

Project 34: In-situ Observation of Phase and Texture Evolution Preceding Abnormal Grain Growth in Ni-based Aerospace Alloys

***Spring 2018 Semi-Annual Meeting
Colorado School of Mines, Golden, CO
April 11-12, 2018***

Student: Byron McArthur (Mines)

Faculty: Amy Clarke (Mines), Kester Clarke (Mines)

Industrial Mentor(s): Eric Payton, Adam Pilchak (AFRL), Kevin Severs (ATI)



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Project 34 - In-situ Observation of Phase and Texture Evolution Preceding Abnormal Grain Growth in Ni-based Aerospace Alloys

- Student: Byron McArthur (Mines)
- Advisors: Amy Clarke, Kester Clarke (Mines)

Project Duration

PhD: Nov. 2017 to Dec. 2020

Problem: Abnormal grain growth in Ni-based superalloys, occurring as a result of forging parameters, significantly reduces mechanical properties.

Objective: Determine the mechanism of abnormal grain growth in Ni-based superalloys using ex-situ and in-situ characterization techniques.

Benefit: Improved mechanical properties for turbine disc alloys.

Recent Progress

- Preliminary literature review
- Initial forgings performed
- Beginning material characterization
- Recreated abnormal grain growth phenomena

Metrics

Description	% Complete	Status
1. Literature review	50%	●
2. Explore abnormal grain growth forging parameters for RR1000	15%	●
3. Ex-situ and interrupted material characterization	10%	●
4. Develop and test theory to explain abnormal grain growth phenomena	5%	●
5. Perform in-situ microscopy with a synchrotron source (HEDM) to demonstrate phenomena	0%	●



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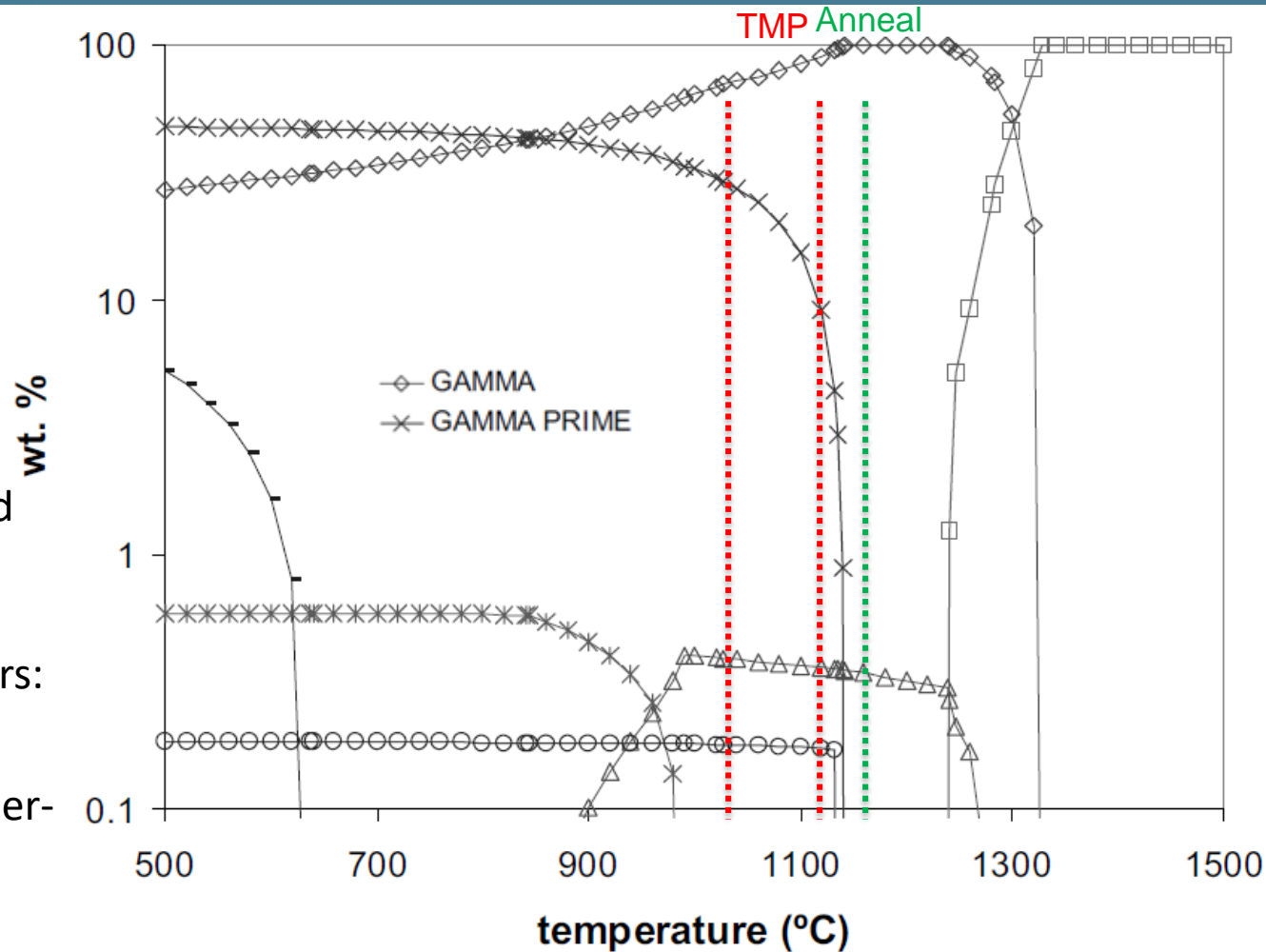
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Industrial Relevance

- Turbine engine discs are flight-critical components
- Forging parameters may induce abnormal grain growth (AGG)
 - Reduction in fatigue life
- Applicable to other superalloys and material systems

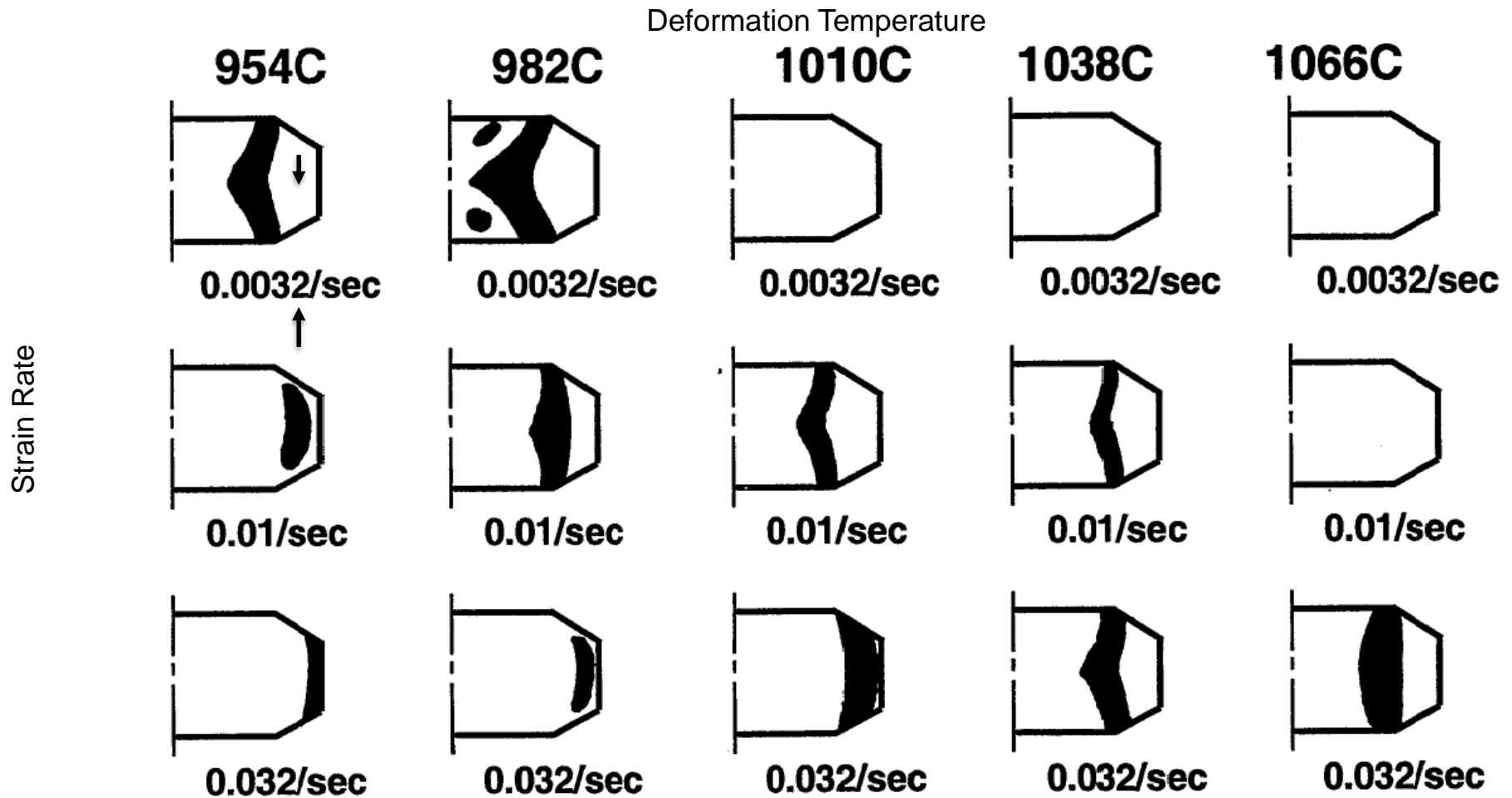
Material Overview

- Material: RR1000
 - $\gamma - \gamma'$
- Processing:
 - Powder-metallurgy
 - Extruded
 - TMP
 - 1035-1110°C
 - Super γ' solvus hold
 - 1150/1170°C
 - < 1 Hr
- Critical AGG parameters:
 - $\dot{\epsilon}$
 - Heating rate to super- γ' solvus hold
 - TMP temperature



Hardy, M. C., Zirbel, B., Shen, G., & Shankar, R. (2004). Developing damage tolerance and creep resistance in a high strength nickel alloy for disc applications. *Superalloys 2004*, 83–90.

Occurrence of AGG (R'88DT) Double-Cone Compression



Raymond, E., Huron, E., & Srivasta, S. (2000). Control of Grain Size Via Forging Strain Rate Limits for R'88DT. *Superalloys*, 49–58.

Possible Contributions

1. γ' coherency loss
 - Dynamic recrystallization
 - Dislocation interactions
 - Orowan looping
 - Primary/secondary/tertiary γ' coherency & dissolution
2. Special boundaries
 - Mobile boundaries
 - Possible twinning contributions
3. Stored energy difference in grains
 - Local gradients in plastic deformation lead to preferential grain growth

Prior literature suggests multiple contributions



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Ex-situ Experiments

- Recreate AGG in Gleeble
 - Controlled T , \dot{T} , ϵ , and $\dot{\epsilon}$
- Obtain load-displacement curves
- Model values in DEFORM[®] software
 - Local strains, strain rates, temperatures
- Ex-situ characterization
 - Search for abnormal grain growth
 - Perform interrupted testing

*note: testing so far has been performed on one of two supplied material conditions

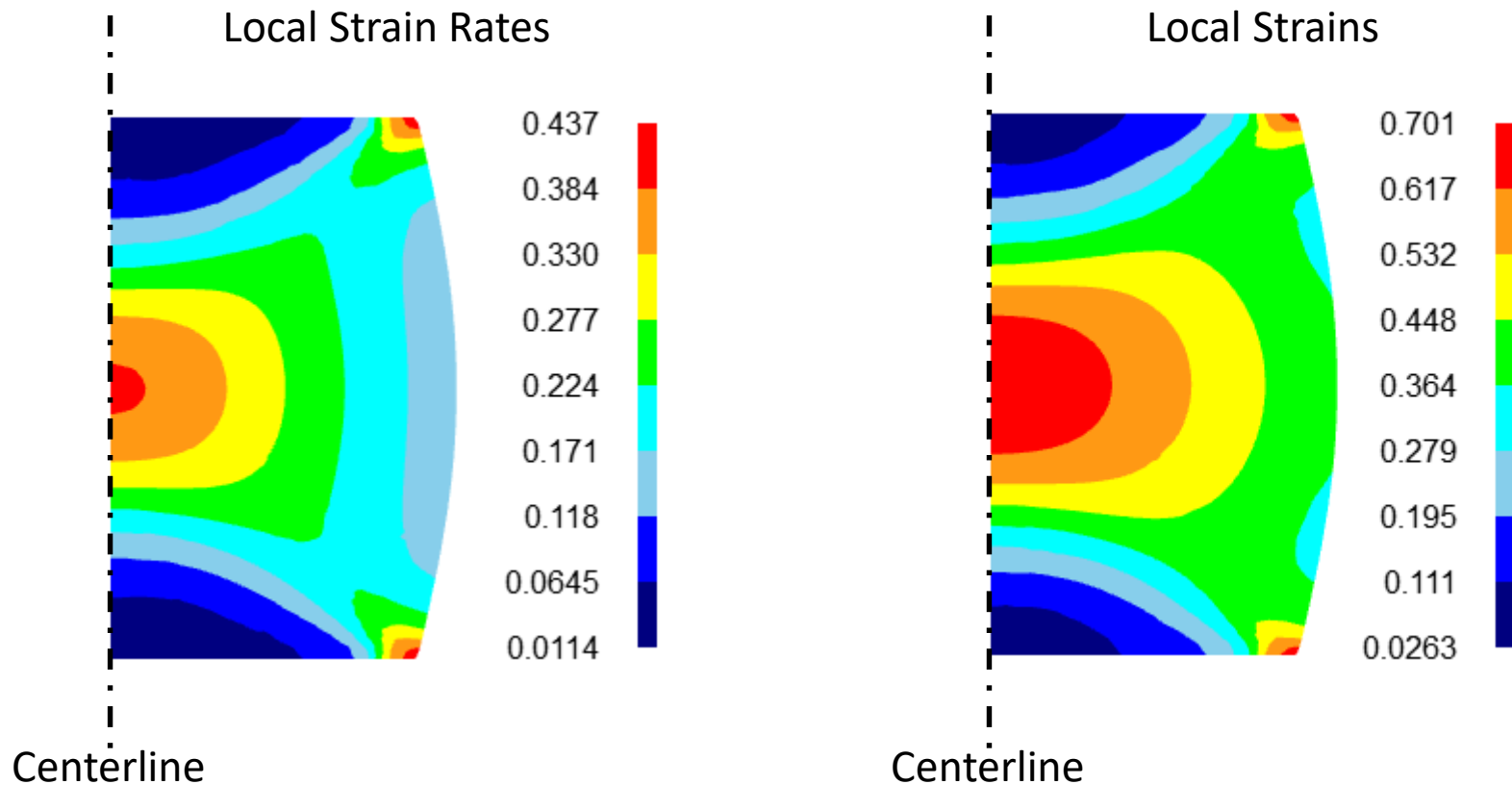


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FEA Modeling - Preliminary



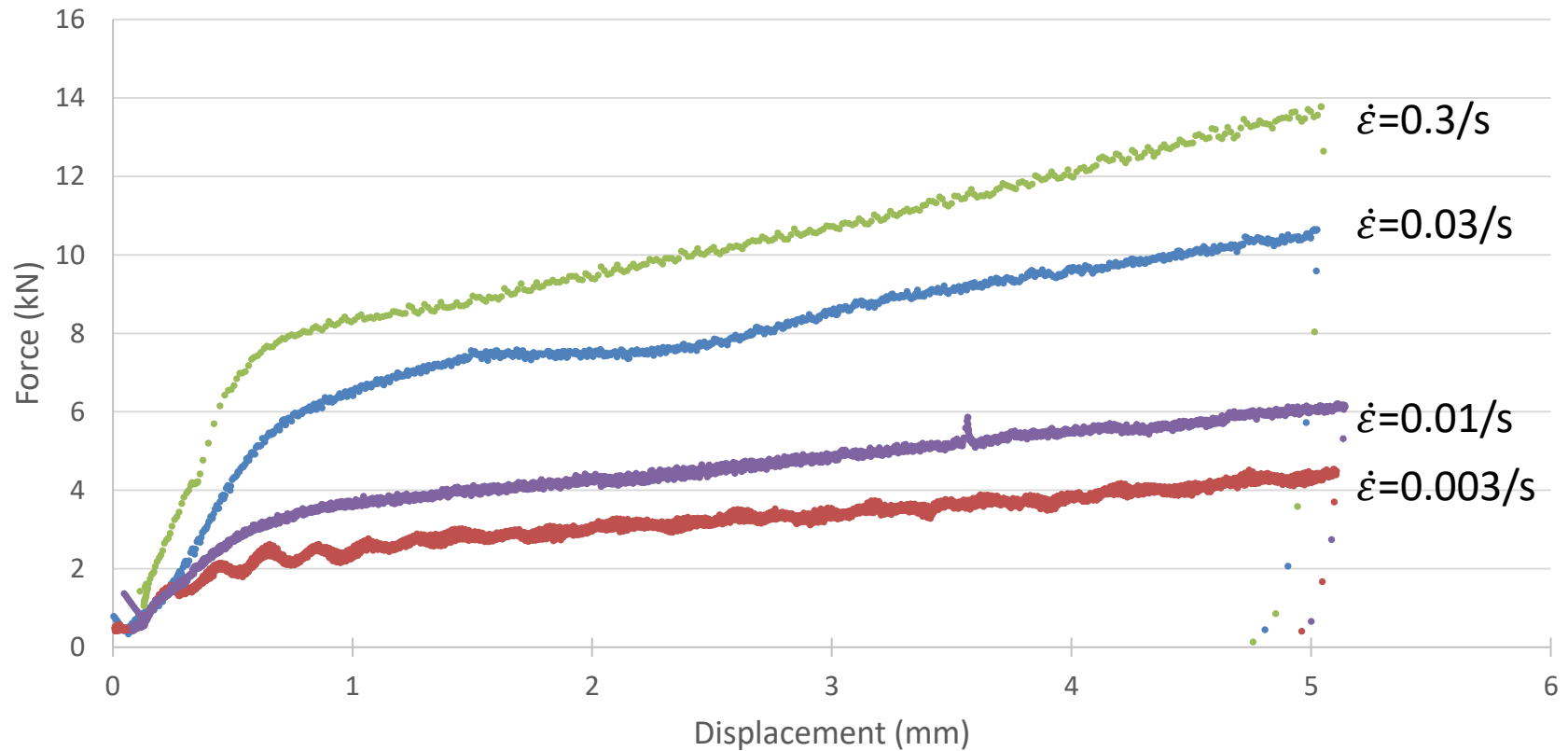
Input parameters of $\epsilon = 0.33$ and $\dot{\epsilon} = 0.3$. Basic material properties (to be refined)

FEA Modeling - Continued

- Local strain accumulation and strain rates greatly influenced by local flow stresses and temperatures
- Accurate modeling requires high fidelity data input
- Need more Gleeble load-displacement data

Gleeble Load–Displacement Curves

Load vs. Displacement RR1000 - 1075°C

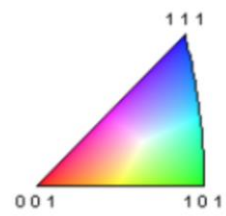


Material Characterization

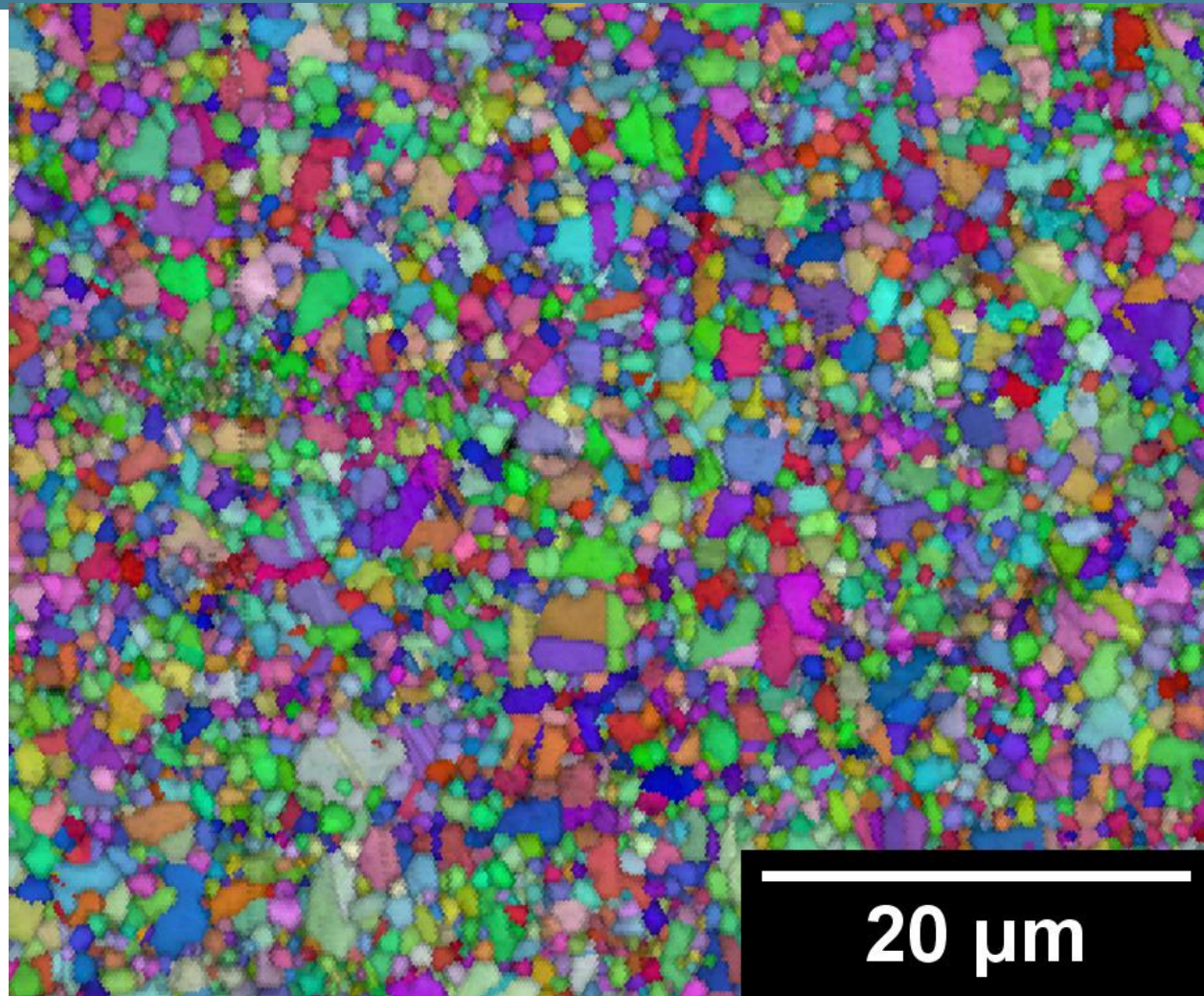
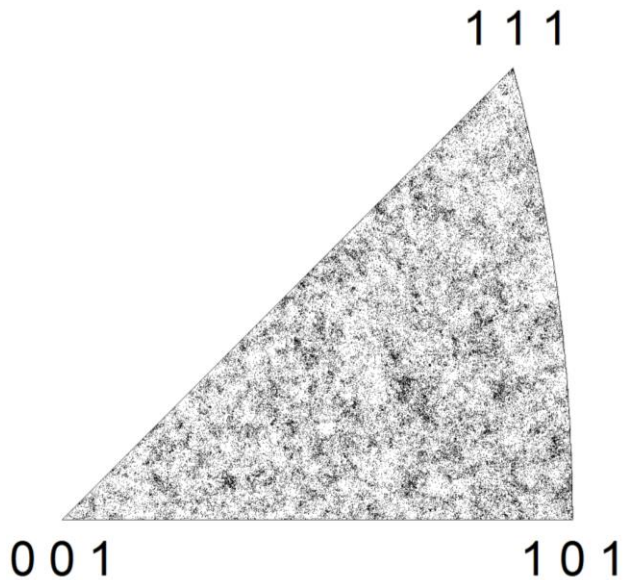
Electron Backscatter Diffraction (EBSD)

- Determine grain size distribution
- Local stored energy
 - Grain Reference Orientation Deviation (GROD)
 - Indication of dislocation density within grain
- Follow TMP processing route:
 - As received
 - Forged
 - Heat treated

As-Received Material

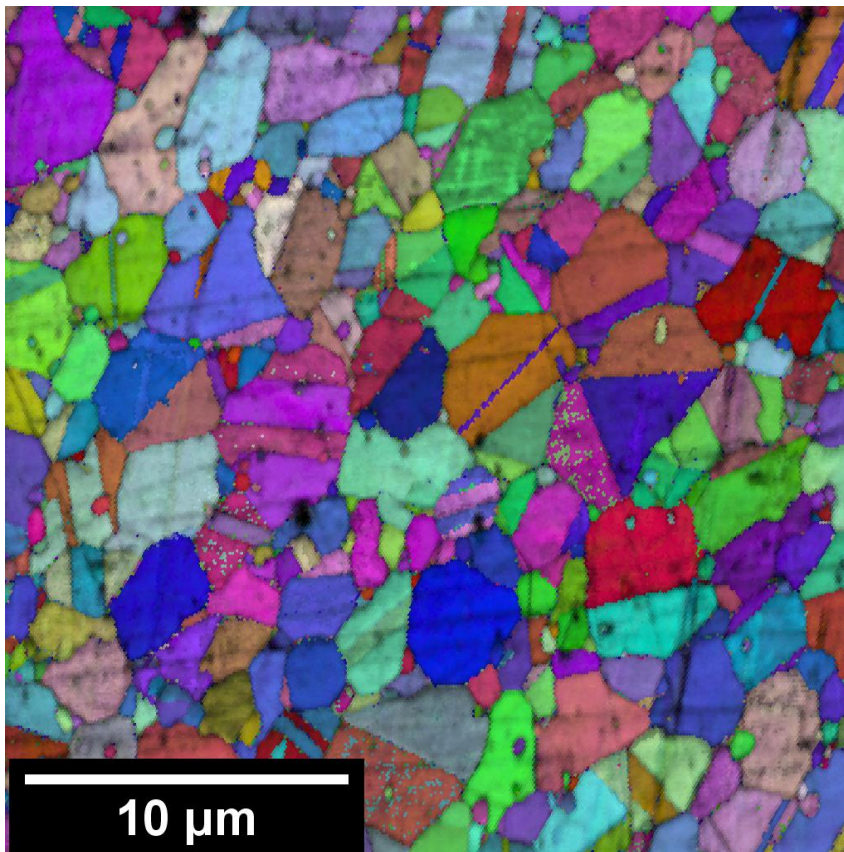


- Isotropic material
- Small starting grain size

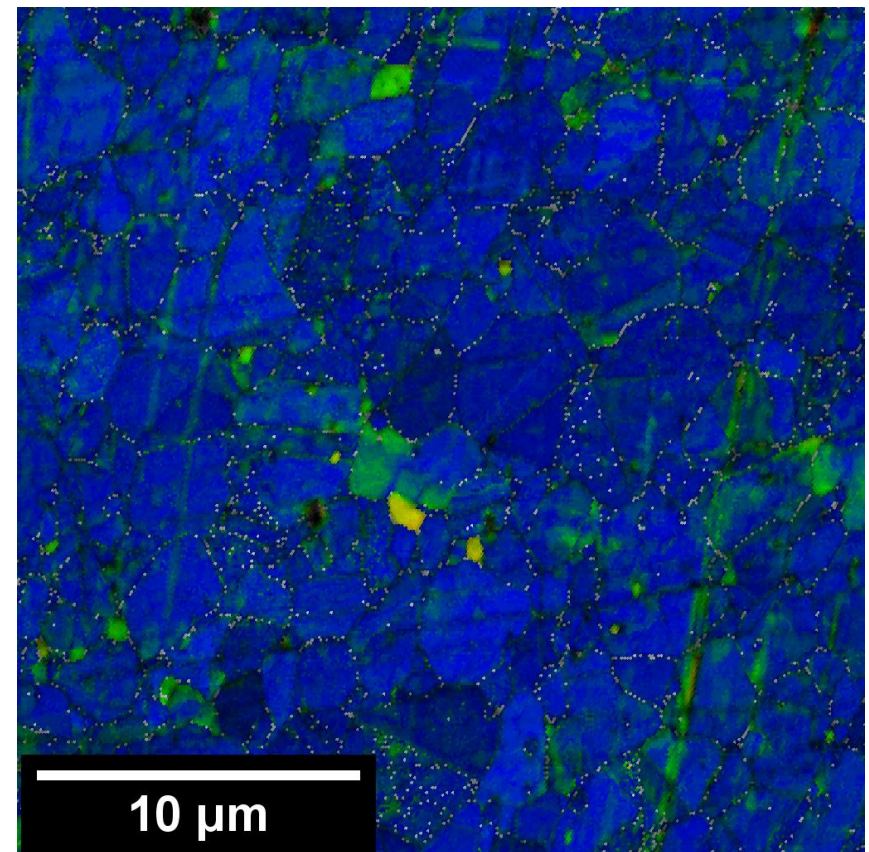


Microstructure – Post Forge, Pre HT

Orientation Map



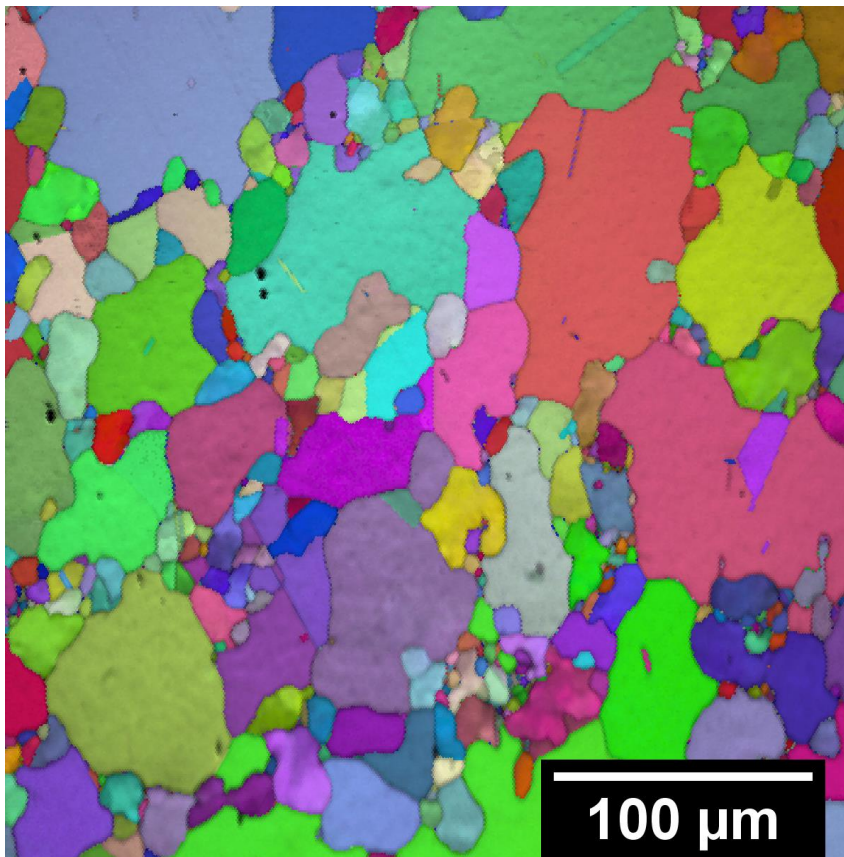
Grain Reference Orientation Deviation



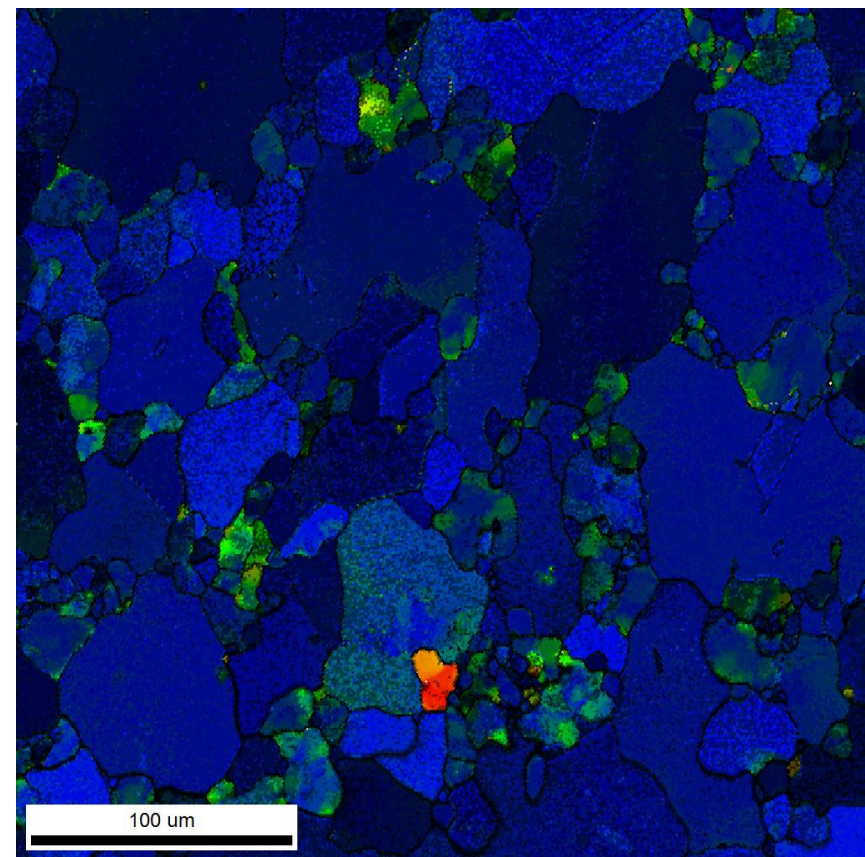
1110°C forging, $\epsilon = 0.3$, $\dot{\epsilon} = 0.003/s$

Microstructure – Post Forge & HT

Orientation Map



Grain Reference Orientation Deviation

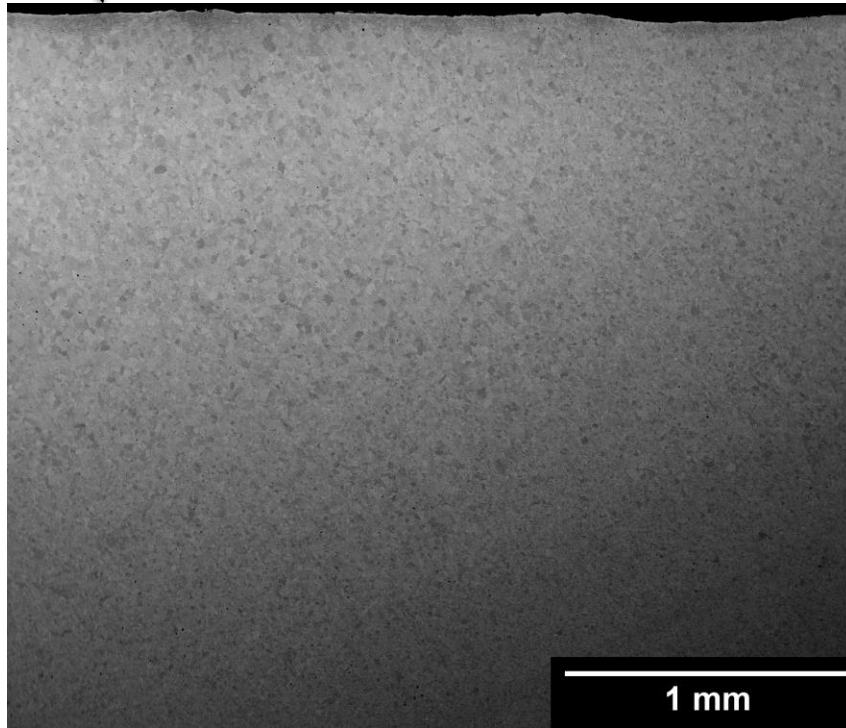


1110°C forging, $\epsilon = 0.3$, $\dot{\epsilon} = 0.003/s$, furnace heating to 1170°C for 20 minutes.

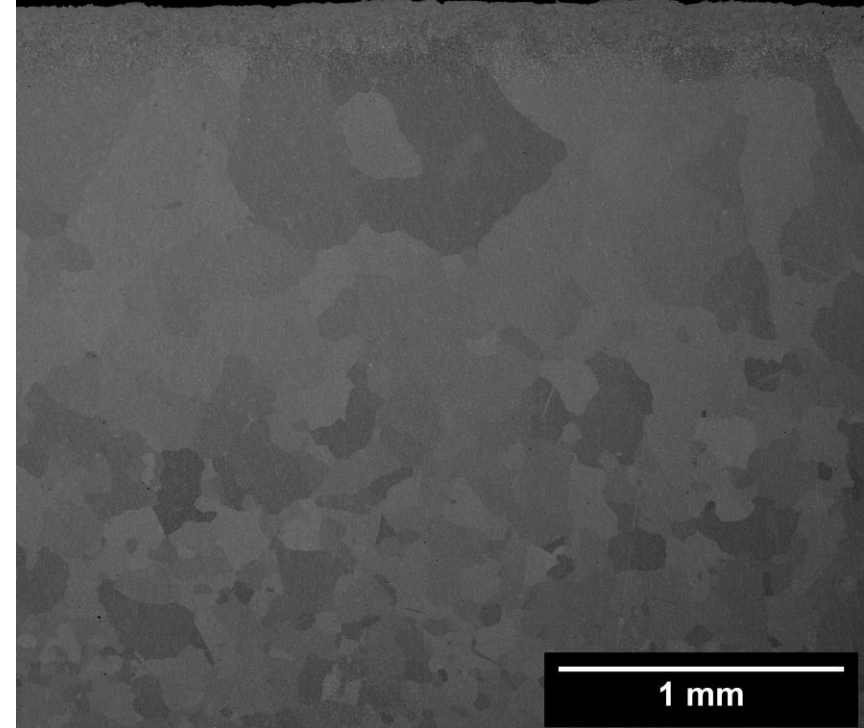
Microstructure – 1075°C Forging, 0.003 /s

Anvil-Specimen
border

Forged (Pre-HT)



Forge & HT



Planned Interrupted Testing

- Utilize forged material at T and $\dot{\epsilon}$ for AGG
- Section & polish
- Perform EBSD
 - Orientation, size distribution, and GROD
- Interrupted heat treatment in dilatometer
 - Determine AGG rate & temperature
- Repeat EBSD analysis
- Continue interrupted heat treatment and EBSD

*Free surface effects may limit validity of test

Planned Gleeble Testing

- Continue exploring forging temperature and strain rate parameters that produce AGG
- Self-impingement of large grains occurs
 - Lower input strain → fewer & larger abnormal grains (?)
- Relatively low strain rate and high forging temperature appear to produce AGG
- Use values for DEFORM[®] modeling

Conditions for AGG – Gleeble Work

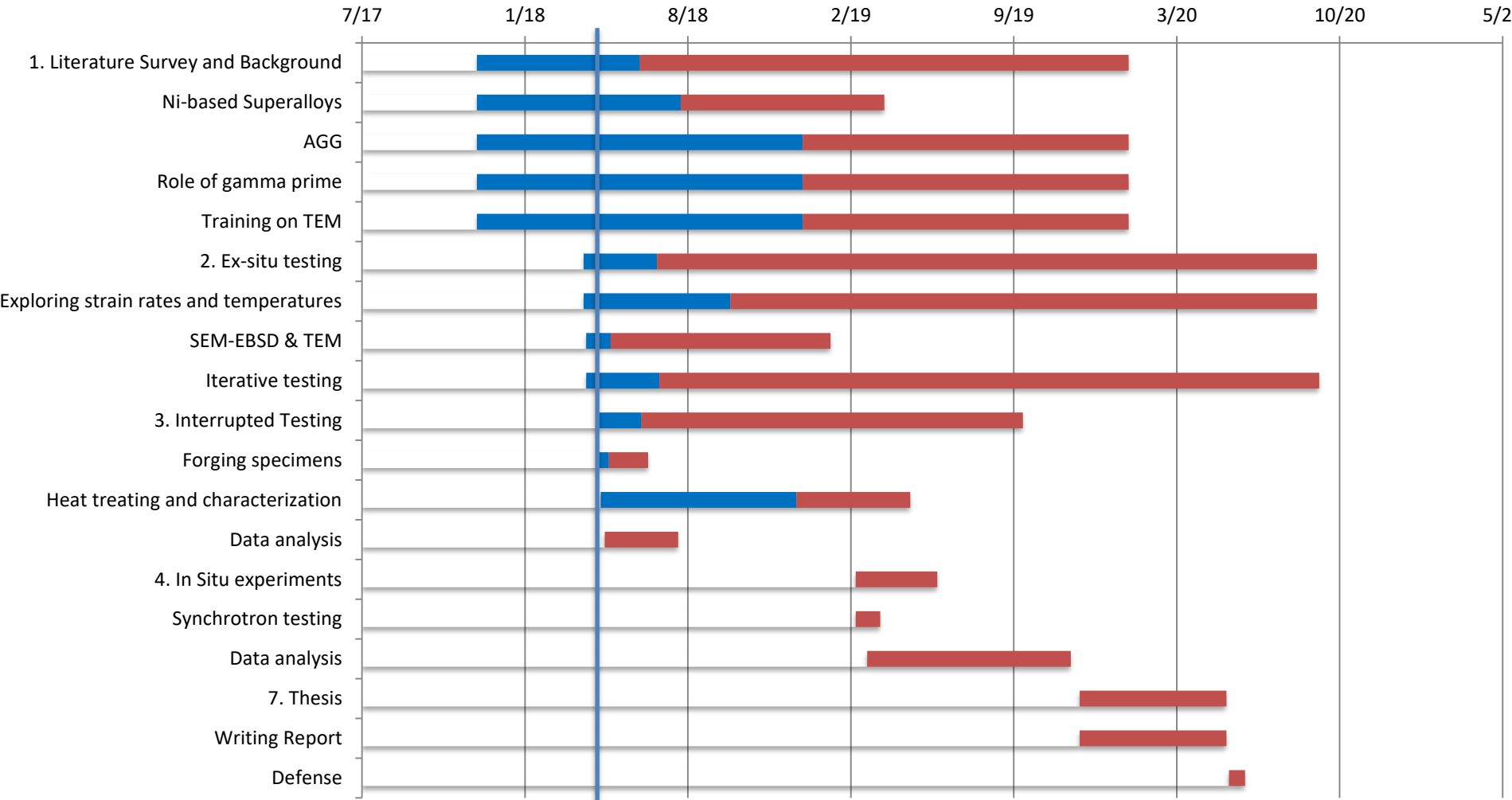
		Temperature (°C)						
		1035	1050	1065	1075	1080	1095	1110
Strain Rate (s ⁻¹)	0.003							
	0.01							
	0.03							
	0.1							
	0.3							

	No AGG
	Partial AGG
	Extensive AGG

Role of γ' in AGG

- The γ' acts to pin γ grain boundaries
- Various sizes form at different stages
- Forging influences the γ'
 - Coherency
 - Size
- Further analysis needed on specimens
 - TEM analysis
 - Incorporate simultaneous EDS-EBSD

Progress



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Thank you!
Questions?

Byron McArthur

bmcarthu@mines.edu



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