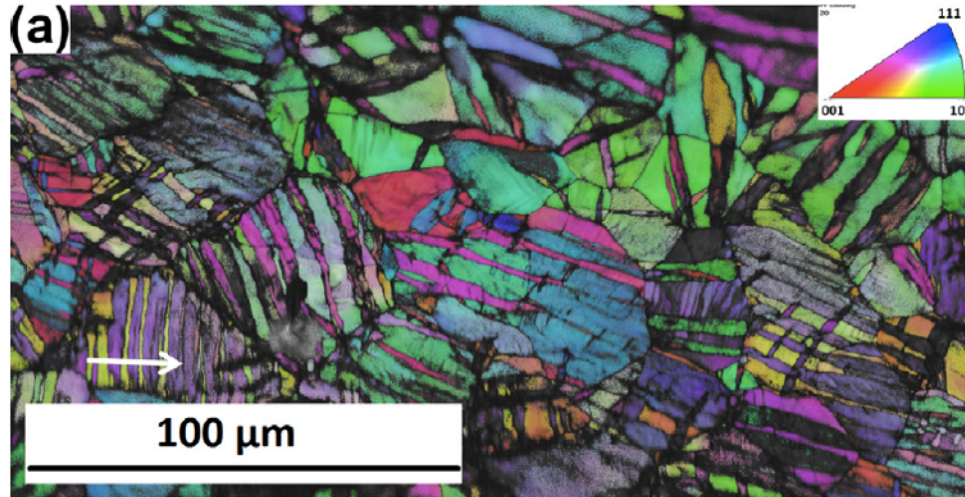
# Industrial Relevance: Development of Blast Resistant Materials for the Navy

- **Cellular Materials Program**
  - Multifunctional structures
  - **Blast resistance**
  - Thermal management
- **Propulsion Materials Program**
  - **Aircraft and marine engines**

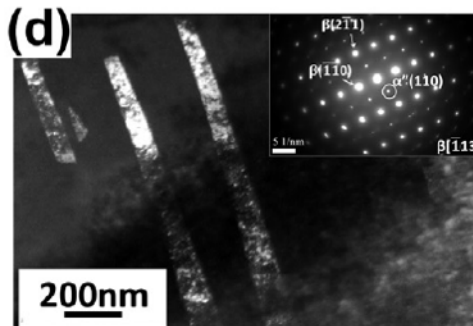


<https://www.onr.navy.mil/Science-Technology/Departments/Code-33>

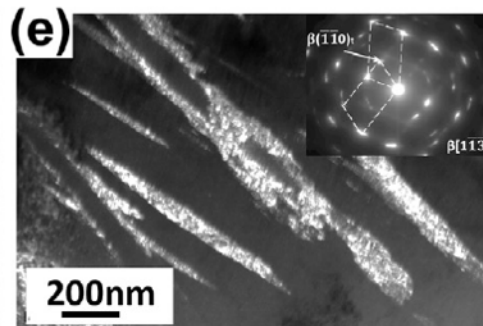
# Ti-25Nb-3Zr-3Mo-2Sn (wt.%) Alloy Microstructure After Deformation at $10^{-3} \text{ s}^{-1}$ and 0.18 True Strain



Deformation-induced  $\omega$  phase



Deformation-induced  $\alpha''$  phase

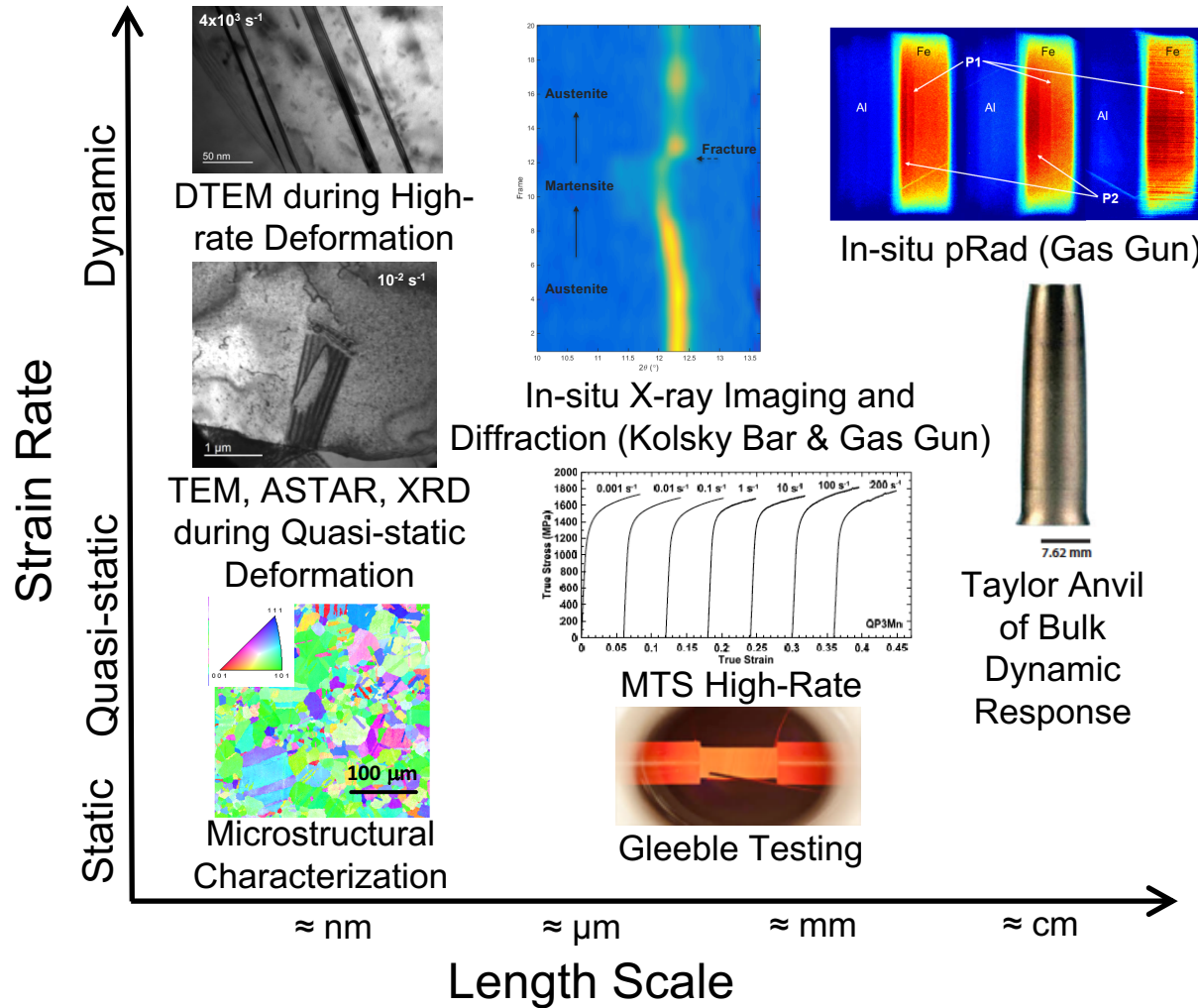


β mechanical twinning

H. Zhan, et al. 107 Scripta Materialia (2015): 34-37



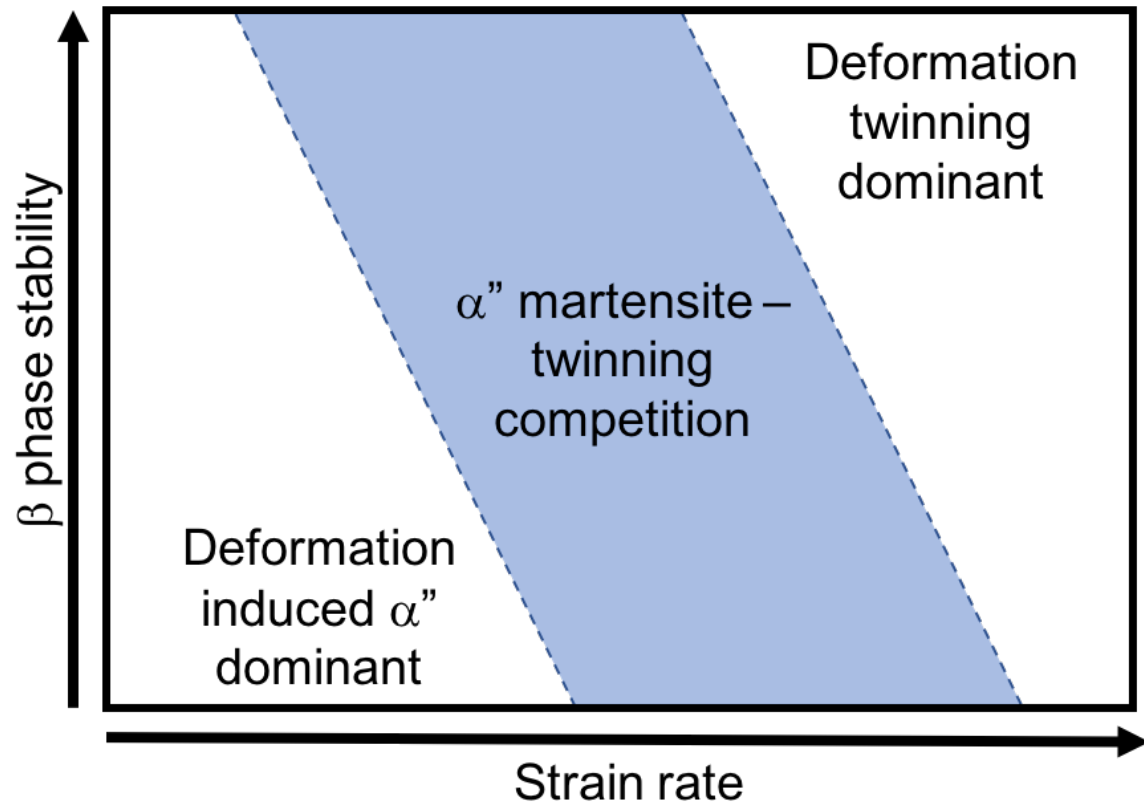
# Multi-scale Studies of TRIP/TWIP during High Rate Deformation



# The Effect of Strain Rate ( $10^{-3}$ , $10^{-1}$ , $10^1$ , $10^2$ s $^{-1}$ ) on Deformation Mechanisms during Compression of a Ti-10V-3Fe-3Al (wt.%) Alloy

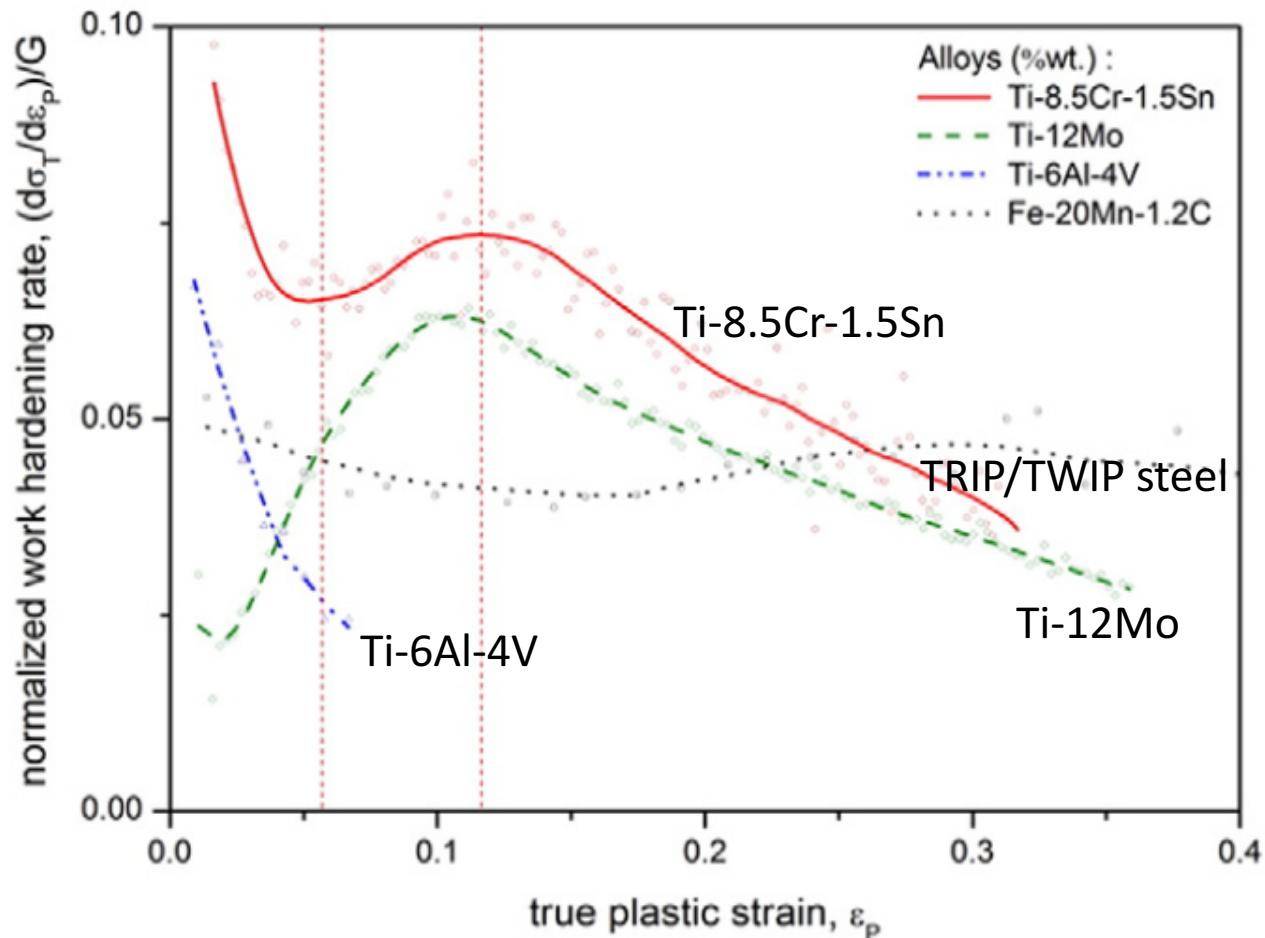
- **As strain rate increases:**

- Stress-induced  $\alpha''$  martensite (dominant) +  $\{332\}\langle 113 \rangle \beta$  twinning + stress-induced  $\omega$  phase + slip
- Stress-induced  $\alpha''$  martensite +  $\{332\}\langle 113 \rangle \beta$  twinning + stress-induced  $\omega$  phase + slip
- Stress-induced  $\alpha''$  martensite +  $\{332\}\langle 113 \rangle \beta$  twinning (dominant) + stress-induced  $\omega$  phase + slip



Ahmed, M., et al. 104 Acta Materialia (2016): 190-200

# Work Hardening and Evidence of TRIP/TWIP



Brozek, C., et al. Scripta Materialia 114 (2016): 60-64

# Improvements in Sample Preparation

- Consulting Buhler, Struers, LECO and Oxford EBSD
- Reading literature

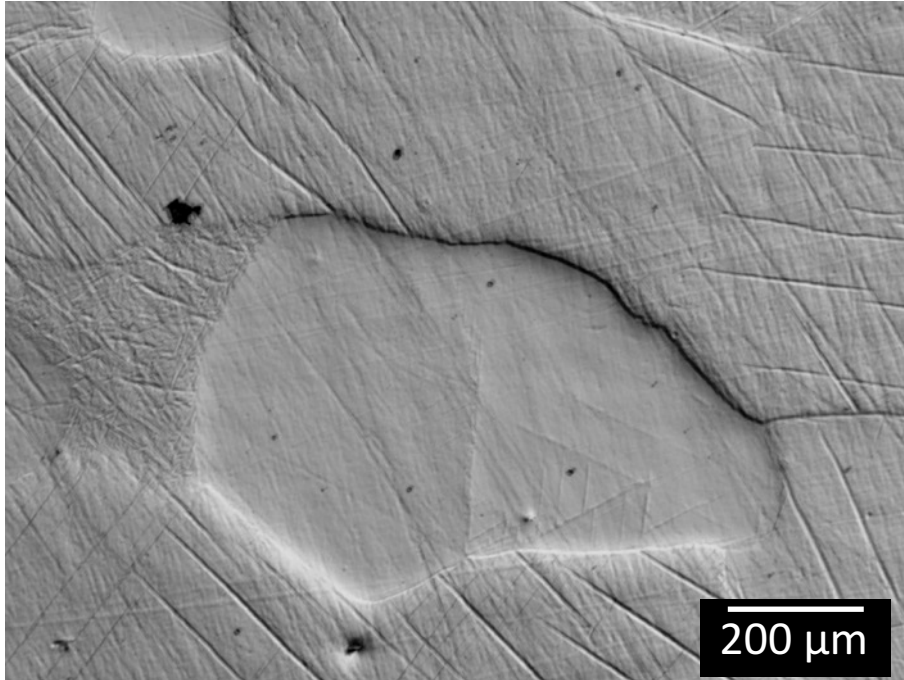
## Automatic mechanical polishing methods

- Fast, high quality and reproducibility

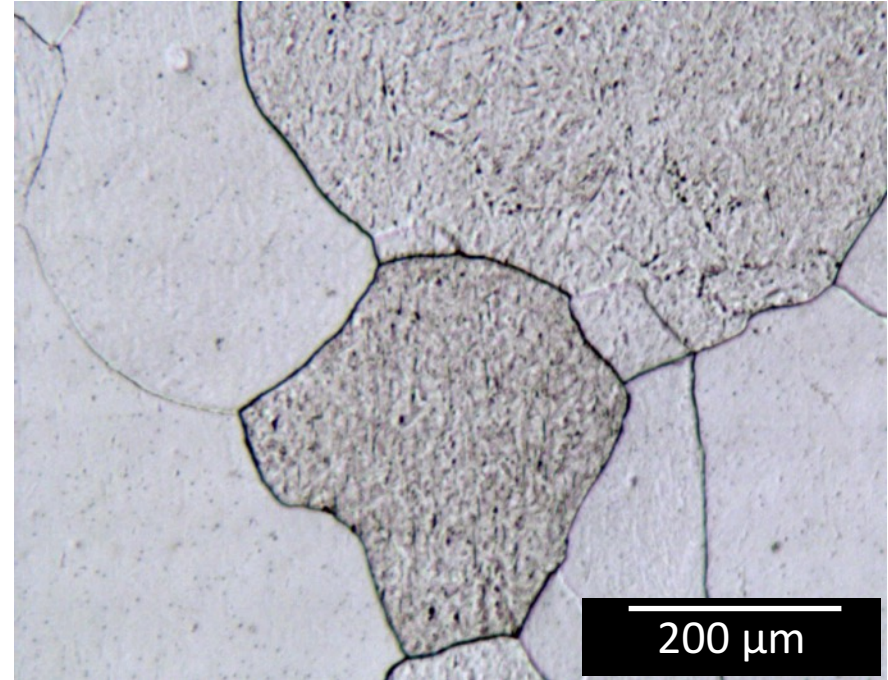
grit/grain size	abrasive	duration (seconds)	load (N)	rotation (rpm)
320 grit	SiC paper	until flat	5	350/175
9 $\mu\text{m}$	diamond suspension	300	5	150/75
0.02 $\mu\text{m}$	5 OP-S + 1 H <sub>2</sub> O <sub>2</sub>	600	10	150/75

Notes: individual force applied and holder rotates in opposite direction with the platen.

# Comparison of Ti-1023 OM Images



**previously**



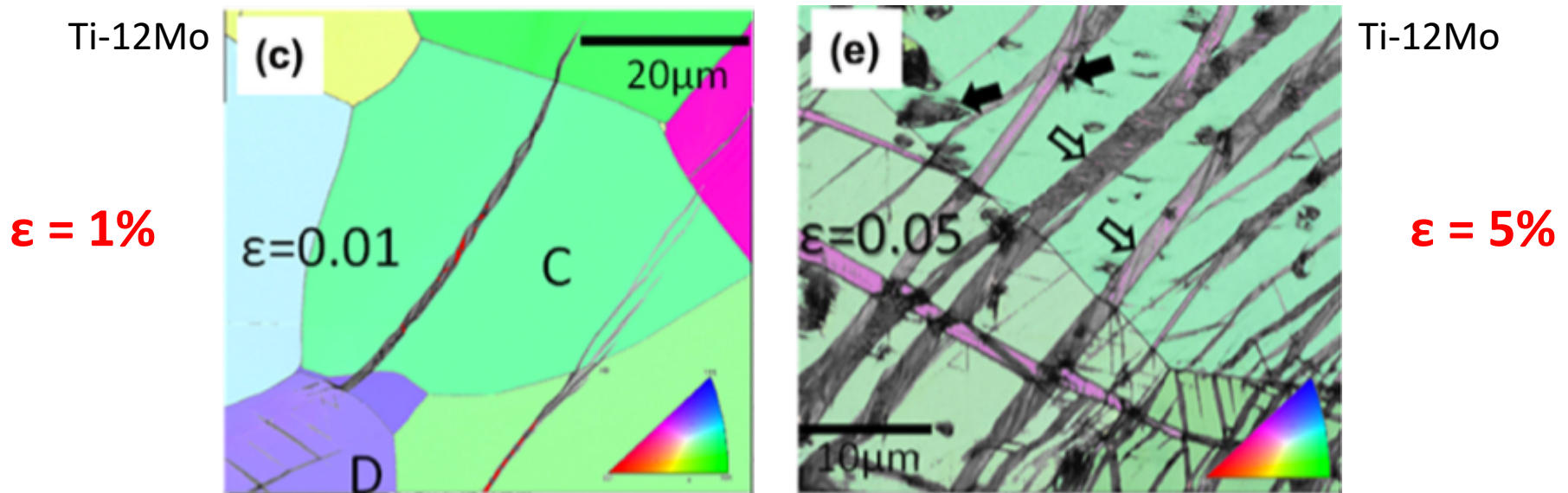
**improved**

- Etchant
  - left: Kroll's reagent (6% HNO<sub>3</sub>, 2% HF and 92% DI water)
  - right: 1% HF in DI water
- **No more stress-induced martensites due to sample preparation**



# Difficulty in Obtaining High Quality EBSD Results in Deformed Samples

The indexing level decreases very rapidly with strain, as illustrated by Fig. 3e and f. Only a few twins and martensitic plates can be indexed in this figure, while most of the deformation areas, e.g. the dark clusters (black arrows)



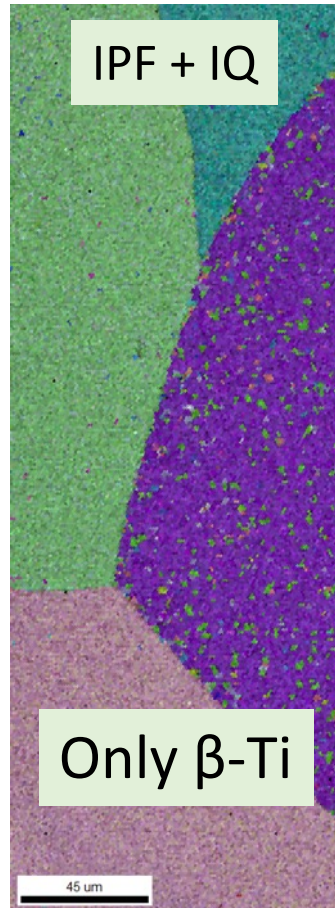
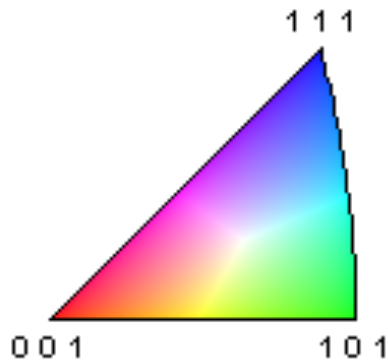
F. Sun et al. / Acta Materialia 61 (2013) 6406

# Preliminary Ti-1023 EBSD Results

**before** deformation

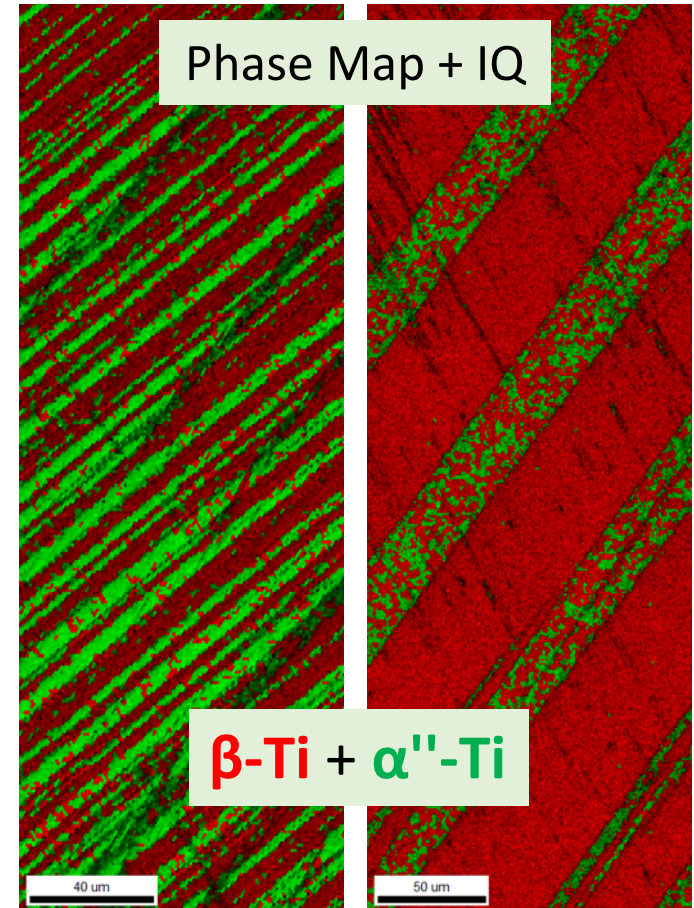
2hrs@1200°C

CI: 0.4 - 0.8

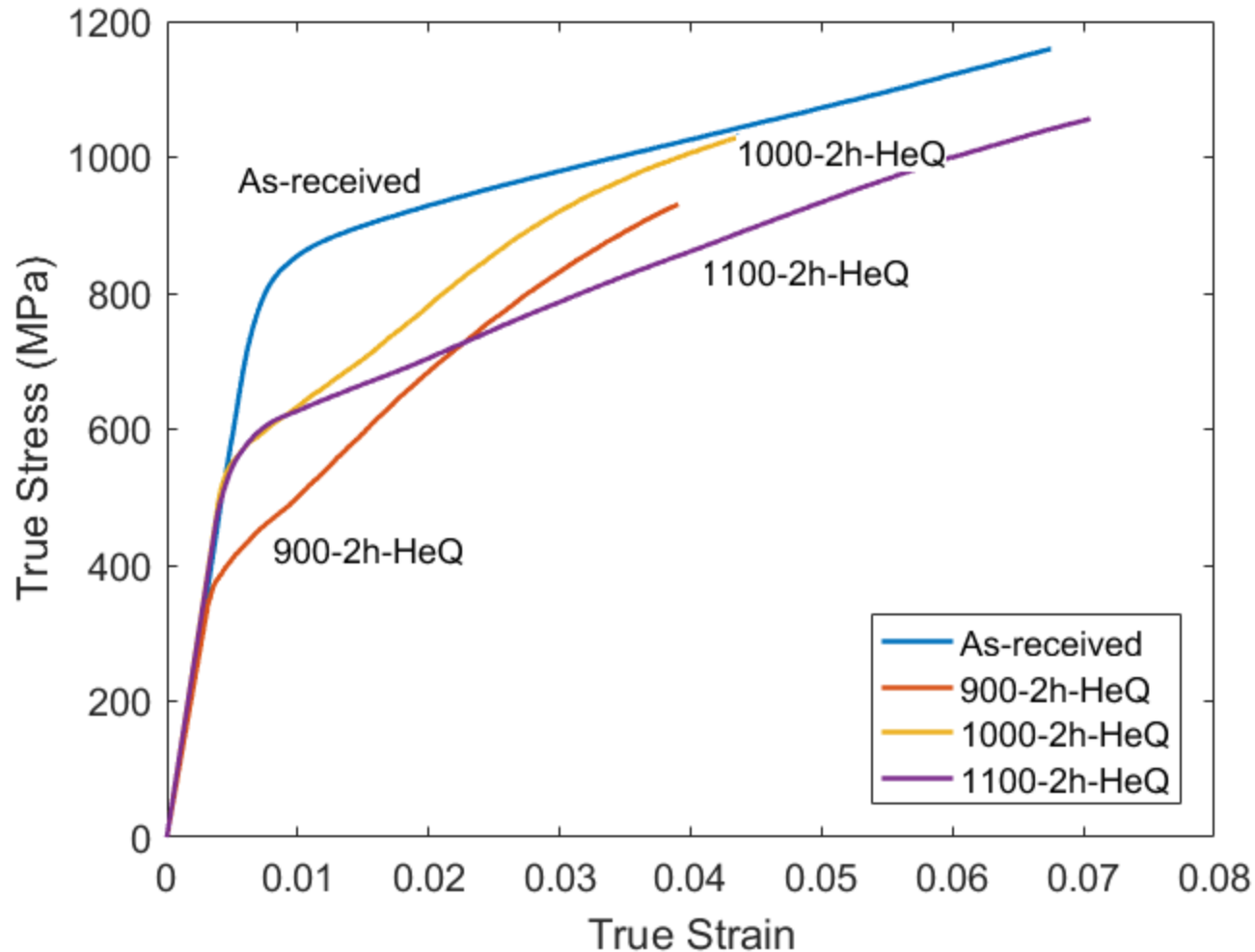


**after** deformation, 5%

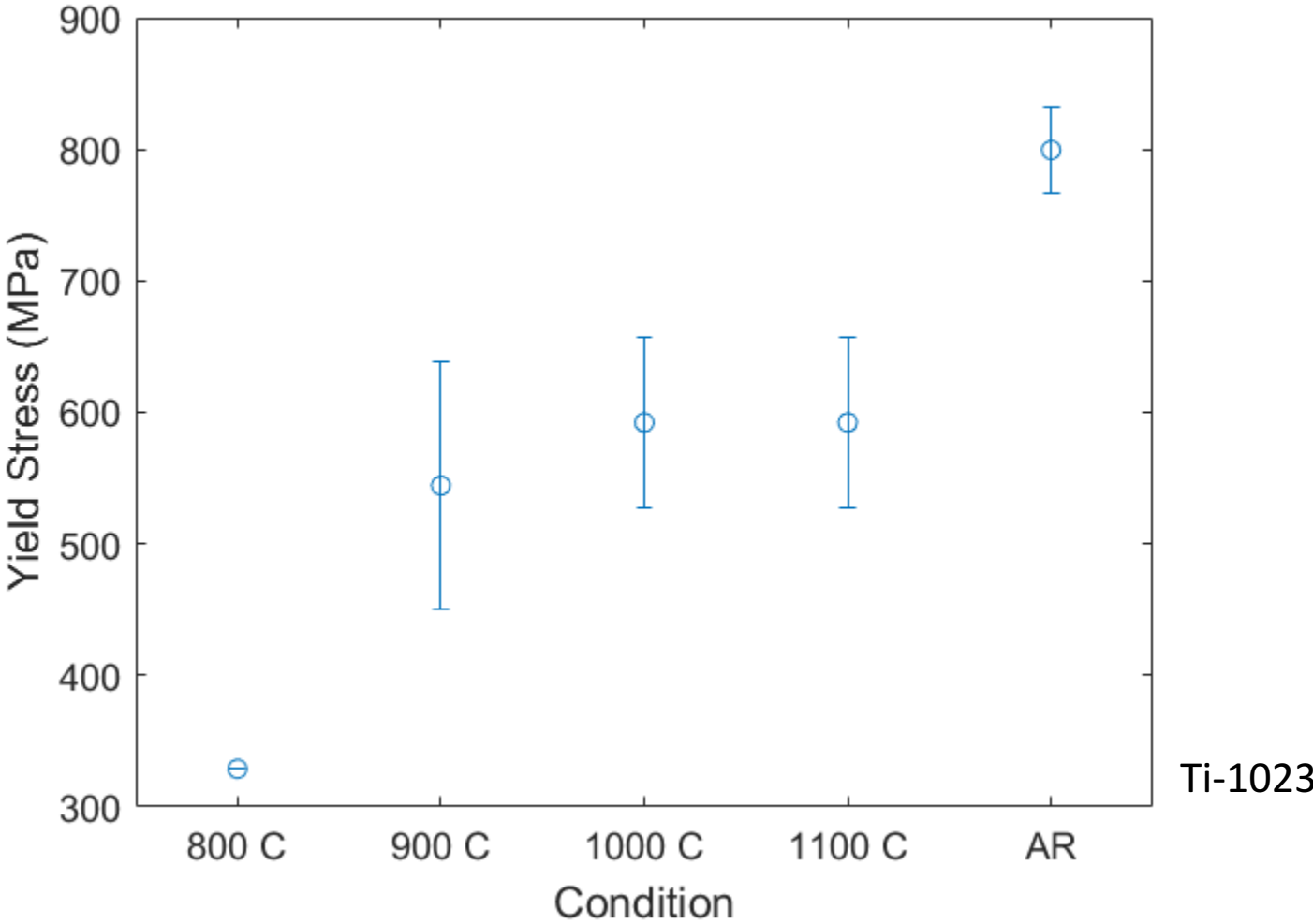
CI: ~ 0



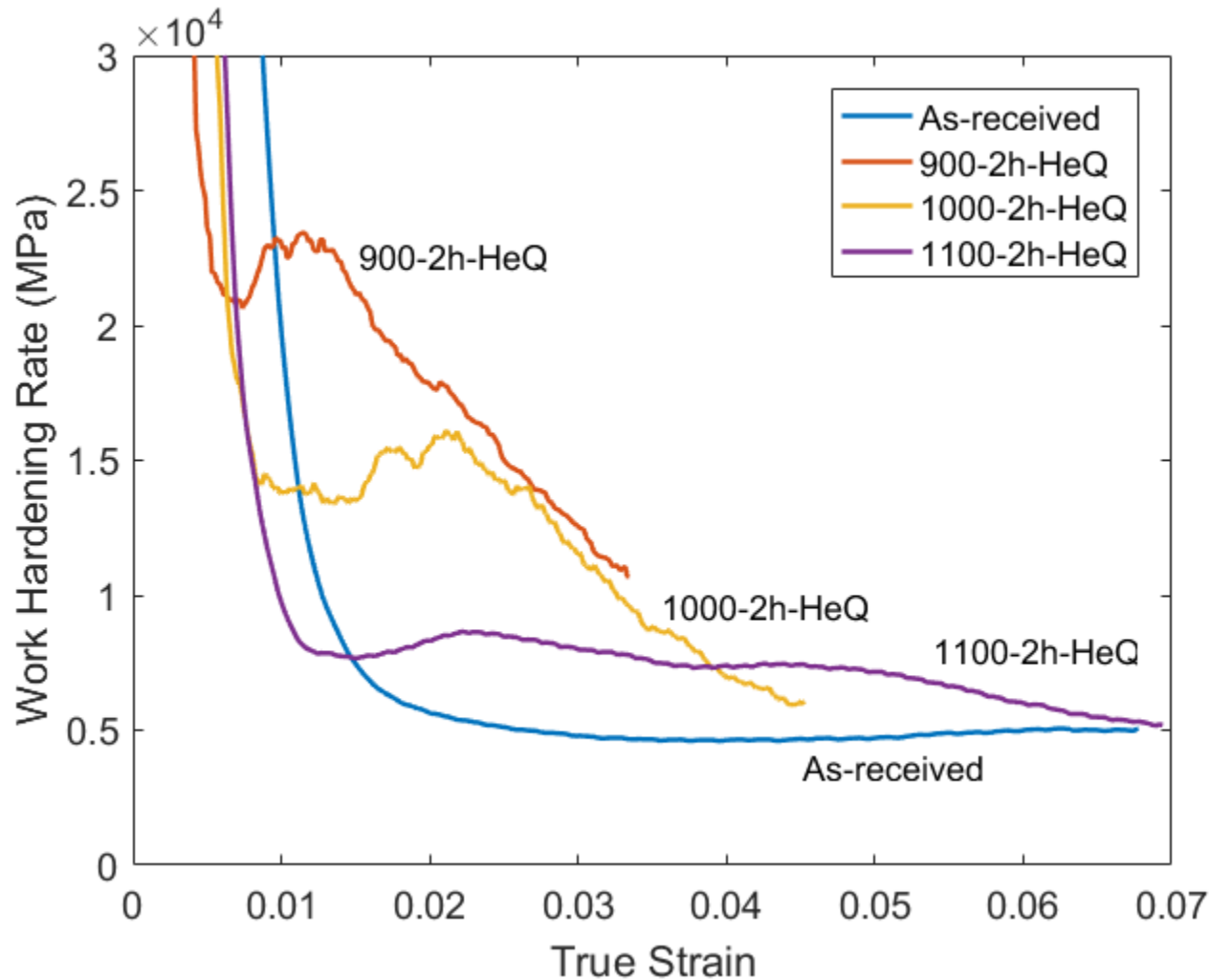
# Comparison of Ti-1023 Stress-Strain Behavior in Compression



# Trend in 0.2% Offset Yield Stress in Compression

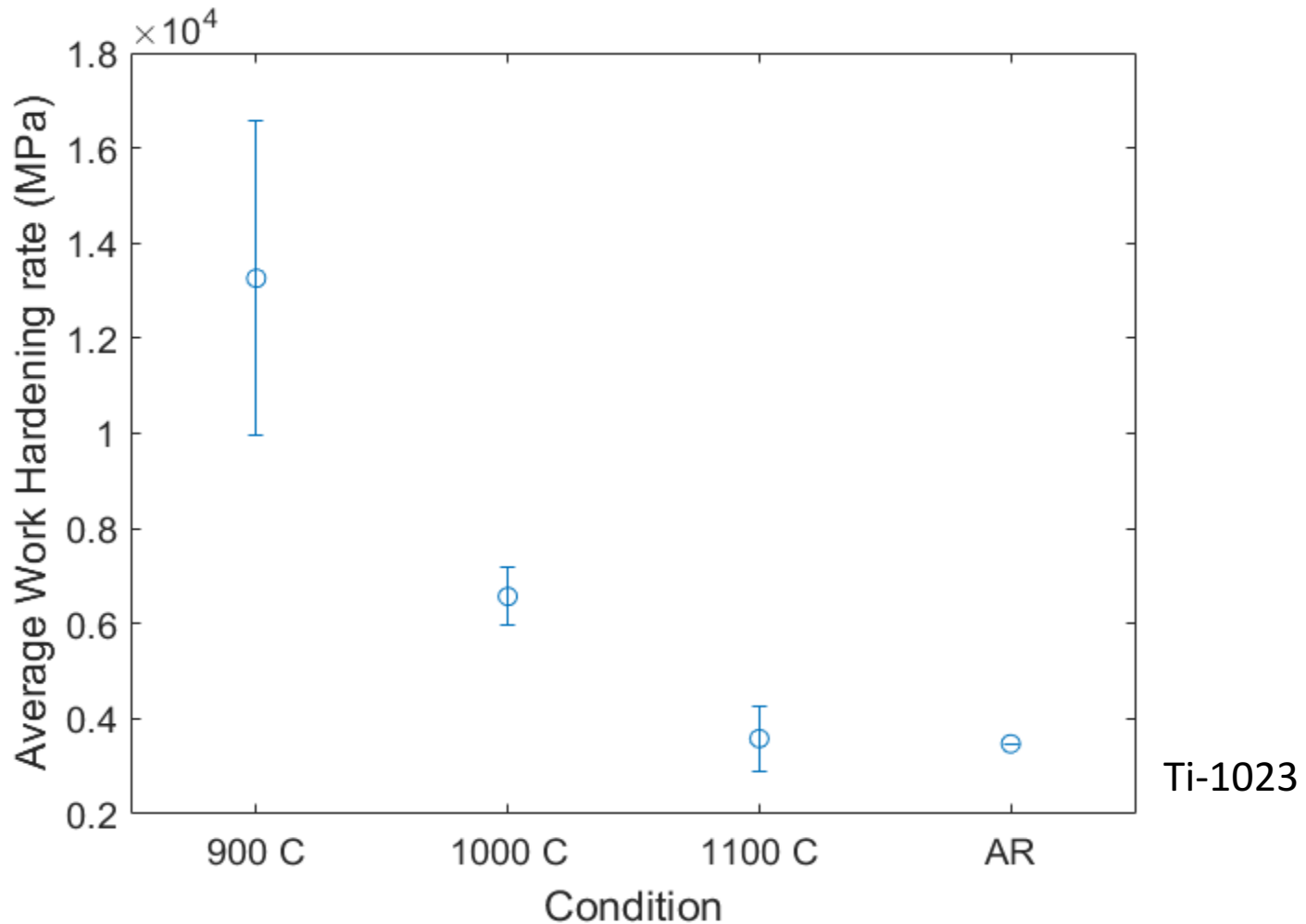


# Instantaneous Work Hardening as a Function of Solution Temperature





# Trend in Average Work Hardening Behavior



# Plan for Initial Ti-1023 Tensile Testing

- Lessons from compression study and literature:
    - **Grain size/solution temperature** has a controlling effect on work hardening behavior
    - Effect of grain size and holding temperature are convoluted
1. 700°C-1h for full annealing
  2. 1100°C-3h for grain growth treatment for large grain sample
  3. 850°C-1h for final homogenizing solution treatment

# Ti-1023 Microstructure and Grain Size

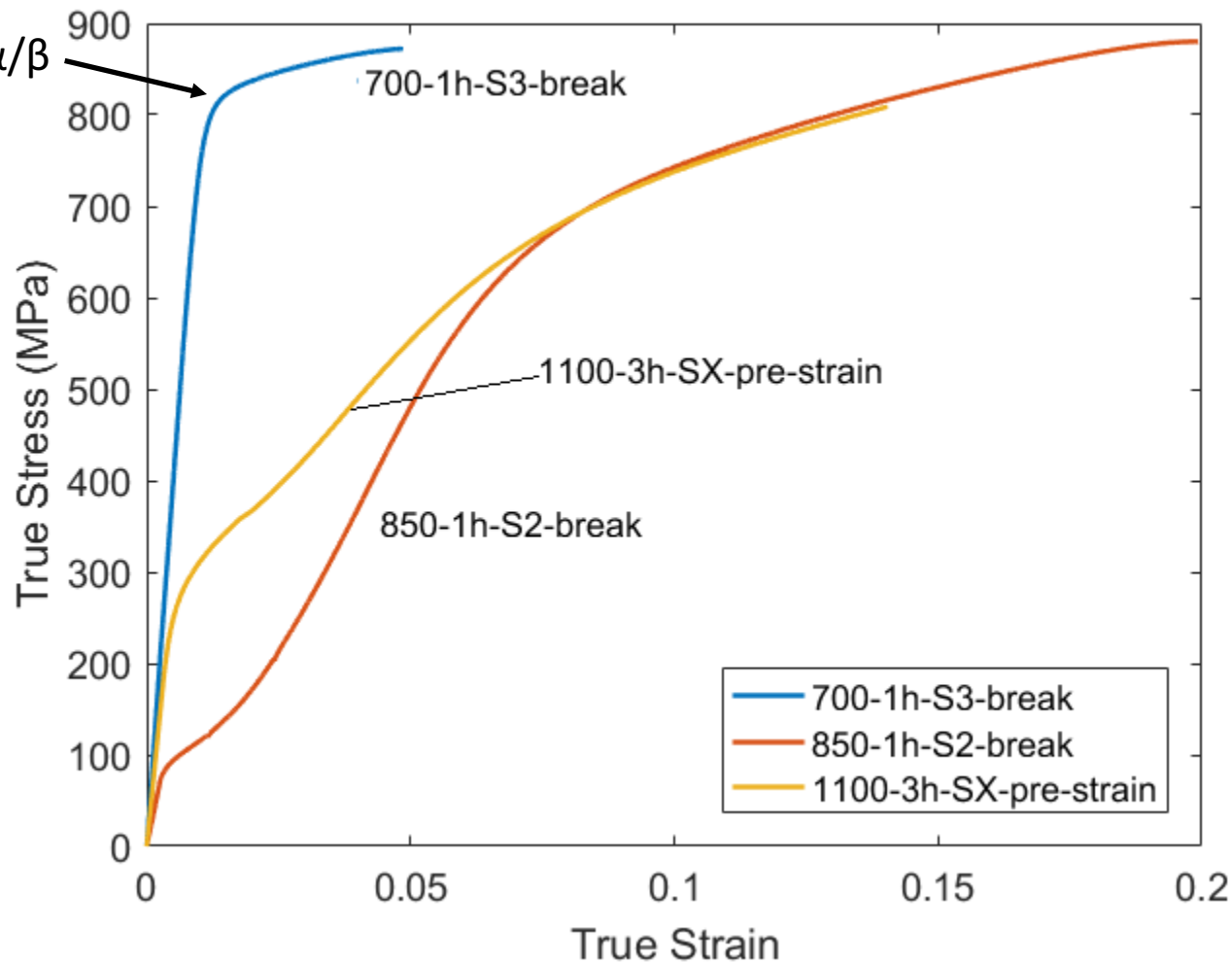
$\alpha/\beta$  treatment, 700 °C for 1h :

- Roughly 70 % primary  $\alpha$  fraction
- Leaves A/R prior  $\beta$  grain boundaries un-modified

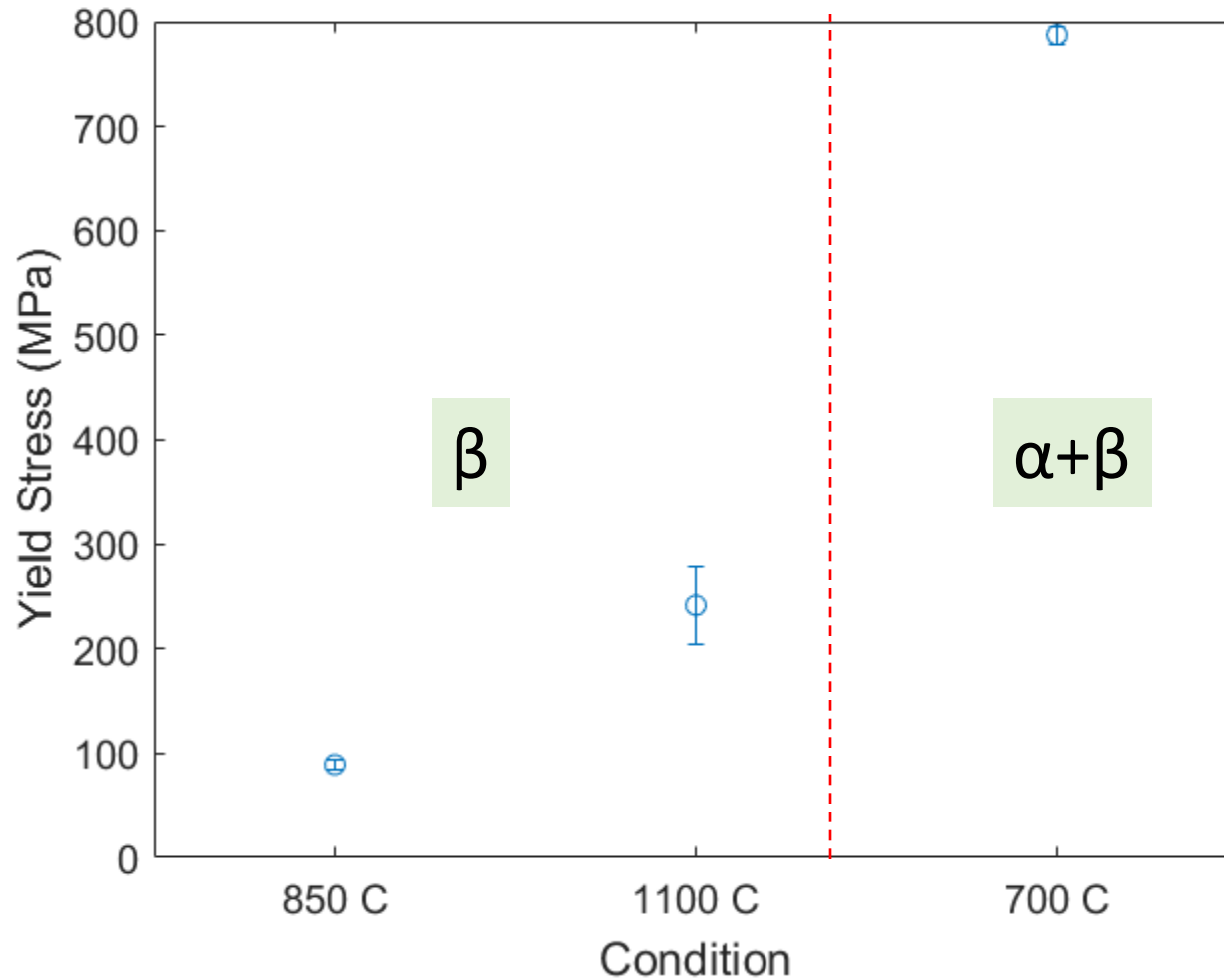
Heat treatment	Grain size
850 °C for 1h, small grain state	~ 150 $\mu\text{m}$
1100 °C for 3h, large grain state	~ 1200 $\mu\text{m}$

# Ti-1023 Stress-Strain Behavior in Tension

Typical behavior  
for fully stable  $\alpha/\beta$

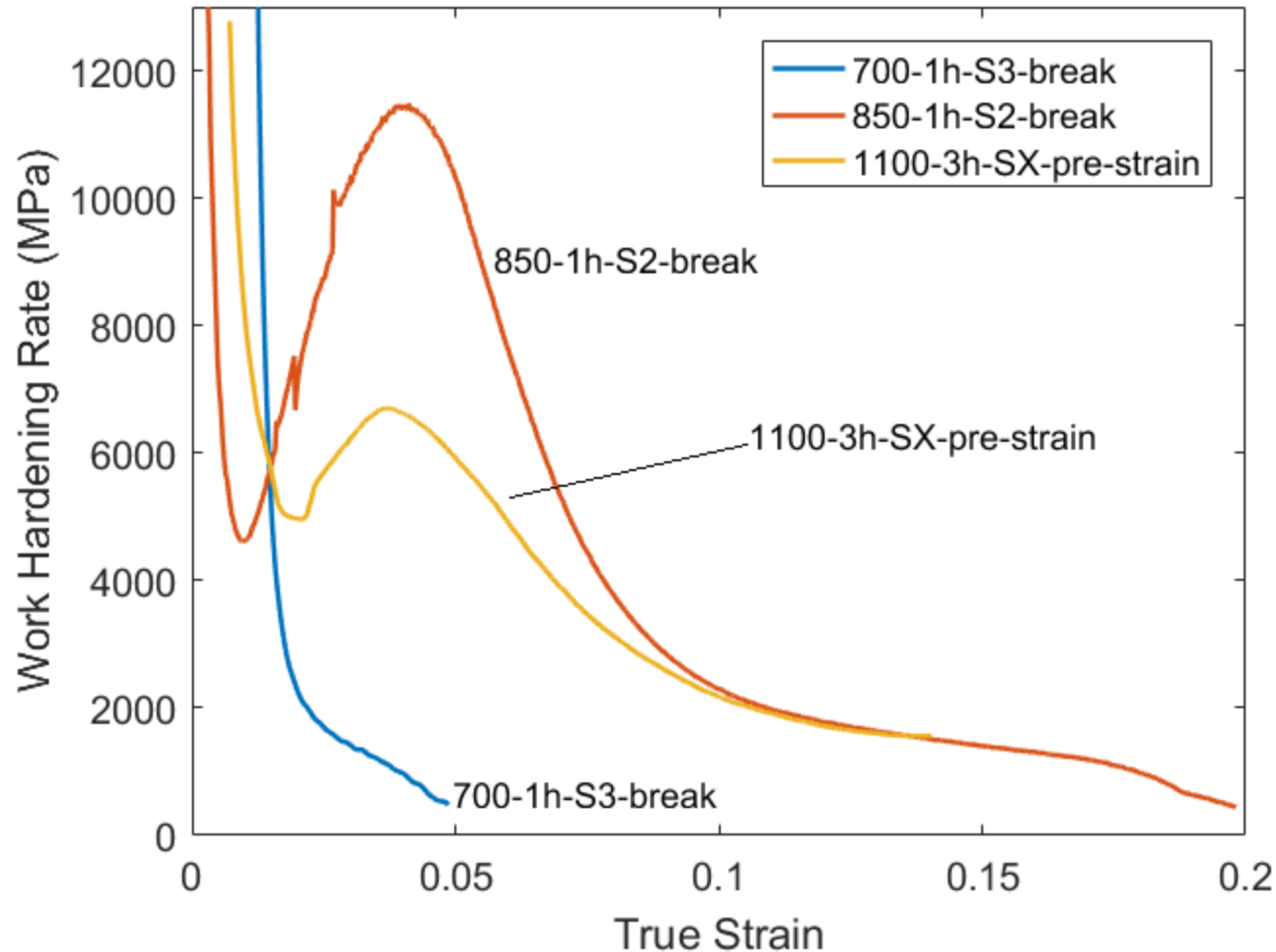


# Trend in Ti-1023 0.2% Offset Yield Stress in Tension

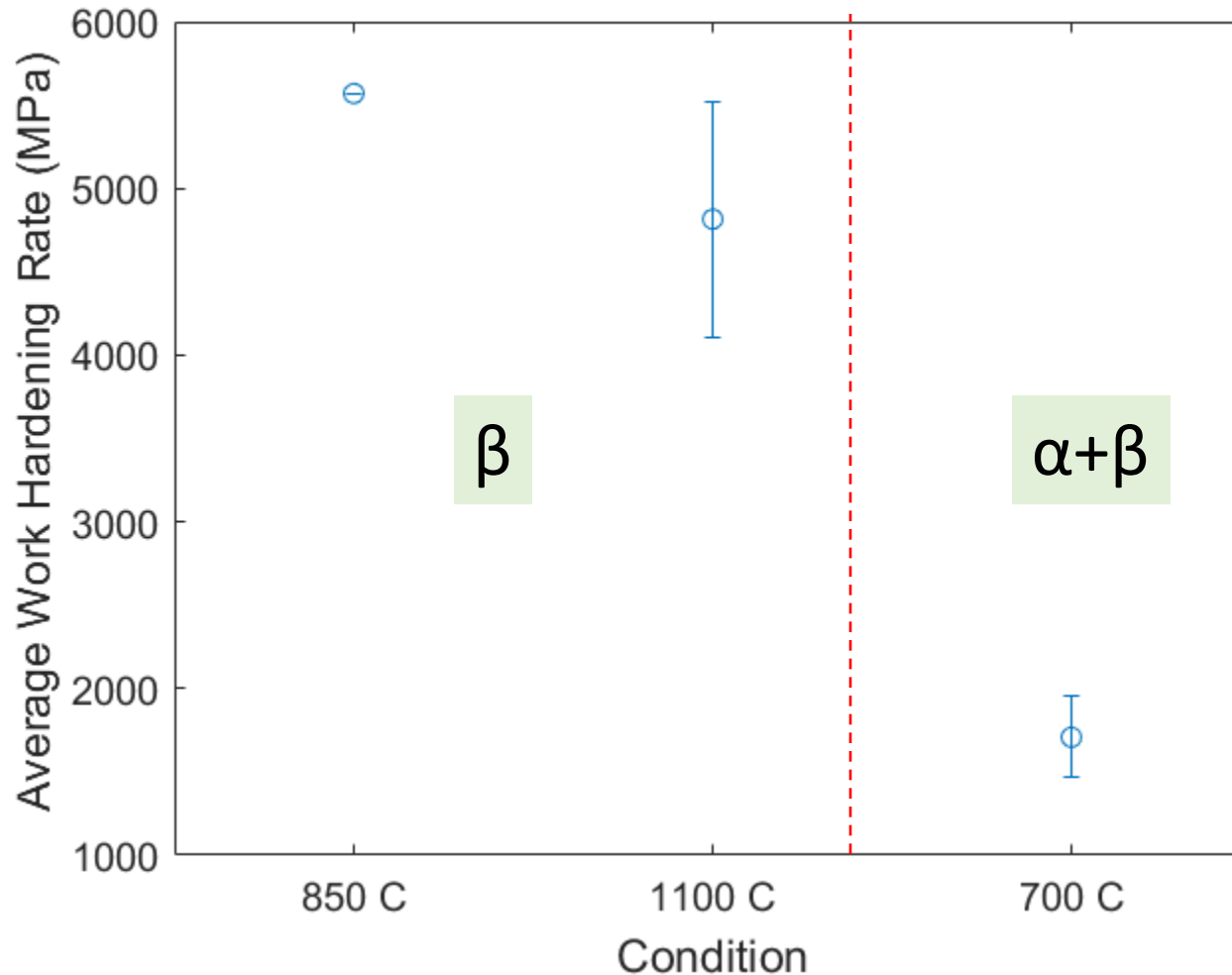




# Ti-1023 Instantaneous Work Hardening Behavior in Tension



# Trend in Ti-1023 Average Work Hardening Behavior



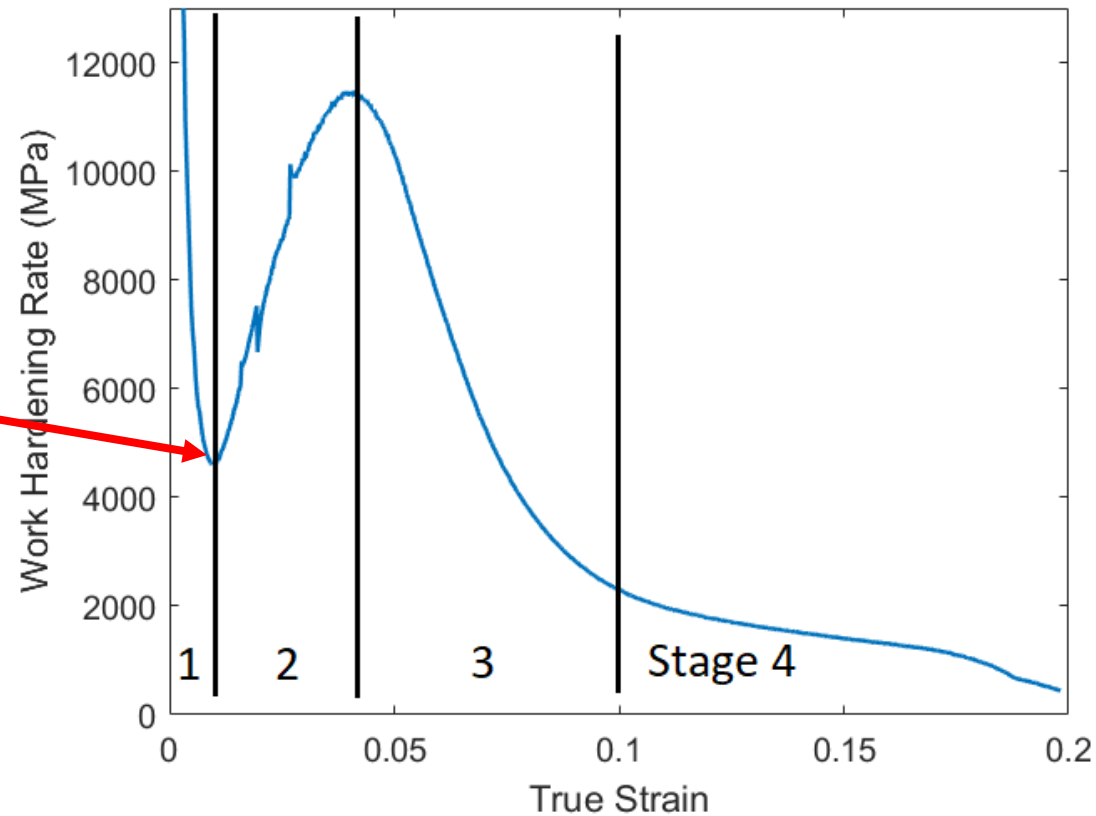
# Ti-1023 Summary

- Compression study shows:
  - Work hardening peak tends to be broader as solution temperature increases
  - Yield stress dependence on solution temperature exhibits a large spread
- Initial tensile testing shows:
  - Grain size and beta chemical stability are controlling factors for work hardening behavior
  - TRIP/TWIP occurs even at large grain sizes
  - Yield stress and work hardening trend with grain size and chemical stability are clear in tension

# Stages of Work Hardening in Ti-1023

TEM characterization of Interrupted tensile tests :

- ✓ 0.5% plastic strain
- 1.7% plastic strain



# Undeformed Ti-1023

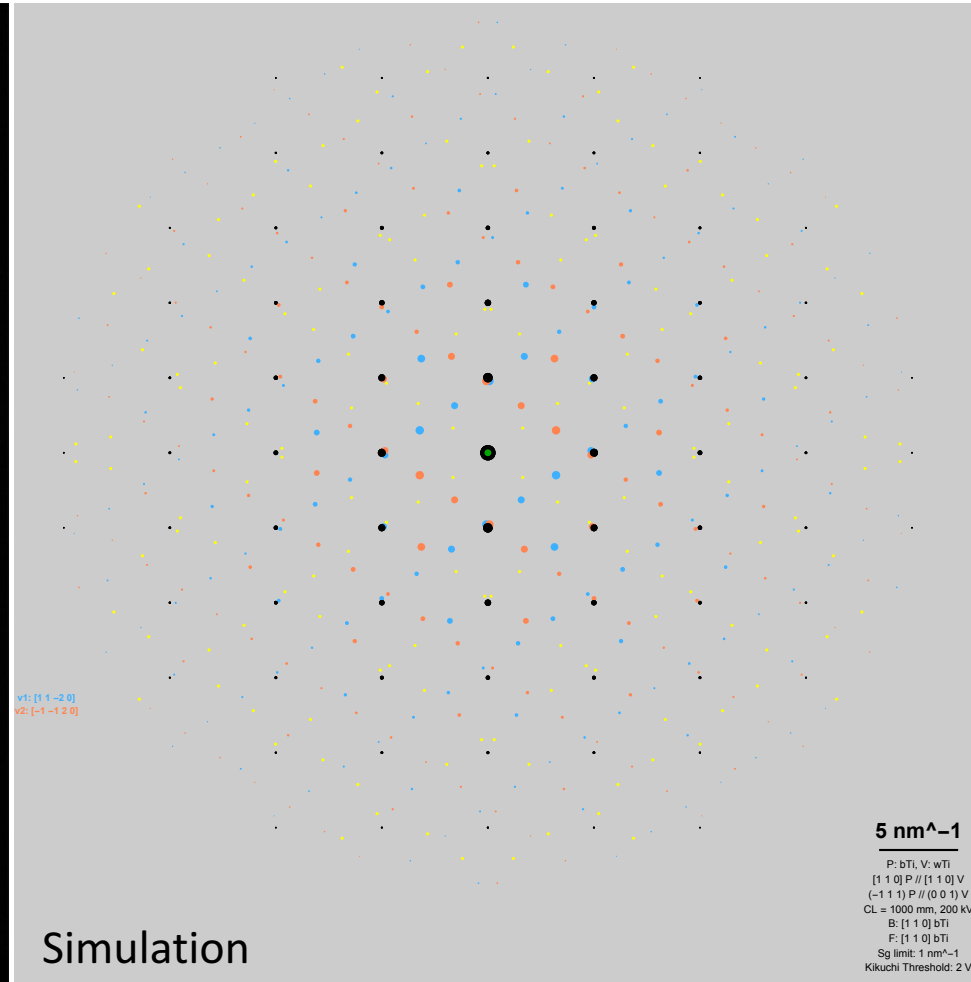
1h@850°C, small grain size

$(-111)_{\text{BCC}} // (0001)_{\omega}$ ,  $[110]_{\text{BCC}} // [11-20]_{\omega}$

-220    -22-2  
-112    -11-2  
          00-2

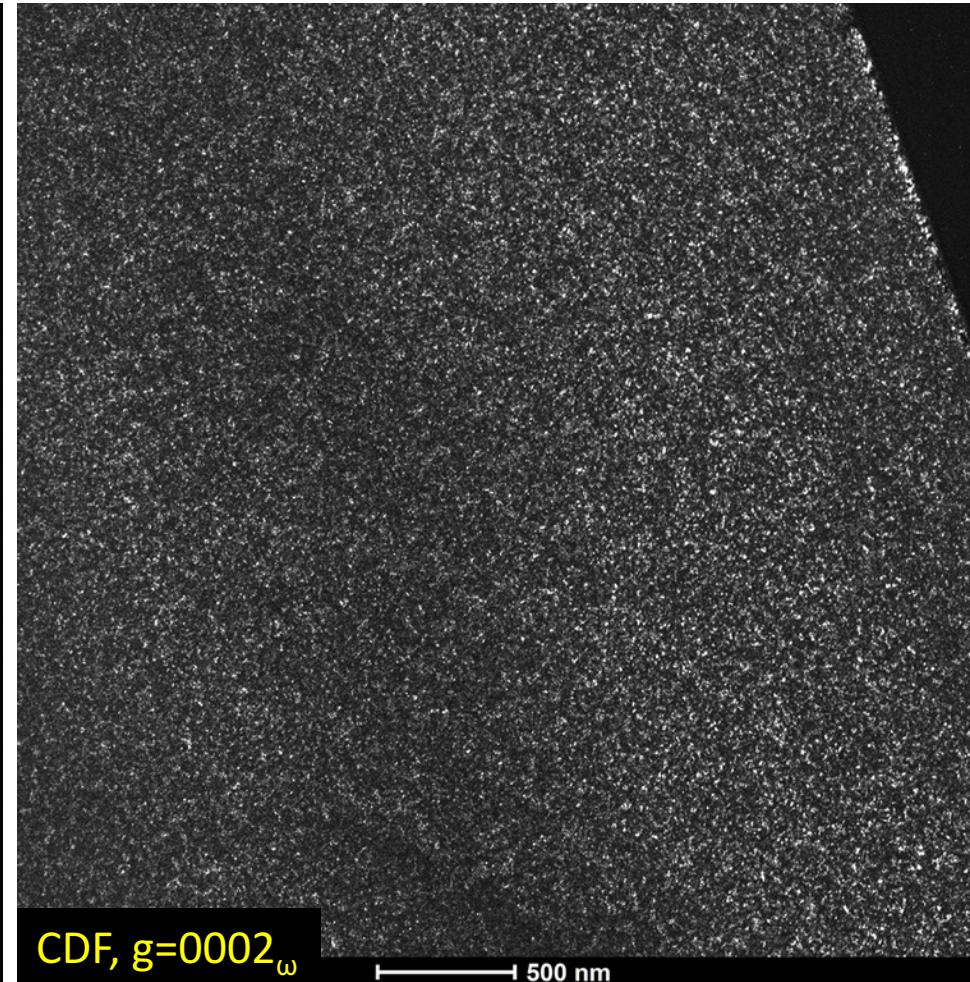
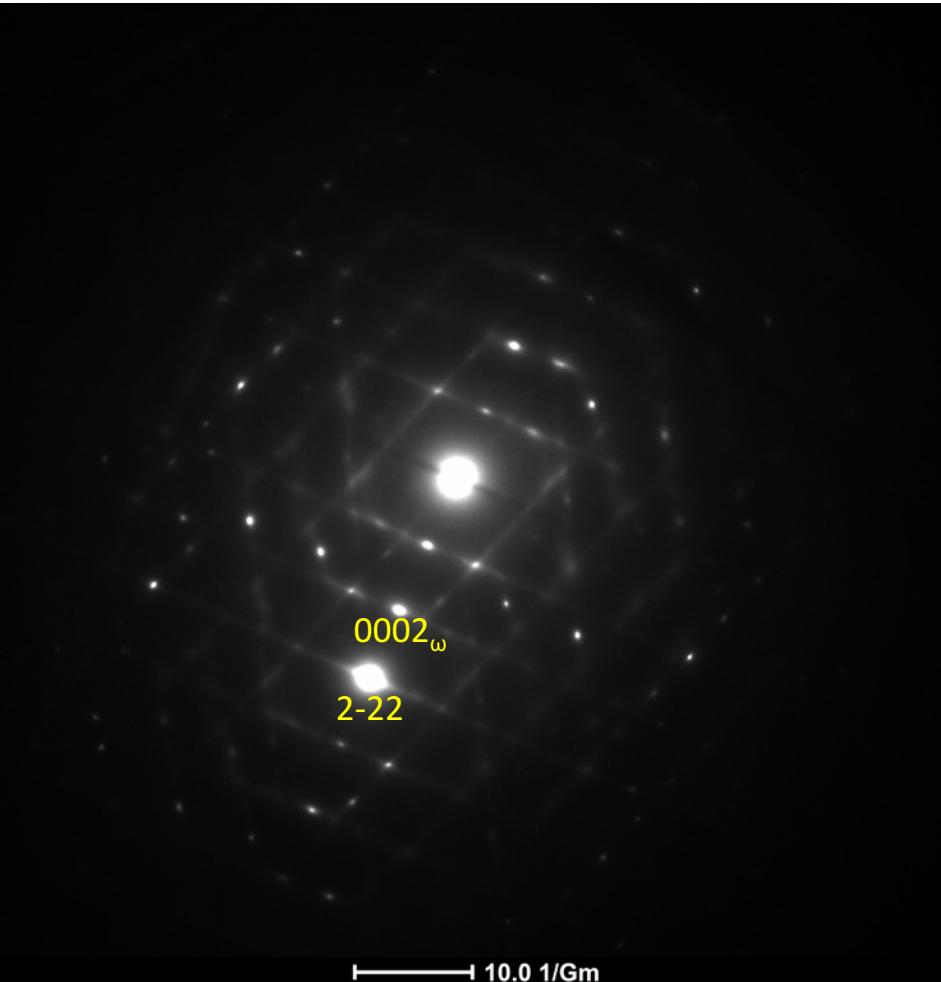
**[110]BCC**

10.0 1/Gm

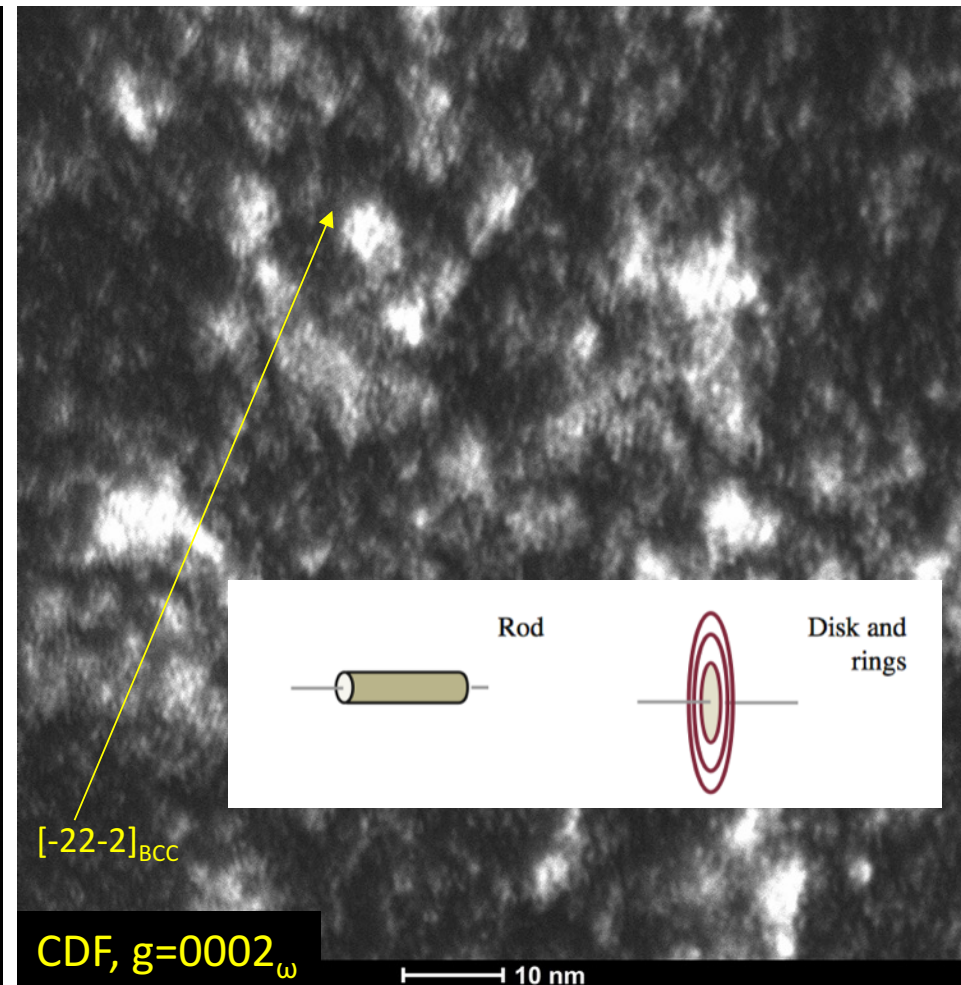
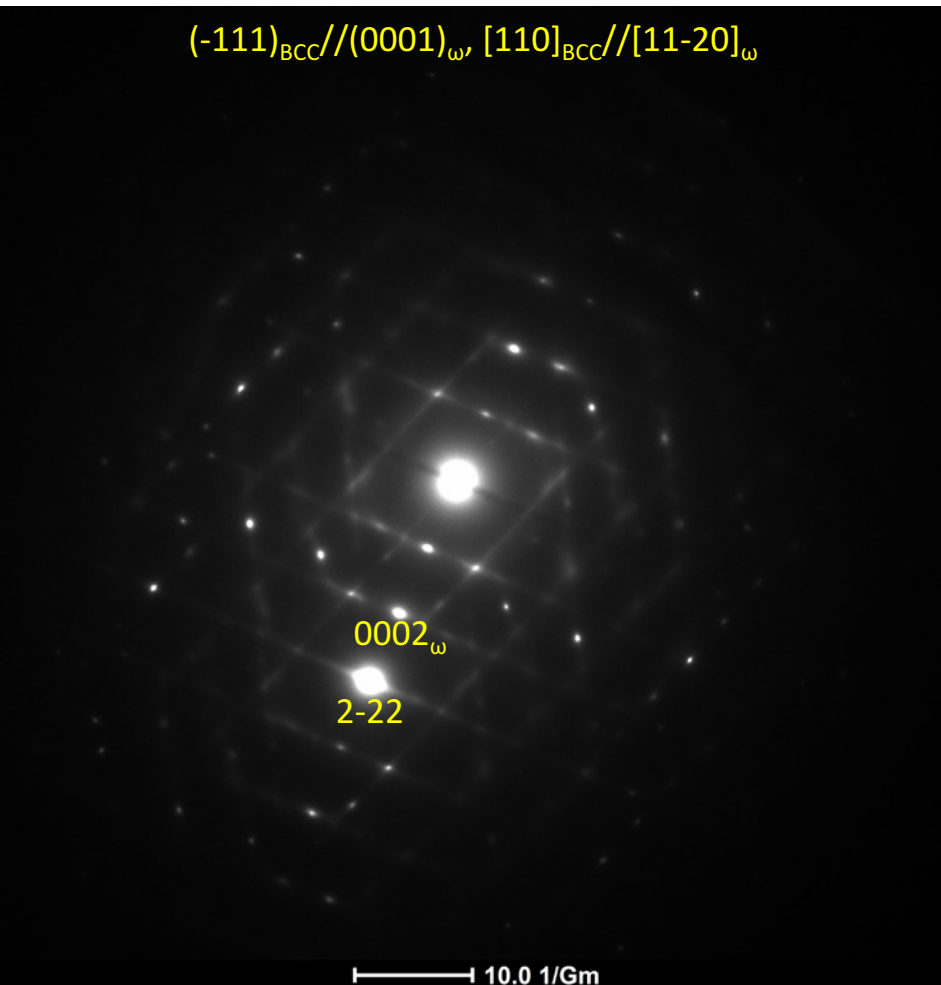




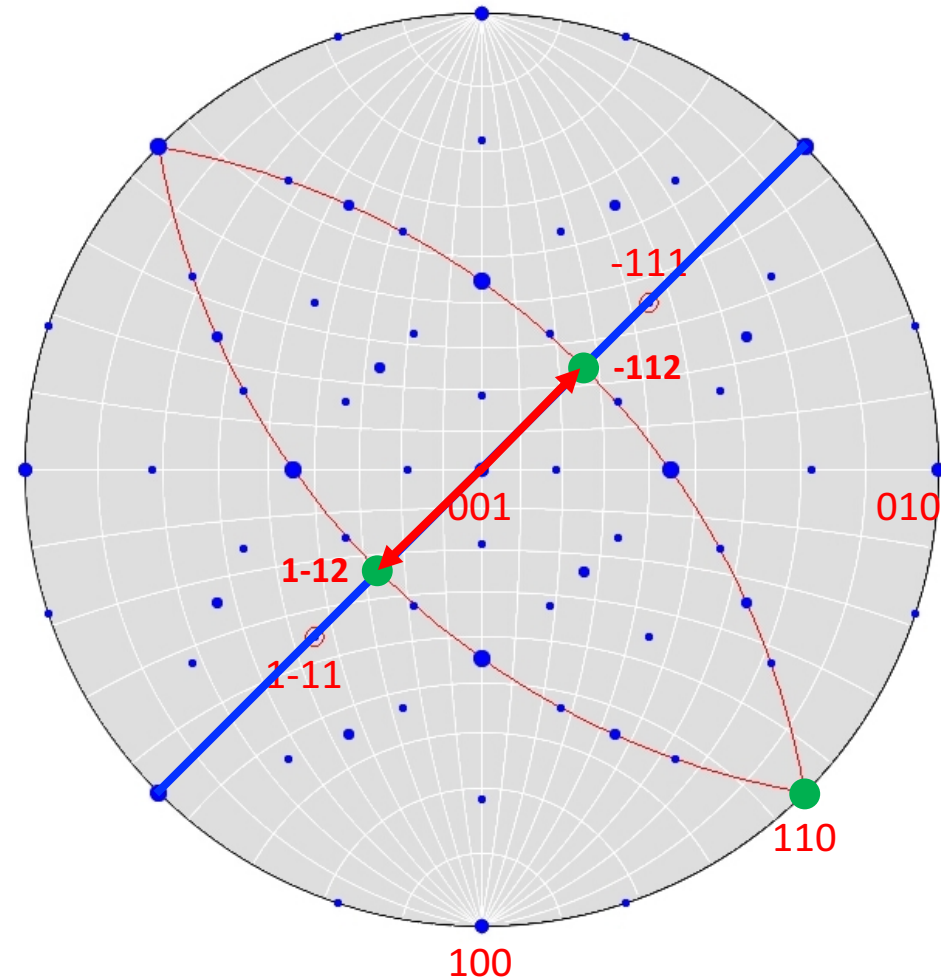
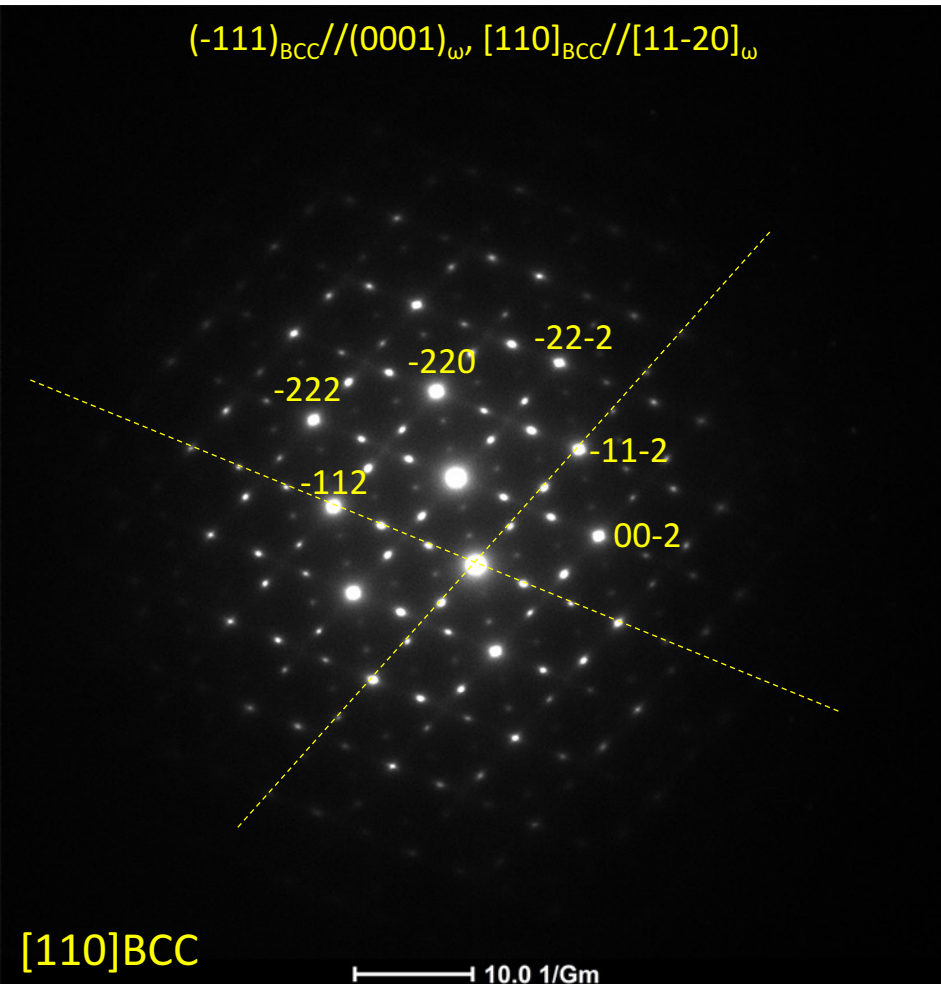
# Distribution of Athermal $\omega$ Phase in Ti-1023



# Morphology of Athermal $\omega$ Phase in Ti-1023

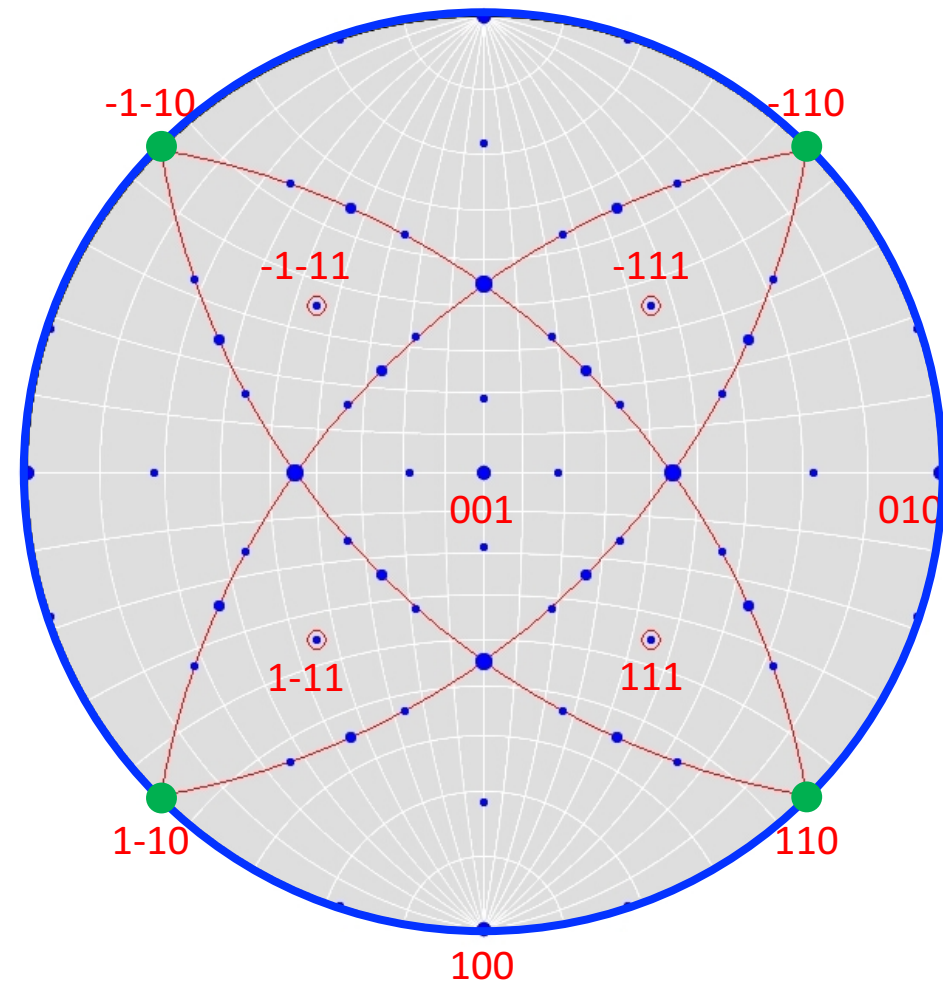
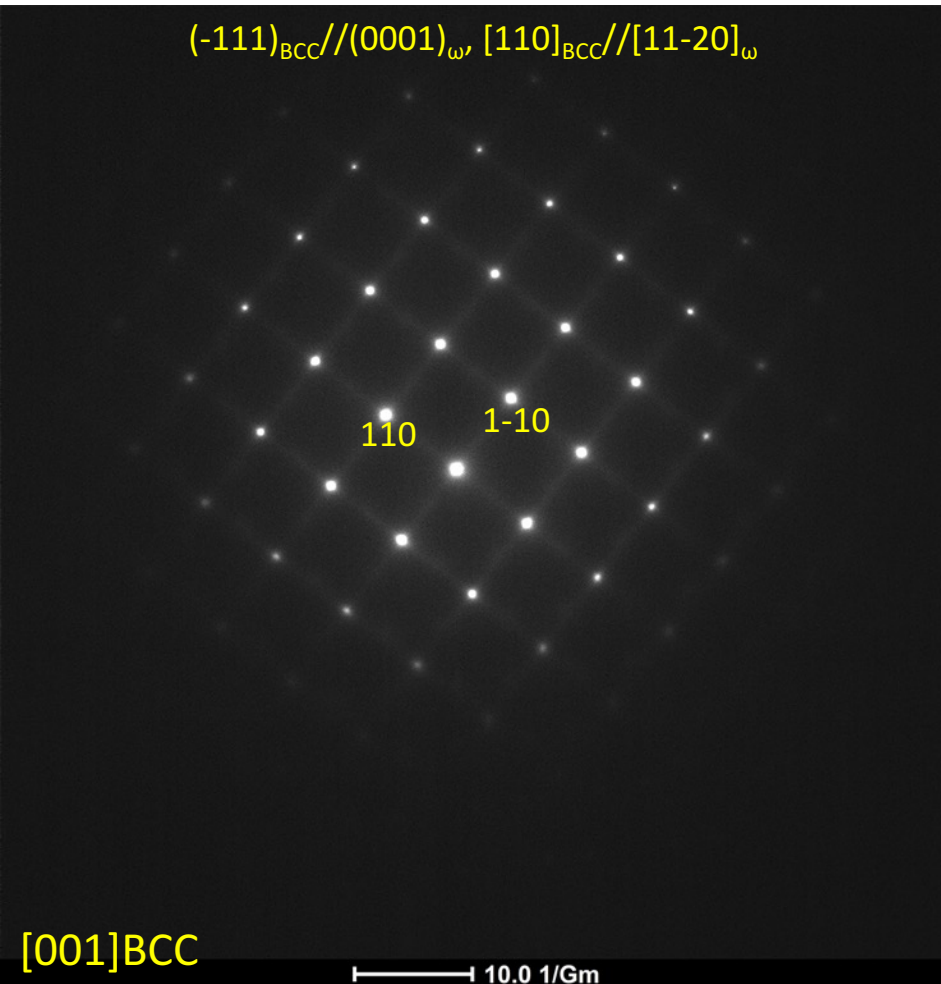


# Streaking in Diffraction Pattern





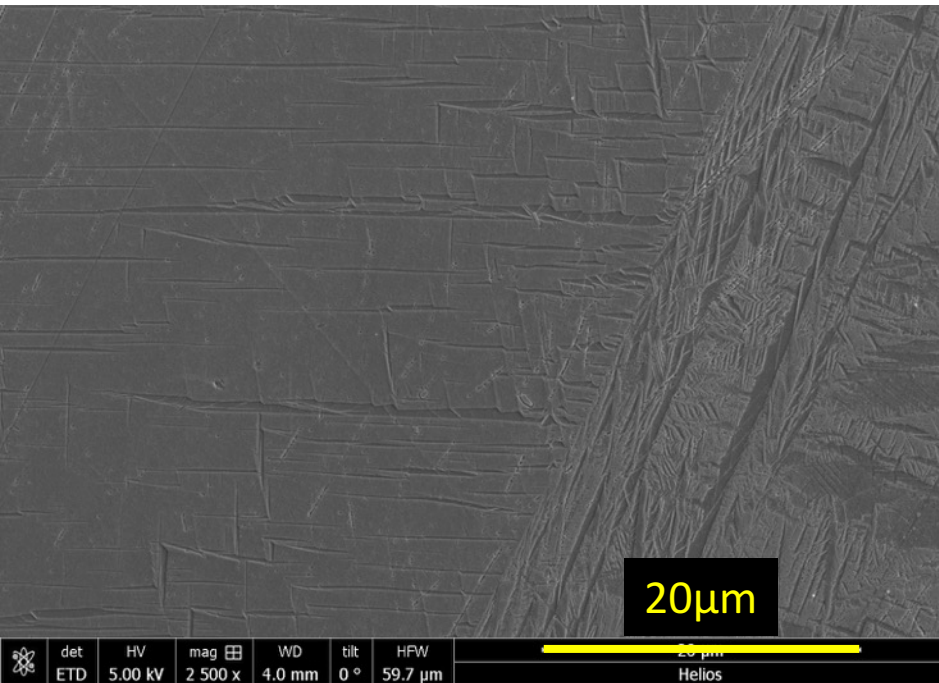
# Streaking in Diffraction Pattern



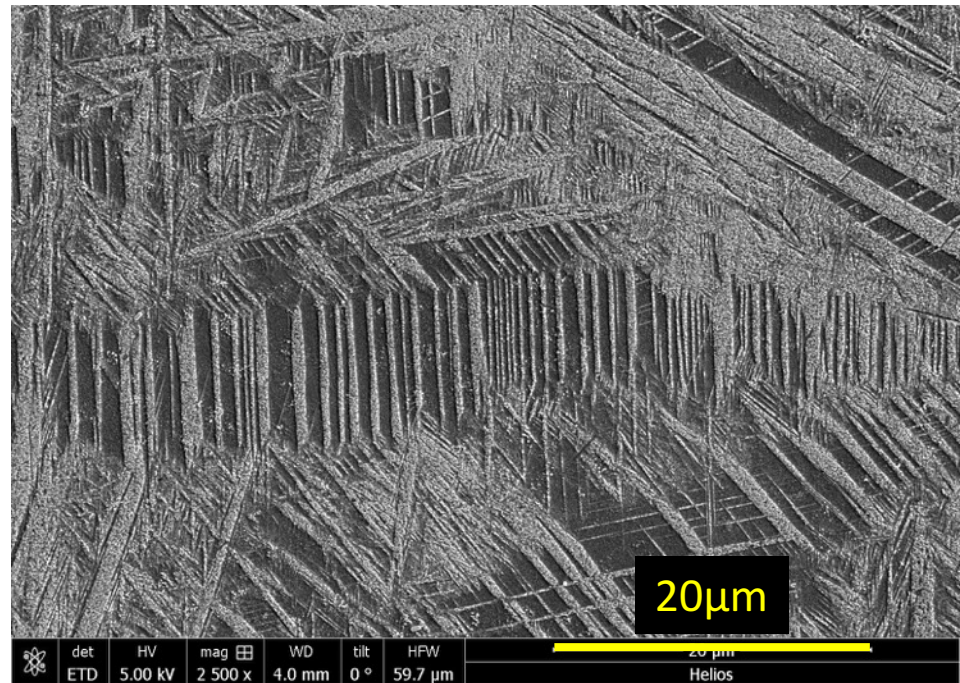
# Deformed Ti-1023 (Tension)

1h@850°C, small grain size

0.5%

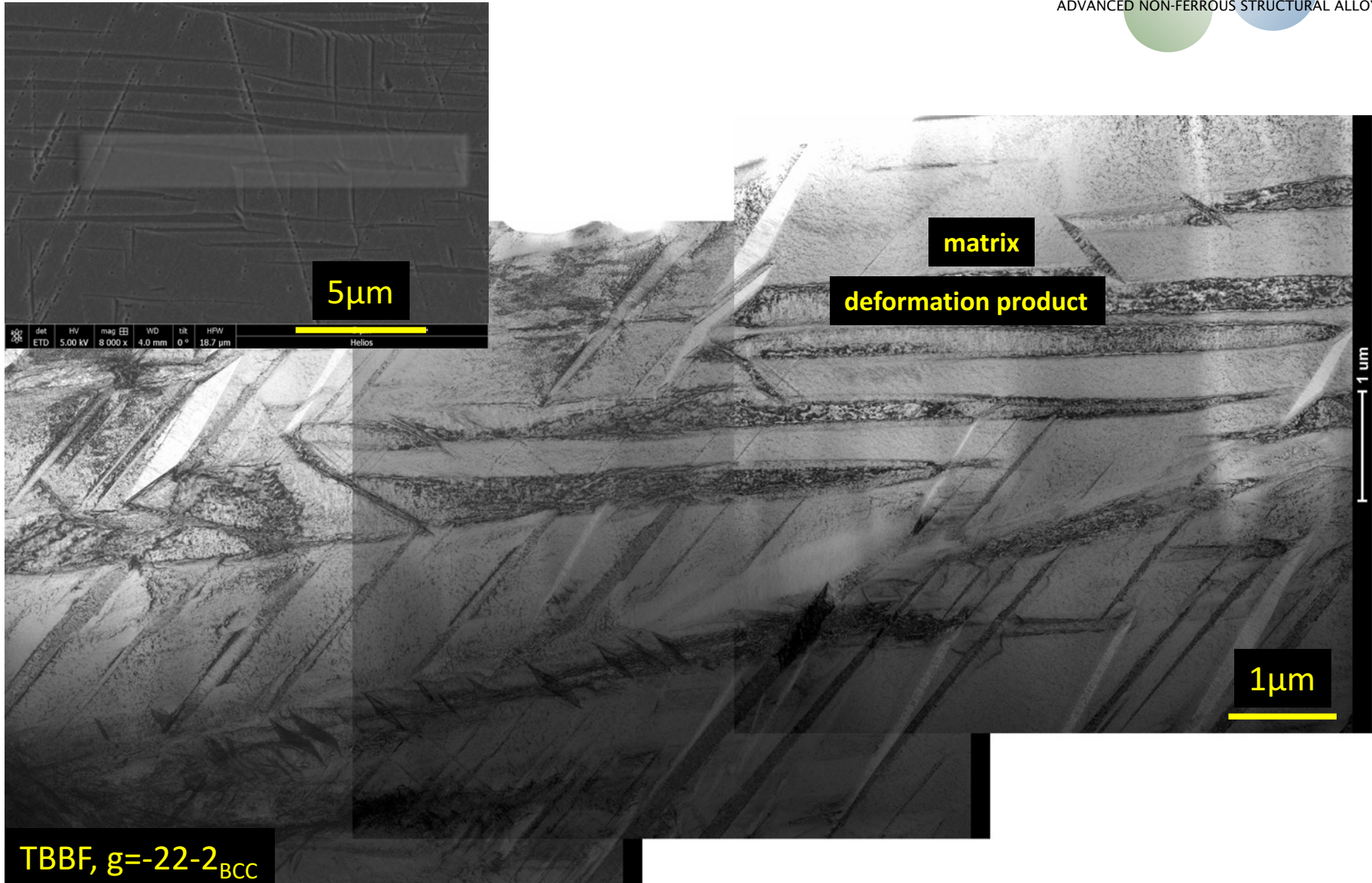


1.7%





# 0.5% Deformed Ti-1023 (Tension)



# Matrix: Undeformed vs 0.5% Deformed

$(-111)_{\text{BCC}} // (0001)_{\omega}$ ,  $[110]_{\text{BCC}} // [11-20]_{\omega}$

-220      -22-2  
-112      -11-2  
00-2

[110]BCC, undeformed

10.0 1/Gm

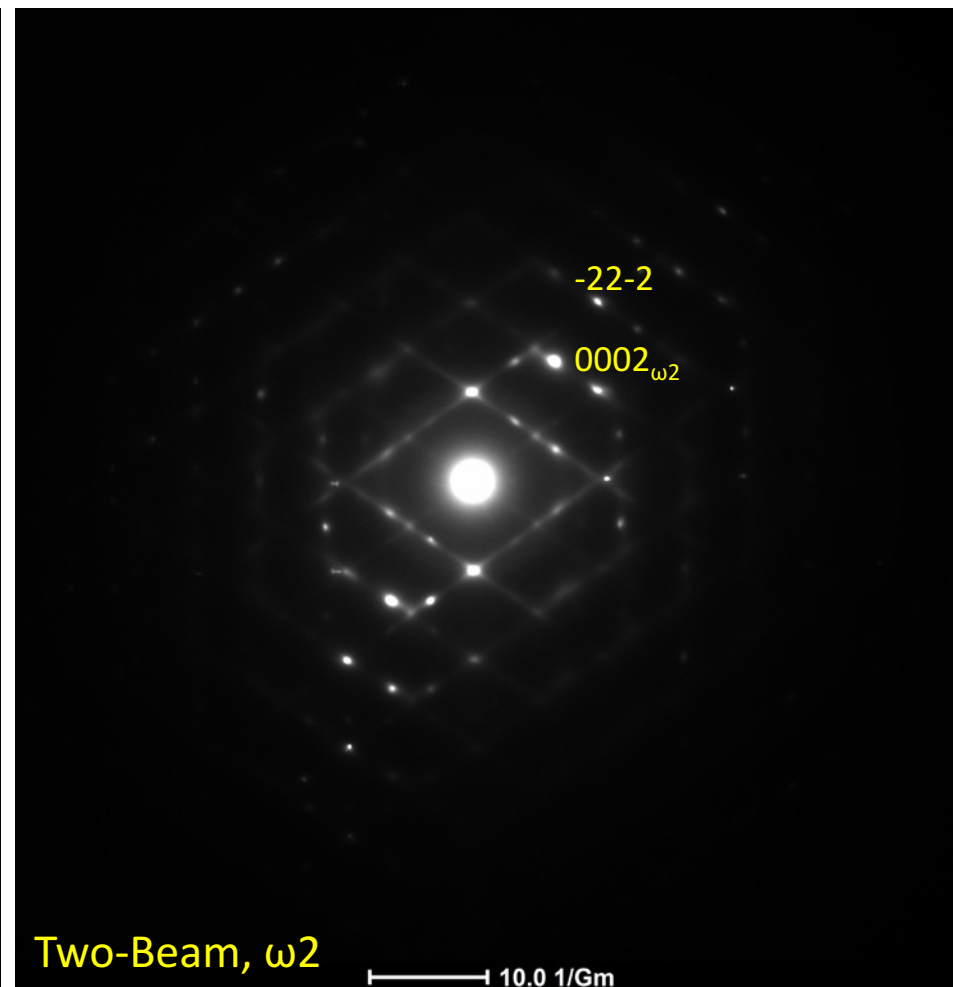
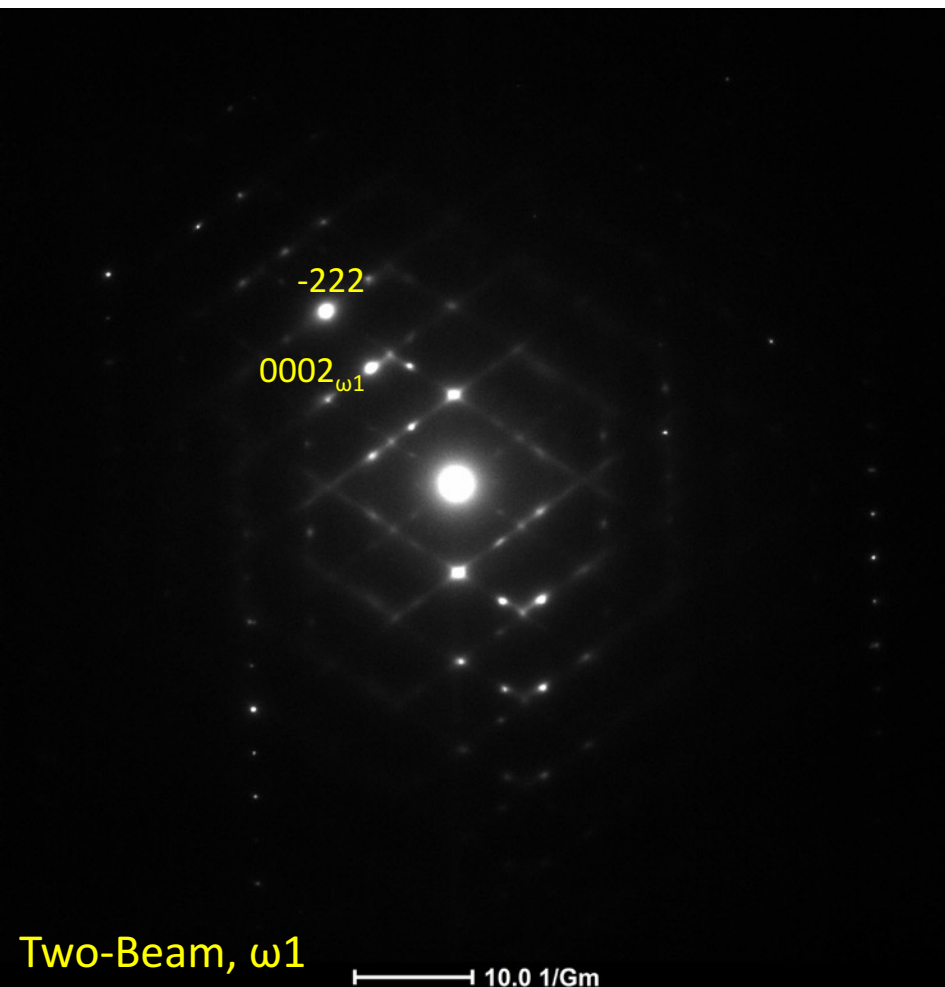
$(-111)_{\text{BCC}} // (0001)_{\omega}$ ,  $[110]_{\text{BCC}} // [11-20]_{\omega}$



[110]BCC, 0.5% deformed

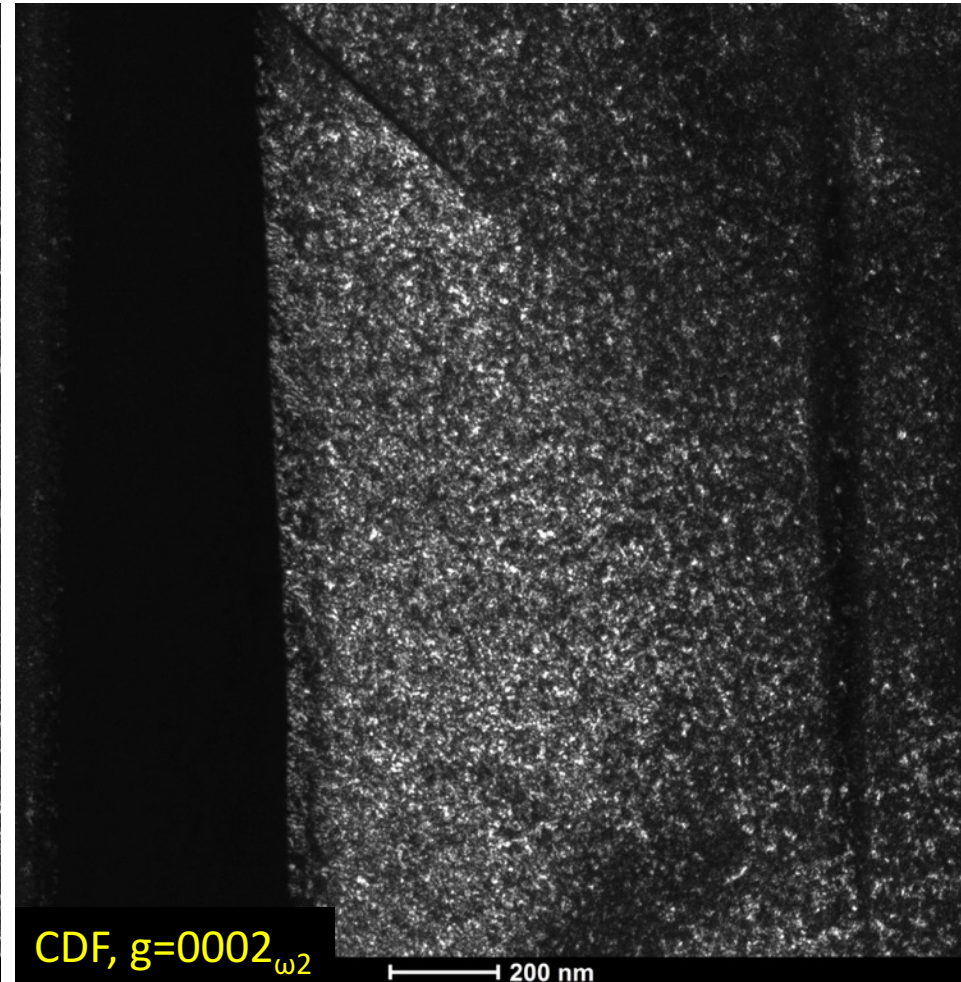
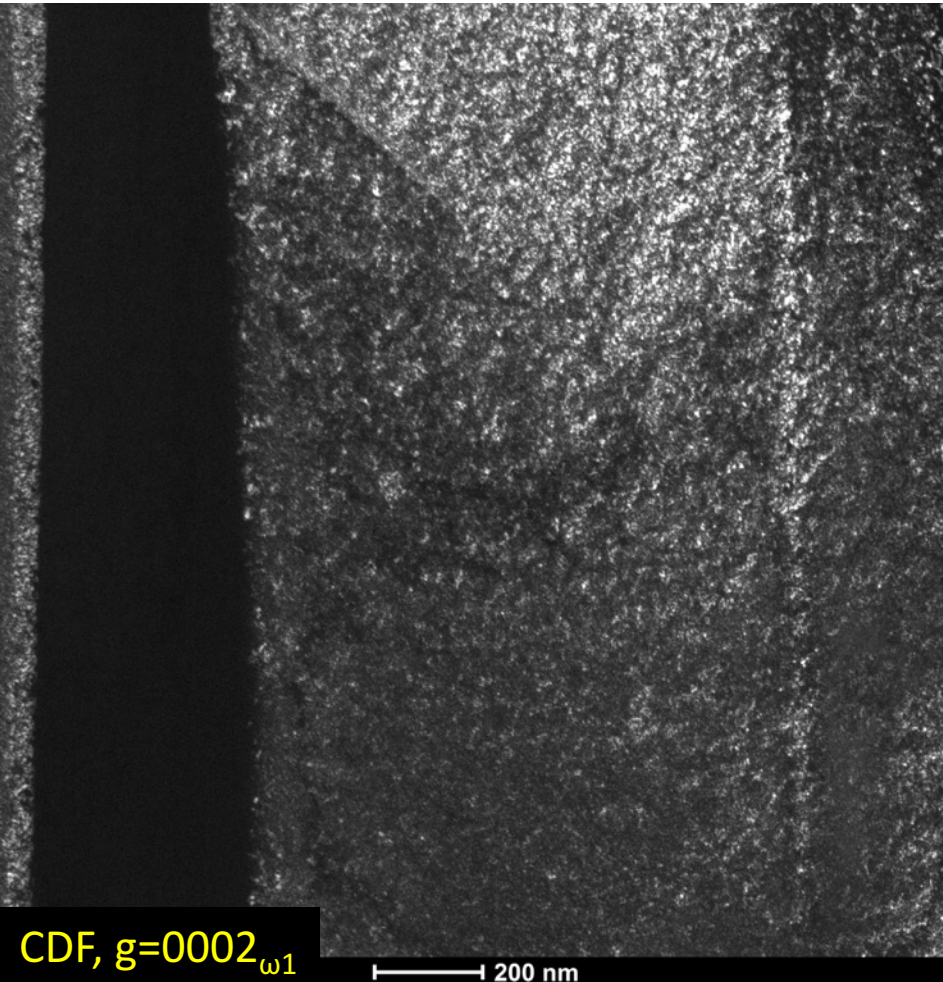
10.0 1/Gm

# Two Variants of Athermal $\omega$ in Ti-1023

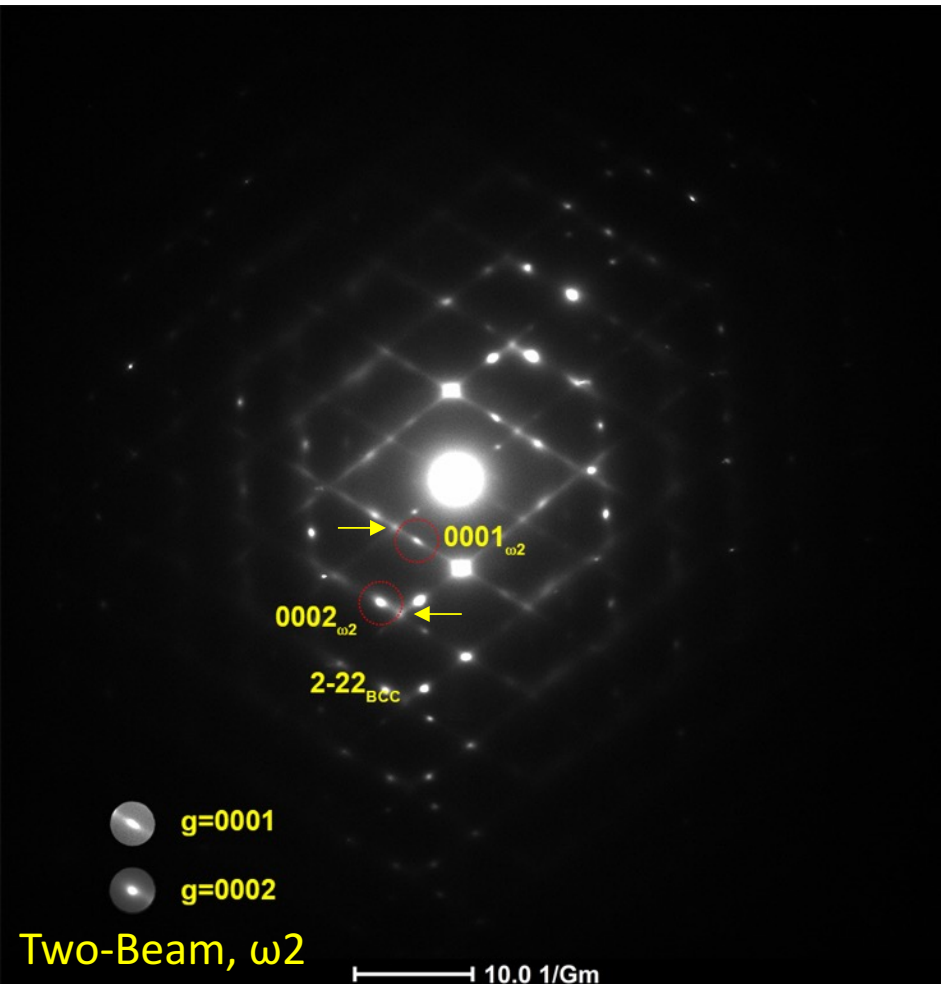




# Distribution of Athermal $\omega$ in 0.5% Deformed Ti-1023

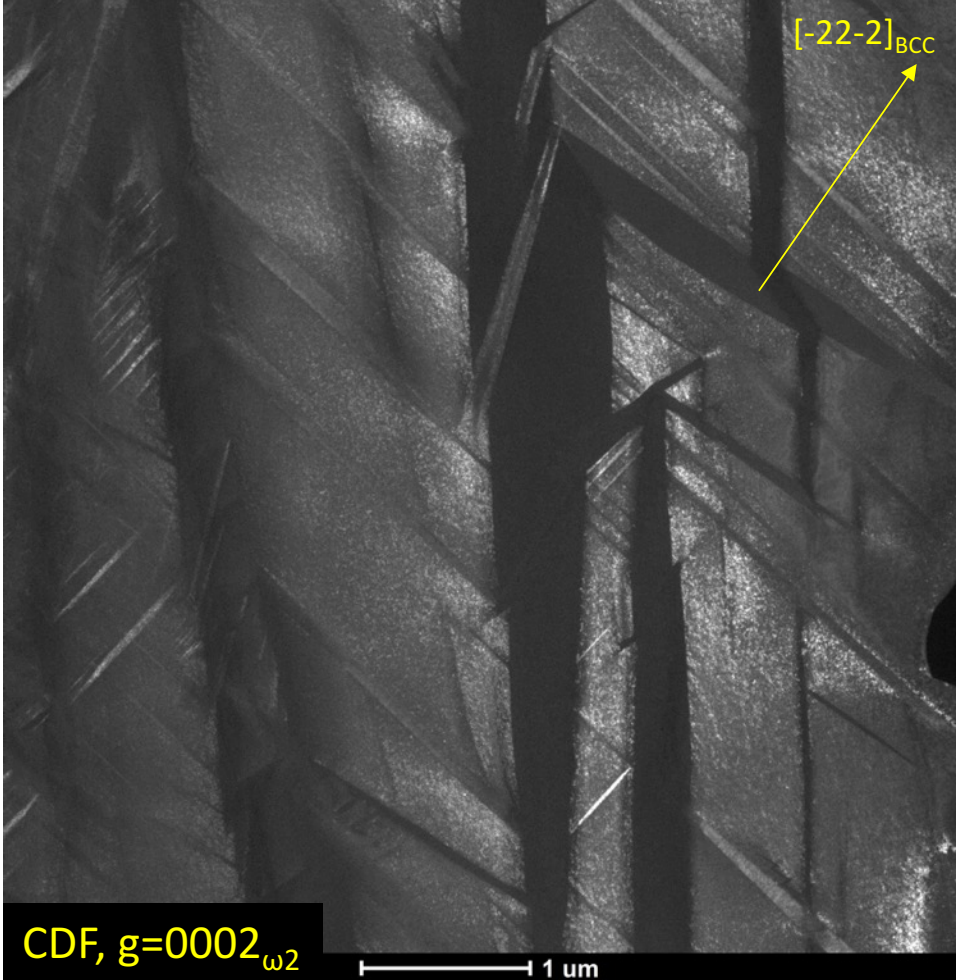
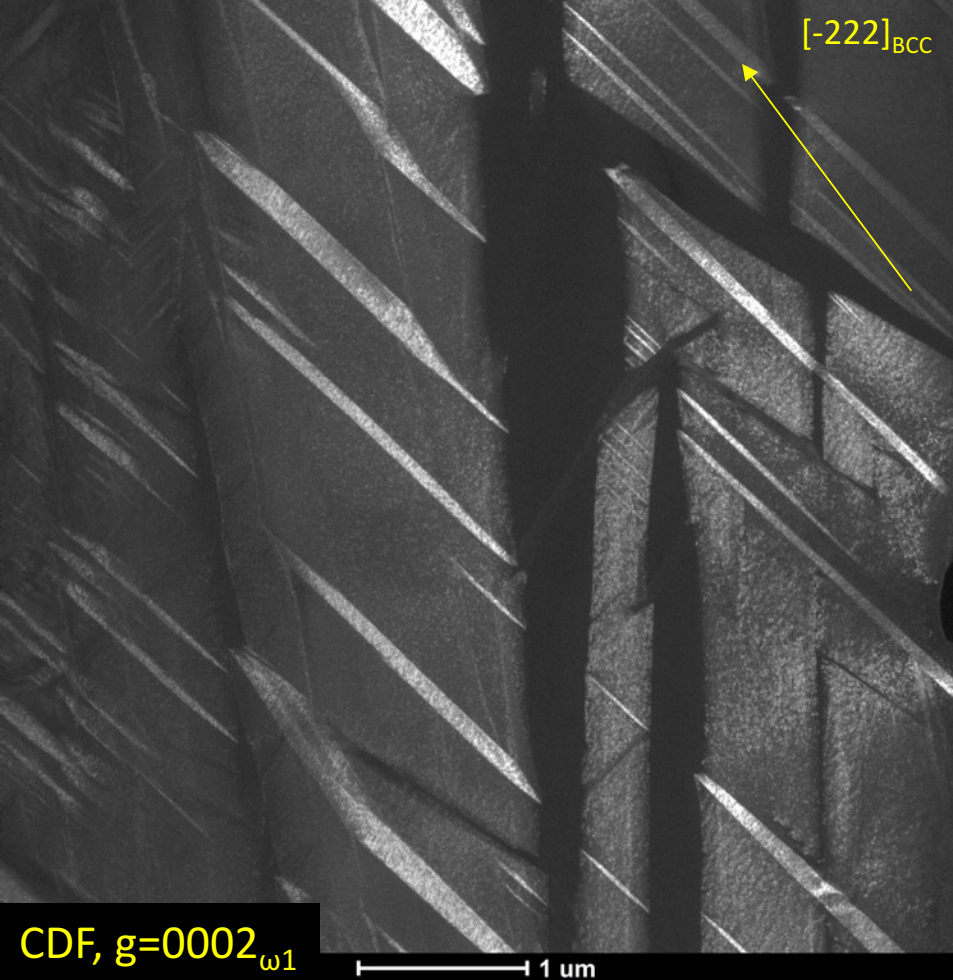


# No $\omega$ Phase Appeared in CDFs



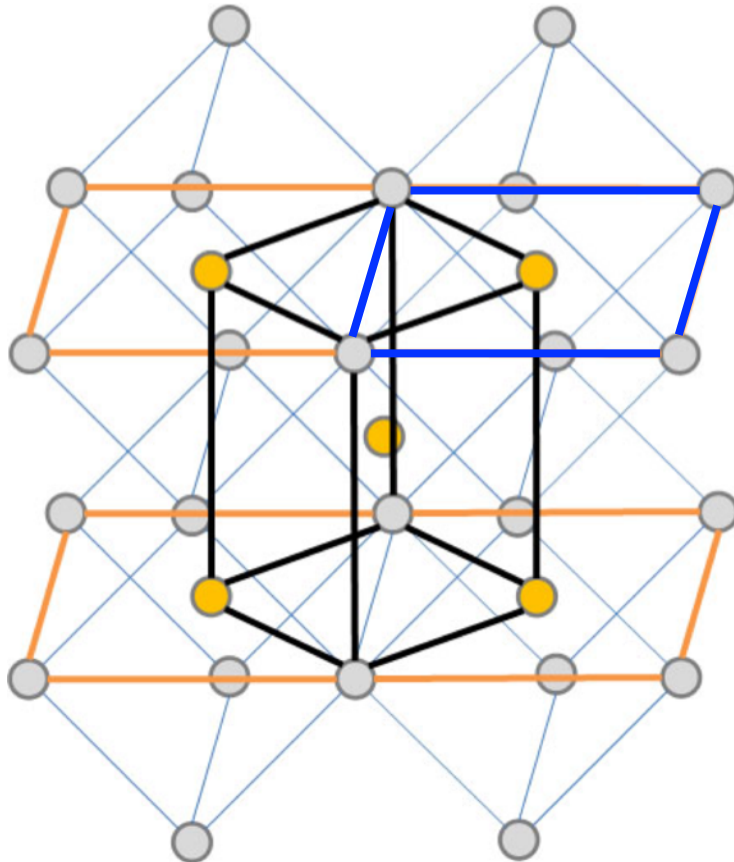


# CDFs using $g=0002_{\omega 1/2}$



# Nature of Martensite: $\alpha'$ or $\alpha''$ ?

- $(110)_{\text{BCC}} // (0001)_{\alpha/\alpha'}$ ,  $[1-11]_{\text{BCC}} // [2-1-10]_{\alpha/\alpha'}$  (Burgers OR)
- $(1-10)_{\text{BCC}} // (010)_{\alpha''}$ ,  $[110]_{\text{BCC}} // [001]_{\alpha''}$
- Duerig\* reported  $\alpha''$  in heat-treated Ti-10-2-3 alloy.

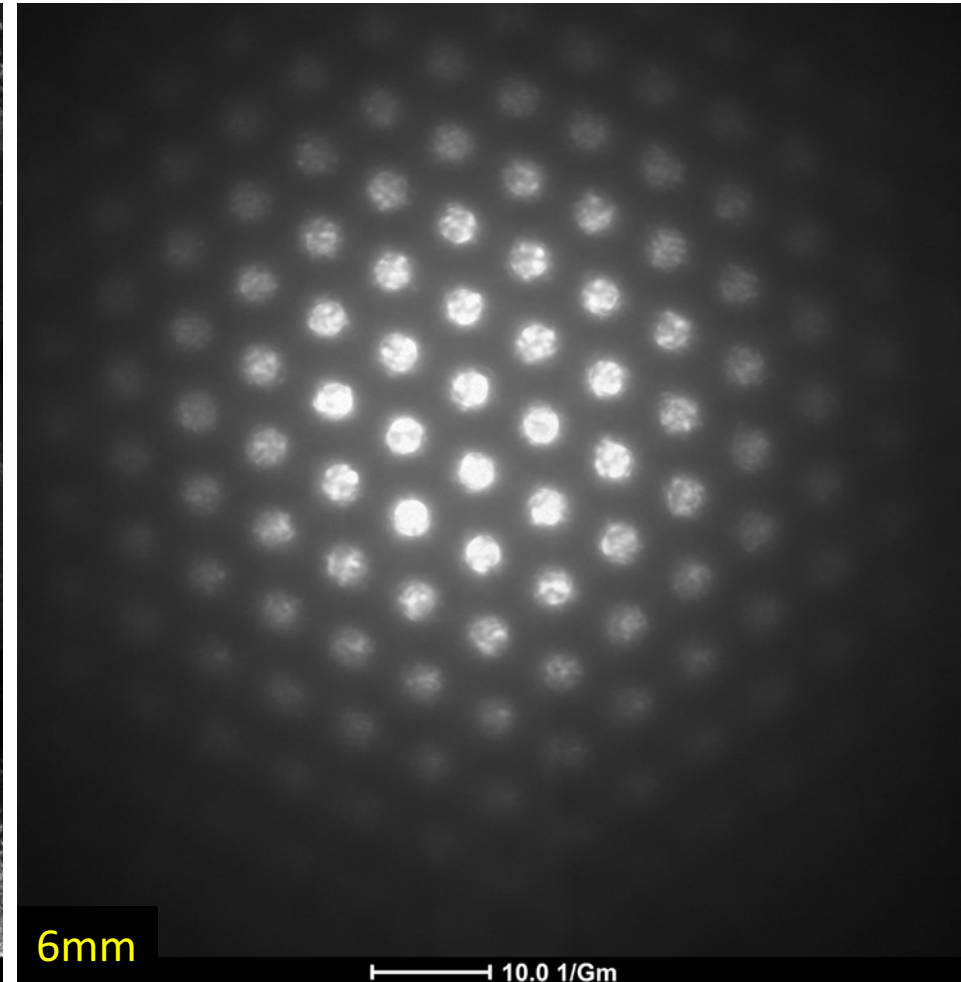
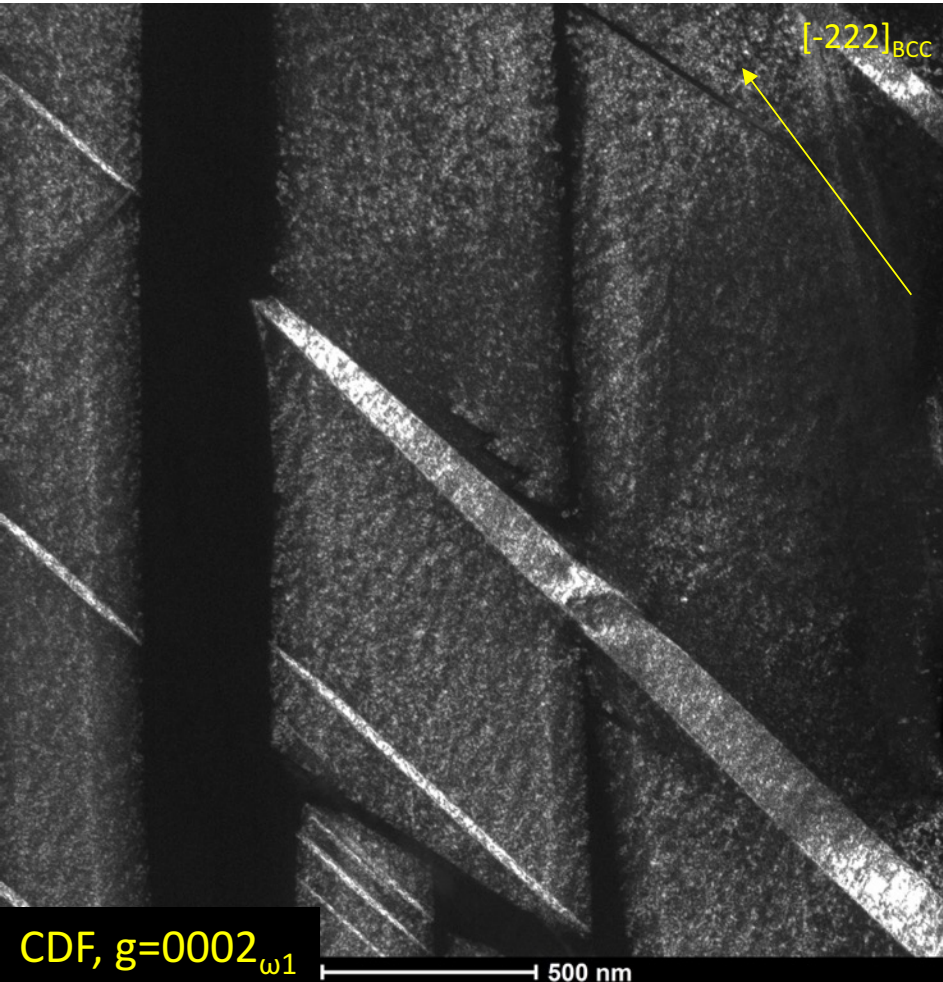


Zone Axis Correspondence between  $\alpha'$  and  $\alpha''$  in  $[110]_{\text{BCC}}$

$\alpha'$ (hex)	$\alpha''$ (ort)
v1, [0001]	v1, [001]
v2, [000-1]	
v3, [10-10]	v2, [010]
v4, [-1010]	
v5, [5-4-13] or [2-11]	v3, [211] v4, [-211] v5, [2-11] v6, [-21-1]
v8, [-541-3] or [-21-1]	
v10, [5-4-1-3] or [2-1-1]	
v12, [-5413] or [-211]	

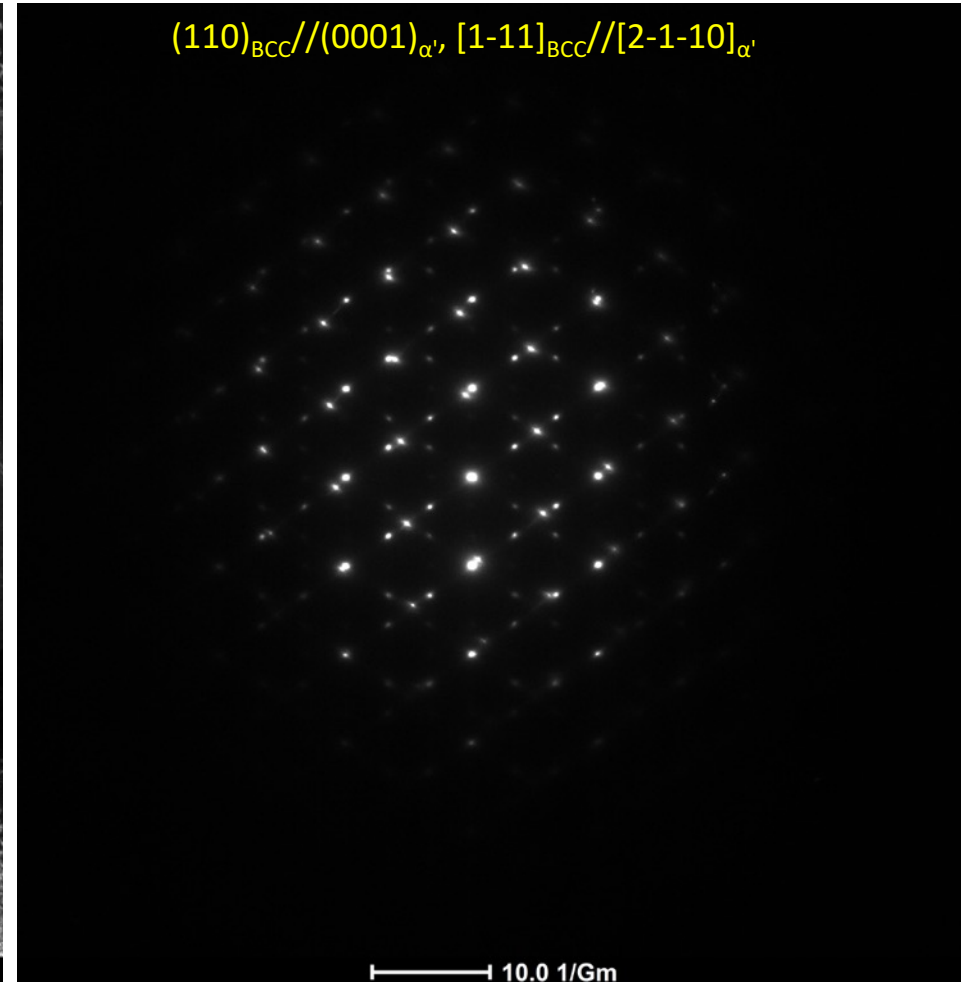
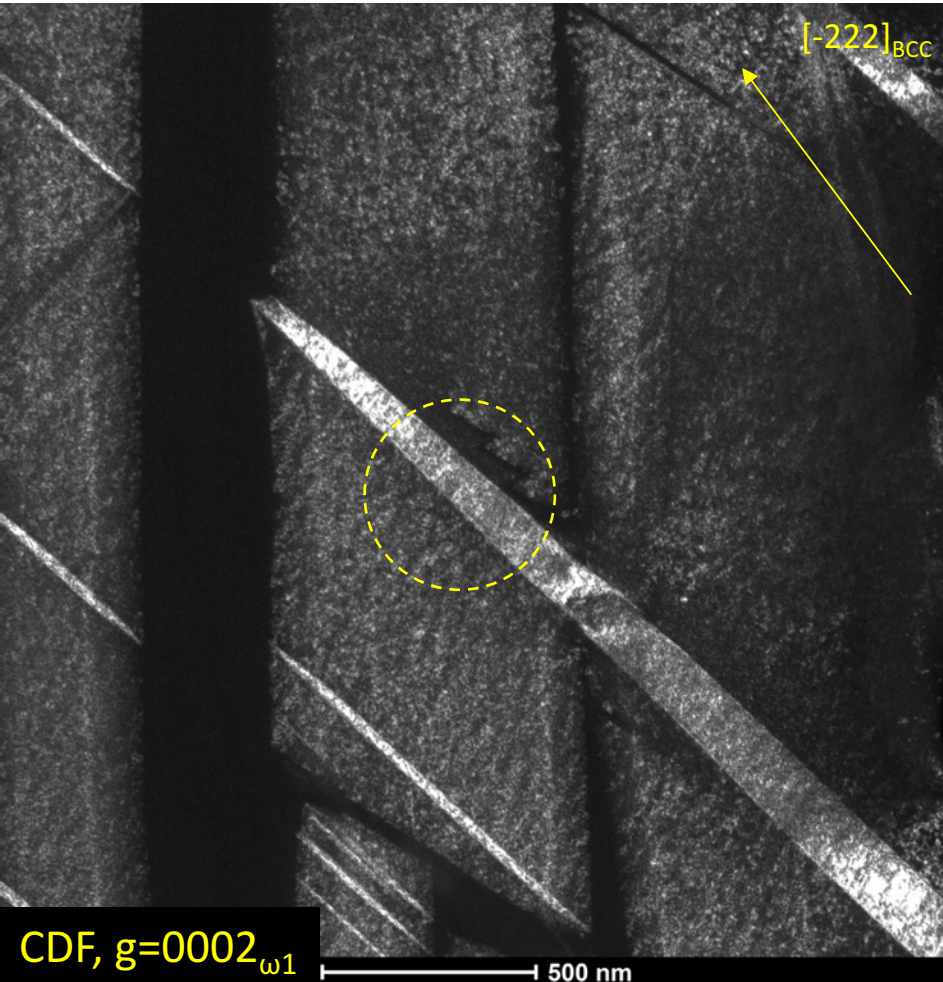
[\*] T.W. Duerig, et al., Acta Metall. 30 (1982) 2161–2172.

# Evidence 1: $[0001]_{\alpha'}$ or $[001]_{\alpha''}$

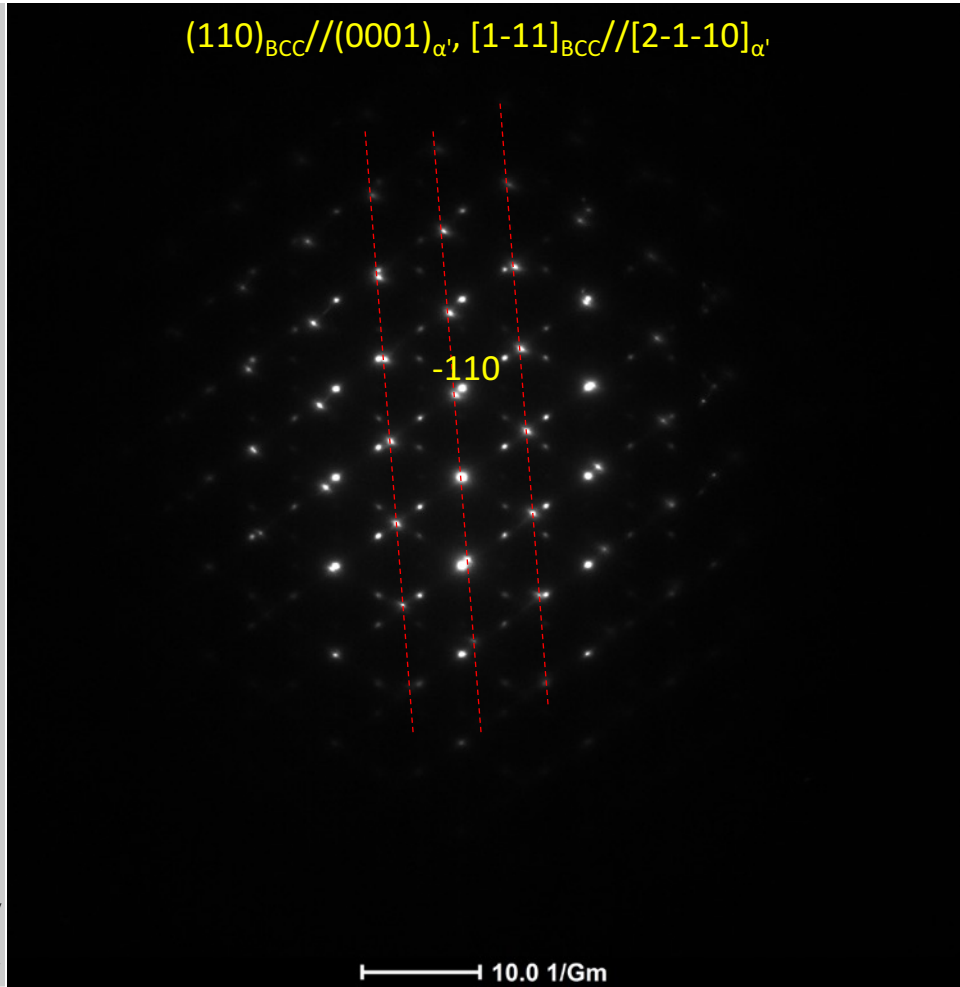
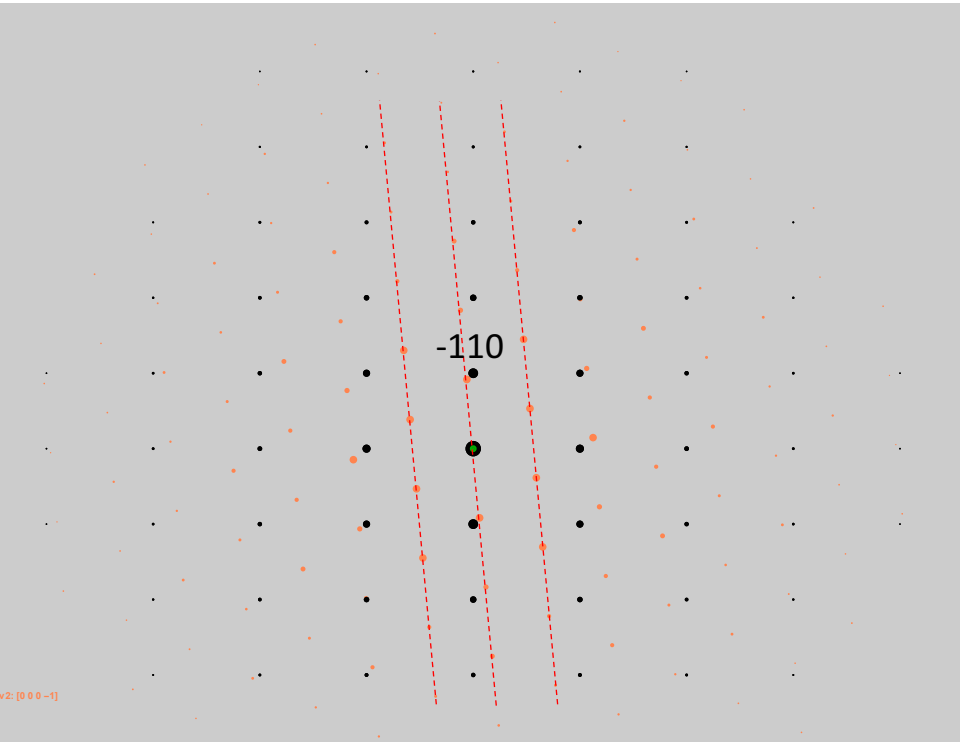




# Evidence 1: $[0001]_{\alpha'}$ or $[001]_{\alpha''}$



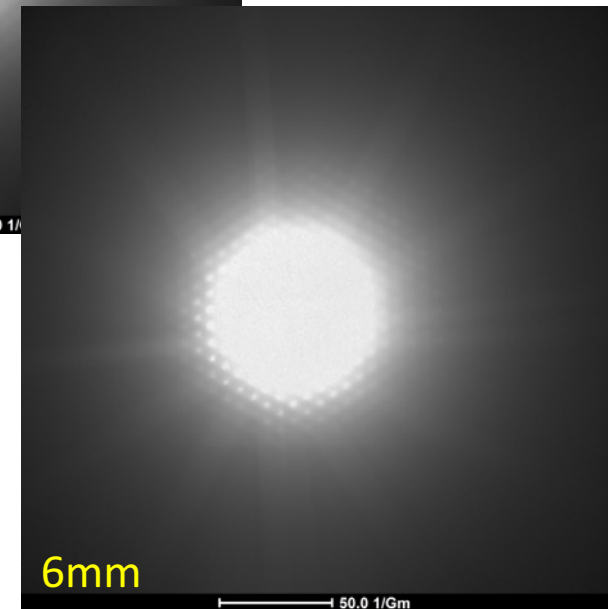
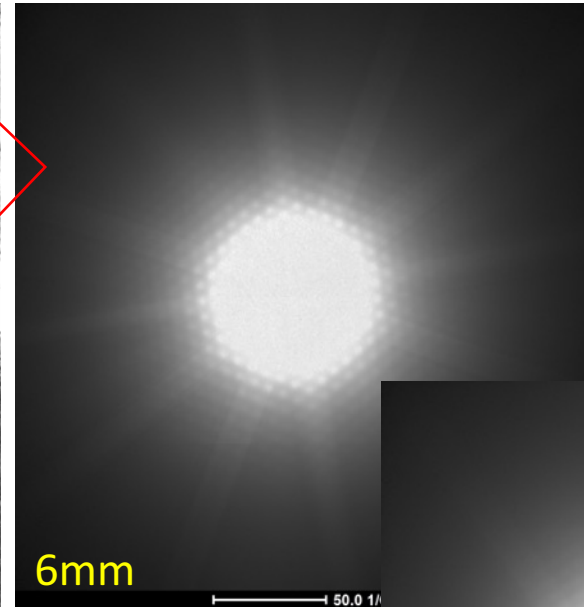
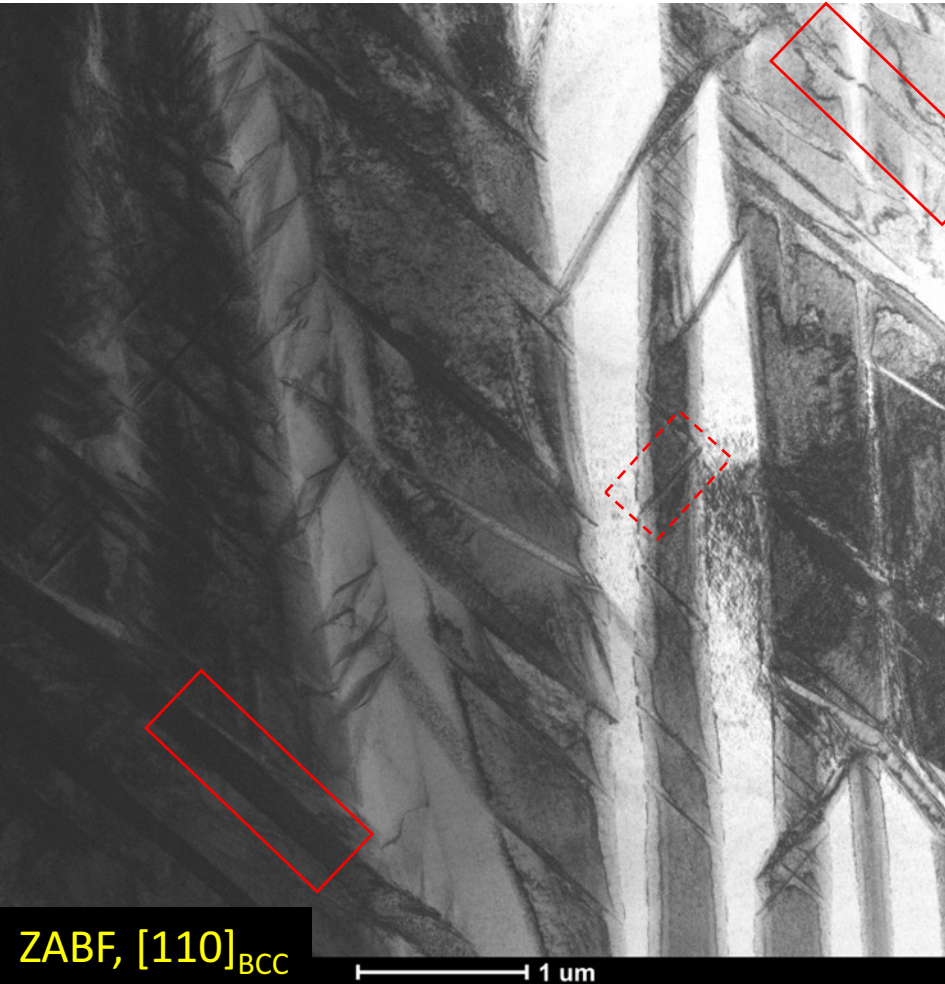
# Evidence 1: $[0001]_{\alpha'}$ or $[001]_{\alpha''}$



Measured (nm)	Reference (nm)
$a_{\text{athermal-}\omega} = 0.45557$	$a_{\omega\text{-Ti}} = 0.4625$
$a_{\gamma} = 0.29080$	$a_{\alpha\text{-Ti}} = 0.295$

5 nm<sup>-1</sup>  
P: bTi, V: aTi  
[1 -1 1] P // [1 0 0] V  
(1 1 0) P // (0 0 1) V  
CL = 1000 nm, 200 kV  
B: [1 1 0] bTi  
F: [1 1 0] bTi  
Sg limit: 1 nm<sup>-1</sup>  
Kikuchi Threshold: 2 V

# Evidence 2: Two Variants

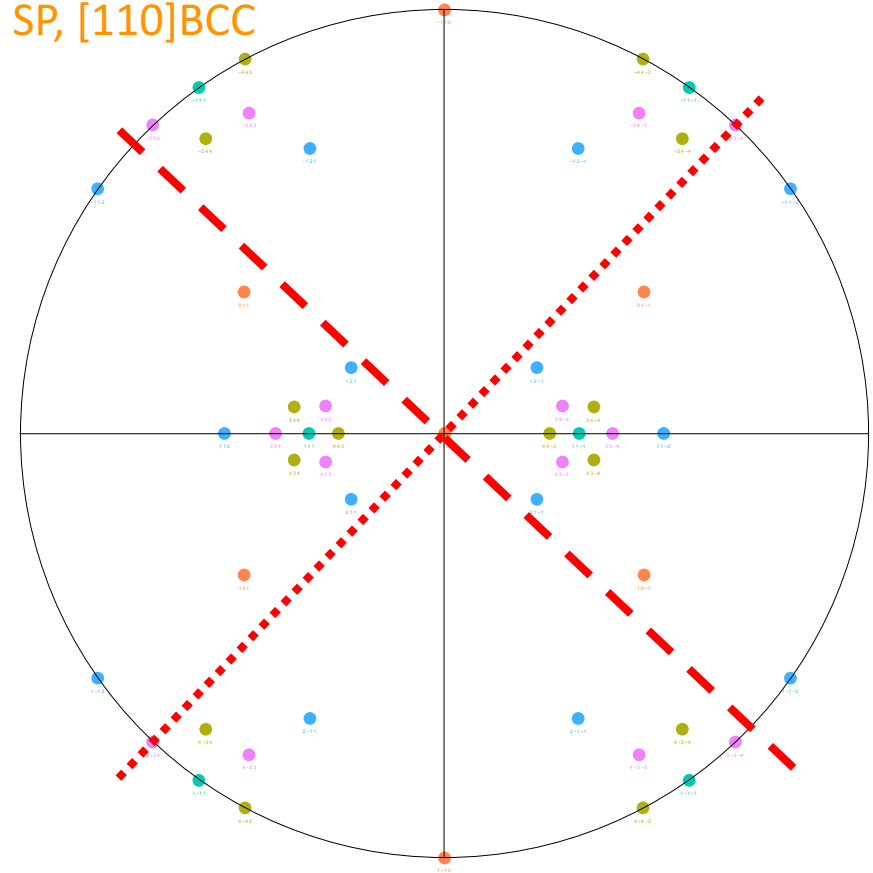




# Evidence 3: Habit Plane



SP,  $[110]_{\text{BCC}}$

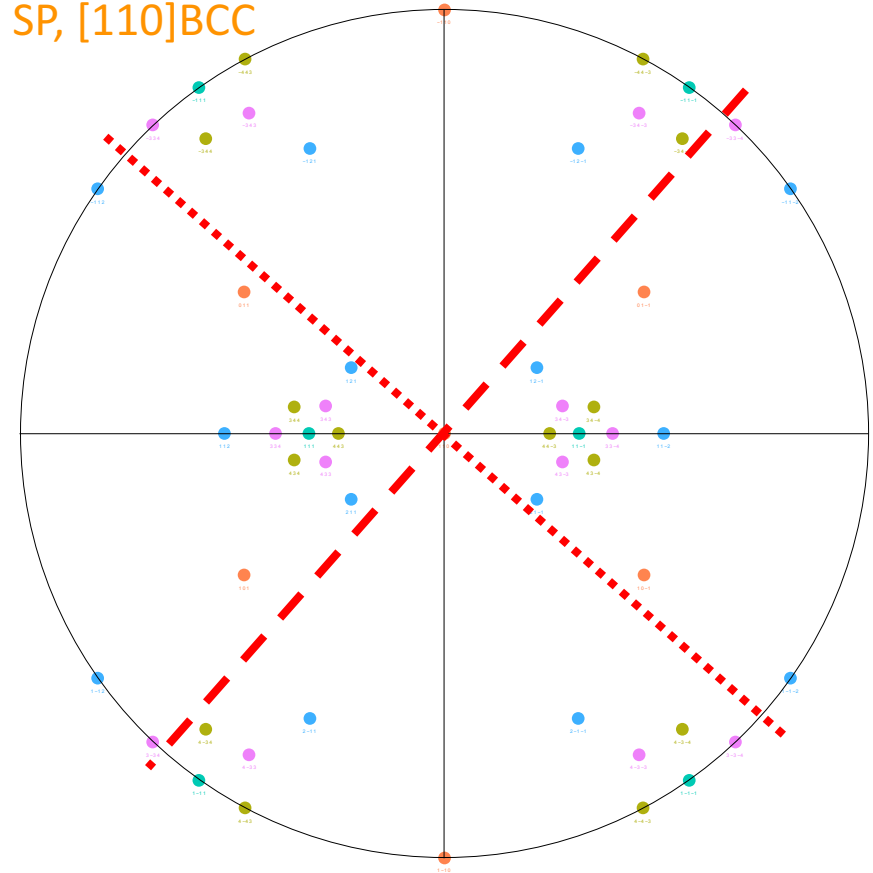


$\{112\}$ ,  $\{110\}$ ,  $\{111\}$ ,  $\{334\}$ ,  $\{344\}$

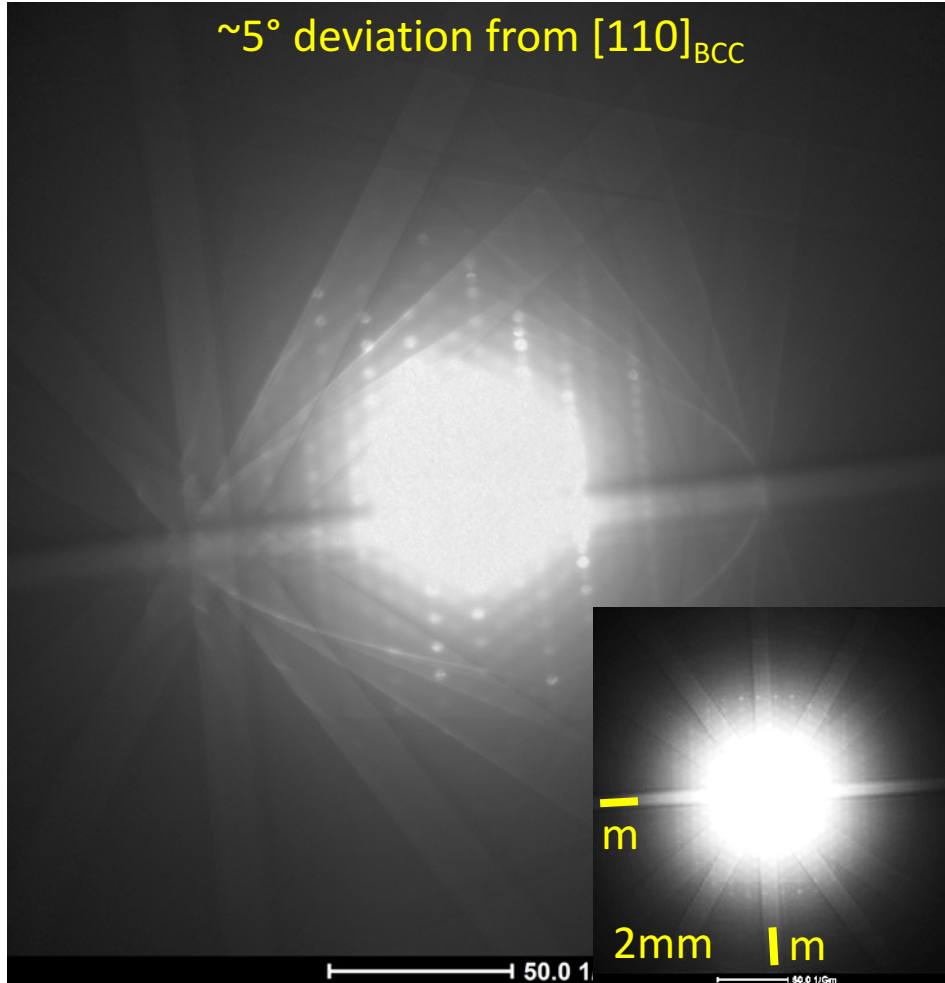
# Evidence 3: Habit Plane



SP,  $[110]_{\text{BCC}}$

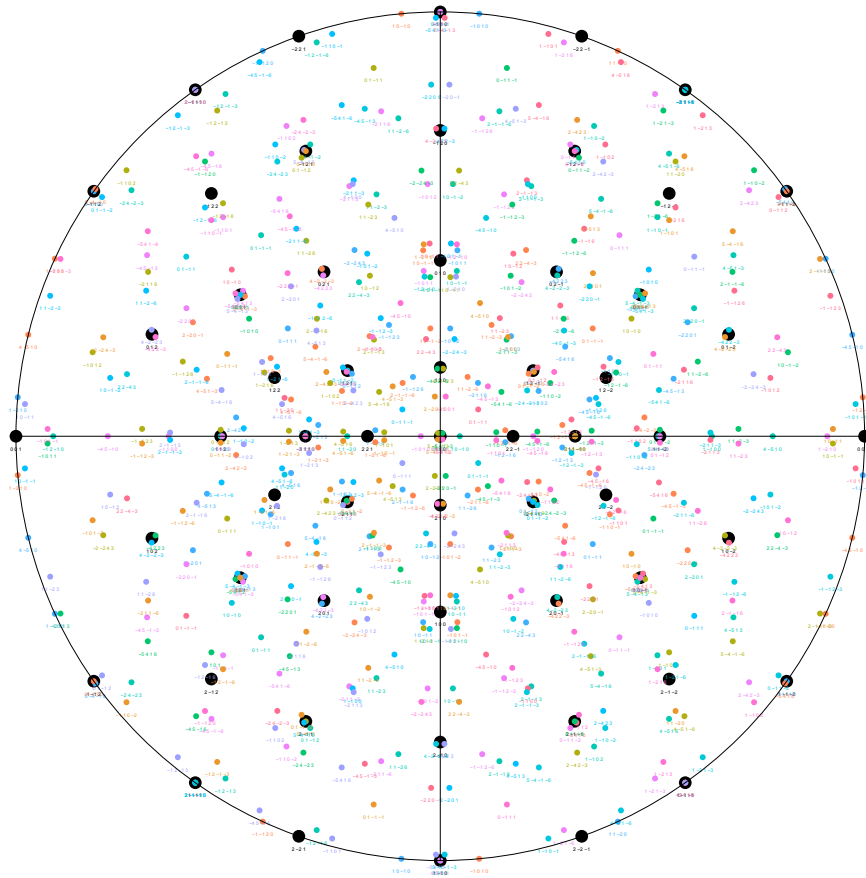


# Evidence 4: $[10-10]_{\alpha'}$ or $[010]_{\alpha''}$



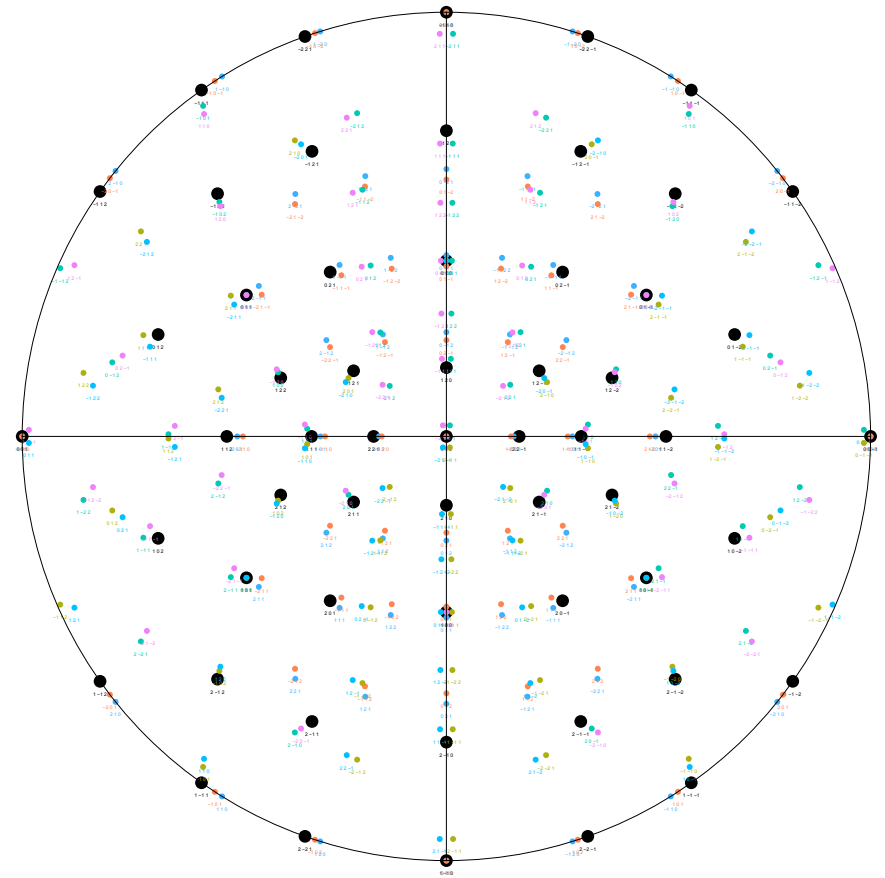
# Evidence 4: $[10-10]_{\alpha'}$ or $[010]_{\alpha''}$

SP,  $[110]_{\text{BCC}}$ , 12  $\alpha'$  variants



view along direction  $[uvw]=[1\ 1\ 0]$  bTl  
upperHS: labeled below  
lowerHS: labeled above and underlined  
directions  $[uvw]$  shown here  
bTl bTl

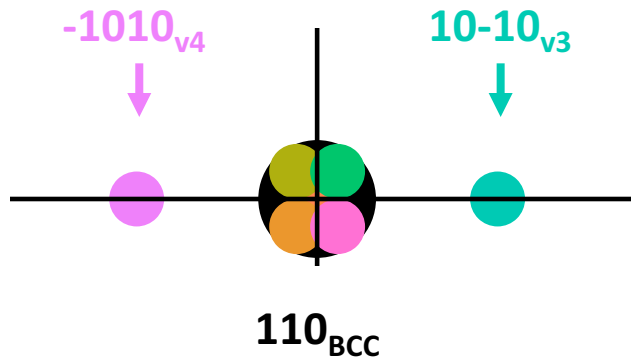
SP,  $[110]_{\text{BCC}}$ , 6  $\alpha''$  variants



view along direction  $[uvw]=[1\ 1\ 0]$  bTl  
upperHS: labeled below  
lowerHS: labeled above and underlined  
directions  $[uvw]$  shown here  
bTl bTl

# Evidence 4: $[10-10]_{\alpha'}$ or $[010]_{\alpha''}$

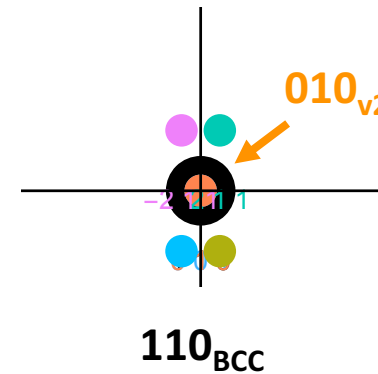
SP,  $[110]_{\text{BCC}}$ , 12  $\alpha'$  variants



~5° deviation

P, ind=1   V4, ind=5   V8, ind=9   V12, ind=13  
 V1, ind=2   V5, ind=6   V9, ind=10  
 V2, ind=3   V6, ind=7   V10, ind=11  
 V3, ind=4   V7, ind=8   V11, ind=12

SP,  $[110]_{\text{BCC}}$ , 6  $\alpha''$  variants

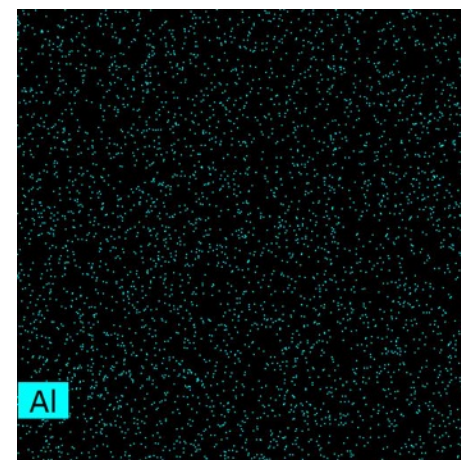
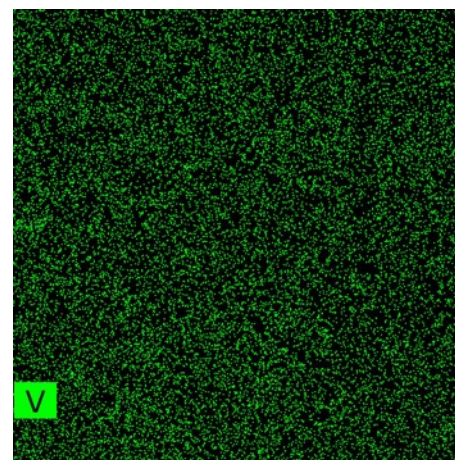
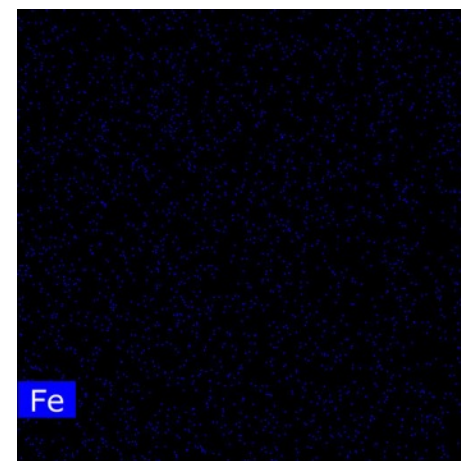
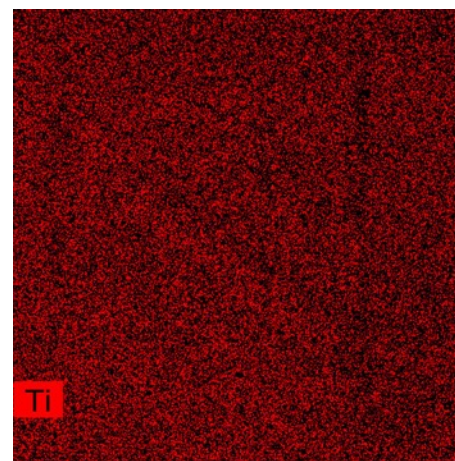
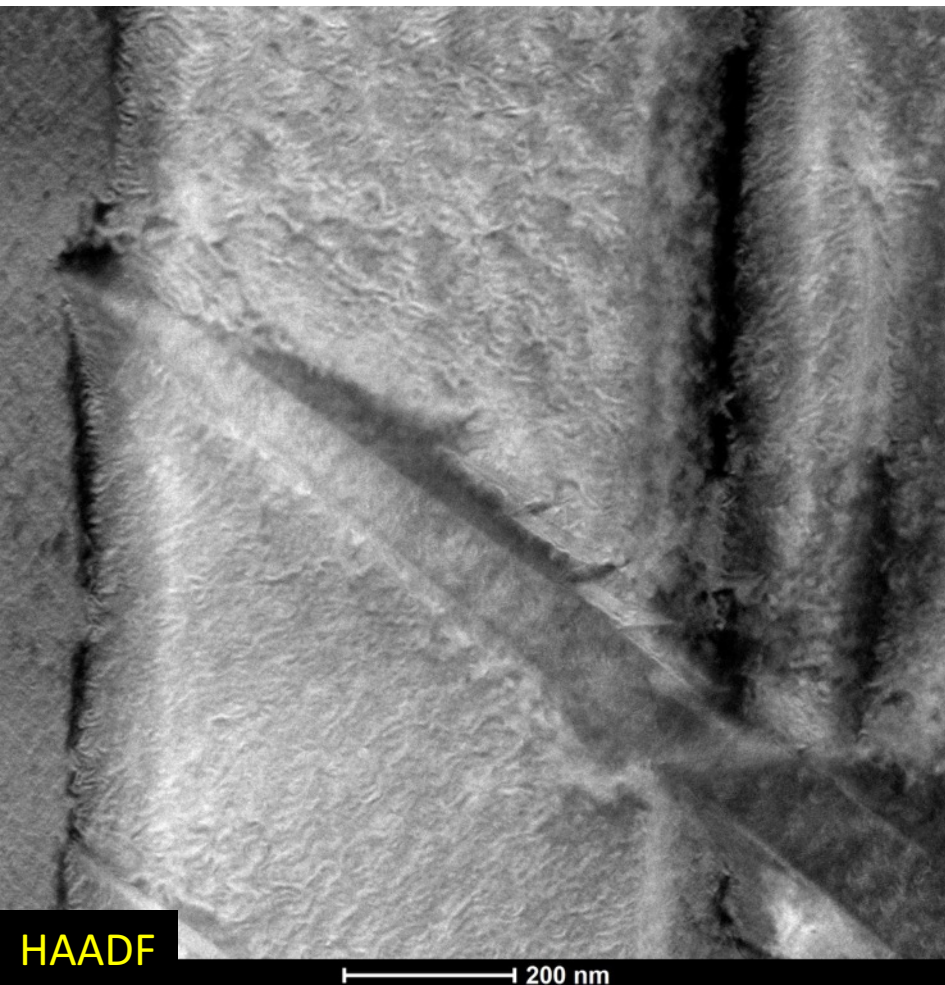


exactly parallel

P, ind=1   V3, ind=4   V6, ind=7  
 V1, ind=2   V4, ind=5  
 V2, ind=3   V5, ind=6

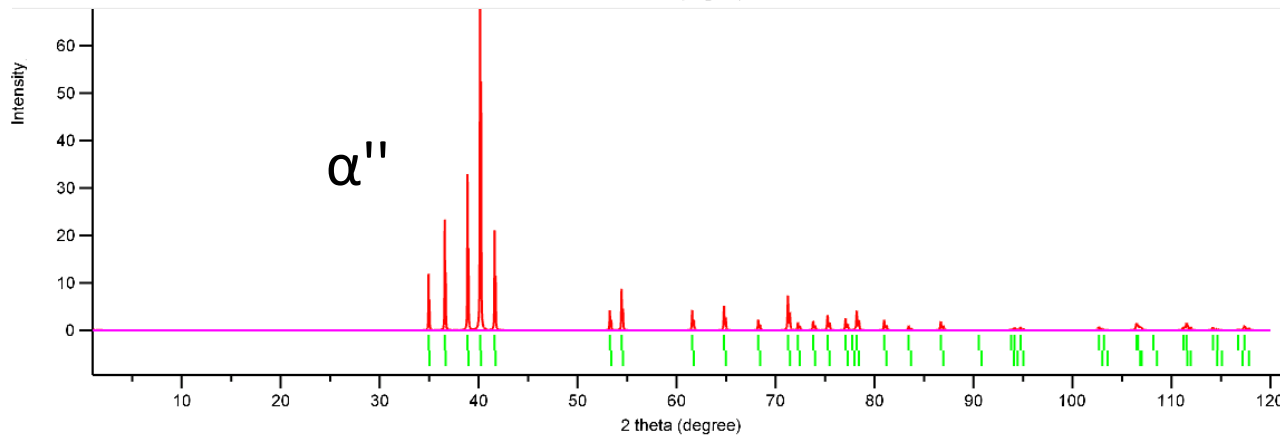
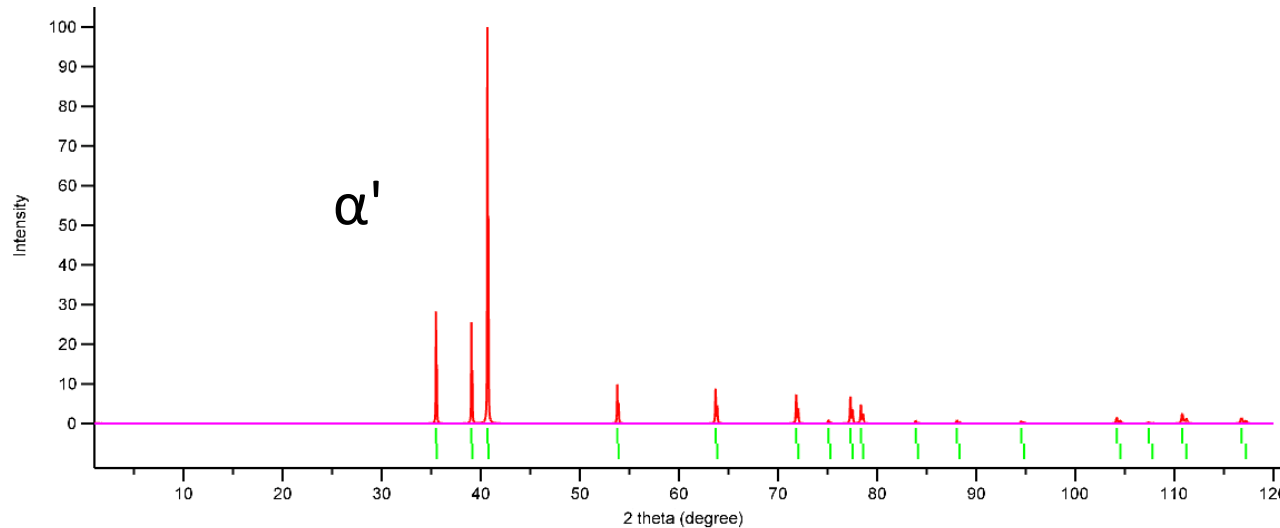


# Evidence 5: Diffusionless Transformation



# Conclusion

- The deformation product is more likely to be hexagonal  $\alpha'$  instead of orthorhombic  $\alpha''$ .
- **XRD** will be done to check whether it is  $\alpha'$  or  $\alpha''$ .

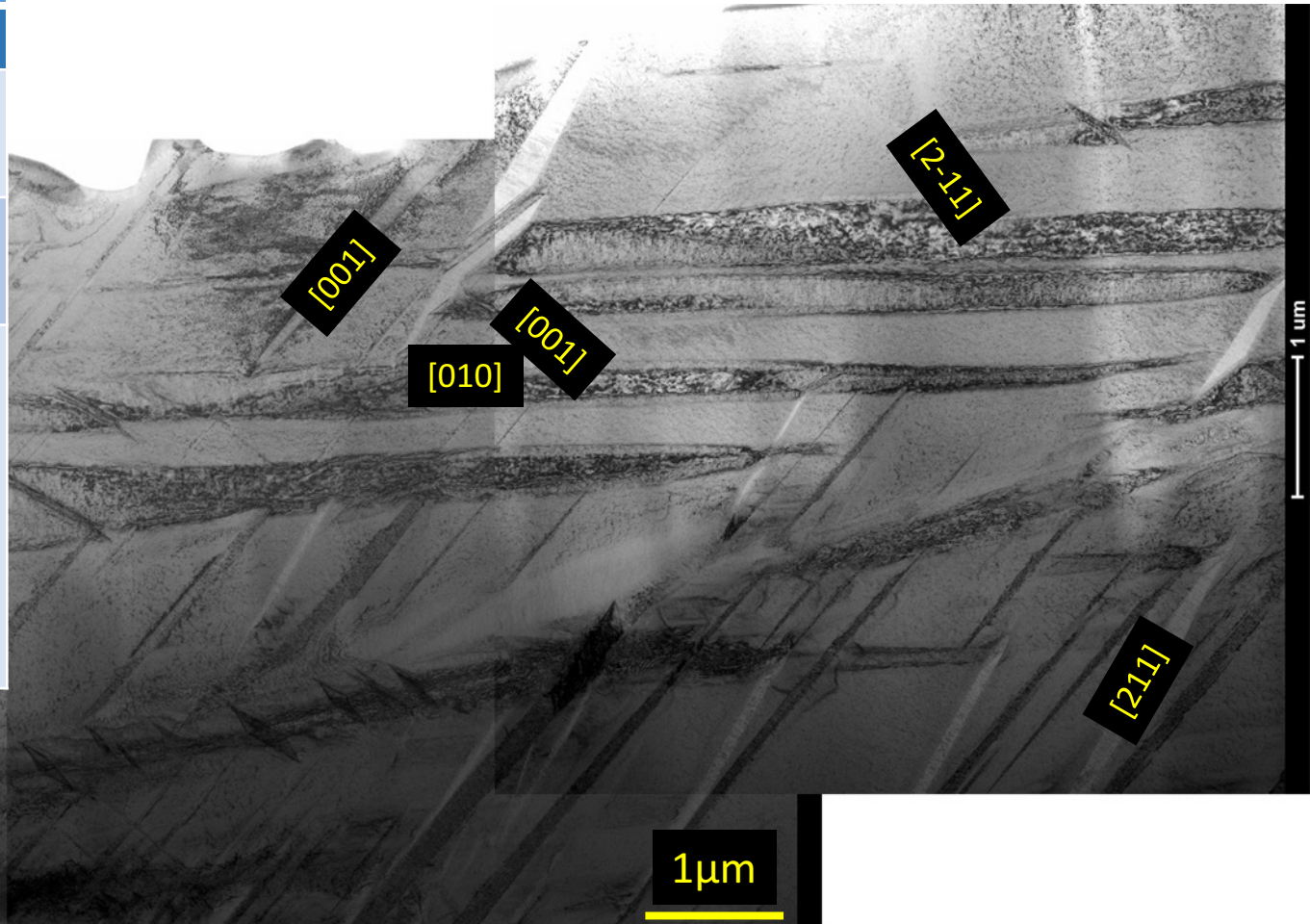




# Different Martensite Variants

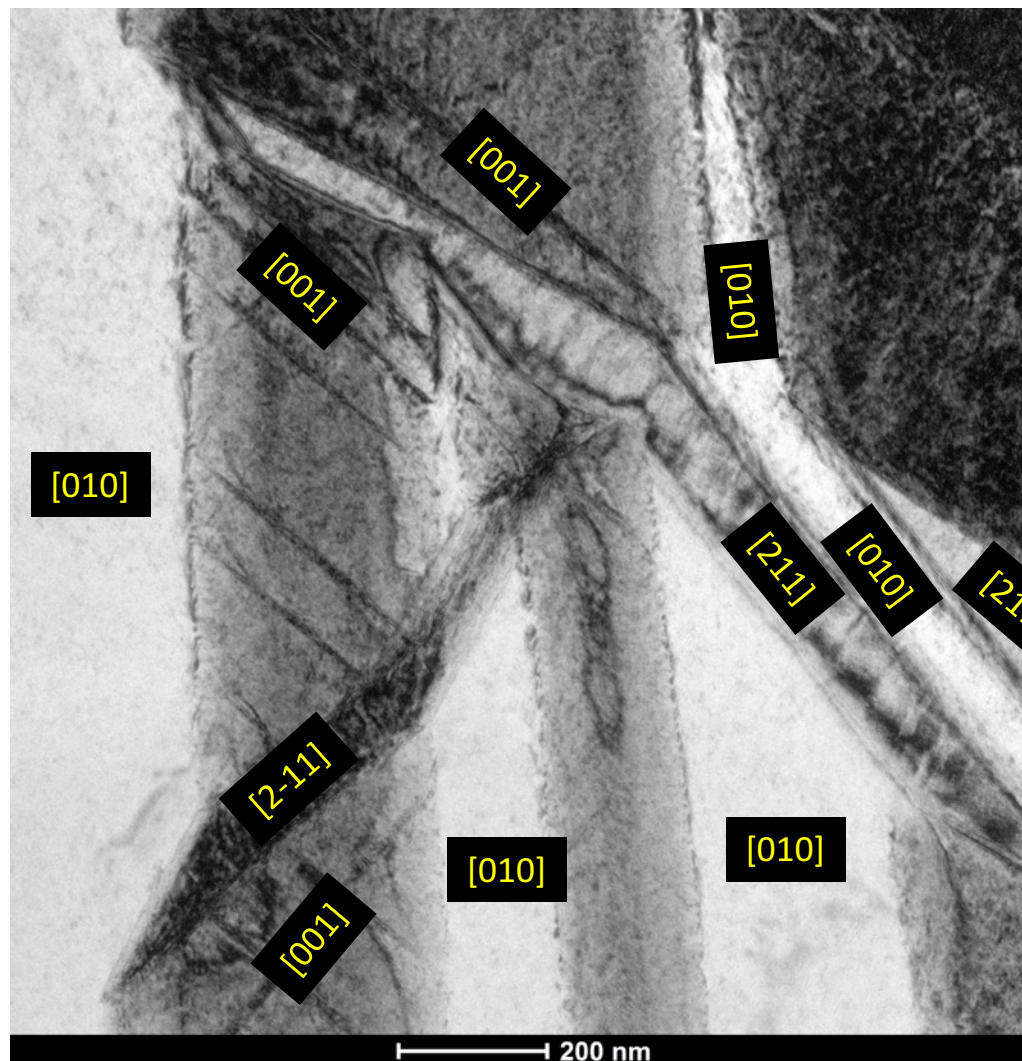
Zone Axis Correspondence  
between  $\alpha'$  and  $\alpha''$  in  
 $[110]_{\text{BCC}}$

$\alpha'$ (hex)	$\alpha''$ (ort)
v1, [0001]	v1, [001]
v2, [000-1]	
v3, [10-10]	v2, [010]
v4, [-1010]	
v5, [5-4-13] or [2-11]	v3, [211] v4, [-211] v5, [2-11] v6, [-21-1]
v8, [-541-3] or [-21-1]	
v10, [5-4-1-3] or [2-1-1]	
v12, [-5413] or [-211]	



TBBF,  $g=-22-2_{\text{BCC}}$

# Interaction of Different Martensite Variants

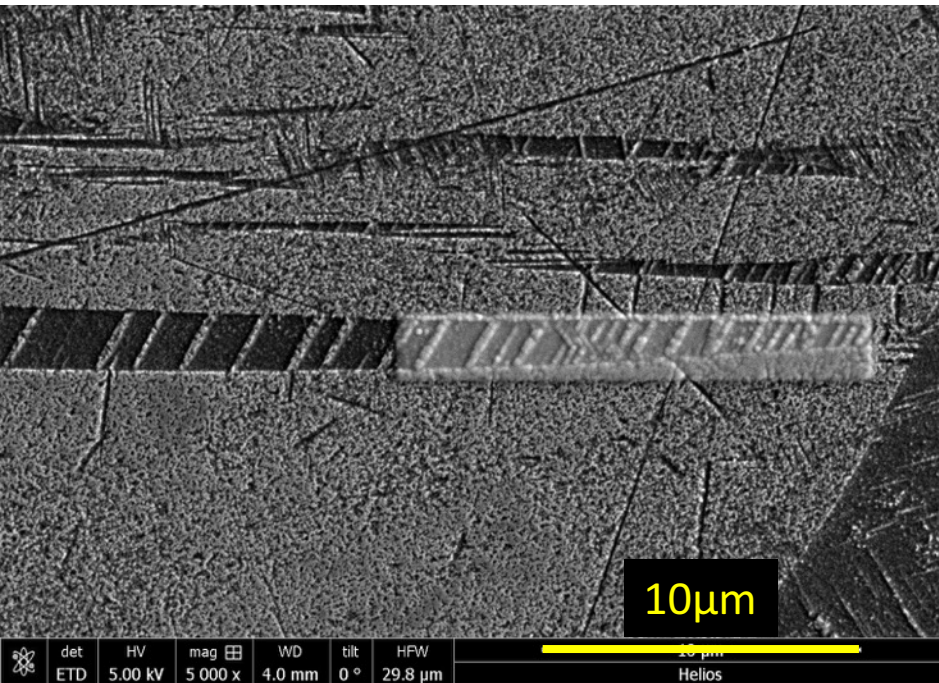




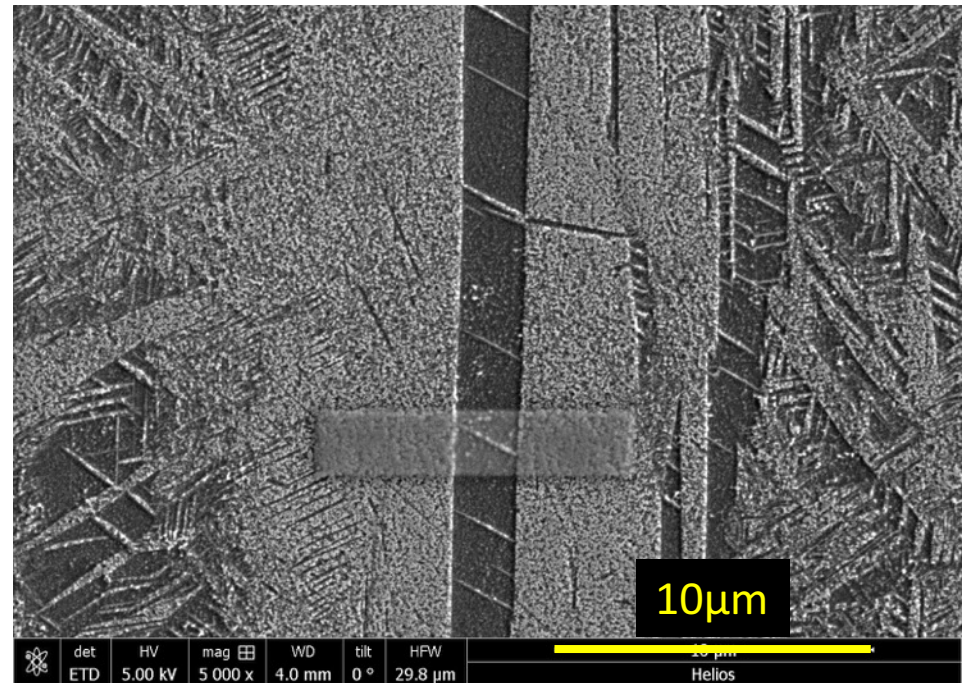
# 1.7% Deformed Ti-1023 (Tension)

1h@850°C, small grain size

lift-out-A



lift-out-C

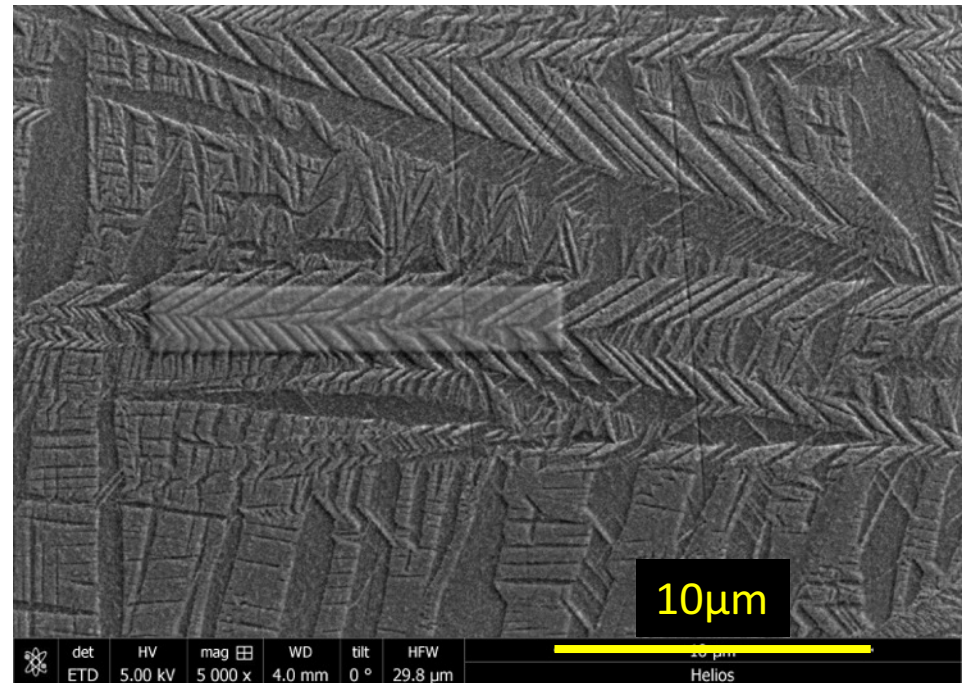
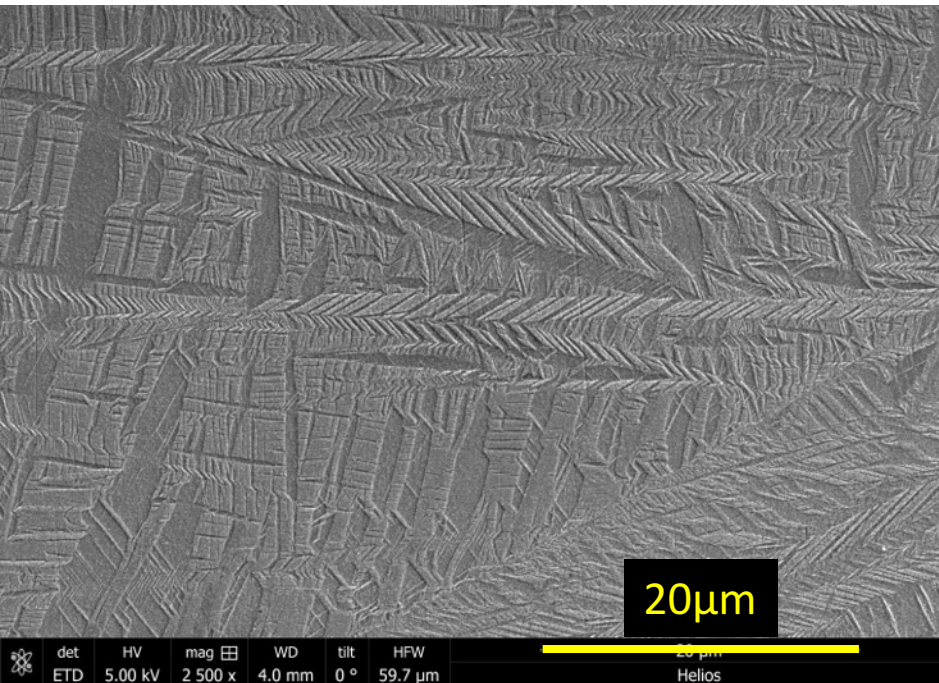


# 1.7% Deformed Ti-1023 (Tension)



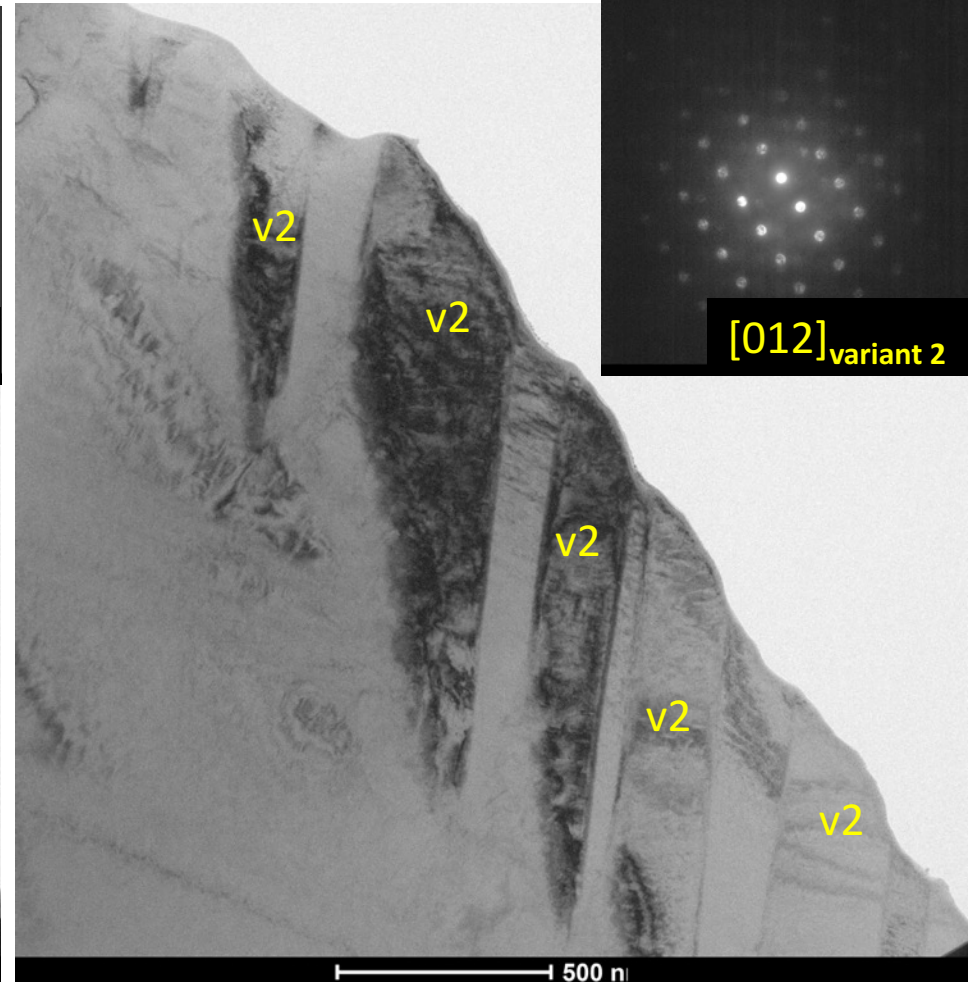
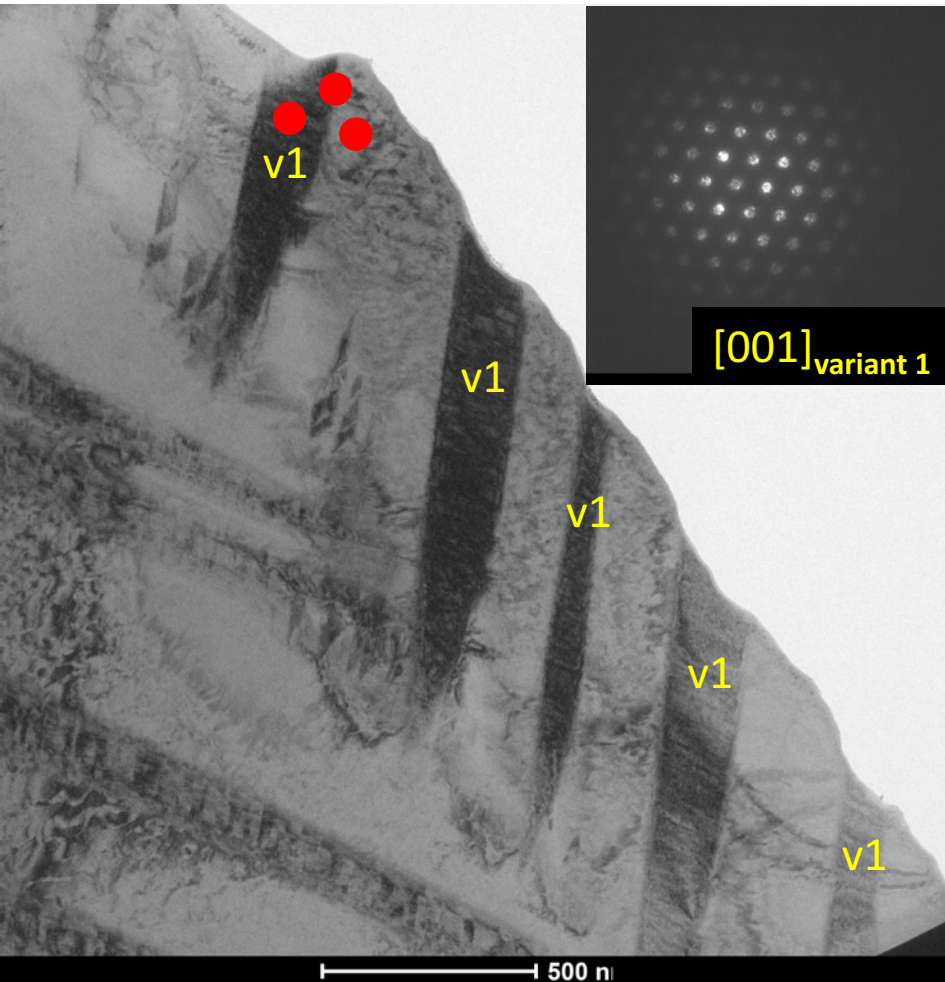
# 5% Deformed Ti-1023 (Compression)

2h@900°C followed by water quenching

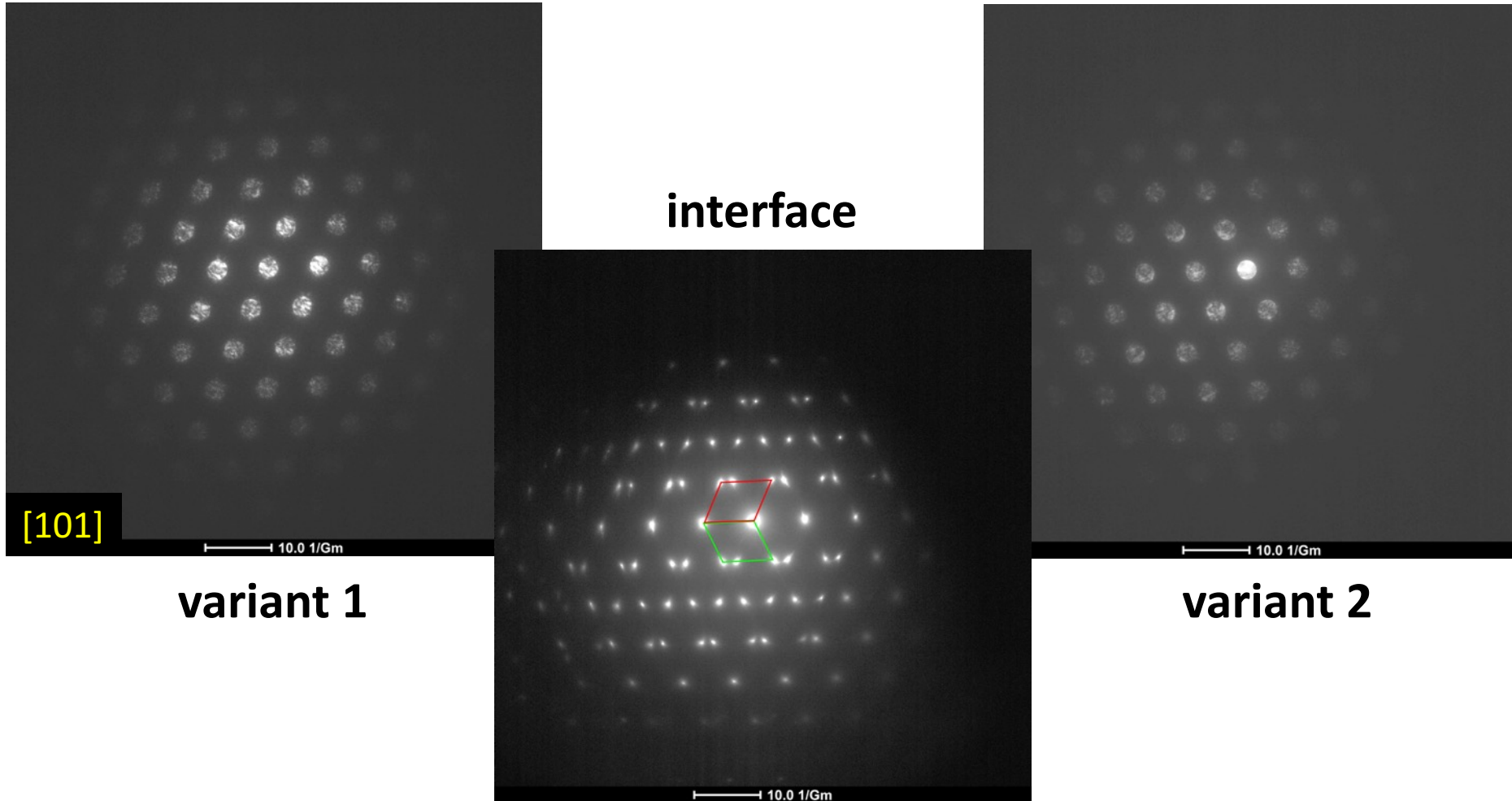




# Fully Martensitic Fishbone

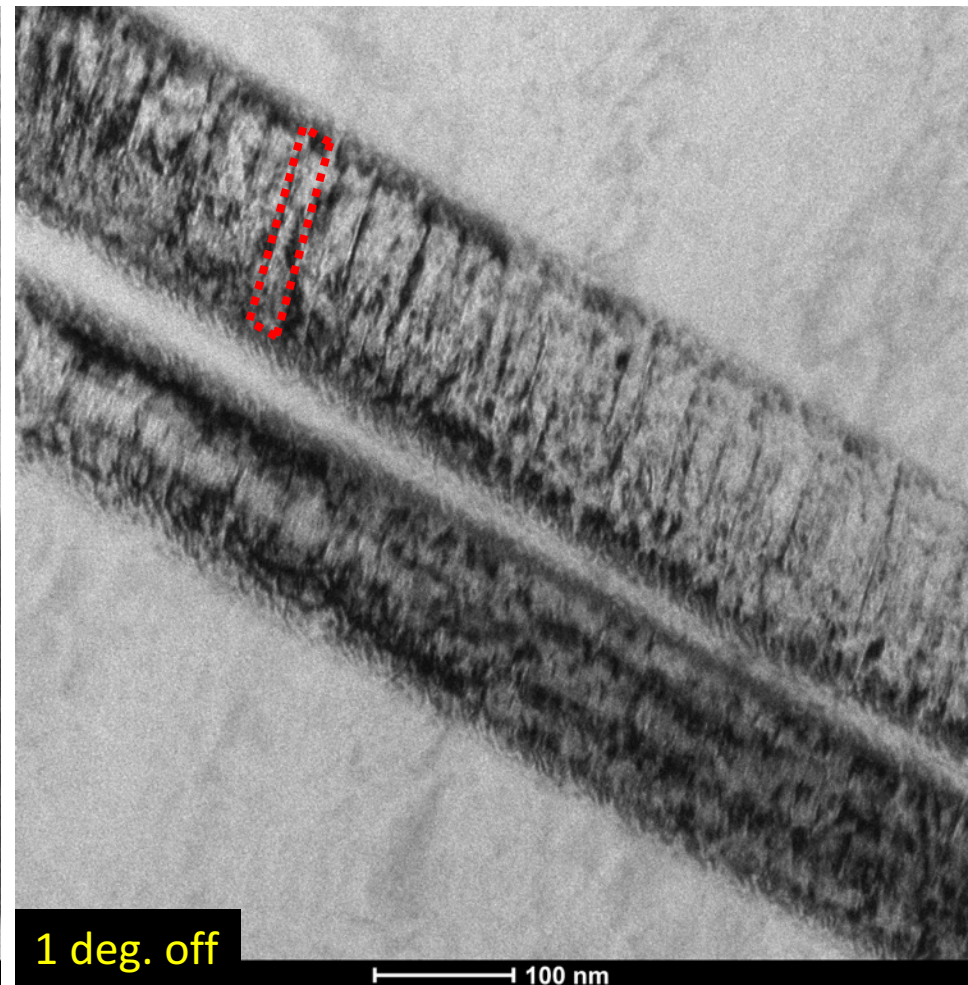


# Twinned Martensites





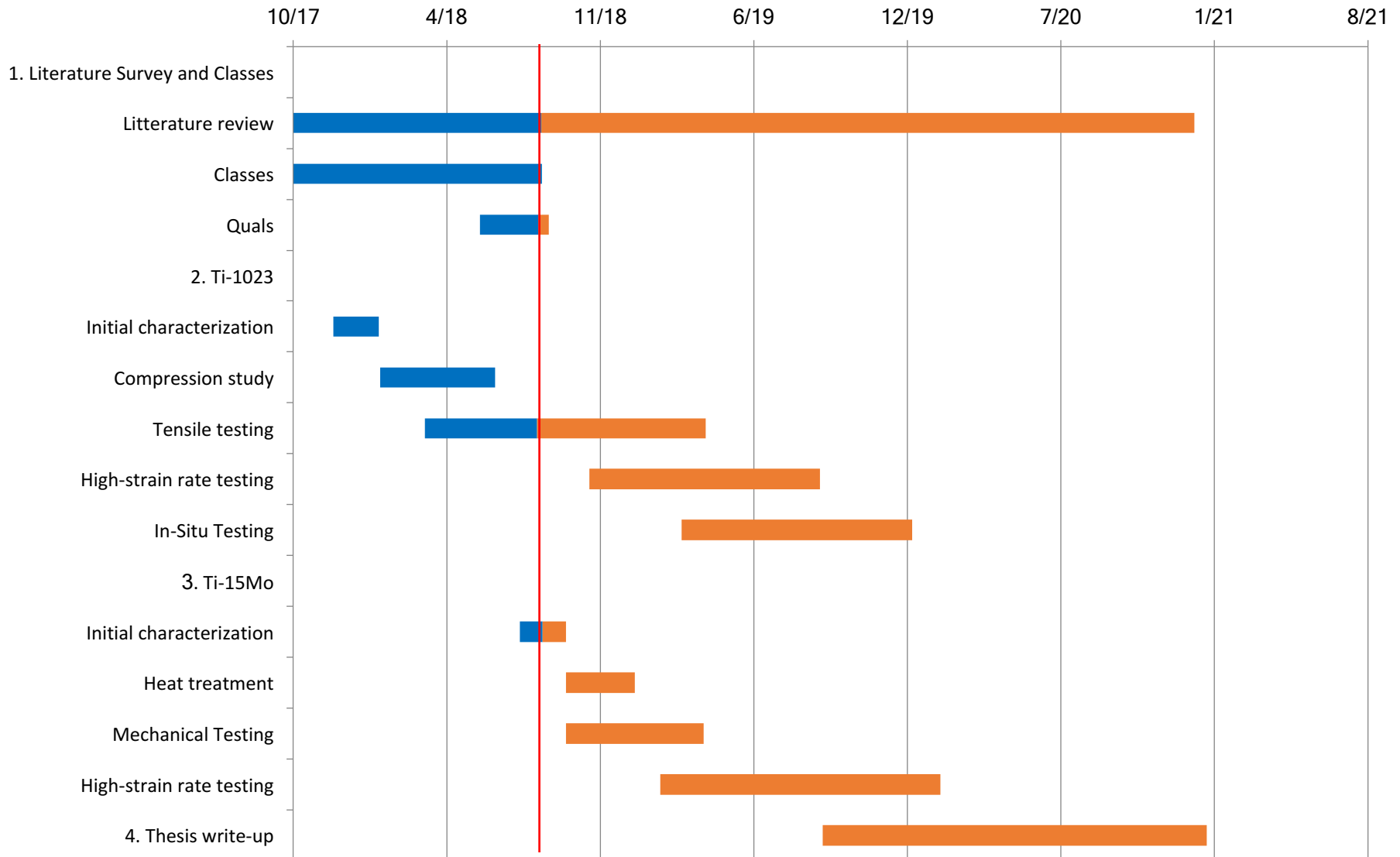
# Substructure within Each Variant



# Summary

- Very **fast** evolution of deformation products at **low** plastic strains
- **No** twinning or stress-induced  $\omega$  was observed
- Martensite is more likely to be  **$\alpha'$**  instead of  $\alpha''$
- Different martensitic variants were indexed

# Updated Progress Summary



Thank you very much!

Benjamin Ellyson

[bellyson@mines.edu](mailto:bellyson@mines.edu)

Yaofeng Guo

[guoyaofeng@mines.edu](mailto:guoyaofeng@mines.edu)



## Project 33a-L: In-situ Studies of Strain Rate Effects on Phase Transformations and Microstructural Evolution in $\beta$ -Titanium

**Student:** Benjamin Ellyson

**Faculty:** Amy Clarke

**Industrial Partners:** Boeing (Austin Mann) ARFL (TBD)

**Project Duration:** Sept. 2017 – May 2021

### Achievement

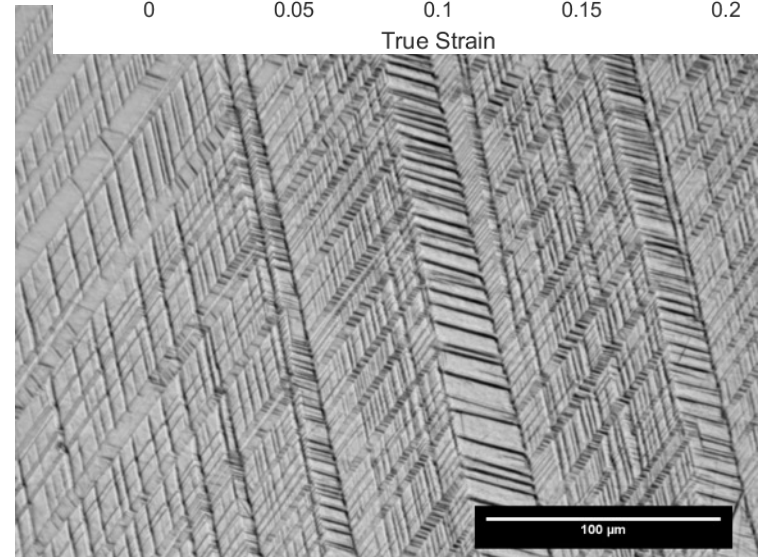
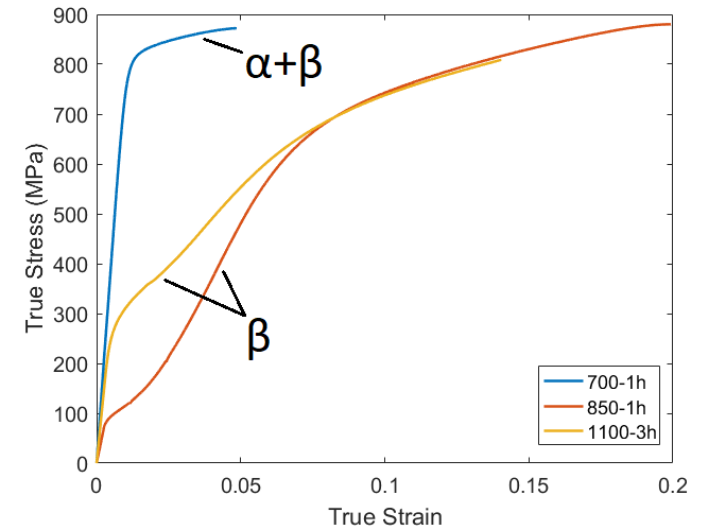
- Develop alloy design methodology to tailor microstructural evolution and mechanical properties for blast resistance

### Significance and Impact

- New TRIP/TWIP metastable  $\beta$ -titanium alloys represent a new class of lightweight blast resistant armor and greatly extend formability to complex structural parts

### Research Details

- Testing over multiple scales (macroscopic to fine scale) and strain rates (quasi-static to dynamic) will inform alloy design



# Project 33a-L: In-situ Studies of Strain Rate Effects on Phase Transformations and Microstructural Evolution in $\beta$ -Titanium

**Student:** *Benjamin Ellyson*

**Faculty:** *Amy Clarke*

**Industrial Partners:** *ORNL (Amit Shyam)*

**Project Duration:** *Sept. 2017 – May 2021*

## Achievement

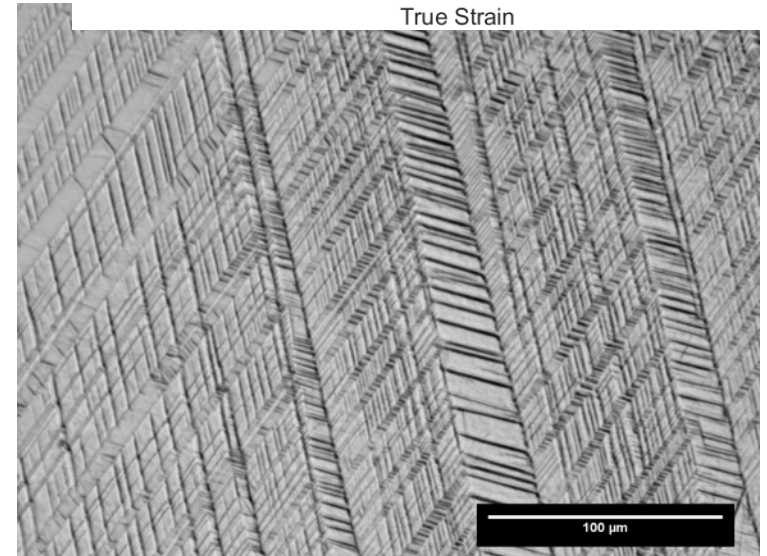
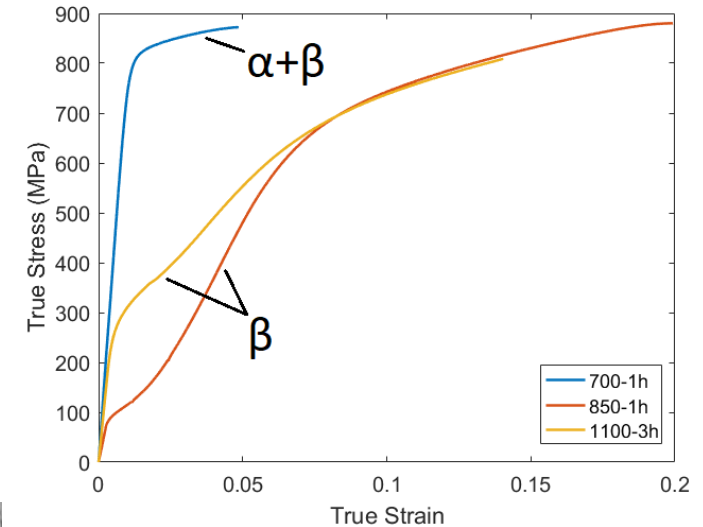
- Develop alloy design methodology to tailor microstructural evolution and mechanical properties for blast resistance

## Significance and Impact

- New TRIP/TWIP metastable  $\beta$ -titanium alloys represent a new class of lightweight blast resistant armor and greatly extend formability to complex structural parts

## Research Details

- Testing over multiple scales (macroscopic to fine scale) and strain rates (quasi-static to dynamic) will inform alloy design



# Project 33a-L: In-situ Studies of Strain Rate Effects on Phase Transformations and Microstructural Evolution in $\beta$ -Titanium

**Student:** *Benjamin Ellyson*

**Faculty:** *Amy Clarke*

**Industrial Partners:** *Boeing (Austin Mann)*

**Project Duration:** *Sept. 2017 – May 2021*

---

## Program Goal

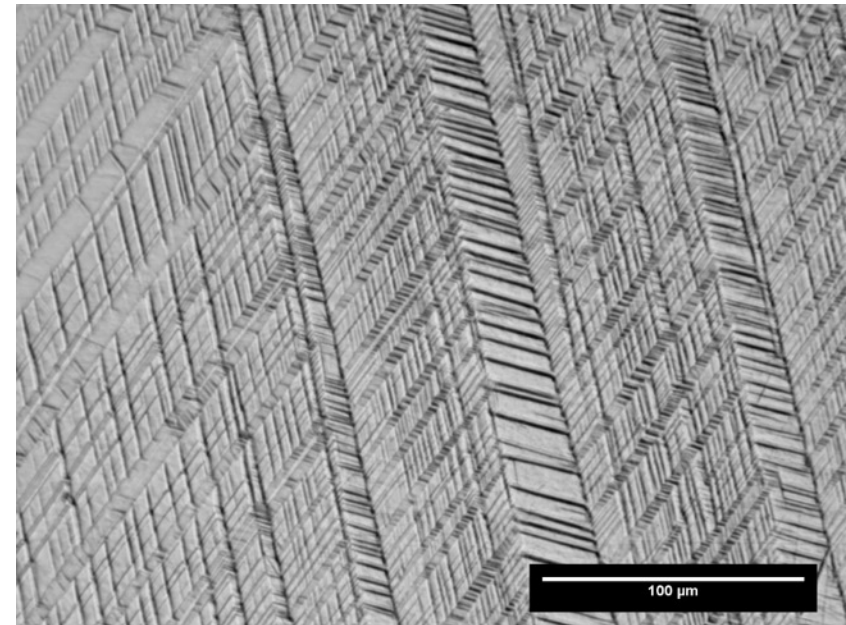
- understand and model microstructural evolution in metastable  $\beta$  Titanium alloys to tailor properties for blast and crash resistance

## Approach

- Utilize testing and in-situ testing and characterization over multiple length scales and decades of strain rates to understand metastable deformation mechanisms

## Benefits

- Improved understanding of microstructural evolution in metastable  $\beta$  Titanium alloys and formulation of novel alloys with tailored performance



Microstructural evolution in Ti-1023 compressed to 5%